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Tunable topological domain structures in high-density PbTiO, nanodots array ⊘

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ABSTRACT

In this work, we demonstrated that tunable topological domain structures, e.g., center-type domains and skyrmion-like polar bubbles, can be generated at room temperature in high-density epitaxial PbTiO₃ nanodots fabricated via the template-assisted tailoring of thin films. These topological domain structures can be manipulated electrically by applying an appropriate bias on the conductive atomic force microscopy tip, allowing for writing, erasing, and rewriting of topological domains into the nanodot. Moreover, ring-shaped conductive channels are observed around the center-type domain states. These findings provide a playground for further exploring their emerging functionalities and application potentials for nanoelectronics.

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Exotic topological domain structures in ferroelectric materials, such as vortices, ¹⁻³ flux-closures, ⁴⁻⁶ center domains, ⁷⁻¹² skyrmions, ¹³⁻¹⁵ bubbles, ¹⁶⁻²¹ and merons, ²² have been extensively studied to comprehend their physical properties as well as potential for high-density memories, ²³⁻²⁸ low-power consumption transistors, ²⁹ and ferroelectric-based nanoelectronic devices. ³⁰ These topological domains exhibit intriguing physical properties, including high stability attributable to topological protection, enhanced piezoelectric properties, ³¹ enhanced conductive channels, ³²⁻³⁴ and negative capacitance ³⁵ among others. However, there functional and structural complexity present challenges in efficiently identifying and manipulating the topological states.

Theoretical studies propose that the formation of topological domains in ferroelectric materials can be attributed to a delicate equilibrium among elastic, electrostatic, and gradient energies. 36,37 Substantial efforts have been made to produce and analyze topological

domains in recent years. For instance, it is reported that polar skyrmions can be stable in confined layers of PbTiO₃ (PTO) in PTO/SrTiO₃ (PTO/STO) superlattices. ¹³ A group of topological center domains, including center-convergent domains, center-divergent domains, and double-center domains, has been discovered in small BiFeO₃ (BFO) nanodots with a lateral size of 60 nm. ⁷ These center-type domains can be reversibly switched between convergent and divergent states, enabling the individual manipulation of the nanodots via electric fields. Ma *et al.* ⁸ further discovered that the center-type quad-domain exhibits a remarkable ability to transition reversibly between a divergent state, characterized by highly conductive confined walls, and a convergent state, characterized by insulating confined walls. This finding highlights the immense potential of this quad-domain in storage applications, as it enables nondestructive readout of topological center-domain states. More recently, there have been

reports of high-density skyrmion-like polar nanodomains in freestanding PTO/STO bilayers that were transferred onto the Si substrate, which indicated the application potential in silicon-based nanodevices. ¹⁴ However, creating and erasing these topological polar domains in precise dimensions at prescribed positions in a simple and controllable way, which is essential for on-demand designs and applications of these exotic domain structures, is still elusive.

It should also be noted that the center-type topological domains reported so far predominantly rely on rhombohedral phase ferroelectric films and nanostructures, for example, isolated BFO and Pb (Zr_{0.7}Ti_{0.3})O₃ (PZT) nanoislands tailored from thin films, ^{7,9} selfassembled BFO nanoislands embedded in tetragonal BFO matrix,8 etc. This greatly restricts the selection of systems available for studying topological center domains. Previous research results suggested that dimension reduction down to nanoscale is an effective strategy to manipulate topological defects. In nanoscale confined systems, e.g., nanodots/nanoislands, the surface or edge effects and flexoelectric effect generated by the possible nonuniform strain can drive the polarization away from their original directions to form more complex domain textures. 7,27,42,43 Therefore, it is intriguing to investigate whether these topological center-type states can maintain stability in structural systems apart from rhombohedral phase via the size effect. For this, we explore the domain structure in the PTO nanodots via tailoring of tetragonal films. Compared to the rhombohedra phase of BFO, PTO systems typically exhibit a-c domain states and have a much higher bandgap of around 3.4 eV, which may bring forth some different aspects in electronic and domain evolution properties.

In this work, we demonstrate the coexistence of different types of topological domain structures, e.g., center-type domains and skyrmion-like polar bubbles, in a high-density array of PTO nanodots fabricated via the template-assisted tailoring of tetragonal PTO thin films. These topological domain states can be electrically written, erased, and rewritten into the nanodot by utilizing an electrically biased atomic force microscopy (AFM) tip. Moreover, ring-shaped conductive channels have been observed surrounding these topological domains. The findings offer a valuable opportunity for in-depth exploration of their emerging functionalities and potential applications in nanoelectronics.

To begin, we start from the PTO nanodots array fabrication. The PTO thin films were first deposited on SrRuO₃ (SRO) buffered (100)oriented SrTiO₃ (STO) substrates via pulsed laser deposition (PLD) using a KrF excimer laser (wavelength $\lambda = 248$ nm) at 680 °C in 15 Pa oxygen at a fluence of 55 mJ at 8 Hz. The microstructures and ferroelectric properties of the as grown PTO thin films are illustrated in Fig. S1. The PTO film is in a single c-domain state with the polarization completely in the out-of-plane direction. The ordered PTO nanodot arrays were patterned via a mask-assisted Ar+ beam etching method. Figure S2(a) shows the schematic diagram of the fabrication process. The etching details can be referred to our previous work.³⁸ In brief, a monolayer of polystyrene spheres (PS, with lateral size of 200 nm) was first transferred onto the as grown PTO epitaxial film to form an ordered and close-packed template. Then, the PS were etched to the desired size by oxygen plasma to form a discrete array. This was followed by an Ar+ ion beam etching process. Finally, the PS template was lifted off with chloroformic solution to obtain ordered PTO nanodot arrays.

We then examined the patterned sample from the PTO film, which exhibits a well-arranged array of nanodots with average lateral size and height of 80 and 35 nm, as illustrated in the scanning electron microscopy (SEM) and atomic force microscopy (AFM) topographical images in Fig. 1(a) and Fig. S2(b), respectively. A rough estimation based on 80 nm lateral size of nanodots gives a high information memory density above 200 Gbit/in². The microstructures of the nanodots sample were then exampled by cross-sectional transmission electron microscopy (TEM) image, as shown in Fig. 1(b). A clear isolated PTO nanodots array was observed in the mesa. The high-resolution TEM image of a selected area in Fig. 1(b) is shown in Fig. 1(c), demonstrating the high-quality epitaxial structure. The x-ray diffraction (XRD) θ -2 θ spectrum in Fig. 1(d) reveals the good epitaxy structure of the nanodots. This is evidenced by the presence of diffraction peaks originating from the substrate STO (002), bottom electrode SRO (002), and PTO (002). Moreover, the epitaxy structure is further confirmed by the (002) reciprocal space map (RSM) shown in Fig. 1(e). The out-ofplane lattice constant is 4.151 Å, close to that PTO thin films.

To see the domain structures, these nanodots were examined using vector piezoresponse force microscopy (PFM). This technique enables the mapping of both vertical and lateral amplitudes (Lamplitude and V-amplitude) as well as phases (L-phase and V-phase) of piezoresponse signals simultaneously. Figures 2(a) and 2(b) show topography and vertical PFM phase images of a selected PTO nanodots array. We can see in V-phase image in Fig. 2(b), some of the nanodots have single dark (bright) phase contrast, suggesting the favorable upward (downward) out-of-plane polarization component. The rest of the nanodots show double contrast, indicating the coexistence of upward and downward out-of-plane polarization in each nanodots. Figures 2(c) and 2(d) show the lateral PFM amplitude and phase images obtained when the cantilever was aligned at 0° and 90° , respectively, relative to the [010] direction. The corresponding PFM image captured at 45° is shown in Fig. S3. It was found that most of the nanodots have the half-dark and half-bright contrast in the lateral phase captured in both two cantilever rotation angles. These features are confirmed by the presence of a distinct, dark line in the lateral amplitude images, displaying a domain wall-like characteristics for the subsequent polarization component along the (010) direction. In addition, a small number of nanodots have single contrast.

As shown in Figs. 2(e) and 2(f), we chose two representative nanodots with magnified PFM images captured at 0°, 45°, and 90° sample rotation angles as paradigms. Based on both lateral phase and amplitude data, one can derive a 2D vector polarization distribution map using the MATLAB program following the method proposed by Kalinin *et al.*³⁹ The result indicates that the domain structures of these two types of PTO nanodots are center-convergent domain with downward polarization and center-divergent domain with upward polarization. These domain structures are somewhat similar to the center domains in small BFO nanodots (60 nm in diameter), where the polarizations rotate continuously. Based on these findings, we also presented a comprehensive three-dimensional polarization distribution of the center domain structures, as illustrated in Figs. 2(g) and 2(h).

Equally exciting is the discovery of skyrmion-like polar bubbles in some PTO nanodots, as illustrated in Fig. 3. Figure 3(a) shows the vertical PFM phase and amplitude images of a selected nanodots array. We can observe that the majority of the out-of-plane polarization components of the nanodots exhibit cylindrical bubble domain structures, as shown in Fig. 3(a). However, a clear ring-shaped domain wall feature is present in the corresponding location in the vertical amplitude image.

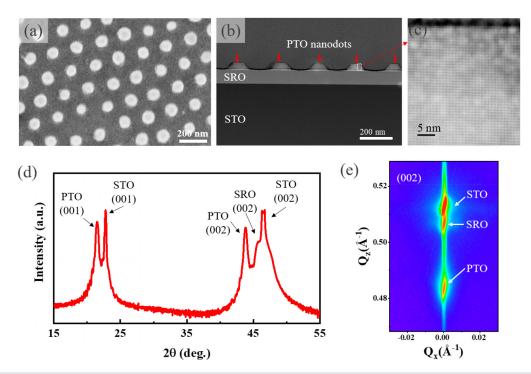


FIG. 1. (a) SEM image of a selected PTO nanodots array. (b) Cross-sectional SEM image of a PTO nanodots array. (c) High resolution TEM magnification image for the selected area in panel (b). The θ -2 θ scan diffraction pattern (d) and reciprocal space map (RSM) (e) measured at around (103) plane direction of the PTO nanodots array.

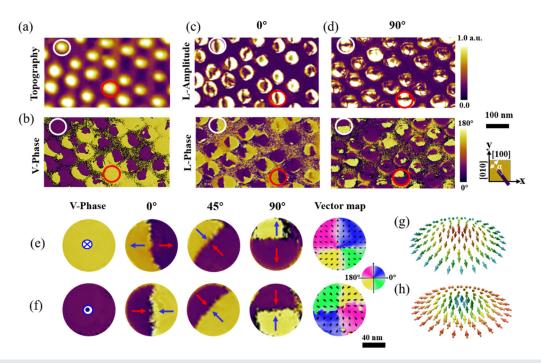


FIG. 2. (a) Topography image of the selected PTO nanodots array measured by AFM. (b) Vertical PFM phase image. (c) and (d) Lateral PFM amplitude and phase images obtained by aligning the cantilever at 0° and 90° to the [010] direction, as shown in the inset. (e) and (f) Magnified PFM images of the nanodots indicated by white circle and red circles in the corresponding images. (g) and (h) Schematic three-dimensional polarization distribution of the PTO nanodots.

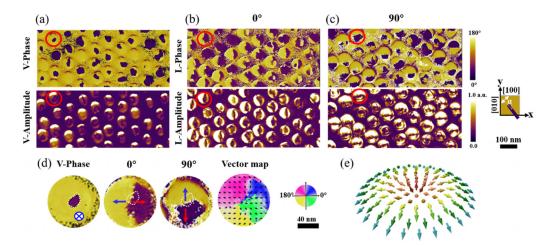


FIG. 3. (a) Vertical PFM amplitude and phase images of the PTO nanodots array. (b) and (c) Lateral PFM amplitude and phase images obtained by aligning the cantilever at 0° and 90° to the [010] direction, as shown in the inset. (d) Magnified PFM images of the nanodots indicated by red circles in the corresponding images. (e) Schematic three-dimensional polarization distribution of the PTO nanodots.

Figures S3(b) and S3(c) present the lateral PFM phase and amplitude images obtained when aligning the cantilever at 0°, 45°, and 90°, respectively, relative to the [010] direction. Clear dark lines are observed in the lateral amplitude images of the bubble domains. These lines separate the lateral phase images of the bubble domain into two halves, exhibiting a 180° phase contrast. It is worth noting that these lines are consistently aligned with the cantilever and rotate in accordance with its movement. This indicates that the bubble domains exhibit an inplane polarization configuration that is antisymmetric, with rotational symmetry around the out-of-plane axis passing through the center of the domain. Combining the vertical and lateral PFM results, one may identify that these are divergent nanodomains, in which the upward polarization at the core gradually diverges away from the center and ultimately transitions to a downward orientation along the periphery of these nanodomains, as shown in the schematic polarization distribution in Fig. 3(d), similar to the spin configuration of a Néel-type skyrmion in ferromagnetics⁴⁰ as well as the reported results in freestanding STO/ PTO bilayers transferred on Si substrates. 14

The observation of these topological domains in PTO nanodots is intriguing, as it reveals a deviation in the polarization from the typically observed out-of-plane orientations in tetragonal phase PTO. The formation of these domains is probably driven by the competitions among depolarization energy, polarization-strain coupling, and surface strain, all of which can greatly change the local anisotropy and thus the polarization distribution. The similar center domains with continuous rotational polarization instead of the earlier reported quadrant center domains have been reported in BFO nanodots with lateral size of 60 nm in our previous work.7 Moreover, the potential existence of nonuniform strain in the nanodots can induce flexoelectric rotations, thereby causing the polarization to deviate from its original orientation. Figure S4 shows the reciprocal space mappings of PTO films and nanodots around (002) diffraction of STO substrate before and after etching, respectively. The out of plane reciprocal vector of the nanodots sample increases compared to the films, indicating the decrease in the out of plane lattice parameter. The nanodots were studied in atomic resolution by observing a cross section through high angle

annular dark field scanning transmission electron microscopy (HAADF-STEM) from the [010] zone axis in Fig. S5(a). The GPA $\varepsilon_{\rm xx}$ (in-plane) map in Fig. S5(b) shows the strain variations along the direction parallel to the interface, indicating the increase in the inplane lattice parameter. This is because the nanodots etched from the thin film reduced the clamping effect from the substrate, resulting in a fast lattice strain relaxion in nanodots. These findings suggested that reducing the dimensions to the nanoscale is a viable approach to manipulate and control topological defects.

In addition, these topological domains in PTO nanodots can be reversibly switched. We next study the evolution of these topological domains by applying scanning bias at various voltages using conductive tip. As demonstrated in Fig. 4(a), the selected PTO nanodots show center-up skyrmion-like polar bubble structures. After scanning the tip biased at $-2 \,\mathrm{V}$, the out-of-plane cylindrical bubble domains grow to form single domains, while the lateral PFM phases show uniform halfdark and half-bright contrast. Combining the vertical and lateral PFM signals, it can be discovered that the center-up skyrmion-like polar bubble domains switched to center convergent domains with upward polarization [Fig. 4(b)]. Then, by scanning the tip biased at 2.5 V, the centerup skyrmion-like polar bubbles appear again. PFM measurements in Fig. 4(c) show that both the vertical and the lateral PFM phases are reversed as compared to the domain states in the as-prepared sample, indicating that these nanodomains are center divergent with the center polarization pointing upward. As the positive bias increases to 3.5 V, the upward *c*-domains in the center of the nanodots are completely switched downward. The center-up skyrmion-like polar bubbles switched to center divergent with the central polarization pointing downward [Fig. 4(d)]. Finally, by scanning the tip biased at $-2\,\mathrm{V}$, the center divergent states switched to center convergent states again [Fig. 4(e)]. These topological domains can be switched multiple times, and both remain stable for over 48 h at room temperature, as shown in Fig. S6.

Conductive atomic force microscopy (CAFM) measurements are performed on the selected nanodots to study the transport characteristics of the center domains. The current map measured by a $-1.5\,\mathrm{V}$ (+1.5 V bottom electrode bias in the CAFM mode) scan shows that

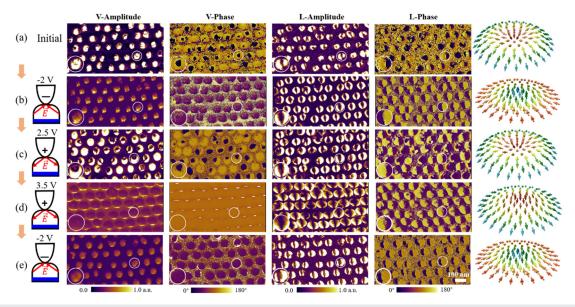


FIG. 4. (a) Vertical and lateral PFM amplitude and phase images and schematic three-dimensional polarization distribution of the nanodots marked with white circles. (b)–(e) The vector PFM images of the same region obtained after scanning the tip biased in sequence at -2.0 V (b), 2.5 V (c), 3.5 V (d), and -2.0 V (e).

ring-shaped conduction channels are distributed in the region of PTO nanodots (Fig. 5). The conduction channels are located on the hillsides of the nanodots, which exhibit large currents of 40 pA. We note that similar ring-shaped conduction channels were also observed in some BFO nanodots and nanoplates. ^{34,41} Turning to the origins of the formation of ring-shaped conduction channels in our PTO nanodots, we mainly consider the following factors. First, it is worth mentioning that

the sample is prepared by converting the film into nanodots, and the surface of the PTO film undergoes etching. This process creates abundant surface defects, such as oxygen vacancies, which can reduce the Schottky barrier at the interface of Pt tip and PTO. Additionally, due to more severe etching on the hillsides of the nanodots compared to the peaks, a greater concentration of surface defects is distributed along the hillsides, making the Schottky barriers there much lower.

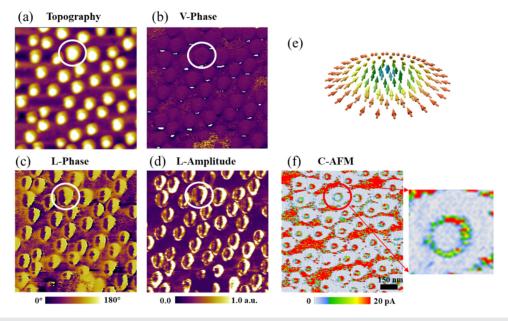


FIG. 5. (a)–(d) Topography (a), vertical PFM phase (b), lateral PFM phase (c), and amplitude (d) images of a PTO nanodots array, obtained after applied a write voltage of $-2.0 \,\mathrm{V}$ ($+2 \,\mathrm{V}$ bottom electrode bias in the CAFM mode). (e) Schematic three-dimensional polarization distribution of the polar nanodomain. (f) CAFM images of the same area shown in panels (a)–(d), obtained after applied a read voltage of $-1.5 \,\mathrm{V}$ ($+1.5 \,\mathrm{V}$ bottom electrode bias in CAFM mode).

This factor appears to play a significant role in the formation of ringshaped conduction channels. Second, we propose that the center-convergent-type topological domain structures of the as-etched PTO nanodots could potentially temporarily stabilize the surface defects located at the hillsides. More specifically, the presence of negative polarization charges near the surfaces of hillsides may contribute to the stabilization of positively charged oxygen vacancies, preventing them from diffusing away or interacting with the atmosphere. Therefore, the Schottky barrier is lowered due to the presence of oxygen vacancies, which in turn contribute to the formation of ring-shaped conduction channels. In addition, there is a large polarization gradient in the center domains region of the nanodot that serve as traps for space charge to partially compensate the bound charges. Carriers associated with these space charges may be set free as long as the potential drop at hillsides of the nanodots bend the conduction band below the impurity levels, leading to an enhanced conductivity.

In summary, tunable center domains and skyrmion-like polar bubbles are observed in high-density PbTiO₃ nanodots on SrRuO₃-buffered (001) STO substrate fabricated via the template-assisted tailoring of PTO thin films. Using vector PFM measurements, the polarization distribution of these nanodots is identified, which is similar to the spin configuration in small sized BFO nanodots (center domain states) and freestanding PTO layer (skyrmion-like polar bubbles). We are able to write and erase these topological domains by adjusting scanning bias applied on the conductive AFM tip. Moreover, ring-shaped conductive channels are observed around the center domain states. These findings suggest that reducing dimensions to the nanoscale is a promising strategy for manipulating and controlling topological defects, which in turn creates opportunities for extensive exploration of their emerging functionalities and application potentials in the field of nanoelectronics.

See the supplementary material for details on domain structures of the PTO film/nanodots, schematic flow chart illustrating the procedures of fabricating the PTO nanodots array, reciprocal space mapping of samples, and HAADF-STEM image and GPA $\varepsilon_{\rm xx}$ (in-plane) map of a part of a PTO nanodots.

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AUTHOR DECLARATIONS

The authors have no conflicts to disclose.

Author Contributions

Conflict of Interest

Hongying Chen: Data curation (equal); Investigation (equal); Writing – original draft (supporting). **Zhiyu Liu:** Data curation (equal). **Guo Tian:** Conceptualization (lead); Funding acquisition (equal);

Project administration (equal); Writing – original draft (equal); Writing – review & editing (equal). Gui Wang: Investigation (equal). Yihang Guo: Investigation (equal). Zongwen Duan: Investigation (equal). Di Wu: Investigation (equal); Resources (equal); Supervision (equal). Yu Deng: Data curation (equal). Guoyu Wang: Investigation (equal); Resources (equal). Zhipeng Hou: Investigation (supporting). Deyang Chen: Investigation (supporting). Zhen Fan: Investigation (supporting). Minghui Qin: Funding acquisition (equal); Investigation (supporting). Ji-Yan Dai: Funding acquisition (equal); Writing – original draft (supporting). Jun-Ming Liu: Investigation (supporting). Xingsen Gao: Funding acquisition (equal); Investigation (equal); Project administration (equal); Writing – original draft (equal).

DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding authors upon reasonable request.

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