# Effect of seawater as mixing water on the hydration behaviour of tricalcium aluminate

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**Abstract:** This study explored the mechanism of the effect of seawater on tricalcium aluminate (C<sub>3</sub>A) hydration. The results showed that seawater retarded C<sub>3</sub>A hydration and reduced the reaction degree of C<sub>3</sub>A. The co-existence or ion pairing of Ca<sup>2+</sup> and SO<sub>4</sub><sup>2-</sup> onto the surface of C<sub>3</sub>A poisoning the reactive sites is the main reason for this retardation. Besides, the precipitation of Mg(OH)<sub>2</sub> on the surface of C<sub>3</sub>A would prolong the induction period for another 30 minutes and then consequently decrease the dissolution rate of C<sub>3</sub>A hydration. Trace amounts of MgSO<sub>4</sub><sup>0</sup> and Mg<sup>2+</sup> present in the alkaline solution had little retardation effect. It was also found that Cl<sup>-</sup> would preferentially react with C<sub>3</sub>A to form Friedel's salt, rather than forming hydroxy-AFm as that in the C<sub>3</sub>A-deionized water paste. The direct formation of Friedel's salt resulted in an accumulation of Al<sup>3+</sup> in solution, which would also hinder the subsequent dissolution of C<sub>3</sub>A.

**Keywords:** Seawater; C<sub>3</sub>A; retardation; hydration; Friedel's salt.

## **1. Introduction**

In Portland cement, tricalcium aluminate ( $C_3A$ ), as the most reactive phase, affects the setting time, workability and rheological property of fresh cement or concrete [1], even though

it only accounts for approximately 5-10 % by mass of ordinary Portland cement. It has been well acknowledged that the reaction rate of  $C_3A$  with water is extremely rapid leading to the precipitation of calcium hydroaluminates. This would lead to flash set [2]. To avoid this phenomenon, gypsum is always incorporated into Portland cement clinkers during the grinding process to delay the hydration.

With respect to the retarding mechanism of gypsum (sulphates) on the C<sub>3</sub>A hydration, the physical barrier effect mainly caused by the precipitation of ettringite (AFt) on the C<sub>3</sub>A surface has been questioned [3]. Geng et al. [4] reported that the needle network of AFt at the dissolution front of C<sub>3</sub>A was highly porous, which was not possible to act as a dense barrier layer. In comparison, more and more researchers support the adsorption mechanism. It has been pointed out that the initial reactivity of  $C_3A$  was reduced through  $SO_4^{2-}$  alone adsorbed onto the active sites of C<sub>3</sub>A surface [5-7] or through  $Ca^{2+}$  and  $SO_4^{2-}$  adsorbed on an aluminiumrich layer that was formed on the surface of C<sub>3</sub>A due to its partially dissolution [8-10]. However, Myers et al [11, 12] put forward that the adsorption behaviour of  $SO_4^{2-}$  on the surface of  $C_3A$ was not supported by the geochemical literature, because little or no adsorption of SO4<sup>2-</sup> onto Al (hydr)oxides above pH 7-10. Therefore, they proposed that the ion pairing of CaSO<sub>4</sub><sup>0</sup> would adsorb on this aluminium-rich layer, which would decrease the local undersaturation of  $Ca^{2+}$ in aqueous solution near C<sub>3</sub>A surface sites and consequently retard the C<sub>3</sub>A dissolution. However, in contrast to the adsorption mechanism of  $Ca^{2+}$  and/or  $SO_4^{2-}$ , Joseph et al. [13] suggested that the precipitation of AFt on the surface of C<sub>3</sub>A could also block the active sites of C<sub>3</sub>A. This process would be consistent with a dissolution-controlled mechanism [10], rather than a diffusion-controlled mechanism which is the case for the coating mechanism of AFt. Referring to these studies above, the retarding mechanism on the initial dissolution of C<sub>3</sub>A still need to be further explored.

Meanwhile, it was also reported that the type of sulphates would determine the delaying effect on the C<sub>3</sub>A hydration. When the amount of SO<sub>4</sub><sup>2-</sup> was the same, MgSO<sub>4</sub> had the greatest retardation effect on the hydration of C<sub>3</sub>A, followed by CaSO<sub>4</sub> and Na<sub>2</sub>SO<sub>4</sub> [14]. For this phenomenon, it can be inferred that the adsorption of  $SO_4^{2-}$  alone on the surface of  $C_3A$  is not the sole reason for retarding the hydration of  $C_3A$ . The cation-specific effect was proposed to explain it [15, 16]. As for the retarding mechanism of MgSO<sub>4</sub> on C<sub>3</sub>A hydration, a simple extension from CaSO<sub>4</sub> to MgSO<sub>4</sub>, i.e., the adsorption of Mg<sup>2+</sup>, SO<sub>4</sub><sup>2-</sup> or MgSO<sub>4</sub><sup>0</sup> onto the reactive sites of C<sub>3</sub>A to inhibit its initial dissolution, was proposed, considering many of the comparable chemical properties between  $Ca^{2+}$  and  $Mg^{2+}$  [14, 15]. Moreover, it was also reported that, when C<sub>3</sub>A was mixed with MgSO<sub>4</sub>, there was a lower ratio of Ca/Al and a higher ratio of Mg/Ca on the surface of  $C_3A$  particles [16], implying that Mg<sup>2+</sup> would partially substitute for  $Ca^{2+}$  in the near surface crystal structure due to a higher affinity on the  $C_3A$ surface and a relatively smaller ionic radii [16, 17]. The replace of Ca<sup>2+</sup> by Mg<sup>2+</sup> would contribute to a relatively longer induction period. Based on the above disputation, the magnesium species present in alkaline environments should be clarified at first, then the role of different magnesium species on C<sub>3</sub>A hydration needs to be further analysed.

Currently, the possible use of seawater for concrete production has gained increasing interest [18-22]. In the above reviewed studies, the effect of sulphate alone on the hydration of  $C_3A$  was emphasized. In comparison, seawater is a complex system, containing different cations and anions. The specific ions in seawater that are responsible for inhibiting the hydration of  $C_3A$  are still unclear. The purpose of this study is thus to fill the research gap, understand the interaction between the  $C_3A$  and seawater and further explore how these interactions influence the initial dissolution and hydration evolution of  $C_3A$ . This work would provide new insights on the role of seawater in regulating the  $C_3A$  hydration process.

However, it has been shown that the C<sub>3</sub>A exhibits polymorphism, which depends on the alkali contents and is also influenced by the cooling rate during the production process of cement [23-25]. In cement clinkers, cubic C<sub>3</sub>A and orthorhombic C<sub>3</sub>A usually exist alone or in combination [23]. The different crystal structure of C<sub>3</sub>A can influence its reactivity. It has been reported by Boikova et al. [26] that, in the absence of sulphates, orthorhombic C<sub>3</sub>A was less reactive than cubic C<sub>3</sub>A. However, when sulphate was present, the hydration of orthorhombic C<sub>3</sub>A was faster than that of the cubic C<sub>3</sub>A [27, 28]. But, as reported in the literatures [29-35], the relative amount of cubic C<sub>3</sub>A to orthorhombic C<sub>3</sub>A in Portland cements indicates that the cubic C<sub>3</sub>A is usually the dominant phase in most commercial cements. Therefore, this study focused on the hydration behaviour of the cubic C<sub>3</sub>A when hydrating in seawater. Isothermal Calorimetry, XRD, SEM, ICP-OES and Ion Chromatography were carried out to characterize the development of cubic C<sub>3</sub>A hydration using deionized water (DI) and simulated seawater (SW) respectively as the mixing water. Thermodynamic modelling software (GEMS) was further used to simulate the evolution of equilibrium phases and the change of solution chemistry of seawater under different pH conditions.

# 2. Materials and experimental methods

### **2.1 Materials**

A synthesized cubic tricalcium aluminate (C<sub>3</sub>A) in powder form (provided by DMT Materials Technology Company in Shanghai of China) was used in this study. The specific surface area of C<sub>3</sub>A measured by N<sub>2</sub> adsorption and desorption was  $611.9 \text{ m}^2/\text{kg}$ , and its particle size distribution and morphology are shown in Fig. 1, indicating that its median particle diameter was 13.7 µm and its shape was angular. The QXRD result in Fig. 2 shows that the purity of C<sub>3</sub>A was higher than 95 %, and the free lime and mayenite (C<sub>12</sub>A<sub>7</sub>) contents were about 0.04 % and 4.30 %, respectively.

According to ASTM D1141-98 [36], a simulated seawater was prepared and its chemical composition and main ionic concentration are shown in Table 1. 0.1 N NaOH solution was used to adjust the pH value of seawater to 8.2. All chemicals used in this work were analytical grades.



Fig. 1 The particle size distribution and morphology of C<sub>3</sub>A



Fig. 2 XRD of C<sub>3</sub>A by Rietveld refinement

Table 1. Chemical composition and main ionic concentration of simulated seawater [36]

Chemicals	Concentration (g/L)	Ions	Ionic concentration (g/L)
NaCl	24.530	$Na^+$	11.020

MgCl <sub>2</sub>	5.200	$Mg^{2+}$	1.326
$Na_2SO_4$	4.090	$Ca^{2+}$	0.418
CaCl <sub>2</sub>	1.160	$\mathbf{K}^+$	0.400
KCl	0.695	Cl <sup>-</sup>	19.830
NaHCO <sub>3</sub>	0.201	$SO_4^{2-}$	2.765
KBr	0.101	HCO <sub>3</sub> -	0.146
H <sub>3</sub> BO <sub>3</sub>	0.027	Br⁻	0.068
$SrCl_2$	0.025	$Sr^{2+}$	0.014
NaF	0.003	-	-

### 2.2 Sample preparation

In this work, DI and SW were used to prepare C<sub>3</sub>A pastes using a water-to-solid ratio of 1.0. At first, the C<sub>3</sub>A powder and DI or SW were placed in 50 mL centrifugal tubes, and then these samples were vigorously agitated in a vortex apparatus for 1 min. After the required reaction durations from 2 min to 12 h, the liquid was collected after centrifugation at 10000 rpm for 5 min and passing through a 0.45  $\mu$ m membrane filter. Then the filtrated liquid samples were stored in a fridge at 5 ± 1 °C prior to testing. Additionally, the solid residues remained in the tubes were immediately immersed in liquid nitrogen for about 15 min. Then the frozen samples were transferred to a freeze-dryer for drying for another 5 h. Once dried, the samples without grinding were stored in a desiccator until further testing.

### 2.3 Experimental techniques

#### 2.3.1 Isothermal calorimetry

The hydration heat evolution of C<sub>3</sub>A mixed with DI or SW was monitored using a TAM Air isothermal calorimeter at 20 °C until 3 days. The water-to-solid ratio of C<sub>3</sub>A pastes was set to 1.0. In this work, internal mixing method was used for experimental groups contained C<sub>3</sub>A-DI paste, because the initial process (0-10 min) of hydration was studied due to the extremely rapid hydration reaction of C<sub>3</sub>A-DI paste. 2 g C<sub>3</sub>A powder was placed in a glass ampoule, and 2 g solution was introduced into a syringe. After the solution was injected, the paste was stirred

by a mechanical rotor for 2 min. However, the internal mixing intensity was set to a very low speed to prevent the disturbance or damage of calorimeter [37]. When the early process was not of interest (<10 min), external mixing was preferred [37]. Thus, when comparing the length of induction period and major exothermic peak of C<sub>3</sub>A-SW and C<sub>3</sub>A-S2 (S2 denotes seawater solution excluded MgCl<sub>2</sub>), the external mixing method was used to prepare the pastes by hand stirring using a glass rod for 2 min, and the sample was quickly introduced into the calorimeter (less than 5 min after mixing).

## 2.3.2 Ion chromatography

The concentrations of  $Cl^{-}$  and  $SO_4^{2-}$  in the collected liquid in Section 2.2 were measured using ion chromatography (Dionex AS-DV, Thermo Scientific). At different reaction periods, the filtrates collected were diluted using Milli-Q water prior to testing.

#### 2.3.3 ICP-OES

The concentrations of cations in the collected filtrates, i.e.,  $Ca^{2+}$ ,  $Al^{3+}$ ,  $Mg^{2+}$ , were characterized by an inductively coupled plasma/optical emission spectroscopy (ICP-OES, FMX36, SPECTROBLUE). Before measurement, the liquid samples extracted from C<sub>3</sub>A and SW or DI mixes in Section 2.2 were digested with concentrated nitric acid and then diluted with 5% nitric acid.

## 2.3.4 X-ray diffraction (XRD)

In order to follow the evolution of hydration products at early ages, in-situ XRD measurements were conducted. After mixing with SW or DI for 1 min, the fresh  $C_3A$  paste was directly casted in the XRD sample holder. A Kapton film was used to cover the  $C_3A$  paste to prevent the loss of water and carbonation during the test. The XRD (Rigaku SmartLab 9kW-

Advance) measurement was carried out in a  $2\theta$  range of 5 ° to 36 ° with a step width of 0.02 °, and the scan speed was fixed at 30 °/min. During measurement, CuK $\alpha$  was used as radiation source.

Due to the fast scanning speed and the contribution of the Kapton film and water in the sample, it was not possible to quantify the absolute amounts of various phases. The freezedried samples in Section 2.2 were co-ground with corundum (10 wt.%), and the quantitative X-ray Diffraction (QXRD) technique with internal standard was used to characterize the amount of unhydrated  $C_3A$  in different periods. The samples were scanned from 5° to 60° at a speed of 2.5 °/min with a 0.02° increment.

## 2.3.5 Scanning electron microscopy (SEM)

The morphology of the hydration products in the C<sub>3</sub>A-DI/SW paste was observed by a scanning electron microscope equipped with an energy dispersive X-ray detector (SEM, TESCAN VEGA3). This equipment was operated at ~0.001 Pa and in secondary electron mode at a 20 kV voltage, a 60  $\mu$ A beam current and a working distance of ~10 mm. After drying, the unground solid residues collected in Section 2.2 were stuck on carbon tape attached on SEM sample holders. These samples were gold coated before observation.

#### 2.3.6 GEMS modelling

By using the PSI-Nagra database [38] and the Cemdata 18 database [39-41], a thermodynamic modelling software (GEMS) was used to assess the variation of solution chemistry of seawater under different pH conditions and the evolution of equilibrium phases in the C<sub>3</sub>A-SW paste. For modelling the development of equilibrium phases in the C<sub>3</sub>A-SW system, firstly, the reaction degree of C<sub>3</sub>A in the C<sub>3</sub>A-SW paste at different reaction time was obtained using QXRD as mentioned in Section 2.3.4. Then, a four-parameter logistic non-linear

regression model (4PL) was used to fit data points of  $C_3A$  dissolution degree [42, 43]. After that, this fitting curve, i.e., the dissolution degree as a function of time, was used as kinetic input for GEMS [42].

# 3. Results and discussion

#### 3.1 Influence of seawater on C<sub>3</sub>A hydration

### 3.1.1 Hydration heat evolution of C<sub>3</sub>A

Fig. 3 shows the hydration heat evolution of  $C_3A$  mixed with DI and SW. For the  $C_3A$ -DI paste and the  $C_3A$ -SW paste, the main exothermic peak appeared at 3.8 min and 1.7 h, respectively, indicating that a delay of hydration of  $C_3A$  was caused by seawater. When comparing the two exothermic peaks, i.e.,  $Q_1$  and  $Q_2$ ,  $Q_2$  was lower and broadened with a relatively smaller rising slope, which demonstrated a relatively slower hydration reaction when  $C_3A$  was mixed with SW.

For the cumulative heat release in Fig. 3 (b), there was a lesser amount of heat release in the  $C_3A$ -SW paste than that in the  $C_3A$ -DI paste, i.e. the corresponding heat release was decreased by 24% at 72 h. In addition, the curve of the  $C_3A$ -SW paste reached a stable value at about 30 h, while the corresponding time for  $C_3A$ -DI paste was about 72 h. These features indicated a relatively lower reaction degree when  $C_3A$  was mixed with SW.



Fig. 3 Development of hydration heat of C<sub>3</sub>A mixed with DI and SW

#### 3.1.2 Hydration products of C<sub>3</sub>A

Fig. 4 shows the in-situ XRD patterns of the C<sub>3</sub>A pastes mixed with DI and SW. It can be noticed that the reaction of C<sub>3</sub>A with DI was extremely rapid. From Fig. 4 (a), plenty of C<sub>4</sub>AH<sub>13</sub> and C<sub>2</sub>AH<sub>8</sub> (hydroxy-AFm) were detected after just 9 minutes. With the increase in hydration time, the intensity of C<sub>4</sub>AH<sub>13</sub> and C<sub>2</sub>AH<sub>8</sub> peaks increased, indicating the continuous hydration reaction of C<sub>3</sub>A. In the first 12 h, the metastable C<sub>4</sub>AH<sub>13</sub> and C<sub>2</sub>AH<sub>8</sub> phases were the major hydration products, even though the stable C<sub>3</sub>AH<sub>6</sub> phase was also detected and its intensity slightly increased with hydration time.

In comparison, when SW was used as the mixing water shown in Fig. 4 (b), during the first 30 min, no crystalline hydrates were detected by XRD. This was due to the retardation effect of seawater on the C<sub>3</sub>A hydration as mentioned above. After 1.5 h, a peak at about  $11^{\circ}$  (2  $\theta$ ) was obviously observed, indicating the formation of Friedel's salt ( $Ca_4Al_2Cl_2(OH)_{12} \cdot 4H_2O$ ). At 4 h, the peak of C<sub>3</sub>AH<sub>6</sub> was also observed, and its amount increased with time. However, the metastable C<sub>4</sub>AH<sub>13</sub> and C<sub>2</sub>AH<sub>8</sub> were no longer detected, revealing that, compared to OH<sup>-</sup>, the Cl<sup>-</sup> in seawater could preferentially react with C<sub>3</sub>A to form Friedel's salt.

In order to compare the hydration degree of  $C_3A$  in DI and SW, the amount of the hydrated  $C_3A$  was calculated from QXRD data and the results are tabulated in Table 2. It was found that, at 30 min and 1 h, only 0.6 % and 12.7 % of  $C_3A$  were reacted when SW was used as the mixing water, while 52.0 % and 54.8 % of  $C_3A$  were consumed for the case of using DI. At 3 d, the reaction degree of  $C_3A$  in SW was still 8.2 % lower than that in DI. These results reflected that the use of seawater would delay the hydration reaction of  $C_3A$  and decrease its reaction degree.



Fig. 4 In-situ XRD patterns of C<sub>3</sub>A mixed with DI or SW for different times ((a): C<sub>3</sub>A-DI system, (b): C<sub>3</sub>A-SW system). A: C<sub>3</sub>A (tricalcium aluminate); #: C<sub>2</sub>AH<sub>8</sub> (calcium hydroaluminate); X: C<sub>4</sub>AH<sub>13</sub> (calcium hydroaluminate); ♣: C<sub>3</sub>AH<sub>6</sub> (hydrogarnet); F: Friedel's

salt;

Table 2 Hydration degree of C<sub>3</sub>A in C<sub>3</sub>A-DI/SW pastes as calculated by QXRD

Items	Hydration degree of C <sub>3</sub> A (%)		
Ages	C <sub>3</sub> A+DI	C <sub>3</sub> A+SW	
30 min	52.0	0.6	
1 h	54.8	12.7	
4 h	59.9	44.5	
8 h	73.5	61.0	
12 h	75.1	62.9	
3 d	94.5	86.3	

Fig. 5 presents the morphology evolution of hydration products in the C<sub>3</sub>A-DI/SW paste. For the C<sub>3</sub>A-DI paste in Fig. 5 (b) and 5 (c), after hydrating for 5 min and 2 h, the formed calcium hydroaluminates were flaky, and most of them were stacked up. In contrast, when SW was mixed with C<sub>3</sub>A for 30 min in Fig, 5 (d), the C<sub>3</sub>A particle still had a relatively smooth surface, indicating less reaction. At 2 h, the hydration products with a honeycomb shape precipitated on the C<sub>3</sub>A surface. These hydrates should be Friedel's salts, because their characteristic peaks were detected by XRD as shown in Fig. 4 (b). The arrangement of the Friedel's salt was disordered, unlike its assemblage reported in some literatures [44-46]. According to Talero [47], a relatively slow precipitation rate of the Friedel's salt would perhaps lead to a more-ordered hexagonal shape. Thus, the precipitation rate of the Friedel's salt in the C<sub>3</sub>A-SW paste could be quicker than that in other reported situations, such as OPC system. Furthermore, after 12 h of hydration, the paste prepared with SW had a similar morphology to that of the C<sub>3</sub>A-DI system in Fig. 5 (c), as a high amount of cubic C<sub>3</sub>AH<sub>6</sub> was formed.



Fig. 5 Morphology of hydrates in the C<sub>3</sub>A-DI/SW paste at different times ((a): pure C<sub>3</sub>A (b): C<sub>3</sub>A+DI-5 min, (c): C<sub>3</sub>A+DI-2h, (d): C<sub>3</sub>A+SW-30 min , (e): C<sub>3</sub>A+SW-2 h, (f): C<sub>3</sub>A+SW-12

## 3.1.3 Variation of ionic concentrations during C<sub>3</sub>A hydration

Fig. 6 displays the change of concentrations of different ions as a function of time in the  $C_3A$ -DI/SW paste. The ionic concentrations at time zero denoted the corresponding concentration in the pure simulated seawater and DI water, respectively.

h)



Fig. 6 Different ionic concentrations as a function of hydration time in the C<sub>3</sub>A-DI/SW paste

Fig. 6 (a) shows when  $C_3A$  was mixed with DI, little  $Al^{3+}$  concentration was detected in the solution, as almost all of the dissolved ions immediately precipitated to form calcium hydroaluminates due to the rapid hydration reaction depicted in Figs. 3 and 4 (a). In comparison, when  $C_3A$  was mixed with SW, within the first hour, a low concentration of  $Al^{3+}$  were observed due to the lower initial reactivity of  $C_3A$  during the induction period. But after 1.5 h, a high concentration of  $Al^{3+}$  was found in the solution, which corresponded to the initial formation stage of Friedel's salt in Fig. 4 (b). This implied that, unlike the formation of calcium hydroaluminates ( $C_4AH_{13}$  and  $C_2AH_8$ ) in the  $C_3A$ -DI paste, the formation of Friedel's salt in the  $C_3A$ -SW paste would result in an accumulation of  $Al^{3+}$  in the solution. The specific reason will be further discussed in Section 4.2. After 6 h, the concentration of  $Al^{3+}$  began to decrease,

and at the same time a noticeably increased XRD peak intensity of  $C_3AH_6$  appeared at 8 h as shown in Fig. 4 (b), implying an increased amount of  $C_3AH_6$  formed in this period.

Compared with  $Al^{3+}$ , a relatively high concentration of  $Ca^{2+}$  was released from  $C_3A$ , especially during the initial contact with DI or SW [15]. From Fig. 6 (b), the concentration of  $Ca^{2+}$  in the  $C_3A$ -SW paste at 2 min was 4 times higher than that of the  $C_3A$ -DI paste. This could be explained by an accelerated initial dissolution of  $C_3A$  as shown in Equation (3-1) due to the consumption of OH<sup>-</sup> by Mg<sup>2+</sup> to precipitate Mg(OH)<sub>2</sub>. Fig. 6 (b) shows a decrease of Mg<sup>2+</sup> concentration by 93% at 2 min. After 1 h, the decrease in the concentration of  $Ca^{2+}$  might be due to the formation of Friedel's salt and hydroaluminates caused by further hydration of  $C_3A$ .

$$Ca_3Al_2O_6 + 6H_2O \to 3Ca^{2+} + 2Al(OH)_4 + 4OH^-$$
 (3-1)

For the  $Mg^{2+}$ , from SEM images in Fig. 7, it can be observed that some magnesium compounds were attached on the surface of C<sub>3</sub>A after 5 min and 30 min of hydration, which could be due to the precipitation of  $Mg(OH)_2$ . The influence of magnesium species on the C<sub>3</sub>A hydration will be further discussed in Section 3.2.2 and 4.1.

Furthermore, in the case of  $C_3A$  mixed with SW, the concentration changes of  $Cl^-$ ,  $SO_4^{2-}$ ,  $Na^+$ ,  $K^+$  and  $Sr^{2+}$  introduced by seawater are shown in Fig. 6 (c) and 6 (d). For  $Cl^-$ , before 1 h, it decreased by only 2%. But after 1 h, the consumption of  $Cl^-$  was caused by the reaction between  $Cl^-$  and  $C_3A$  to form the Friedel's salt as revealed by XRD pattern in Fig. 4 (b). After 6 h, the concentration of  $Cl^-$  became stabilised, while the amount of  $C_3AH_6$  was significantly increased (Fig. 4 (b)), indicating the formation of  $C_3AH_6$  after the Friedel's salt.

Regarding the concentration of  $SO_4^{2^-}$ , during the first 1 h, its concentration deceased by 37%. Together with the modest decrease of  $Ca^{2+}$  during the first 1 h, this could be due to the adsorption of  $Ca^{2+}$  and  $SO_4^{2-}$  on the surface of  $C_3A$ , which resulted in the retardation of  $C_3A$  hydration [6, 10, 11, 14]. After 1 h, the concentration of  $SO_4^{2^-}$  significantly dropped to close to zero, but no related new products were detected from XRD. The reason might be that a small amount of  $SO_4^{2-}$  in the C<sub>3</sub>A-SW paste, i.e., only about 0.3 wt. % of the mass of C<sub>3</sub>A, was incorporated in the hydration products, like the Friedel's salt or C<sub>3</sub>AH<sub>6</sub> phase. The specific reason still needs to be further explored.

With respect to the evolution of other cations shown in Fig. 6 (d), the concentration of Na<sup>+</sup>,  $K^+$  and  $Sr^{2+}$  remained stable in the first 1 h of hydration, indicating that these cations in seawater were neither adsorbed on the C<sub>3</sub>A surface nor substituted Ca<sup>2+</sup> that had been adsorbed on the C<sub>3</sub>A surface to retard its hydration. After 1 h, the concentrations of Na<sup>+</sup> and K<sup>+</sup> slightly increased owing to the depletion of free water due to C<sub>3</sub>A hydration.



Fig. 7 SEM and EDS results of C<sub>3</sub>A mixed with SW for 5 min (a) and 30min (b)

# 3.2 Role of SO4<sup>2-</sup> and Mg<sup>2+</sup> in seawater on C<sub>3</sub>A hydration

## 3.2.1 Effect of SO4<sup>2-</sup>

Many literatures have reported the retardation effect of  $SO_4^{2-}$  on  $C_3A$  hydration when gypsum or other forms of sulphate alone was present [9, 13, 14, 16]. In order to further confirm the role of  $SO_4^{2-}$  in seawater on the C<sub>3</sub>A hydration, the sulphate salt was selectively excluded when preparing the seawater for this part of the study, and the corresponding change of heat evolution of the C<sub>3</sub>A paste (C<sub>3</sub>A-S1) is shown in Fig. 8. Compared with the original C<sub>3</sub>A-SW paste, when the SO<sub>4</sub><sup>2-</sup> was removed (C<sub>3</sub>A-S1), there was no induction period in the heat evolution curve. This indicated that SO<sub>4</sub><sup>2-</sup> in seawater had a determinative effect on the occurrence of the induction period. Comparing with the C<sub>3</sub>A-DI paste, after the removal of SO<sub>4</sub><sup>2-</sup> from the seawater, the occurrence of the major heat evolution peak was accelerated by about 3 min. This could be due to the slight acceleration effect derived from the reaction of C<sub>3</sub>A and Cl<sup>-</sup> in seawater. To further demonstrate this effect, a NaCl solution with the same concentration in the simulated seawater was mixed with C<sub>3</sub>A, the similar accelerating effect for the heat flow curve further validated the action of Cl<sup>-</sup> in Fig. 8.



Fig. 8 Effect of  $SO_4^{2-}$  in seawater on the hydration rate of C<sub>3</sub>A. The solution name of S1 in the specimen of C<sub>3</sub>A+S1 denotes the simulated seawater without Na<sub>2</sub>SO<sub>4</sub>.

## 3.2.2 Effect of Mg<sup>2+</sup>

In order to explore the effect of  $Mg^{2+}$  in seawater on the C<sub>3</sub>A hydration, a seawater solution without adding the magnesium salt (S2) was used to prepare the C<sub>3</sub>A paste, and its evolution of hydration heat is shown as plotted by the red dashed line in Fig. 9. In contrast, the corresponding change in the heat flow of C<sub>3</sub>A mixed with the original seawater (C<sub>3</sub>A-SW) is

shown as the black line in Fig. 9. It was found that when  $Mg^{2+}$  was not involved into the seawater (C<sub>3</sub>A-S2), the induction period was shortened by approximately 30 minutes. It can be deduced that the  $Mg^{2+}$  in seawater would contribute to the hydration retardation of C<sub>3</sub>A as well. In the Section 4.1, the possible reasons will be further discussed.

Furthermore, the slope of the hydration heat curve of the C<sub>3</sub>A-S2 paste in both the acceleration and deceleration periods were steeper than that in the C<sub>3</sub>A-SW paste. In addition, the peak intensity of the C<sub>3</sub>A-S2 paste was also higher than that of the C<sub>3</sub>A-SW paste. This could be inferred that, in the C<sub>3</sub>A-SW paste, the presence of Mg<sup>2+</sup> in seawater would decrease the dissolution rate of C<sub>3</sub>A and the formation rate of hydration products, which will be further discussed in Section 4.1.



Fig. 9 Effect of  $Mg^{2+}$  on the hydration rate of C<sub>3</sub>A. The solution name of S2 in the specimen of C<sub>3</sub>A+S2 denotes the simulated seawater without MgCl<sub>2</sub>. (Note: An external mixing method was adopted in this test. The initial peak could thus be affected due to the disturbance of moving samples into the calorimeter. This peak was not analysed in the work.)

## 4. Discussion

#### 4.1 Effect of seawater on the C<sub>3</sub>A hydration process

As illustrated in Fig. 3, seawater had a retardation effect on the C<sub>3</sub>A hydration. After excluding  $SO_4^{2^-}$  in seawater (Fig. 8), this retardation phenomenon disappeared. This demonstrated that the critical and determinative role of  $SO_4^{2^-}$  in seawater on the retarded hydration of C<sub>3</sub>A, even though the amount of  $SO_4^{2^-}$  in this system only accounted for 0.3 wt. % of the mass of C<sub>3</sub>A.

In order to further scrutinize the chemical form of  $SO_4^{2-}$  in seawater, GEMS (Gibbs Energy Minimization Software) was used to simulate the concentration variation of sulphate species in seawater under different pH conditions, and the results are illustrated in Fig. 10 (a). It was found that, when  $C_3A$  contacts with seawater, the leaching of  $Ca^{2+}$  and  $OH^{-}$  would increase the pH value in this system. In this case, the amount of ion pairing of CaSO<sub>4</sub><sup>0</sup> in this system was increased to about 4.7 mmol/L. Thus, the complexation of the ion pairing of CaSO<sub>4</sub><sup>0</sup> in the solution onto the Al-rich leached layer of C<sub>3</sub>A could poison the reactive sites and retard its hydration reaction. This theory has been experimentally supported by other researchers who explored the hydration behaviour of C<sub>3</sub>A with sulphates [12, 15]. In addition, another possible reason could be that the  $Ca^{2+}$  is first chemisorbed onto the Al-rich leached layer of  $C_3A$ , forming a positive charge, and then the negatively charged  $SO_4^{2-}$  would be adsorbed on the surface [9]. Therefore, the adsorption of  $CaSO_4^0$  or the successive adsorption of  $Ca^{2+}$  and  $SO_4^{2-}$ on the Al-rich layer of  $C_3A$  would explain the retarding effect of seawater on the  $C_3A$  hydration. Moreover, no AFt was detected in this low concentration of SO4<sup>2-</sup> in Fig. 4. In this case, the retardation phenomenon could mainly derive from the adsorption of the above ions or ion pairing, rather than the adsorption of AFt onto the active sites of C<sub>3</sub>A [13]. This further confirms the important role of the adsorption of  $Ca^{2+}$  and  $SO_4^{2-}$  or  $CaSO_4^{0}$  on  $C_3A$  surface in retarding the C<sub>3</sub>A hydration.

However, it has been acknowledged that seawater contains more than one kind of cation. In this case, except for the role of  $Ca^{2+}$  and  $SO_4^{2-}$ , the interaction of  $Mg^{2+}$  and  $SO_4^{2-}$  is another concern. The thermodynamic tendency of Mg<sup>2+</sup> species in seawater with the increase of pH value is also presented in Fig. 10. The initial pH value in the Figs. 10 (a) and 10 (b) was that of the simulated seawater at 8.2. From these two figures, it can be found that, pH value would increase due to the dissolution of  $C_3A$ , consequently the amount of ion pairing of MgSO<sub>4</sub><sup>0</sup> and Mg<sup>2+</sup> reach zero due to the precipitation of Mg(OH)<sub>2</sub>. Besides, the evolution of magnesium species in the pore solution of C<sub>3</sub>A-SW paste in the first 5 min of hydration was further simulated using GEMS and shown in Fig. 10 (c). The decrease of  $Mg^{2+}$  and  $MgSO_4^0$  and the increase of  $Mg(OH)_2$  (s) were obvious, and at about 1.5 min, the concentration of  $MgSO_4^0$  was close to zero. The experimental data shown in Fig. 6 (b) in the Section of 3.1.3 also showed a reduction of the concentration of Mg<sup>2+</sup> by 93% at only 2 min of hydration, which was in agreement with the simulation result in Fig. 10 (c). Therefore, the adsorption of  $MgSO_4^0$  or the successive adsorption of  $Mg^{2+}$  and  $SO_4^{2-}$  on the Al-rich layer of C<sub>3</sub>A that has been reported in the C<sub>3</sub>A mixed with single magnesium salt in the solution (e.g. MaSO<sub>4</sub>) [14, 15] had little effect on the retarded hydration of C<sub>3</sub>A under this alkaline environment.





Fig. 10 Variation of sulphate species (a) and magnesium species (b) in seawater under different pH conditions (Ca(OH)<sub>2</sub> as the pH regulator). (c): the change of main magnesium species in C<sub>3</sub>A-SW system with the increase of hydration time

Furthermore, it was also found that when magnesium salt was removed from the seawater, the induction period was shortened for about 30 minutes (Fig. 9). This also demonstrated that the presence of the magnesium salt also played a role on the retardation effect. But in Fig. 10 (c), it was found that the amounts of  $MgSO_4^0$  and  $Mg^{2+}$  were decreased to nearly zero at 1.5 min and 2.5 min of hydration, respectively. Therefore, the effects of  $MgSO_4^0$  and  $Mg^{2+}$  would become insignificant. Considering that  $Mg(OH)_2(s)$  was the major species after 3 min of hydration as shown in Fig. 10 (c), the precipitation of  $Mg(OH)_2$  on the C<sub>3</sub>A surface could also block parts of the active sites of C<sub>3</sub>A surface to extend the induction period, but its significance was less than the adsorption effect of Ca<sup>2+</sup> and SO4<sup>2-</sup>. Besides, the attachment of  $Mg(OH)_2$  on the C<sub>3</sub>A surface would to some extent decrease the dissolution rate of C<sub>3</sub>A in both the acceleration and deceleration period as shown in Fig. 9.

Compared with  $Ca^{2+}$  and  $Mg^{2+}$  ions, the concentrations of  $Na^+$  and  $K^+$  ions introduced by the seawater remained relatively stable in the solution during the first 1 h of hydration. Therefore, the possibility that these cations combined with  $SO_4^{2-}$  to retard  $C_3A$  hydration would be small. Consequently, the effect of seawater on the retardation of C<sub>3</sub>A hydration may be summarised by a schematic diagram in Fig. 11. The initial dissolution of C<sub>3</sub>A forms an Al-rich layer with negative charge on the surface of C<sub>3</sub>A [12]. This would allow 1) Ca<sup>2+</sup> to preferentially adsorb on the surface of C<sub>3</sub>A, and afterwards, SO4<sup>2-</sup> in seawater would migrate to the surface, or 2) the ion pairs of CaSO4<sup>0</sup> present in seawater are directly adsorbed on the C<sub>3</sub>A surface. These possible adsorption behaviours could poison the reactive sites of the C<sub>3</sub>A surface, which would hinder the further dissolution of C<sub>3</sub>A. Besides, the precipitation of Mg(OH)<sub>2</sub> from seawater onto the surface of C<sub>3</sub>A also would to some extent decrease the dissolution of C<sub>3</sub>A and prolong the induction period. In contrast, the adsorption of Mg<sup>2+</sup> and SO4<sup>2-</sup> or the ion pairs of MgSO4<sup>0</sup> on the C<sub>3</sub>A surface had less effect on retarding the hydration of C<sub>3</sub>A due to the transformation of Mg<sup>2+</sup> and MgSO4<sup>0</sup> to Mg(OH)<sub>2</sub>(s) under the alkaline environment.



Fig. 11 Schematic diagram of the C<sub>3</sub>A hydration and retardation in DI and seawater

After induction period, the concentration of  $SO_4^{2-}$  rapidly decreased to close to zero (Fig. 6 (c)), but no related new product was detected as shown in the XRD data (Fig. 4 (b)). This  $SO_4^{2-}$  might be adsorbed on the formed hydration products, but this still needs to be further explored.

#### 4.2 Effect of seawater on the hydration products of C<sub>3</sub>A

 $C_3A$  reacts rapidly with freshwater to form metastable hydroxy-AFm phases (Figs. 3 and 4 (a)), i.e.,  $C_4AH_{13}$  and  $C_2AH_8$ . Finally, these products can be converted to a more stable hydrogarnet phase ( $C_3AH_6$ ). However, when seawater is mixed with  $C_3A$ , after the induction period, the Friedel's salt would be formed first. After that, the amount of  $C_3AH_6$  began to increase (Fig. 4 (b)). The simulated result using the GEMS software in Fig. 12 also confirmed the preferential formation of the Friedel's salt. It was also found that, during this process, no metastable hydroxy-AFm was detected according to the in-situ XRD data (Fig. 4 (b)).

Furthermore, as shown in Eq. (3-1), the molar ratio of  $Ca^{2+}$  to  $Al(OH)_4^-$  produced by the dissolution of  $C_3A$  is 3:2, meaning that the dissolution of 2 mol of  $C_3A$  would produce 1 mol of  $C_4AH_{13}$  and 1 mol of  $C_2AH_8$  referring as indicated by Eqs. (4-1) and (4-2). Therefore, no significant amount of  $Al^{3+}$  was detected in the solution due to the rapid precipitation of  $C_4AH_{13}$  and  $C_2AH_8$  in the  $C_3A$ -DI paste (Fig. 6 (a)). In contrast, when there is a direct reaction between  $C_3A$  and  $Cl^-$  to form Friedel's salt in the  $C_3A$ -SW paste, according to the molar ratio of  $Ca^{2+}$  to  $Al(OH)_{4^-}$  in Equations (3-1) and (4-3), the amount of  $Ca^{2+}$  is insufficient. This would result in an accumulation of  $Al^{3+}$  and decrease of  $Ca^{2+}$  as shown in Figs. 6 (a) and 6 (b). This further illustrates that, when  $C_3A$  was mixed with seawater,  $Cl^-$  would directly adsorb into the interlayers of the principal layers ( $[Ca_2Al(OH)_6]^+$ ) of AFm structure to form Friedel's salt, rather than forming  $C_4AH_{13}$  in which  $OH^-$  is replaced by  $Cl^-$  through an ion-exchange mechanism [48], because this would not lead to a large amount of  $Al^{3+}$  in solution. Besides,

this accumulation of  $Al^{3+}$  in solution could also hinder the subsequent dissolution of  $C_3A$ , leading to a slower hydration rate of  $C_3A$  at the later stage of  $C_3A$ -SW paste.

$$4 Ca^{2+} + 2 Al(OH)_{4}^{-} + 6 OH^{-} + 6H_2O \rightarrow Ca_4Al_2(OH)_{14} \cdot 6H_2O (C_4AH_{13})$$
(4-1)

$$2 Ca^{2+} + 2 Al(OH)_{4}^{-} + 2 OH^{-} + 3H_{2}O \rightarrow Ca_{2}Al_{2}(OH)_{10} \cdot 3H_{2}O (C_{2}AH_{8})$$
(4-2)

$$4 \operatorname{Ca}^{2+} + 2 \operatorname{Al}(OH)_{4}^{-} + 2\operatorname{Cl}^{-} + 4 \operatorname{OH}^{-} + 4H_2O \longrightarrow \operatorname{Ca}_4\operatorname{Al}_2\operatorname{Cl}_2(OH)_{12} \cdot 4H_2O \text{ (Friedel's salt)}$$
(4-3)

Therefore, even though the hydroxy-AFm and Friedel's salt are in the similar calcium aluminium layered double hydroxide (Ca-Al-LDHs) family [2] which is composed of positively charged hydroxide layers ( $[Ca_2Al(OH)_6]^+$ ) and anions, such as OH<sup>-</sup>, SO<sub>4</sub><sup>2-</sup>, CO<sub>3</sub><sup>2-</sup>, Cl<sup>-</sup>, intercalating the interlayer spaces [2, 49-51], when a substantial amount of Cl<sup>-</sup> is present in this reaction system, as compared to OH<sup>-</sup>, Cl<sup>-</sup> would preferentially intercalate the interlayers of [Ca<sub>2</sub>Al(OH)<sub>6</sub>]<sup>+</sup> to form the Friedel's salt. After most of Cl<sup>-</sup> is consumed (Fig. 6 (c)), C<sub>3</sub>AH<sub>6</sub> starts to form (Fig. 4 (b)), along with a decrease in the concentration of Al<sup>3+</sup>.



Fig. 12 Modelled evolution of hydration phases at equilibrium in the C<sub>3</sub>A-SW paste (Note: The Brucite and Al(OH)<sub>3</sub> were not found in XRD data in Fig. 4, which could be due to their relatively small amounts .)

# **5.** Conclusions

This study aims to elucidate the effect of seawater on the cubic  $C_3A$  hydration. Isothermal Calorimetry, XRD, SEM, Ion Chromatography and ICP-OES were used to investigate the evolution of hydration heat and hydration products, as compared to that of the  $C_3A$ -DI paste. The development of equilibrated hydration phases in the  $C_3A$ -SW paste and the variation of ionic species were simulated by GEMS. The following findings can be summarized:

- Using seawater as mixing water retarded the hydration of the cubic  $C_3A$ . When  $SO_4^{2-}$  was excluded from the seawater, the induction period disappeared. The co-existence or ion pairing of  $Ca^{2+}$  and  $SO_4^{2-}$  in seawater adsorbed on the surface of  $C_3A$  would be the main reason for this retardation.
- Under an alkaline environment, the precipitation of Mg(OH)<sub>2</sub> would block parts of the active sites on the surface of C<sub>3</sub>A to decrease the dissolution of C<sub>3</sub>A and prolong the induction period.
- In the C<sub>3</sub>A-SW paste, the Friedel's salt was first detected, followed by C<sub>3</sub>AH<sub>6</sub>. This was different from the hydration products formed in the C<sub>3</sub>A-DI paste, in which the metastable hydroxy-AFm, i.e., C<sub>4</sub>AH<sub>13</sub> and C<sub>2</sub>AH<sub>8</sub>, were found at the beginning and then they were gradually converted into C<sub>3</sub>AH<sub>6</sub>. Additionally, the direct formation of Friedel's salt in C<sub>3</sub>A-SW paste resulted in an accumulation of Al<sup>3+</sup> in solution, which would also hinder the subsequent dissolution of C<sub>3</sub>A.

As mentioned in the introduction, the crystal structures of  $C_3A$  in cement clinker is also an important factor influencing its hydration process. This study mainly focused on the hydration behaviour of the cubic  $C_3A$  mixed with seawater, considering that the cubic  $C_3A$  is predominant in most Portland cements. However, as the orthorhombic  $C_3A$  is also present in some cements, the impact of seawater on its hydration behaviour is worthwhile to be explored. Based on the results of the cubic  $C_3A$  mixed with seawater obtained in this work, a possible effect of the orthorhombic  $C_3A$ -seawater paste can be inferred: seawater could retard its hydration, but the degree of retardation could be less than that on the cubic  $C_3A$ -seawater paste due to the faster hydration behaviour of the orthorhombic  $C_3A$  in presence of sulphate [27, 28]. But the above needs to be further verified.

# Acknowledgments

This study work was supported by a grant from the Research Grants Council of the Hong

Kong SAR Government (Project No. T22-502/18-R) and The Hong Kong Polytechnic

University.

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