

24 **Upcycling of Air Pollution Control Residue waste into cementitious product through**
25 **geopolymerization technology**

26 **Abstract**

27 This study explores the possibility of using geopolymerization technology (GT) to immobilize the
28 potentially toxic elements (PTEs e.g., Zn, Cu, Cr, As) in the APCr and convert it into useful
29 cementitious product. To maximize its recycling, the amount of APCr in the designed product was
30 increased gradually from 20% to 80% by the total solid mass. Leaching test showed that GT can
31 effectively immobilize the PTEs in the APCr solidified samples without any health and
32 environmental concerns. The compressive strength of samples can exceed 18 MPa at 28 days at a
33 highest amount of 80% APCr through GT. Thermogravimetric analysis (TGA) results showed that
34 solidified samples underwent mass loss due to evaporation of free and physically bound water at
35 low temperatures (<200°C) and melting and evaporation of soluble salts in APCr at high
36 temperatures (>800°C). Characterization of solidified samples conducted through the X-ray
37 diffraction (XRD), Fourier transform infrared (FTIR) and Scanning electron microscopy-Energy
38 dispersive analysis (SEM) revealed the formation of C-A-S-H and N-A-S-H gels in solidified
39 bodies and verified that APCr was successfully solidified and embedded into the geopolymer
40 network structure.

41 **Keywords**

42 Air pollution control residue; Geopolymer; solidification; Leaching; sewage sludge

43 **1 Introduction**

44 Incineration is a widely used and successful technique for the treatment of dewatered sewage
45 sludge and municipal solid wastes (MSW) as it can effectively reduce volumes of wastes and
46 generate energy. This process can eliminate the organic matter while reducing the volume of waste

47 by 85-90% and mass of waste by 60-90% (Leckner, 2015), which greatly helps to reduce the
48 demand of land required for the disposal of MSW and sewage sludge waste and its related carbon
49 footprint, especially in the heavily populated areas like Hong Kong (Woon and Lo, 2013; Xi et al.,
50 2021; Zhuang et al., 2020). As per Hong Kong Environmental Bureau (EBHK, 2013), Hong Kong
51 is expected to reach the designed capacities of present landfills in the near future due to the high
52 disposal rate of wastes, and extension of these landfills will become inevitable. Hong Kong has
53 one of the world's largest incinerator facilities to treat the dewatered sewage sludge that can
54 incinerate nearly 2000 tonnes/day of sewage sludge. However, incineration ashes produced from
55 these so-called waste-to-energy (W-t-E) incinerators contains several potentially toxic elements
56 (PTEs) (Chen et al., 2021; Tang et al., 2020; Lei Wang et al., 2020). Hence, disposal of such kind
57 of ashes into landfill without prior treatment should be evaluated.

58 Air pollution control residue (APCr) is regarded as a kind of industrial waste produced from these
59 incinerators and is normally transported to waste landfill sites. It is produced during the cleaning
60 process of flue gas in the incinerators and contains solids and fly ash which are captured
61 downstream of acid gas treatment. It is highly alkaline and corrosive and consists of soluble anions
62 and pollutants in a large concentration. This combination further poses problems in the treatment,
63 recovery, and disposal of APCr (Gomes et al., 2016) and hence it is classified as a hazardous waste
64 in several jurisdictions (Bogush et al., 2015; Chandler et al., 1997). It is estimated that the
65 production of APCr will continue to rise due to the increase in numbers of incinerators and W-t-E
66 plants. Thus, it has become necessary to find out the recovery and recycling options of APCr.
67 Treatment of hazardous waste materials through solidification/stabilization (S/S) into valuable
68 construction products is a viable and cost-effective approach (Sun et al., 2022).

69 S/S of industrial hazardous wastes is often conducted by using cementitious materials mainly
70 ordinary Portland cement (Alderete et al., 2021; Clavier et al., 2021; Geng et al., 2020; Kumar and
71 Garg, 2022; Loginova et al., 2021; Ma et al., 2020; Qian et al., 2022). Recently, GT has received
72 increasing attention as a substitute of solidification through ordinary Portland cement (OPC) due
73 to its potential benefits as compared to OPC e.g. (1) release of a large amount of carbon gas due
74 to the production of Portland cement and (2) carbonation (Lange et al., 1995; Wang et al., 2019;
75 Y.-S. Wang et al., 2020). Several studies have reported the use of geopolymer based binder for the
76 S/S of radioactive contaminated sewage sludge ash (Kozai et al., 2021) and borate waste (Kim et
77 al., 2021), heavy metals (Zn, Pb, and Cr) (Kozai et al., 2021; Pu et al., 2021; Xuan and Poon,
78 2018), municipal solid waste incineration (MSWI) fly ash (Cyr et al., 2012; Kozai et al., 2021; Liu
79 et al., 2021; Wang et al., 2015), bottom ash (Boca Santa et al., 2016), bauxite residue and coal ash
80 (Nguyen et al., 2022) and lead glass and lead slag (Long et al., 2021a; Nath et al., 2022; Pan et al.,
81 2019; Sun et al., 2020; Van De Sande et al., 2020). Moreover, a comparison of sewage sludge ash
82 and MSWI fly ash S/S through the use of geopolymer and Portland cement showed that GT was
83 far superior in immobilization of heavy metals and the produced geopolymer based solidified
84 products showed higher uniaxial compressive strength than its Portland cement counterpart (Fan
85 et al., 2021; Kozai et al., 2021).

86 There are numerous studies available on the S/S of different types of hazardous wastes, however,
87 very limited research is available on the recycling of APCr and its utilization as construction
88 material in civil engineering (Bogush et al., 2020; Quina et al., 2014; Stegemann, 2014). Kourti et
89 al., used glass-forming additives and DC plasma technology to treat the APCr and produce
90 precursors for the geopolymers (I. et al., 2011; Kourti et al., 2010). A low-strength cement (<10
91 MPa) was produced by solidification of APCr through alkali-activation by mixing it with the co-

92 fired fuel ash (Shirley and Black, 2011). Geopolymerization technology was also helpful in
93 immobilizing the several heavy metals (Ba, Pb, and Ni) present in the APCr (Mustard et al., 2016).
94 Recycling of waste materials through S/S does not only reduces the burden on landfills but also
95 eliminates the potential environmental risks associated with the disposal of hazardous wastes
96 occurring due to leaching of the PTEs. Meanwhile, recycled waste materials through S/S can also
97 partially reduce the demand for conventional construction materials leading to the path of
98 sustainability and low carbon footprint (Ahmad et al., 2020c, 2020b, 2020a; Chang et al., 2022;
99 Zentar et al., 2021; Zhang et al., 2022, 2020). In this regard, this study explores the possible use
100 of APCr for the production of a cementitious product by immobilization of heavy metals through
101 the geopolymerization technology (GT) (Li et al., 2017).

102 Therefore, this study aims to evaluate the feasibility of using GT for the upcycling of APCr waste.
103 In recent years, a significant amount of research has been conducted on the reuse of MSWI fly ash,
104 bottom ash, and sewage sludge ash in the cementitious system, however, very few researchers have
105 tried to utilize the APCr through the use of GT. Hence, this study is focused to use the GT for the
106 sustainable utilization of APCr. First, the produced APCr geopolymer pastes through S/S depict
107 low to medium strength and can be used for the application as construction materials. Second, S/S
108 of APCr will help to curb the landfill disposal issues and its potential environmental risks
109 associated with the leaching of heavy metals. Furthermore, it will directly reduce the demand for
110 virgin construction materials used in the cement and concrete application.

111 2 **Materials and Methods**

112 2.1 *Raw materials*

113 Raw materials used to prepare the solidified APCr based geopolymer paste included fly ash (FA),
114 ground granulated blast-furnace slag (GGBS), APCr, and anhydrous sodium metasilicate (SS;

115 Na₂SiO₃-Anhydrous). Particle size distribution of raw materials determined by the Laser
 116 diffraction technique is plotted in Fig. S1 (supplementary information). The mean diameter of FA,
 117 GGBS, and APCr was 17.9, 12.4, and 8.5 μm with the specific surface area of 1392, 1488, and
 118 6651 m²/kg, respectively. The chemical compositions of raw materials are provided in Table 1.
 119 The FA mainly consisted of SiO₂, Al₂O₃, and Fe₂O₃ (84.8%) and is classified as Class F FA as per
 120 ASTM C618-19. Major compounds in GGBS are CaO (42.4%), SiO₂ (32.6%), and Al₂O₃ (14.7%),
 121 whereas Na₂O (38.3%) and SO₃ (31.9%) are the main compounds in APCr.

122
 123

124 Table 1. Oxide composition of FA, GGBFS and APCr (%)

Oxides	Fly ash	GGBS	APCr
SiO ₂	49.7	32.6	6.19
Al ₂ O ₃	25.1	14.7	3.57
Fe ₂ O ₃	9.58	0.35	4.65
CaO	7.84	42.4	3.27
MgO	2.8	6.53	1.71
Na ₂ O	-	-	38.3
K ₂ O	1.53	0.41	0.57
TiO ₂	1.15	0.61	0.19
P ₂ O ₅	0.84	0.18	3.02
SO ₃	0.96	1.82	31.9
Others	0.5	0.4	6.64

125

126 X-ray diffraction (XRD) patterns of FA, GGBS, and APCr are plotted in Fig. S2. APCr was
 127 obtained from the T-park incinerator in Hong Kong. It was dried in an oven and then ground to
 128 powder in a ball mill for four hours before solidification. The main phases in FA are crystalline
 129 mullite (3Al₂O₃·2SiO₂) and quartz (SiO₂) and APCr consists of thenardite (Na₂SO₄), halite (NaCl),
 130 monosodium phosphate (NaH₂PO₄), and trisodium phosphate (Na₃PO₄). GGBS showed highly
 131 amorphous behavior through the broad hump around 25-35° 2θ. SS of industrial-grade containing
 132 2.73% of impurities, 46.52% of SiO₂, and 50.75% of Na₂O was used as an alkaline activator in this

133 study. The modulus ratio ($M_r = \text{SiO}_2/\text{Na}_2\text{O}$) of SS was 0.92. The loose bulk density of SS was 1.31
 134 g/cm^3 and size of particles was between 0.25-1.0 mm. Morphology and distribution of raw
 135 materials observed by SEM are shown in Fig. S3. FA consists of sphere shape particles with
 136 smooth surface. GGBS has angular/irregular-shaped particles while the shape of APCr particles
 137 consists of partially rounded corners and a fluffy appearance.

138

139 2.2 *Mix proportions and solidification*

140 In this study, different dosages of APCr ranging from 20% to 80% were solidified by the GT. Total
 141 five mix proportions were prepared as shown in Table 2. One reference geopolymer mixture
 142 (G100-APCr0) without APCr was also prepared to compare the mechanical and microstructure
 143 properties with the APCr based solidified mixtures. The percentage of fly ash to slag in the
 144 reference mixture was 80:20. The amounts of GGBS and SS were kept constant for all mixtures to
 145 achieve the higher solidification efficiency whereas FA was gradually replaced APCr. The
 146 mixtures notation followed the weight percentage of geopolymer and APCr, respectively, e.g.,
 147 G20-APCr80 means mixture containing 20% of geopolymer and 80% of APCr by total weight.
 148 Geopolymer consists of GGBS, FA, and SS. Here, FA, GGBS and SS act as the binder materials
 149 for the APCr. The percentage of alkali content (Na_2O) for chemical activation was kept 5.1% by
 150 the total weight of solids for all mixtures.

151 Table 2. Mix proportions of APCr solidified bodies

Mix ID	G:APCr (solids)	water/solids	Na_2O (% of solids)
G100-APCr0	100:0	0.33	5.1
G80-APCr20	80:20	0.33	5.1
G60-APCr40	60:40	0.33	5.1
G40-APCr60	40:60	0.33	5.1
G20-APCr80	20:80	0.33	5.1

152

153 The mixing process was started by dry mixing the raw materials (FA, GGBS, APCr and NS) in
154 the Hobart mixer at a low speed. After dry mixing for 3 minutes, water was gradually introduced
155 into the mixer, and mixing was carried out at a medium speed for another 3 minutes. Finally,
156 mixing was continued at a high speed for 3 minutes until a homogenous and uniform mixture was
157 obtained. Fresh mixtures were poured into plastic molds of dimensions 50×50×50 mm³ and
158 covered with the plastics sheets to avoid the loss of moisture. All the samples were demolded after
159 24 hours of casting and stored in plastic sheets at ambient air conditions until testing.

160 2.3 *Experimental programme*

161 *Immobilization efficiency of GT (TCLP method)*

162 Immobilization efficiency of GT was characterized through the TCLP method 1311 (Toxicity
163 Characteristic Leaching Procedure for heavy metals) as per United States EPA protocol (EPA,
164 1992; Liuwei Wang et al., 2020). Before TCLP, raw APCr was digested through the aqua-regia
165 digestion medium and total heavy metal concentration was determined with the Inductively
166 Coupled Plasma-Optical Emission Spectrometer (ICP-OES, SpectroBlue). After it was found that
167 the concentration of heavy metals in APCr is higher than critical limits, solidified samples were
168 further tested for the TCLP test. Solidified samples were crushed to small sizes and were passed
169 through a 4 mm sieve. Extraction was performed by rotating the samples capped in polypropylene
170 bottles for 18 ± 2 h at 30 ± 2 rpm in a TCLP rotator. The extraction liquid was glacial acetic acid
171 and liquid to solid ratio was 20:1. Liquid was separated by filtering the extracted liquid through
172 the 0.45 μm glass fiber filter. Liquid samples were further diluted by concentrated HNO₃ (sample
173 5:2 HNO₃) and digested. After the digestion, samples were diluted with 5% HNO₃ and filtered
174 again. The concentration of heavy metals was determined using the ICP-OES instrument. After
175 extraction, pH of leachates was also measured through the pH meter.

176 *Mechanical testing*

177 Samples were removed from the plastic sheets for the determination of compressive strength and
178 tested using a compression machine (MATEST, 3000 kN) at 3 days and 28 days as per ASTM
179 C109. All the samples were tested in triplicate at the loading rate of 1 MPa/s (Xu et al., 2021).
180 Water absorption of samples was measured by immersing the samples into the water for 24 hours
181 (to calculate saturated weight) and then drying in an oven at $105\pm 2^{\circ}\text{C}$ for 24 hours (to calculate
182 dry weight).

183 *Phase and microstructure characterization*

184 Solidified samples were ground to fine powder after drying in the freeze-dryer and XRD analysis
185 was performed using Rigaku SmartLab-Advance instrument at 200 mA and 45kV using Cu-K α
186 radiations. Scanning rate for all powder samples was kept at 0.02° per step using Bragg-Brentano
187 focusing mode. Scanning of samples was carried out for the 2θ value from 5° to 75° . Samples for
188 the XRD analysis were prepared after 28 days of ambient curing. To characterize the
189 microstructure of solidified samples by scanning electron microscope (SEM), 28 days ambient
190 cured samples were broken into small pieces and embedded into epoxy for polishing purposes.
191 After polishing of samples using AutoMet 300 Grinder, samples were freeze-dried. The
192 microstructure of samples was observed through the SEM machine (Tescan Vega 3 XMU)
193 equipped with an EDS facility. For characterization of hydration products and unreacted phases,
194 elemental and mapping analysis of samples was performed. The SEM analysis was performed in
195 high vacuum conditions at a voltage value of 20 kV and a working distance of 18 mm.
196 Thermogravimetric analysis (TGA) on the dried powder samples was performed using the Rigaku
197 Thermo Plus EVO2 instrument. The mass of each sample was controlled around 8 mg and placed
198 in an alumina crucible. The heating rate of samples was maintained at $10^{\circ}\text{C}/\text{min}$ while samples

199 were heated from 30° to 1000 °C in argon gas conditions. Fourier-transform infrared (FTIR)
200 analysis was conducted on the powder samples using the PerkinElmer UATR-two to identify the
201 functional groups. The analysis was performed in the range of 400 cm⁻¹ to 4000 cm⁻¹ at 2 cm⁻¹
202 resolutions.

203 **3 Results and Discussions**

204 *3.1 Immobilization efficiency of APCr through GT*

205 The results of total concentrations for different metals are provided in Fig. S4. Heavy metals
206 present in APCr were Ni, Pb, Zn, As, Cd, Cr, Ba, Ag, and Cu. Some of the heavy metals e.g. Hg
207 and Se were not detected and hence were of no concern. Among all heavy metals, the highest
208 concentration was shown by Zn was around 1019 mg/L. The concentrations of Cu, Ba, Cr, and As
209 were 266, 173, 99.4, and 75.5 mg/L respectively. Moreover, Pb, Ni, Ag, and Cd were also present
210 in little amounts. Hence, it is highly suggested that APCr must undergo treatment when it is aimed
211 to be reused for construction purposes due to the presence of high metal contents. As health and
212 environmental risks are more associated with the presence of soluble heavy metals rather than the
213 total concentration of heavy metals, leaching test was carried out by TCLP method to study the
214 leaching characteristics of APCr samples solidified through GT.

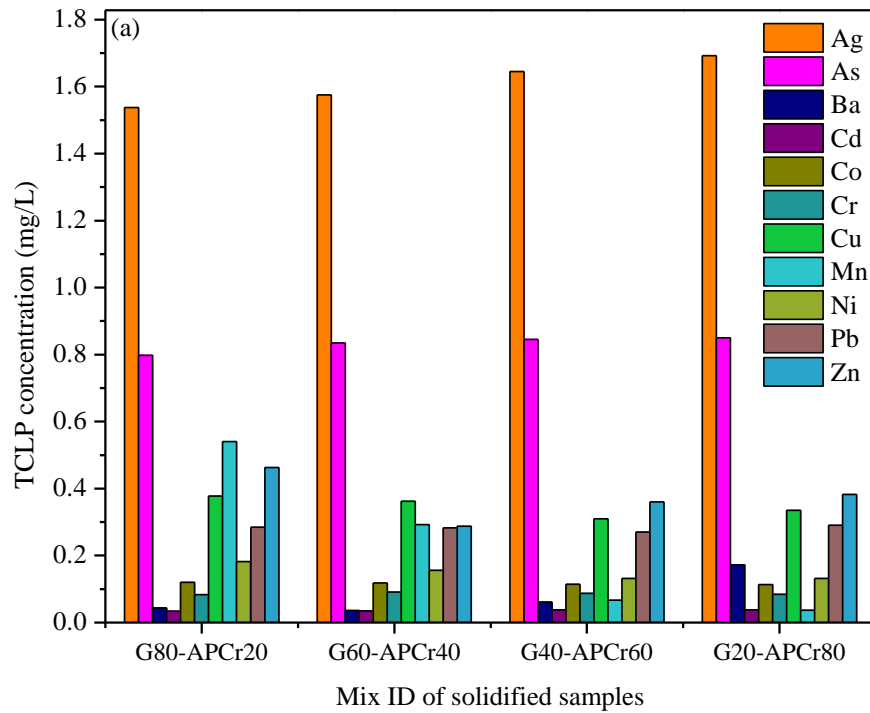
215 The TCLP method simulates the leaching of PTEs from the landfill and is often used to measure
216 the efficiency of solidification approaches to immobilize heavy metals. The results of TCLP
217 method for APCr solidified bodies through GT are shown in Fig. 1a. The concentration of heavy
218 metals leached from the APCr samples was greatly reduced when solidified through the GT.
219 Importantly, concentrations for all leached metals were far below the regulatory limits advised by
220 the U.S. EPA (US EPA, 2004) and Chinese standard (CN-GB, 2007) as shown in Table 3. This
221 indicates that geopolymer is very efficient to immobilize heavy metals even at a higher percentage

222 of APCr of 80%. Geopolymer consists of 3D structure of aluminosilicate network of AlO_4 and
 223 SiO_4 tetrahedral units which can effectively seal the contaminants. Moreover, it has been reported
 224 that fly ash based geopolymer skeleton is normally stable under the strong acid-base environment.
 225 Besides the physical encapsulation, double-layered hydrate phases in geopolymer (e.g. calcium
 226 alumino/ferric) can interact with these metals chemically and replace them with interlayer
 227 hydroxyl (Long et al., 2021b; Luna Galiano et al., 2011).

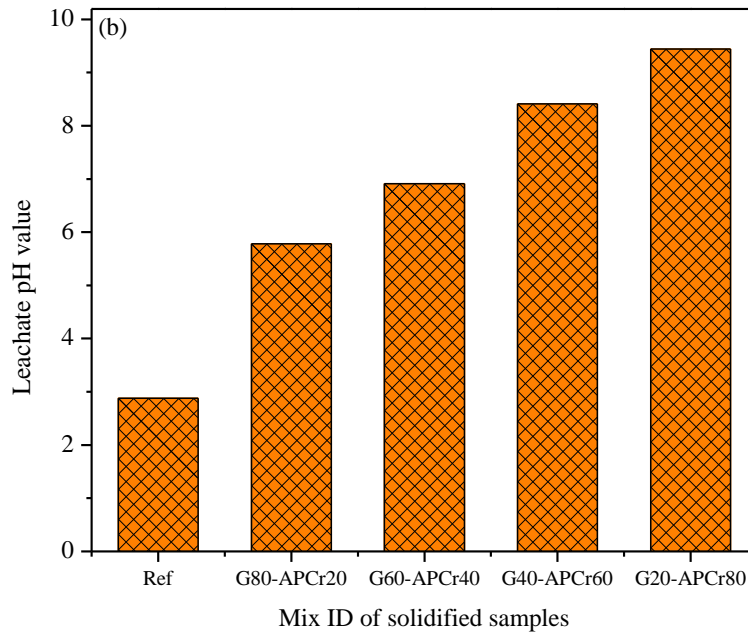
228 Table 3. TCLP concentration of heavy metals and regulatory limits

Metal	TCLP concentration of heavy metals (mg/L)				Upper Limitation (mg/L)	
	G80-APCr20	G60-APCr40	G40-APCr60	G20-APCr80	U.S. EPA	CN-GB
Ag	1.54	1.58	1.65	1.69	5.0	-
As	0.80	0.84	0.85	0.85	5.0	5.0
Ba	0.04	0.04	0.06	0.17	100	100
Cd	0.03	0.04	0.04	0.04	1.0	-
Co	0.12	0.12	0.11	0.11	-	-
Cr	0.08	0.09	0.09	0.08	5.0	5.0
Cu	0.38	0.36	0.31	0.34	-	100
Mn	0.54	0.29	0.07	0.04	-	-
Ni	0.18	0.16	0.13	0.13	-	-
Pb	0.29	0.28	0.27	0.29	5.0	5.0
Zn	0.46	0.29	0.36	0.38	-	100

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232 Fig. 1. TCLP results of Solidified samples (a) total concentration of heavy metals; (b) pH value
 233 of leachate

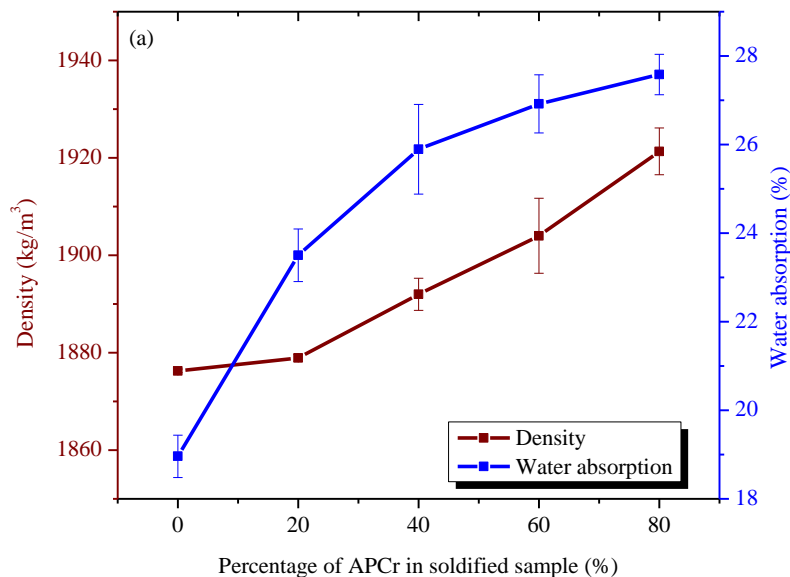
234 The pH values of leachates of TCLP extracted samples are shown in Fig. 1b. Leachate pH value
 235 is one of the key factors and provides important information about the immobilization efficiency
 236 of cementitious materials. The pH value of Ref. solution (buffer acetic acid solution) was 2.88 ± 0.5 .

237 After TCLP extraction, pH value was gradually increased with the increase in percentage of APCr
238 and was in the range of 5.78 to 9.44. Similar results have been reported in past studies and an
239 increase in pH value is associated with the acid buffering capacity of the alkaline nature of APCr
240 waste and geopolymer. According to past studies, the heavy metals (Mn, Cu, Cd, Zn, and Pb)
241 which follow the cationic leaching behavior showed the reduced leaching concentration under the
242 higher pH environment (Luo et al., 2019; Zhang et al., 2016) whereas concentration of Ba and As
243 increased with the increase in alkalinity. Results of the present study are in agreement with the
244 previous findings as with the increase in pH, Mn and Zn leaching was reduced and Ba and As
245 leaching was slightly increased. However, there were marginal differences in the leaching
246 concentration of all solidified samples and all values were well below the upper limits.

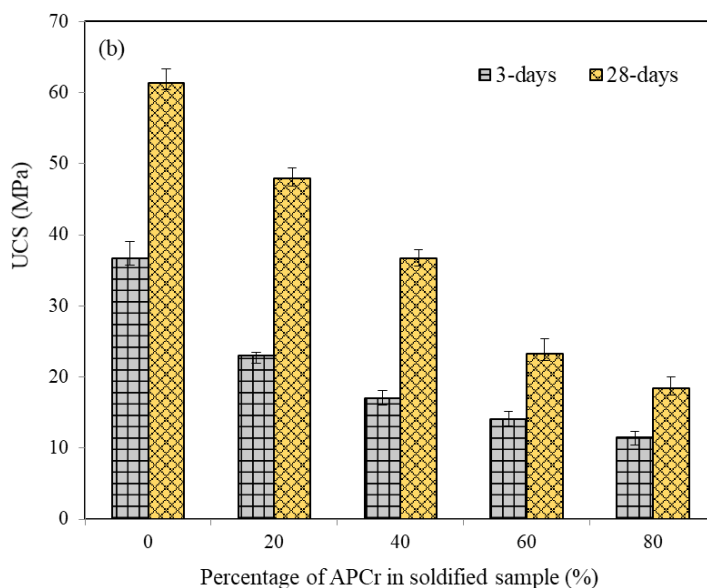
247 *3.2 Physical and mechanical properties of APCr solidified samples*

248 Bulk density and water absorption of control and solidified samples measured after 28 days of
249 ambient curing are shown in Fig. 2a. There was a marginal increase in density when the amount
250 of APCr was increased from 0% to 80%, however, the addition of APCr considerably increased
251 the water absorption of solidified samples. The density and water absorption of controls samples
252 without APCr (G100-APCr0) were 1876 kg/m³ and 18.9%, which were increased to 1921 kg/m³
253 (2.3% increase) and 27.58% (31% increase) for the G20-APCr80. The density of geopolymer
254 pastes in a similar range has been observed in previous studies. (Hui-Teng et al., 2021; Song et al.,
255 2019). The increase in density of APCr solidified samples can be linked to the smaller particle
256 size of APCr as compared to FA and GGBS as it can fill into micropores and slightly increase the
257 packing density of solidified mixtures. The increase in water absorption of APCr samples is mainly
258 due to the filler nature of APCr as it does not react in a geopolymer system. When the amount of
259 relatively inert APCr is increased, reactive aluminosilicate precursors are reduced in similar

260 proportions and a relatively weaker geopolymer network structure with the lower strength (see Fig
261 2(b)) is formed, leading to the higher water absorption of samples.



262



263
264 Fig. 2. Physical and mechanical properties of APCr solidified samples (a) Density and Water
265 absorption; (b) Unconfined compressive strength (UCS)The results of unconfined compressive
266 strength (UCS) for the APCr solidified samples are plotted in Fig. 2b. The pure geopolymer matrix
267 (G100-APCr0) without APCr had the highest strength among all mixtures both at the age of 3 days

268 and 28 days. The UCS of G100-APCr0 was 36.7 MPa and 61.3 MPa at 3 days and 28 days,
 269 respectively. During the process of geopolymer reaction, there are several types of ions
 270 $[\text{SiO}_3(\text{OH})]^{3-}$, $[\text{SiO}_2(\text{OH})]^{-2}$, $[\text{SiO}(\text{OH})]^{-}$ and $[\text{AlO}(\text{OH})_4]^{-}$ available to dissolve in an alkaline
 271 environment, which undergo polycondensation with Na and Ca cations to produce the three-
 272 dimensional network structure of geopolymer gels (N-A-S-H or C-A-S-H). The geopolymer gels
 273 are normally more compact than OPC based C-S-H gel and hence, make the geopolymer a stronger
 274 candidate for the solidification of hazardous wastes (Rasaki et al., 2019; Tian et al., 2020).

275 For the solidified APCr waste through GT, the UCS was significantly reduced gradually from
 276 61.3 MPa to 18.4 MPa. The UCS for APCr solidified samples containing 20%, 40%, 60%, and
 277 80% APCr was 47.9 MPa, 36.7 MPa, 23.3 MPa, and 18.4 MPa respectively. The decrease in the
 278 strength of APCr solidified samples was linked to the very low reactivity or filler nature of APCr
 279 (Li et al., 2017). This was because FA was gradually replaced by APCr for the solidification
 280 purpose and it contained a much lower amount of aluminosilicate phases (3.6% SiO_2 and 6.2%
 281 Al_2O_3) as compared to FA which could facilitate to form geopolymer gels (N/Ca-A-S-H) and
 282 contribute to strength development. Hence, as FA is replaced by APCr, the amount of
 283 aluminosilicate is also reduced in the solidified mixture and a lower amount of sodium
 284 aluminosilicate hydrates (N-A-S-H) were formed (Ali Shah et al., 2021; Yousefi Oderji et al.,
 285 2019). Results of the present study are compared with similar past studies conducted on the
 286 solidification of hazardous lead slag (Long et al., 2021a) and MSWI-FA (Fan et al., 2021)
 287 through FA-slag based geopolymer and FA based geopolymer respectively in Table 4. The
 288 comparison shows that GGBS-FA based geopolymer is more efficient in solidification of APCr
 289 than that of hazardous lead slag and MSWI-FA given that APCr solidified samples showed a
 290 superior strength even at a higher dosage of APCr waste (80%) and lower alkali content
 291 (5%). Table 4. Comparison of strength of APCr solidified samples with literature

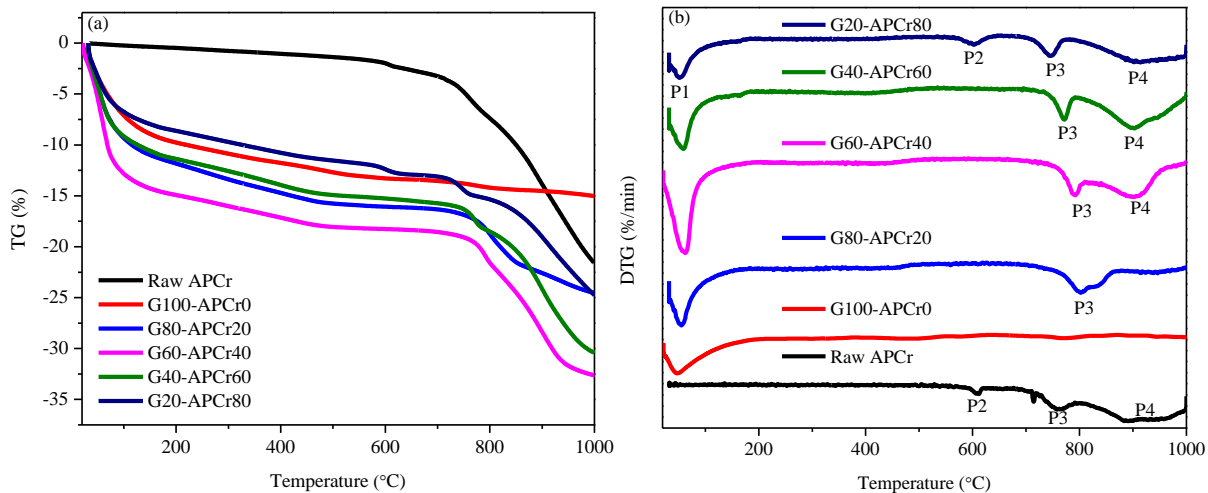
Type of waste	% of waste by total mass	Precursor ratio (GGBS:FA)	UCS (MPa)	Na ₂ O/Precursor (%)
Hazardous lead slag (Long et al., 2021a)	33	1:0	18	10
	33	1:0.43	18	10
	33	1:1	17	10
	33	1:2.33	14.5	10
	33	0:1	5	10
MSWI-FA (Fan et al., 2021)	50		16.1	10
	60		18.1	10
	70	0:1	22.4	10
	80		18.7	10

	90		14.3	10
This study (APCr)	20	1:3	47.9	5
	40	1:2	36.7	5
	60	1:1	23.3	5
	80	1:0	18.4	5

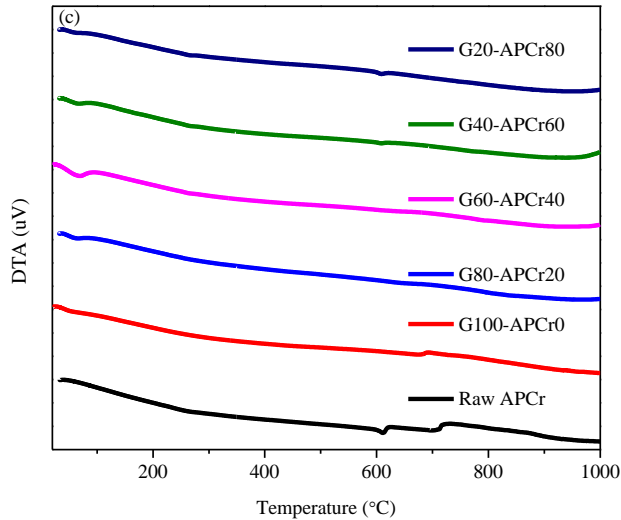
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293 *3.3 TG/DTA results of APCr solidified bodies*

294 The thermal stability of raw-APCr and APCr solidified samples was analyzed by the
 295 thermogravimetric analysis (TG-DTG-DTA curves) and results are shown in Fig. 3. The total mass
 296 loss for the raw-APCr from 30°C to 1000°C was 21.6%, The similar amount of (21%) mass loss
 297 was reported by the recent TGA study on APCr (Bogush et al., 2019). The mass of raw-APCr was
 298 quite stable up to 600°C and a sharp decrease was observed beyond that. DTG curve of raw-APCr
 299 showed three distinct peaks around 606°C (P2), 760°C (P3), and 891°C (P4). This loss in mass is
 300 linked to the endothermic peaks as shown in Fig. 3c around those temperatures. The loss in mass
 301 in between 600-760°C is primarily due to the decomposition of calcite (CaCO₃). Raw APCr had
 302 a mass loss of 3.39% in this temperature range. A broad peak around 891°C is associated with
 303 evaporation of Na, K, and Ca-based soluble salts containing sulfates and chlorides (Amutha Rani
 304 et al., 2008; Bogush et al., 2019).



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307 Fig. 3. Thermogravimetric analysis of APCr solidified samples (a) TG analysis; (b) DTG
 308 analysis; (c) DTA analysis. DTG and DTA curves are shifted for comparison purpose

309 For the control samples without APCr, there is only one distinct and broad peak (P1) as shown in

310 DTG curve. The major loss in mass occurred from 30°C to 180°C. This loss in mass in control

311 sample is attributed to evaporation of free water initially at lower temperatures and physically

312 bound water (in C-S-H or N-(Ca)-S-H gels) at higher temperature up to 180°C (Hager et al., 2021;

313 Kuri et al., 2021). The rate of mass loss was significantly reduced in the temperature range of

314 180°C to 1000°C. Mass loss for control sample was 9.5% from 30°C to 180°C and 5.5% from

315 180°C to 1000°C with the lowest total mass loss value of 15% among all samples. TG curves for

316 the all APCr solidified samples showed the combination of raw-APCr and control sample and two

317 prominent stages of loss in mass were seen in all solidified samples; (i) 30°C to 180°C and (ii)

318 beyond 600°C. Mass loss was first increased with the increase in amount of APCr from 0% to 40%

319 and then decreased with the further increase in amount of APCr from 40% to 80%. A sharp

320 endothermic peak (P1) appeared in the G80-APCr20, becoming more deeper in G60-APCr40 and

321 then again reduced in G40-APCr40 and G20-APCr80 (DTG curve Fig. 3b). Beyond 600°C,

322 endothermic peaks (P2, P3, P4) started to reappear in the APCr solidified samples due to melting

323 and evaporation of soluble salts present in the APCr. Between the 600-750°C, for the reference

324 mixture without APCr, mass loss was 0.58%. For the mixtures containing 20, 40, 60 and 80%
325 APCr, mass loss was gradually increased to 0.99%, 1.02%, 1.43 and 2.42% respectively. Hence,
326 this increase in mass loss is associated with the phase changes from APCr. It is important to
327 mention that these peaks were shifted to the right side of their original position (although more
328 distinct) due to geopolymerization process in solidified samples. The final loss in mass of solidified
329 samples was higher than the control samples due to combined loss in mass of (i) free and physical
330 bound water from geopolymer below 200°C and (ii) evaporation of soluble salts from the APCr
331 above 800°C.

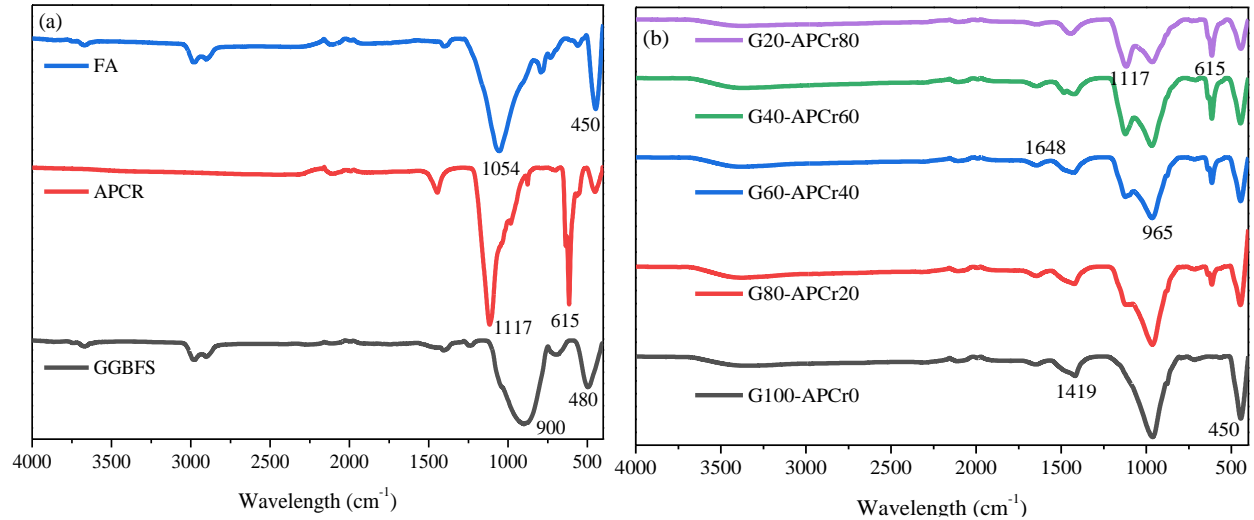
332 *3.4 XRD analysis of APCr solidified samples*

333 The phases in the solidified samples were characterized through the XRD analysis and results are
334 shown in Fig. S5. The XRD pattern of control samples (G100-APCr0) without APCr showed only
335 two types of distinct peaks namely quartz and calcite/C-S-H. The previous studies have also
336 reported the presence of hump peak around 30° associated with C-(A)-S-H (Ababneh et al., 2019;
337 Ahmad et al., 2021a), which could be formed due to the reaction of Ca (from CaO in GGBS) with
338 the $[\text{SiO}_4]^{4-}$ either from raw materials or alkaline activator. Comparing the XRD patterns of
339 precursor FA with control samples reveals that aluminosilicate present in raw FA were absent in
340 the control sample, this indicates the dissolution of these aluminosilicate phases to form the
341 geopolymer gel (N-A-S-H) in the presence of an alkaline activator. Partial dissolution of quartz
342 and mullite in fly ash has been also reported in the literature as it further helps to improve the
343 strength by formation of amorphous silica gel (Li et al., 2021). However, the geopolymer gel was
344 hard to identify from the XRD pattern due to its amorphous nature and coexistence with the other
345 crystalline phases (Xu et al., 2014). After FA was replaced with APCr into the geopolymer binder,
346 new peaks appeared in the APCr solidified samples. Quartz peaks were gradually reduced and

347 eventually disappeared when amount of APCr was increased. For the 2θ value around 29° , calcite
348 peak is present in the reference mixtures (0% APCr), and mixtures containing 20%, 40% and 80%
349 APCr. Another calcite peak is identified in raw APCr around $55.6^\circ 2\theta$. However, there is no calcite
350 peak observed in the reference mixture (Figure S5) at the 2θ value of 55.6° . This calcite peak starts
351 appearing as the percentage of APCr was gradually increased. This finding also corroborates the
352 results of TGA analysis where an increase in loss mass was observed when the amount of APCr
353 was increased (associated with the decomposition of calcite). New peaks in solidified samples
354 originated from the APCr and became sharper with the increase in quantity of APCr. Peaks of
355 thenardite (Na_2SO_4 peak around 18° and 34°), halite (NaCl peak around 32°) declined significantly
356 as compared to peaks of raw APCr, which is linked to their partial dissolution into the water.
357 Although, the use of Na_2SO_4 as a weak alkaline activator has been also reported in the literature
358 (Alrefaei et al., 2020; Yang et al., 2021) however, Na_2SO_4 from APCr did not contribute to the
359 improvement in strength in this study. This conclusion can be drawn from the linear decrease in
360 strength of solidified samples with the increase in percentage of APCr.

361 3.5 FTIR analysis of APCr solidified samples

362 FTIR analysis results for the raw materials and APCr solidified samples are present in Fig. 4. Both
363 FA and GGBS contain two bands associated with the tetrahedral AlO_4 and SiO_4 functional groups.
364 A smaller wavelength band occurs around 450 cm^{-1} in FA and 480 cm^{-1} GGBS. The large
365 wavelength bond occurs at 1054 cm^{-1} in FA and 900 cm^{-1} in GGBS. The shift of this band is linked
366 to the higher degree of cross-linking of amorphous phase present in GGBS due to its high calcium
367 amount (Pan et al., 2018). Two major bands in the APCr were observed at 1117 cm^{-1} and 615 cm^{-1}
368 ¹ which were associated with asymmetrical stretching and asymmetrical bending vibration of SO_4
369 group respectively (Das et al., 2014).



370

371 Fig. 4. FTIR analysis for (a) raw materials; (b) control sample and APCr solidified samples

372 For the control and APCr solidified samples, the presence of asymmetric stretching peak around

373 965 cm^{-1} was associated with the presence of geopolymer products (Ahmad et al., 2021b). This

374 peak was due to presence of Si-O-T (T: tetrahedral Al or Si) and can be also viewed in the raw

375 material (FA), however shifted to the lower wavelength for the geopolymer samples. This shift of

376 T-O bands reveals that Al-Si or Si phases from the precursors had taken part in hydration reaction

377 and produced geopolymer gel. The degree of geopolymerization in different samples can be

378 accessed by the range of Si-O-T stretching peak (Nadeem et al., 2021) which is different for all

379 samples as observed in Fig. 4b. Control samples had the highest degree of geopolymerization and

380 the broader Si-O-T stretching peak. The range of stretching peak was gradually reduced with the

381 increase of APCr amount, which confirms that the degree of geopolymerization was reduced.

382 However, the presence of Si-O-T affirms the formation of geopolymer in all the samples, which

383 played an important role in the solidification of APCr waste. Similar phenomenon was observed

384 at the geopolymer band around 450 cm^{-1} . The absorption band at 1419 cm^{-1} corresponds to anti-

385 symmetric stretching vibration of O-C-O bond of CO_3^{2-} . This band reflects the existence of

386 carbonate as also spotted from the XRD analysis. A broad absorption band between the 3700 cm^{-1}

387 ¹ and 3200 cm⁻¹ is due to -OH stretching vibration whereas an absorption band around 1648 cm⁻¹
388 is associated with the -OH bending vibration. For the samples containing APCr, peaks around 1117
389 cm⁻¹ and 615 cm⁻¹ started emerging and became more prominent when amount of APCr was
390 increased from 20% to 80%.

391 3.6 SEM-EDS analysis of APCr and solidified samples

392 Distribution of different elements present in the APCr is captured by the SEM-EDS analysis and
393 the elements spectrum and mapping are shown in Fig. S6. Two major elements (Na and S) were
394 observed to be present in raw APCr powder and uniformly distributed. Other elements (Fe, Ca and
395 Cl) were present in minor quantities. Mapping results of APCr are in line with the XRF, XRD, and
396 FTIR results of raw-APCr as major elements and phases from these analysis techniques were also
397 composed of Na and S.

398 Back-scattered electron images of control and APCr solidified samples analyzed through the SEM-
399 EDS are provided in Fig. 5. The elemental compositions and spectrums for different points and
400 areas are given in Table S1 and Fig. S7 (see supplementary data). For the control sample, presence
401 of unreacted FA and reacted FA can be observed from Fig. 5a. Reaction of FA proves its
402 dissolution in the alkaline environment which leads to the formation of geopolymer gel. While
403 unreacted FA acts as a filler. Some of the unreacted GGBS particles are also observed. Formation
404 of geopolymer gel was observed in area 1. The elemental composition showed that major elements
405 present in area 1 are Ca, Na, Al, and Si. This indicates the co-existence of geopolymer gels (N-A-
406 S-H and C-(A)-S-H). The co-existence of geopolymer gel (N-A-S-H) along with C-S-H gel has
407 been reported in the metakaolin-GGBS based geopolymer previously (Yip et al., 2005). Sample
408 containing 20% APCr (G80-APCr20) was analyzed for the given potential areas as marked in Fig.
409 5b. The presence of hematite, unreacted GGBS, mullite, and carbon was observed at the areas 2,

410 3, 5, and 6 respectively, while major elements at area 4 are Ca, Na, Al and Si which suggests the
411 formation of geopolymer gel as also observed in control sample. Area 7 in G60-APCr40 samples
412 (Fig. 5c) contains phases consisting of Ca-Al-Mg with ratio of 1:0.8:1.1. This phase seems to be
413 weaker in strength and could be associated with elements from unreacted precursor. The formation
414 of Fe-rich C-A-S-H and Fe-rich (N, C)-A-S-H gels are observed at area 8 and area 9 respectively.
415 The elemental composition of FA is also obtained at area 10 to compare it with the geopolymer
416 gels (Khedmati et al., 2018). The Si/Al and Na/Al ratios of FA are 1.25 and 0.03, respectively. In
417 the geopolymer gel N-A-S-H, the Si/Al ratios at area 1, 4, 9 are 1.05, 1.22, and 0.64 while the
418 Na/Al ratios are 0.29, 0.37 and 0.71, respectively. This shows that some of silica from the FA and
419 GGBS precursors (area 3) has reacted with the alkaline activator. As percentage of APCr is further
420 increased from 40% to 80% (Fig. 5d and Fig. 5e), presence of major elements (Na and S) from the
421 APCr has increased in the solidified samples. The elemental composition at areas 11 and 13
422 indicates the combined presence of geopolymer gels and higher amounts of S, Na, P elements.
423 This proves the interlocking effect of geopolymer gel to stabilize the APCr solidified waste even
424 at a higher dosage. As APCr contained higher amount of Na element in its raw form, ratio of Na/Al
425 was also increased from 0.29 to 1.39 gradually with the increase in percentage of APCr from 0%
426 to 80% (area 1, 4, 9, 11, 13). The higher Si/Al ratio at area 11 and 13 is linked due to replacement
427 of FA with the APCr as FA is major source of Al (while one source of silica (SS) was kept
428 constant).

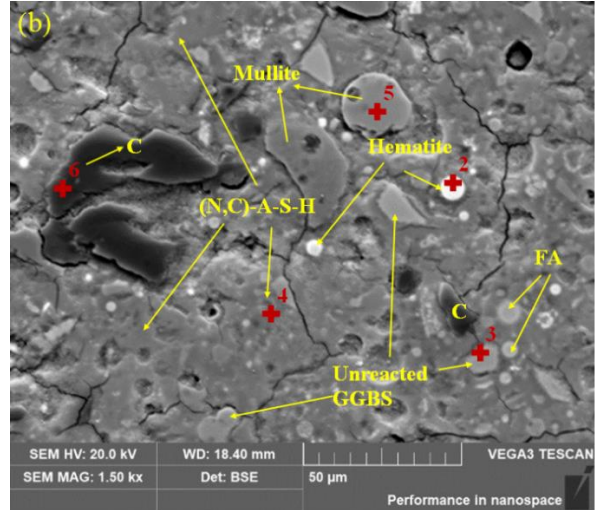
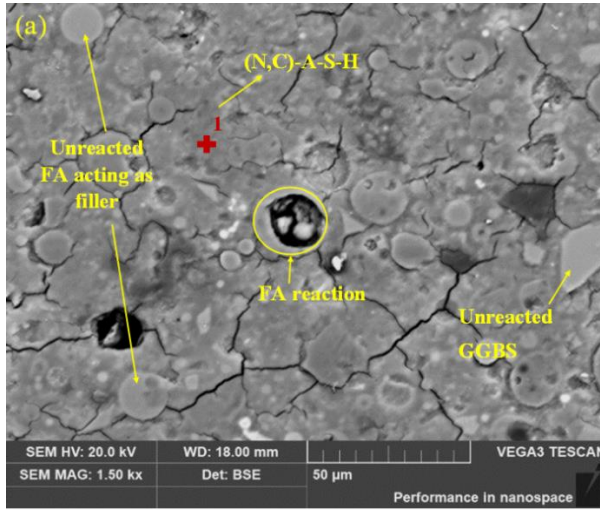
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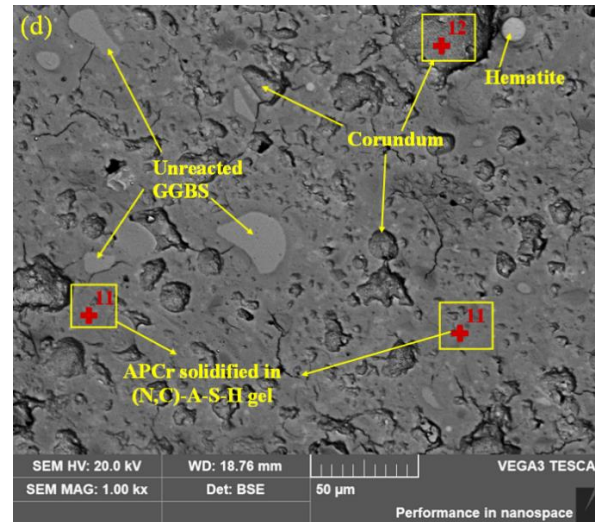
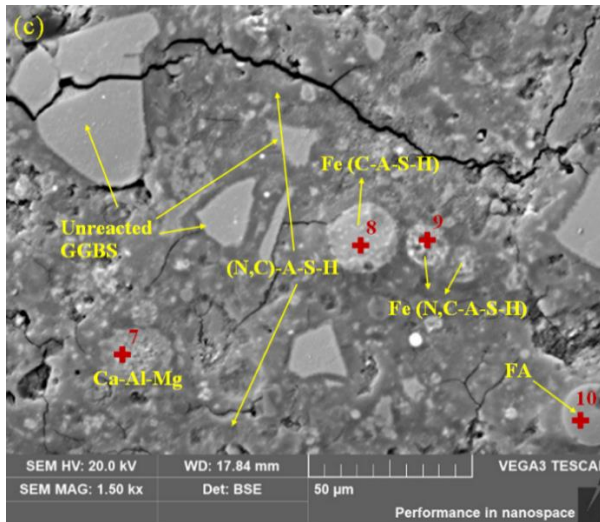
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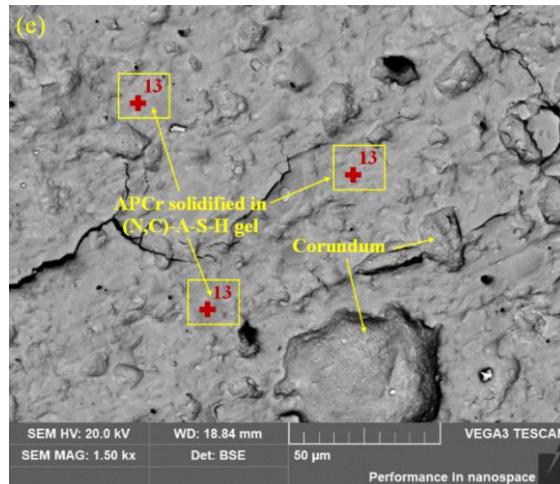
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437 Fig. 5. SEM-EDS analysis of control and APCr solidified samples (a) G100-APCr0; (b)
 438 G20-APCr80; (c) G40-APCr60; (d) G60-APCr40 and (e) G80-APCr20 (Elemental compositions
 439 and EDS-spectrum of points/areas for Fig. 5 are shown in supplementary data information (Table
 440 S1 and Fig. S7)

441 3.7 EDS mapping of APCr solidified samples

442 Unreacted phases and homogeneity of geopolymer gel (distribution of Ca, Na, Al and Si element)
 443 were further investigated through the SEM-EDS mapping technique. The results for reference
 444 sample (G100-APCr0, 0% APCr) are shown in Fig. 6 while the mapping images of solidified
 445 samples containing 20%, 40%, 60% and 80% APCr are also shown in Fig. S8, S19, S10, and S11
 446 respectively. The elemental composition and EDS spectrum of mapping for the control sample and
 447 all APCr solidified samples are given in Table S2 and Fig. S12. The major elements present in the
 448 G100-APCr0 are Ca (10.4%), Na (7.5%), Al (23.7%) and Si (48.9%) making almost 90% of the
 449 mapping area. These four elements are uniformly distributed at similar locations and potentially
 450 form the geopolymer gel (N, Ca-Al-S-H). Some unreacted Ca was also observed in the mapping
 451 indicating the lower amount of formation of Ca-based hydration products (C-S-H/C-A-S-H) as
 452 compared to geopolymer gel (N-A-S-H). The Si/Al and Na/Al ratios in control sample are 2.07
 453 and 0.32, respectively. Other unreacted phases consisted of Fe and Mg elements. Solidified
 454 samples containing 40% and 60% APCr closely followed the similar trend with the slightly

455 reduced Si/Al ratio of 1.32 and 1.67 as compared to control sample. This was due to replacement
456 of APCr with FA which is major source of Si. However, the Na/Al ratio was increased to 0.39 and
457 0.67. The increase in Na/Al ratio was due to the higher content of alkaline content (Na_2O) in the
458 APCr. This Na/Al ratio was further increased to 0.76 in G20-APCr80 sample. Solidified sample
459 containing 80% APCr consists of Ca (12.2%), Na (18.3%), Al (24%) and Si (19.5%). These four
460 elements make 74% of the mapping area. The presence of unreacted Ca was reduced whereas the
461 amount of unreacted Al was increased in mapping area (Fig. S11.). This could indicate the lower
462 degree of geopolymerization in the solidified samples containing a higher amount of APCr (80%)
463 due to absence of aluminosilicate source (Fly ash). The higher content of Na appeared in the APCr,
464 however, its contribution to improve the strength and formation of geopolymer product was
465 negligible as compared to sodium metasilicate (i.e., the main alkaline activator). The elemental
466 composition of G20-APCr80 sample suggests the presence of both C-(A)-S-H gel and N-A-S-H
467 gel. The higher amount of S (11.8%) and P (5.7%) elements was observed and thoroughly
468 distributed in the APCr solidified sample showing that geopolymer even at a lower content could
469 still potentially act as a strong stabilizing agent.

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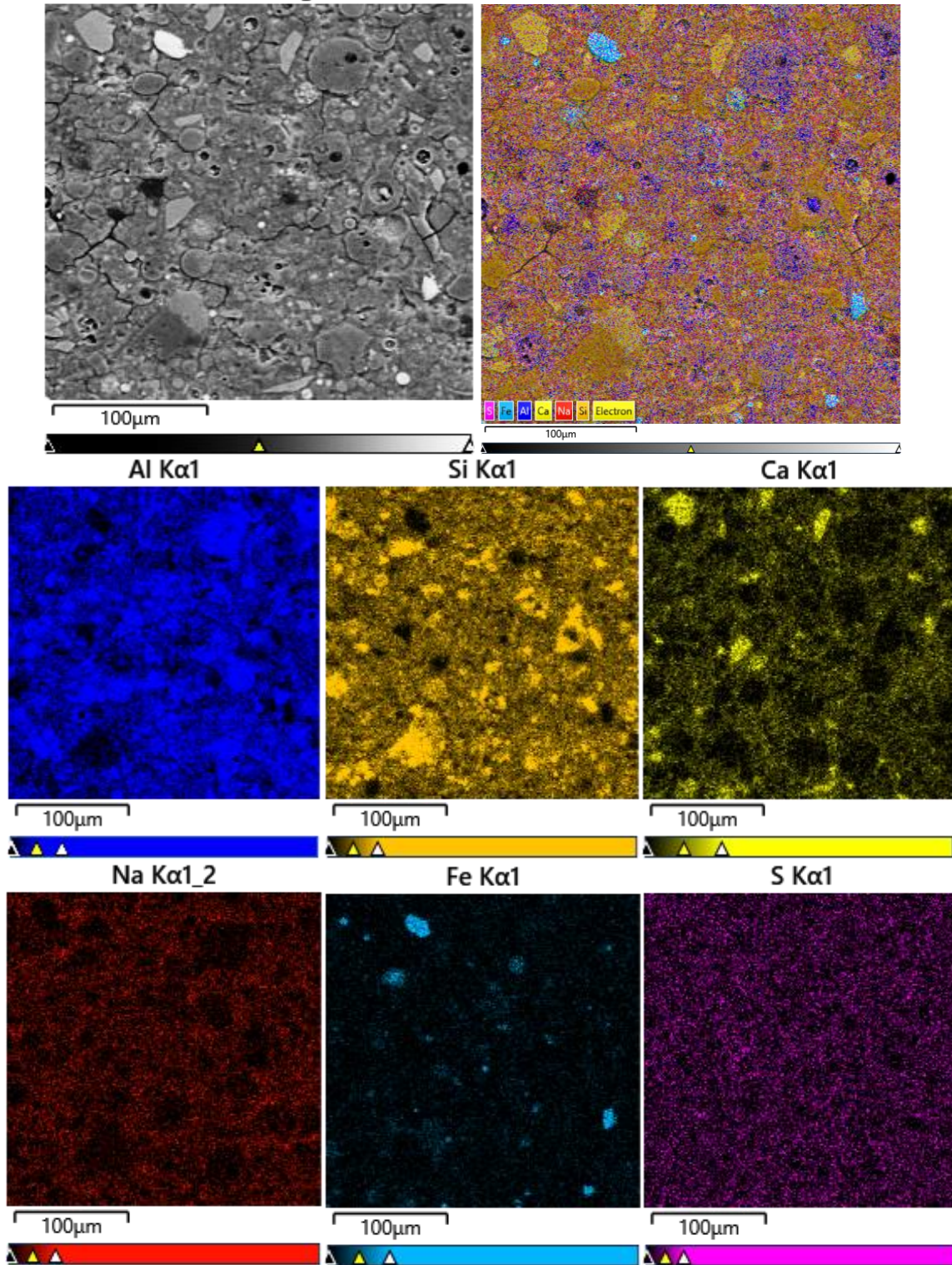
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Fig. 6. SEM-EDS Mapping of control sample without APCr (G100-APCr0)

478

479 The solidified samples containing 80% APCr showed the crack of width of approximately 10 μm .
480 A further investigation was conducted at the interface of crack through the EDS line scanning as
481 shown in Fig. S13. It can be observed that the nature of hydration products are similar on both
482 sides of the crack. Moreover, the formation of low strength geopolymer gel was suggested by EDS
483 spectrum on the both side of crack. Hence, the appearance of such crack could be attributed to (a)
484 crushing of sample during the compressive strength test, (b) force applied on the sample during
485 the sample preparing or polishing process or (c) shrinkage of paste during drying process.

486 **4 Conclusion**

487 In this study, heavy metals in the APCr waste were stabilized/solidified using geopolymer
488 technology. The percentage of APCr waste was increased from 20% to 80% by total weight of
489 solids in the solidified samples. Characterization of solidified samples was performed using TCLP,
490 UCS, XRD, TGA, FTIR, and SEM-EDS analyses. It was found that geopolymer based materials
491 could help to solidify the larger amount of APCr (up to 80%) with high efficiency. Raw-APCr
492 contained high concentrations of PTEs (e.g., Zn, Cu, Ba, Cr, and As). TCLP results showed the
493 geopolymer was able to suppress the activity of these PTEs through S/S technique and the leaching
494 values of all solidified samples fell well below the limits. An increase in the amount of APCr waste
495 to replace FA had a negative influence on UCS. However, GT helped to retain 18.4 MPa even at
496 the higher content of 80% of APCr by total weight in solidified samples, making them still
497 applicable in engineering applications as low-medium strength construction materials. XRD
498 results showed the dissolution of aluminosilicate phases of fly ash that formed the geopolymer gel,
499 which helped to efficiently stabilize the PTEs. Thermal analysis results showed that APCr
500 solidified samples started to lose mass significantly around 800°C due to melting and evaporation
501 of soluble salts. The presence of geopolymer gel was observed in the control sample and all APCr

502 solidified samples by FTIR analysis. SEM-EDS analysis revealed the co-existence of N-A-S-H
503 and C-(A)-S-H gels in the control samples. EDS analysis of solidified samples also indicated the
504 formation of Fe-rich C-A-S-H and Fe-rich (N, C)-A-S-H gels at 20% and 40% of APCr content.
505 A clear presence of unreacted S, P and Na elements was seen at the higher content of APCr, which
506 were surrounded by the geopolymer products. Further research on the utilization of APCr as a
507 replacement of filler materials or a weak alkaline activator (owing to its higher Na content) should
508 be conducted in the future.

509 **Author contribution statement**

510 **MR Ahmad:** Conceptualization, Methodology, Investigation, Formal analysis, Writing - Original
511 Draft. **JC Lao:** Methodology, Investigation, Writing. **JG DAI:** Conceptualization, Funding
512 Acquisition, Supervision, Writing - Review & Editing. **DX Xuan:** Writing - Review & Editing **CS**
513 **Poon:** Writing - Review & Editing

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