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# Tailoring small-molecule acceptors through asymmetric side-chain substitution for efficient organic solar cells

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ABSTRACT Side chain engineering of small-molecule acceptors (SMAs) is a promising strategy for improving device efficiency in organic solar cells (OSCs). This study investigates the parent SMAs of BT-BO and BT-TBO, along with the newly synthesized asymmetric SMA, BT-ASY, which features branched alkyl chains and thiophene side chains substituted at the β positions of the thiophene units, respectively. Despite exhibiting comparable optical and electrochemical properties, the PM6:BT-ASY-based device achieves a power conversion efficiency (PCE) of 18.08% representing a significant improvement over its symmetric counterparts. This enhancement is primarily attributed to improved charge mobility, extended carrier lifetimes, optimized molecular packing, and effective phase separation, as confirmed by grazing incidence wide-angle X-ray scattering measurements. Our findings highlight that asymmetric side-chain strategy enhances  $\pi$ - $\pi$ stacking and electronic coupling, offering a simple yet effective approach to improving photovoltaic performance. This work underscores the potential of asymmetric structural modifications in SMAs for advancing OSC technology and renewable energy solutions.

**Keywords:** organic solar cells, side chain engineering, active layer, asymmetric acceptors, thiophene side chains

#### INTRODUCTION

Organic solar cells (OSCs) are widely researched due to their potential of low-cost production, flexibility, lightweight properties, and the ability to be tailored for specific applications, which makes them promising candidates for sustainable energy solutions [1–3]. In OSCs, the photoactive layer is typically composed of p-type donor polymers (e.g., PM6) and n-type small-molecule acceptors (SMAs), determining the light absorption and power conversion efficiencies (PCEs) of devices [4–8]. Recent breakthroughs have propelled OSCs with PCEs surpassing 20% [9–11], primarily due to the development of

SMAs featuring crescent-shaped architecture like Y6 [12]. Although numerous derivatives based on the Y6 structure have been developed, achieving remarkable progress, a clear and effective understanding of the structure-property relationships remains lacking, which poses significant challenges for the optimization of OSC devices and further enhancement of performance [12–22].

Numerous studies have focused on optimizing the branching points, dimensions, length, and shape of alkyl chains in SMAs [23-31]. Side-chain engineering, including variations in branching points, size, length, and shape, is crucial alongside tuning the end groups and central core [32,33]. Substituents with aromatic groups (e.g., phenyl or thiophenyl) offer superior packing and charge transport properties compared to conventional saturated alkyl chains, leading to record-breaking efficiencies [34,35]. Additionally, introducing asymmetric side chains combines the advantages of different configurations, effectively fine-tuning the packing morphology and optoelectronic properties of materials, ultimately enhancing photovoltaic performance [35-37]. In the realm of organic photovoltaics, thiophene is frequently utilized in the development of highperformance polymer donors like PM6 and D18, but few works demonstrate their impact as side chains in SMAs [38]. As for the building block with side chain substitution in Y-series SMAs, the β-positions of thiophene units are crucial for optimizing molecular stacking and electronic interactions. Previous studies have shown that even minor modifications to thiophene can significantly impact the morphological and optoelectronic characteristics of these materials, leading to notable variations in performance [39-42]. Currently, there is a lack of systematic studies examining the impact of β-position modifications of thiophene side chains with asymmetric strategy on the performance of the SMAs [25,36,43-45]. Thus, investigating the intramolecular and intermolecular dynamics of SMAs with asymmetric side chains of alkyl and thiophene chains could provide valuable correlations between fundamental properties and overall device performance [46-49].

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In this study, we systematically explore the impact of asymmetric versus symmetric side chains on the physicochemical properties, and OSCs performance of SMAs. Based on the highefficiency SMA, BT-BO (with symmetric alky chains), and its derivative BT-TBO (with symmetric thiophene chain), we developed a new asymmetric SMA, BT-ASY, with asymmetric alkyl chain and thiophene chain substitution, respectively, shown in Fig. 1a. The three SMAs display comparable optical properties, with molecular energy levels increasing progressively from BT-BO to BT-ASY and finally to BT-TBO. BT-ASY shows improved molecular stacking along the conjugated backbones, as confirmed by grazing incidence wide-angle X-ray scattering (GIWAXS) measurements, resulting in enhanced electron mobility in both neat acceptor films and donor-acceptor (D-A) blend films. The higher PCE observed in the PM6:BT-ASY device (PCE = 18.08%, short-circuit current density ( $J_{SC}$ ) =  $27.15 \text{ mA cm}^{-2}$ ,  $V_{OC} = 0.849 \text{ V}$ , and fill factor (FF) = 78.44%), which can be attributed to BT-ASY's superior charge transport capabilities, enhanced molecular interactions, and suitable phase separation. Overall, our results indicate that the asymmetric side-chain strategy is an effective approach for fine-tuning photoelectric properties and improving the photovoltaic performance of OSCs.

#### **EXPERIMENTAL SECTION**

The synthesis of BT-ASY is outlined in Scheme S1 (available in the Supporting Information), with detailed procedures provided in the same section. The key intermediate, compound 2, was synthesized through a Pd(PPh<sub>3</sub>)<sub>4</sub>-catalyzed Stille coupling reac-

tion using two different organotin compounds, achieving a yield of 40%. Following this, an intramolecular Cadogan reductive cyclization with PPh<sub>3</sub> and *N*-alkylation with 2-ethylhexyl were performed to create a central core with asymmetric alkyl side chains (compound 3). Compound 3 then underwent a Vilsmeier-Haack reaction, followed by a Knoevenagel condensation, resulting in the target molecule BT-ASY. It is noteworthy that the synthesis of BT-ASY involves the same number of steps as the symmetric compounds BT-BO and BT-TBO. we utilized two different organotin reagents to achieve an asymmetric coupling product (compound 2) via a one-pot method. The chemical structures of the intermediates and final products were confirmed using <sup>1</sup>H NMR, <sup>13</sup>C NMR, and MALDI-TOF techniques.

#### **RESULTS AND DISCUSSION**

#### Optical and electrochemical properties

The absorption spectra of BT-BO, BT-TBO, and BT-ASY in chloroform (CF) solutions and the solid state are presented in Fig. 1c, d, with corresponding optical parameters summarized in Table 1. In solution, the three SMAs exhibit nearly identical spectra, with maximum absorption wavelengths ( $\lambda_{max}$ ) at 730 nm for BT-BO, 731 nm for BT-ASY, and 732 nm for BT-TBO. This indicates that altering the side chains of Y6-like SMAs has minimal impact on the intramolecular charge transfer (ICT) effect [50,51]. When transitioning from solution to solid films, BT-BO and BT-ASY show similar redshifts of 69 and 72 nm, respectively, compared to a more significant redshift of 86 nm for BT-TBO. These differences are attributed to the unique

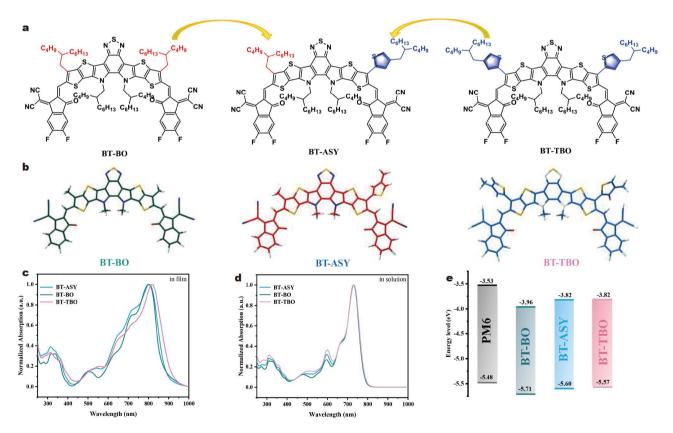


Figure 1 (a) Molecular structures of BT-BO, BT-TBO, and BT-ASY. (b) The monomolecular single crystallographic structures in the top-view of BT-BO, BT-TBO, and BT-ASY. Absorption spectra of BT-BO, BT-TBO, and BT-ASY (c) in the thin-film state, and (d) in dilute chloroform solutions. (e) Energy level diagrams of PM6, BT-BO, BT-TBO, and BT-ASY in thin films.

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Table 1 Optical and electrochemical parameters of BT-BO, BT-TBO, and BT-ASY

Acceptor	$\lambda_{\text{max}}^{\text{sol}}$ (nm)	$\lambda_{max}^{ film} (nm)$	$\lambda_{\text{onset}}^{\text{film}}$ (nm)	$E_{\rm g}^{\rm opt} ({\rm eV})^{\rm a}$	LUMO <sup>CV</sup> (eV)	HOMO <sup>CV</sup> (eV)	LUMO <sup>DFT</sup> (eV)	HOMO <sup>DFT</sup> (eV)
BT-BO	730	800	873	1.42	-3.96	-5.71	-3.50	-5.61
BT-ASY	731	803	898	1.38	-3.81	-5.60	-3.49	-5.53
BT-TBO	732	817	922	1.34	-3.82	-5.57	-3.49	-5.56

a) Calculated from  $E_{\rm g}^{\rm opt} = 1240/\lambda_{\rm onset}$ .

aggregation behaviors of the three SMAs in the solid state. The optical bandgaps for BT-BO, BT-TBO, and BT-ASY were estimated to be 1.42, 1.34, and 1.38 eV, respectively, based on the absorption edge onset, determined by the intersection of the tangent to the peak absorption edge with the vertical axis (Y = 0).

The molecular energy levels of these acceptors were assessed using cyclic voltammetry (CV) (Fig. 1e). The lowest unoccupied molecular orbital (LUMO) and highest occupied molecular orbital (HOMO) values are -3.96/-5.71 eV for BT-BO, -3.81/-5.60 eV for BT-ASY, and -3.82/-5.57 eV for BT-TBO. The slightly elevated LUMO values for BT-ASY, and BT-TBO compared with BT-BO, can be associated with the weak electron-donating properties of thiophene, which contributes to achieving a higher  $V_{\rm OC}$  in the resulting devices. Density functional theory (DFT) calculations were performed at the B3LYP/ 6-31G(d,p) level to understand the molecular structure properties of these NFAs. For simplicity of the calculations, all the alkyl chains of the four NFAs were substituted with methyl groups. From the optimized geometries (Fig. 1b) and Table 1, the LUMO and HOMO values are -3.50/-5.61 eV for BT-BO, -3.49/-5.53 eV for BT-ASY, and -3.49/-5.62 eV for BT-TBO. The results from DFT calculations are consistent with those from CV. As we progress from a side chain without thiophene to an asymmetric structure with one thiophene, and then to a small molecule acceptor (SMA) with two thiophene side chains, the HOMO levels continuously decrease while the LUMO levels increase slightly. This indicates that the introduction of thiophene side chains can effectively regulate molecular energy levels.

#### Performance characterization

To investigate the impact of asymmetric side chains on cell efficiency, we prepared conventional binary OSCs with the conventional architecture ITO glass/PEDOT: PSS/PM6/PNDIT-F3N/Ag. The polymer PM6, a commonly used commercial donor was selected as the donor in the active layer due to its deep HOMO energy level and complementary absorption spectra with the three acceptors. The best J-V curves and corresponding external quantum efficiency (EQE) spectra are presented in Fig. 2a, b. As shown in Table 2, increasing the number of thiophene-based side chains in the SMAs leads to a gradual decrease in  $V_{\rm OC}$ , with values of 0.865 V for BT-BO and 0.849 V for BT-ASY, and 0.828 V for BT-TBO. This trend aligns with the differing LUMO values. Notably, the asymmetric BT-ASY-based device achieves an impressive PCE of 18.08%, accompanied by an outstanding FF of 78.44% and a respectable short-circuit current density ( $J_{SC}$ ) of 27.15 mA cm<sup>-2</sup>, which are higher than those of the symmetric acceptor-based devices (PM6:BT-BO, PCE = 15.27%,  $J_{SC}$  = 24.04 mA cm<sup>-2</sup>, FF = 73.46%; PM6:BT-TBO, PCE = 17.21%,  $J_{SC}$  = 26.90 mA cm<sup>-2</sup>, FF = 77.27%). The enhanced J<sub>SC</sub> and FF of the BT-ASY-based device are attributed to the improved molecular stacking and superior charge transport capabilities in the active layer. Fig. 2b shows that both optimal cells exhibit strong, broad photovoltaic responses in the wavelength range of 500 to 900 nm. The superior EQE response in the PM6: BT-ASY device is likely due to its more favorable morphology compared to the PM6: BT-BO, and PM6: BT-TBO blends. The integrated  $J_{\rm SC}$  value of 26.51 mA cm $^{-2}$  from the EQE spectra in the PM6:BT-ASY device further confirms the reliability of the device data.

Next, we plotted the curves of  $J_{\rm SC}$  or  $V_{\rm OC}$  against different light intensities ( $P_{\rm light}$ ) to analyze charge recombination behaviors. As shown in Fig. 2d, e, and the parameters shown in Table S1, all devices exhibited a similar slope (A) of ~1 in the equation  $J_{\rm SC} \propto P^{\alpha}$  (0.95 for BT-BO, 0.96 for BT-TBO, and 0.97 for PM: BT-ASY), which is closest to 1, indicating that the bimolecular recombination is minimal. When examining the relationship between  $V_{\rm OC}$  and  $P_{\rm light}$ , the slopes for the BT-BO, BT-ASY, and BT-TBO-based cells were 1.38, 1.31 and 1.32 kT/q, respectively. This indicates that the PM6: BT-ASY system experiences more significant trap-assisted recombination.

To investigate the charge dissociation properties of the three device groups, we examined the relationship between effective voltage ( $V_{\rm eff}$ ) and saturated photocurrent density ( $J_{\rm ph}$ ) (Fig. 2f). Under short-circuit conditions, the exciton dissociation probabilities (P(E, T)) were found to be 98.8% for the PM6: BT-BO device, 99.6% for the PM6: BT-TBO device, and 100% for the PM6: BT-ASY device. This indicates that the BT-ASY-based device has superior exciton dissociation capability, contributing to its higher FF.

#### Charge transport and film morphology

To investigate the effect of asymmetric substitution on charge mobility, we conducted space-charge limited current (SCLC) measurements to assess the electron mobility  $(\mu_e)$  of neat acceptors and the hole mobility  $(\mu_h)$  and  $\mu_e$  of blends. As shown in Fig. 3, the measured  $\mu_e$  values were  $2.3 \times 10^{-4}~\text{cm}^2~\text{V}^{-1}~\text{s}^{-1}$  for BT-BO,  $6.4 \times 10^{-4}$  cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> for BT-ASY, and  $5.6 \times$  $10^{-4}$  cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> for BT-TBO. The highest  $\mu_e$  value for BT-ASY aligns with its highly interconnected improved molecular packing, as indicated by GIWAXS results (in the following section). When blended with PM6, the mobility trends in the three blend films mirrored those of the neat acceptors. Notably, the PM6 blend achieved a higher  $\mu_h$  value of and an  $\mu_h$  value of  $1.8 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  for BT-BO,  $5.0 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  for BT-ASY, and  $4.5 \times 10^{-4}$  cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> for BT-TBO (from Table S2). Furthermore, the balance between  $\mu_e$  and  $\mu_h$  of each blend was calculated to correlate with the FF of the corresponding OSC. The PM6: BT-ASY blend exhibits a significantly better  $\mu_e/\mu_h$ (1.10) than the PM6: BT-BO blend (1.36), and PM6: BT-TBO blend (1.15). The best mobility balance of the PM6: BT-ASY blend minimizes the accumulation of either hole or electron within the active layer, thus contributing to the high FF of the

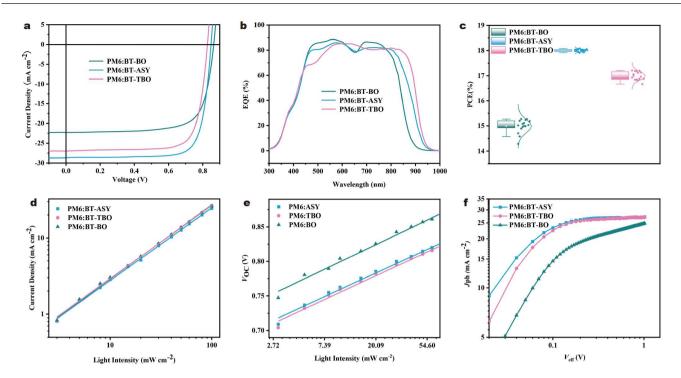


Figure 2 (a) J-V curves of OSCs AM 1.5 G illumination of 100 mW cm $^{-2}$ . (b) EQE spectra of the corresponding devices. (c) Normal distribution of PCEs. (d) J<sub>SC</sub> and (e) V<sub>OC</sub> variation versus light intensity. (f) Photocurrent density (J<sub>ph</sub>) as a function of effective bias (V<sub>eff</sub>) for the devices.

Table 2 Photovoltaic parameters of optimized binary OSCs under 100 mW cm<sup>-2</sup>

PM6: NFA	$V_{\rm OC}$ (V)	$J_{\rm SC}~({\rm mA~cm^{-2}})$	$J_{\rm cal}~({\rm mA~cm}^{-2})$	FF (%)	PCE (%) <sup>a</sup>
BT-BO	0.865	24.04	23.00	73.46	$15.27 \ (14.96 \pm 0.24)$
BT-ASY	0.849	27.15	26.51	78.44	$18.08 \ (18.01 \pm 0.05)$
BT-TBO	0.828	26.90	26.12	77.27	17.21 (17.02 ± 0.14)

a) The brackets contain average and standard errors of PCEs based on 10 devices.

#### corresponding OSCs.

The surfaces of the blend films were analyzed using atomic force microscopy (AFM), as shown in Fig. 3d. All PM6:SMA binary blend films exhibited smooth surfaces, with root-mean-square roughness values of 1.38 nm for PM6:BT-BO, 1.46 nm for PM6:BT-ASY, and 1.55 nm for PM6:BT-TBO. Notably, the PM6:BT-ASY blend displayed the most pronounced fibril structures, indicating a favorable morphology for charge transport.

To investigate the relationship between charge mobility and molecular crystallinity, both pure and blend films were examined using grazing incidence wide-angle X-ray scattering (GIWAXS) [52]. The 2D GIWAXS patterns for the pure SMAs and blend films are presented in Fig. 4a, b, with corresponding line-cut profiles showing in-plane (IP, black lines) and out-of-plane (OOP, red lines) characteristics in Fig. 4c. The 2D patterns reveal that all pure and blended films based on these three SMAs predominantly exhibit a face-on orientation. For the three pure SMA films, the (010) stacking peaks were observed in the OOP direction at  $q_z$  values of 1.80, 1.71, and 1.69 Å<sup>-1</sup>, corresponding to  $\pi$ - $\pi$  stacking distances of 3.50, 3.67, and 3.72 Å for BT-BO, BT-ASY, and BT-TBO, respectively. As summarized in Table S3, the results indicate that the varying numbers of thiophene side chains in the SMAs do not significantly affect the  $\pi$ - $\pi$  stacking

distance in pure films. However, Fig. 4a, b clearly show that BT-ASY exhibits the highest (010) diffraction intensity in both pure and blend films, followed by BT-BO and BT-TBO, indicating superior material crystallinity for BT-ASY, particularly when blended with PM6. Therefore, the trend in material crystallinity among the three SMAs aligns well with their charge mobility. Notably, the PM6:BT-ASY blend displayed an increased d-spacing of 22.05 Å for in-plane and a crystalline coherent length (CCL) of 38.7 Å for  $\pi$ - $\pi$  stacking, compared to the other two blends. This suggests that the BT-ASY blend retains a chain-extended structure and possesses a larger crystalline domain [53].

#### **CONCLUSIONS**

In summary, we have successfully developed a new asymmetric SMA, BT-ASY, derived from BT-BO and BT-TBO. Morphological analyses reveal that the asymmetric BT-ASY exhibits enhanced  $\pi$ - $\pi$  interactions between its terminal accepting units, improved molecular stacking along the conjugated backbones, and greater crystallinity than the symmetric BT-BO and BT-TBO. Among the three PM6-based D-A blends, the PM6: BT-ASY blend demonstrates superior and more balanced charge mobility and enhanced molecular packing. Consequently, the PM6 device achieves a PCE of 18.08%, with a high FF of 78.44%

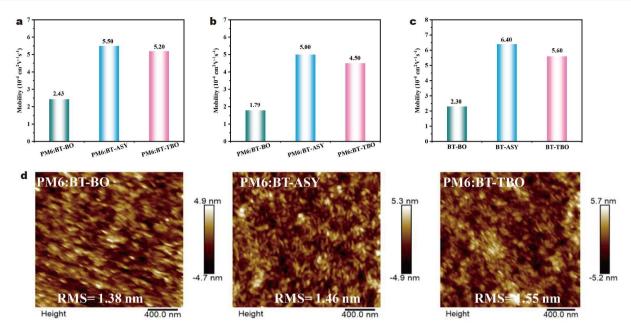


Figure 3 Hole-only and electron-only device results. (a) Electron-only mobility ITO/ZnO/active layer/ PNDI-F3N/Ag. (b) Hole-only /PEDOT:PSS/active layer/MoO<sub>3</sub>/Ag. (c) Electron-only mobility ITO/ZnO/acceptor/Ag. (d) AFM height images of PM6: BT-BO, PM6: BT-ASY, and PM6: BT-TBO.

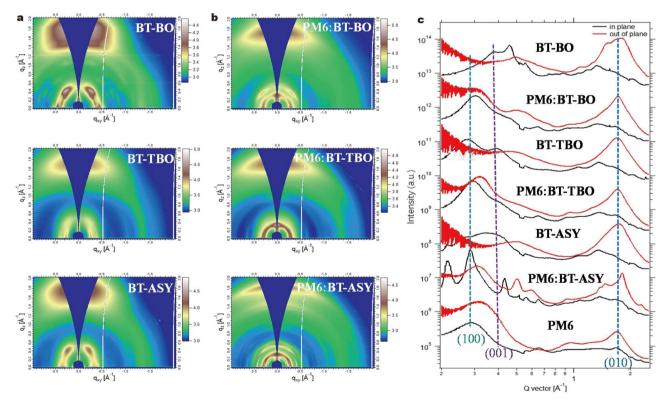


Figure 4 (a) 2D-GIWAXS patterns for pure films of BT-BO, BT-TBO, and BT-ASY. (b) 2D-GIWAXS patterns for blend films of PM6:BT-BO, PM6:BT-TBO, PM6:BT-ASY. (c) 1D line-cuts of the BT-BO, PM6:BT-BO, BT-TBO, PM6:BT-TBO, BT-ASY, PM6:BT-ASY and PM6.

and a respectable  $J_{SC}$  of 27.15 mA cm<sup>-2</sup>. Overall, our systematic study indicates that asymmetric side-chain substitution is a simple yet effective strategy for optimizing the molecular structure of SMAs. This approach improves  $\pi$ - $\pi$  stacking and crystallinity, increases electronic coupling, and ultimately

enhances device performance. We believe this work not only offers a valuable strategy for designing efficient SMAs but also provides a comprehensive understanding of the relationship between molecular structure, morphology, and performance, contributing valuable insights for the OSC community.

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### 通过不对称侧链取代调控小分子受体用于高效有机 太阳能电池

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摘要 小分子受体(SMAs)的侧链工程是一种显著提升有机太阳能电池 (OSCs)器件效率策略. 本工作研究了BT-BO和BT-TBO的母体SMAs, 以及新合成的不对称SMA BT-ASY, 该分子一侧具有带支化烷基链而另一侧具有在噻吩单元β位点取代的噻吩侧链. 尽管它们的光学和电化学性质相似,但基于PM6:BT-ASY的器件实现了18.08%的光电转换效率 (PCE),显著优于其对称结构的光伏体系. 这一提升主要归因于改善的电荷迁移率、延长的载流子寿命、优化的分子堆积和有效的相分离形貌,这些都得到了广角X射线散衍射测量的证实. 我们的研究结果强调,不对称侧链策略增强了π-π堆积和电子耦合,提供了一种简单而有效的方法来提高光伏性能. 这项工作突显了不对称结构修饰在小分子受体中推动有机太阳能电池技术和可再生能源解决方案的潜力.