1 Performance Evaluation of a Novel Photovoltaic-electrochemical Hybrid System

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9 Abstract: To harvest the relatively high wavelength sunlight, a novel hybrid system coupling a thermally regenerative electrochemical cycle to a dye-sensitized solar cell is proposed. Efficiencies 10 11 and power outputs of dye-sensitized solar cell and thermally regenerative electrochemical cycle are calculated, and the mathematical relationship between the electric current of thermally regenerative 12 13 electrochemical cycle and the working current density of dye-sensitized solar cell is deduced. The 14 power output and efficiency of the hybrid system are also derived considering multiple irreversible 15 losses. The feasibility and effectiveness of the proposed hybrid system will be assessed by 16 comparing the performances between the hybrid system and the single dye-sensitized solar cell. Numerical calculations show that the maximum efficiency and power density of the hybrid system 17 allow 32.04% and 32.18% greater than that of the single dye-sensitized solar cell, respectively. 18 19 Comprehensive parametric studies are undertaken to examine the dependences of the hybrid system 20 performance on some operating conditions and microstructure parameters, including electrode 21 porosity, photoelectron absorption coefficient, Schottky barrier, film thickness and internal resistance

- of thermally regenerative electrochemical cycle. The derived results may offer new insights intodesign and optimization of such an actual hybrid system.
- 24
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30 **1. Introduction**

31 Energy shortage and environmental pollution are two major challenges of the 21st century. So 32 far, fossil fuels are still playing the most important role in the global energy structure. With the growth of the world population and economic development, it is a global consensus to vigorously 33 34 promote the energy revolution by replacing fossil fuels by renewable energies, such as wind energy, 35 biomass energy, geothermal energy, wave energy and solar energy [1]. Herein, solar energy is the 36 most potential alternative to meet the increasing clean energy demand because it is abundant, free 37 and clean. Solar energy can be actively used through photovoltaic cell, solar thermal system and 38 concentrated solar power system [2]. Photovoltaic cells enable direct conversion of the solar energy into electricity without any intermediate energy conversion processes. Photovoltaic cells have made 39 a great progress in the past 30 years and can be mainly classified into organic, inorganic and hybrid 40 solar cells [3]. 41

42 As one kind of organic solar cells, dye-sensitized solar cells (DSSCs) convert visible light into electricity using dye molecules for light absorption. Then, the excited dye molecules inject electrons 43 44 into the titanium dioxide (TiO₂), which subsequently conducts the electrons to the current collector. 45 DSSCs have unique advantages such as abundant raw materials, low cost, simple technology and 46 achievable large-scale industrialized production [4]. In addition, all raw materials and production process are non-toxic and non-polluting, and a part of the materials can be fully recycled. For DSSCs, 47 considerable studies have been carried out both theoretically and experimentally [5] on some aspects, 48 49 including sensitizer development [6], photo electrode fabrication [7], prototype design [8] and 50 performance modeling [9].

51 However, the present energy conversion efficiency of DSSCs is not adequately high, and up to 52 80% of the absorbed energy is wasted into the surroundings due to the interfacial transference, 53 relaxation and recombination losses. The energy conversion efficiency can be also enhanced if the 54 relatively high wavelength sunlight is collected and recovered for combined cooling, heating and 55 power (CCHP) applications. For example, Su et al. [10] proposed a hybrid device consisting of 56 DSSC and semiconductor thermoelectric refrigerator to harvest solar energy for cooling. Pounraj et 57 al. [11] experimentally studied a novel Peltier based hybrid photovoltaic/thermal (PV/T) active solar 58 still to increase the freshwater yield and to improve the efficiency of the solar PV panel. Prince 59 Winston et al. [12] further investigated a hybrid PV/T active still incorporating a Nickel-Chromium (NiCr) heater to increase the production of distillate water and enhance the PV efficiency. Owing to 60 the relatively low operating temperature, most of the available researches have focused on 61 photovoltaic-thermoelectric (PVTE) hybrid systems [13], which apply thermoelectric generators 62 63 (TEGs) to reuse the waste heat from photovoltaic cells for extra electrical power generation [14]. For instance, Babu et al. [15] presented a novel PVTE system and found that overall efficiency of 64 65 PV-TEG system was 6% higher than that of the sole PV. Wang et al. [16] created a PVTE hybrid 66 device and found that the overall energy conversion efficiency of the hybrid device was 13% greater 67 than that of the single DSSC. Xu et al. [17] demonstrated a PVTE hybrid device that integrated the hole-conductor-free perovskite solar cell based on a TiO2/Zirconium dioxide (ZrO2)/carbon structure 68 69 PV and a TEG. Under the irradiation of 100 mW cm⁻², the optimized hybrid device achieved a 70 maximum efficiency of 20.3% and an open-circuit voltage (OCV) of 1.29V. Kil et al. [18] fabricated 71 a concentrating PVTE hybrid device consisting of a single-junction GaAs-based solar cell and a TEG, 72 and their hybrid generator gave rise to the conversion efficiency by about 3% compared to the single 73 concentrating photovoltaic cell. Yin et al. [19] studied the coupling selection principle and optimal 74 design methods for a concentrating PVTE hybrid system. Yin et al. [20] also evaluated the feasibility 75 of a photovoltaic-thermoelectric-thermal cogeneration system and found that the energy efficiency 76 and exergy efficiency were, respectively, 43.06% and 31.43%. Mahmoudinezhad et al. [21] 77 investigated the transient response of a hybrid system comprised of a concentrated PV cell and a 78 TEG. Park et al. [22] tried to fully use the spectrum solar energy via coupling TEGs to harness the 79 waste heat from thermalization and transmission losses from PV cell. Their method found that the 80 overall conversion efficiency was 30% under a 15°C temperature gradient. Bjørk et al. [23] compared the maximum efficiencies between a hybrid system that the TEG was directly mounted on 81 82 the PV back and a tandem system that the short and long wavelength sunlight were sent to PV and 83 TEG, respectively. The maximum increases in efficiency for the combined case and tandem case 84 were found to be 4.5% and 1.8%, respectively.

As an alternative heat-to-electricity device, a thermally regenerative electrochemical cycle 85 86 (TERC) constructs a thermodynamic cycle using the temperature-dependence of electrochemical cell 87 voltages [24]. An electrochemical cell is charged at a temperature with lower voltage and then 88 discharged at a different temperature with higher cell voltage, thereby converting heat into electricity by varying temperature [25]. TREC is regarded as a promising low-grade heat harvesting technology 89 90 because of its high efficiency [26], high temperature coefficient and low cost [27]. Recently, TRECs 91 have attracted much attention in the scientific community [28]. Long et al. have evaluated the 92 performances of TREC using objective functions such as power output and efficiency [29],

93 ecological [30] as well as maximum power output and exergy efficiency [31]. Wang et al. [32] optimized a low-grade heat driven TREC with continuous power output using the maximum 94 95 efficiency-power product criterion. Guo et al. [33] studied the performance of an upgraded TREC 96 considering the external heat leakage, heat transfer irreversibility and non-ideal regeneration losses. 97 Lee et al. [25] demonstrated an efficient TREC based on the thermogalvanic effect for harvesting 98 low-grade heat energy. Yang et al. [34] fabricated a novel charging-free TREC, and the 99 heat-to-electricity conversion efficiency of 2.0 % could be reached for the TREC operating between 20 °C and 60 °C. Yang et al. [35] also reported a membrane-free TREC with a nickel 100 101 hexacyanoferrate (NiHCF) cathode and a silver/silver chloride anode. In addition to the studies on TREC itself, some studies have used TREC to harness the waste heat from low-temperature power 102 103 sources. For example, Long et al. [36] and Zhang et al. [37] have integrated the TERC with proton 104 exchange membrane fuel cell (PEMFC) and alkaline fuel cell (AFC) for waste heat harvesting, and 105 found that the performances of the hybrid systems are better than that of the single fuel cells [38]. 106 However, seldom research has been reported on using TREC to harvest the waste heat from 107 photovoltaic cells.

To fill the research gap, a novel photovoltaic-electrochemical hybrid system is proposed by integrating a DSSC and a TREC, where the TREC harvests the waste heat from the DSSC for further electric power generation. Considering various irreversible losses within DSSC and TREC as well as between them, the performances of DSSC, TREC and proposed system will be theoretically derived and studied. The electric current relationship between DSSC and TREC will be derived. The feasibility and effectiveness of the proposed approach will be illustrated by comparing the performances between the proposed system and the single DSSC. Moreover, the effects of the film thickness of electrode material, porosity, Schottky barrier, photoelectron absorption coefficient and internal resistance of TREC on the proposed system performance will be discussed in detail.

117 **2. System description**

As depicted in Fig. 1, the proposed hybrid system is composed of a transparent DSSC, a solar 118 119 selective absorber (SSA) [39] and a TREC, where the SSA is placed between the DSSC and the 120 TREC, and the DSSC and the TREC are electrically connected in series. The DSSC converts the 121 short wavelength sunlight (smaller than 920 nm) into electricity [40], while the relatively high 122 wavelength sunlight not utilized by the DSSC (larger than 920 nm) is absorbed by the SSA and then 123 converted into heat energy [41]. The SSA is a carbon nanotube-based tandem one, which acts as 124 near-perfect blackbody absorber with the thickness larger than 10 µm [42]. The carbon nanotubes 125 have good optical properties (i.e., absorptance and emittance close to 1), high thermal stability (i.e., 126 they exhibit thermal stability at 580 °C in vacuum for 30 days) and high thermal conductivity (>3000 127 W m⁻¹ K⁻¹) [42], which enable rapid heat transfer from the SSA (< 60 °C) to the TREC for electricity 128 generation. Therefore, the performance of the proposed hybrid system is expected to be better than 129 that of a single DSSC. In Fig. 1, Error! Reference source not found. is the power output of DSSC; P_{TREC} is the power output of TREC; Q_{2n} is the heat flow from SSA to TREC; Q_{1n} is the heat flow 130 131 lost to the environment from TREC; Q_L is the rate of heat leakage from the DSSC into the 132 environment; T is the DSSC operating temperature; T_e is the environment temperature; T_1 and T_2 are, respectively, the heat-sink and heat-source temperatures of the TREC. 133

134 To simplify the subsequent analyses with accuracy, the following assumptions are adopted:

- 135 (1) All inlet wavelength sunlight is fully utilized [43];
- 136 (2) Zero energy losses from the SSA to the environment due to the small temperature difference
- 137 between the SSA and the environment [39];
- 138 (3) Transport of carriers in semiconductor is governed by diffusion motion due to the negligibly
- 139 small drift motion [10];
- 140 (4) Diffusion length of carriers is constant since the carriers have an average lifetime [10];
- 141 (5) The heat-source temperature of TREC is equal to the operating temperature of DSSC because the
- 142 SSA has high thermal conductivity [44];
- 143 (6) The heat-sink temperature of TREC is equal to the ambient temperature [44];
- 144 (7) The charging time is equal to the discharging time for the TREC [32].
- 145 It should be pointed out that the above assumptions are widely used in the current literatures, and the
- 146 influences of these assumptions on the hybrid system performance are negligibly small.
- 147 **2.1. Dye-sensitized solar cell**

148 The DSSC main structure is based on a semiconductor formed between an electrolyte and a 149 photo-sensitized anode. The detailed working processes of DSSC can be briefly described as follows 150 [45]. When sunlight irradiates the porous nano- TiO_2 thin film, the dye molecules absorb photons and 151 then generate electrons. The generated electrons are injected to TiO₂ and transported from the 152 nano-TiO₂ thin film to the external electrical load, as described by Eq. (1) \sim Eq. (3) [46]. 153 Afterwards, electrons move through an external electric circuit to the counter electrode from which 154 electrons are transferred to the redox electrolyte. The oxidized dye is restored by extra electron to 155 regenerate Iodide, and the Iodide is going to be oxidized into tri-iodide, as described in Eq. (4) [46].

In the end, the tri-iodide keeps travelling through the electrolyte until it arrives the counter electrodeto get the lost electron, as described by Eq. (5) [46].

158
$$\Theta + \text{photon} \rightarrow \Theta^*$$
 (1)

159
$$\Theta^* + \operatorname{TiO}_2 \to e^-(\operatorname{TiO}_2) + \Theta^+$$
 (2)

160
$$e^{-}(TiO_2) \rightarrow e^{-}(counter electrode) + electric energy$$

$$161 \qquad \Theta^{+} + \frac{3}{2}I^{-} \rightarrow \Theta + \frac{1}{2}I_{3}^{-}$$

$$\tag{4}$$

(3)

162
$$\frac{1}{2}I_3^- + e^-(\text{counterelectrode}) \rightarrow \frac{3}{2}I^-$$
 (5)

Since electron transfer in the nano- TiO_2 thin films is mainly governed by diffusion, one can effectively simulate the performance of DSSC by establishing electron diffusion model [47]. Under steady-state conditions, the generation, transport [48] and recombination [49] of electrons in DSSC can be described by

167
$$D\frac{\partial^2 n(x)}{\partial x^2} - \frac{n(x) - n_0}{\tau^*} + \Phi \alpha e^{-\alpha x} = 0$$
(6)

168 where x is the coordinate starting from the TiO₂/transparent conducting oxide (TCO) interface; 169 n(x) is the excessive electron concentration at x; n_0 is the electron concentration under a dark 170 condition; τ^* is the electron lifetime; Φ is the light intensity; α is the light absorption 171 coefficient of the electrode; *D* is the electron diffusion coefficient.

172 With the boundary conditions (i.e., Eq. (7) and Eq. (8)), the short-circuit current density J_{sc} 173 [50] can be obtained by solving Eq. (6), as shown by Eq. (9) [51]

174
$$n(0) = n$$
 (7)

$$175 \quad \left(\frac{dn}{dx}\right)_{x=d} = 0 \tag{8}$$

176
$$J_{sc} = \frac{q\Phi_0 L\alpha \left[-L\alpha \cosh\left(\frac{d}{L}\right) + \sin\left(\frac{d}{L}\right) + L\alpha \exp\left(-d\alpha\right) \right]}{\left(1 - L^2 \alpha^2\right) \cosh\left(\frac{d}{L}\right)}$$
(9)

177 where

$$178 \qquad L = \sqrt{D\tau^*} \tag{10}$$

179 is the electron diffusion length; Φ_0 is the light intensity at 1 sun condition; q is the charge of an 180 electron; d is the thickness of the thin film.

181 Neglecting the losses at the interface between electrode and electrolyte, the photovoltage of the182 DSSC is given by [50]

183
$$V = V_0 - V_1$$
 (11)

184 where V_0 is the difference of the redox potential between the electrolyte and the TiO₂ Fermi level;

185 V_1 is the voltage loss at the TiO₂/TCO interface.

186 The potential difference V_0 can be expressed as

187
$$V_0 = \frac{kTm^{\#}}{q} \ln \left[\frac{(J_{sc} - J)L\cosh\left(\frac{d}{L}\right)}{qDn_0\sinh\left(\frac{d}{L}\right)} + 1 \right]$$
(12)

188 where $m^{\#}$ is the ideality factor, k is the Boltzmann constant.

189 Referring to the thermionic-emission theory, the voltage loss at the TiO_2/TCO interface can be

190 calculated by

191
$$J = A^* T^2 \exp\left(\frac{-q\Phi_b}{kT}\right) \left[\exp\left(\frac{qV_1}{kT}\right) - 1\right]$$
(13)

192 where

193
$$A^* = \frac{4\pi m^* q k^2}{h^3}$$
(14)

is the Richardson constant of TiO₂; h is the Planck constant; m^* is equal to 5.6 times the electron 194 mass for TiO₂; Φ_b is the Schottky barrier height. 195

196 Rearranging Eq. (13), the voltage loss can be further expressed as

٦

197
$$V_{1} = \frac{kT}{q} \ln \left[1 + \frac{J}{A^{*}T^{2} \exp\left(-\frac{q\Phi_{b}}{kT}\right)} \right]$$
(15)

198 Substituting Eq. (12) and Eq. (15) into Eq. (11), the photovoltage of DSSC can be calculated as

199
$$V = \frac{kTm^{\#}}{q} \ln\left[\frac{(J_{sc} - J)L\cosh\left(\frac{d}{L}\right)}{qDn_0\sinh\left(\frac{d}{L}\right)} + 1\right] + \frac{kT}{q} \ln\left[1 + \frac{J}{A^*T^2\exp\left(-\frac{q\Phi_b}{kT}\right)}\right]$$
(16)

200 With the help of Eq. (13) and Eq. (16), the power output and efficiency of the DSSC can be, 201 respectively, given by

$$202 P_D = VJA_D (17)$$

$$203 \qquad \eta_{\rm D} = \frac{P_D}{GA_D} = \frac{VJ}{G} \tag{18}$$

where G is the solar irradiation under 1 sun condition; A_D is the effective area of the DSSC. 204

205 Since the internal recombination reaction and diffusion process [52] of DSSC are very sensitive 206 to temperature changes [53], the influence of operating temperature on the DSSC performance must be taken into account. Considering the effects of operating temperature, the deviations of the power 207 208 output and efficiency can be, respectively, given by

$$209 \qquad P_{dev} = \lambda \left(T - T_{ref} \right) \tag{19}$$

$$210 \qquad \eta_{dev} = \beta \left(T - T_{ref} \right) \tag{20}$$

where λ and β are, respectively, the decreasing rates of the power output and efficiency with respect to the DSSC operating temperature; and β is interrelated with λ . When λ is 0.00506 mW K⁻¹, the theoretical results of power output density fit well with the experimental results in Ref. [54]. Similarly, according to Eq. (15) ~ Eq. (18) and Ref. [55], β can be calculated as 0.0114 K⁻¹. Therefore, Eq. (17) and Eq. (18) should be, respectively, modified into

$$216 \qquad P_D = VJA_D - \lambda \left(T - T_{ref}\right) \tag{21}$$

217
$$\eta_D = \frac{VJ}{G} \left[1 - \beta \left(T - T_{ref} \right) \right]$$
(22)

218 **2.2. Thermally regenerative electrochemical cycle**

219 TREC within the hybrid system consists of *n* TREC elements, which can generate a continuous power output in the temperature range of 293 ~ 333 K [34]. The positive electrode of a TREC 220 221 element is made of solid copper hexacyanoferrate (CuHCF) immersed in sodium nitrate (NaNO₃) 222 aqueous solution and the negative electrode is made of metal copper (Cu) immersed in 3 M copper nitrate (Cu NO_{3} ₂) aqueous solution. The positive and negative electrodes are separated by an 223 anion membrane [25]. The relevant electrochemical reactions at the cathode and anode are, 224 $Na_{0.71}Cu[Fe^{III}(CN)_{6}]_{0.72} + a(Na^{+} + e^{-}) = Na_{0.71-a}Cu[Fe^{III}(CN)_{6}]_{0.72-a} + [Fe^{II}(CN)_{6}]_{0.72+a}$ 225 respectively, and $Cu^{2+} + 2e^{-} = Cu$ [32]. The TREC elements can be divided into three parts [29]: one part 226 227 functions as hot cells that tightly contact with the hot reservoir, i.e., the SSA; another part works as cold cells that tightly contact with the cold reservoir, i.e., the environment; and the rest part conducts 228 229 the regenerative processes.

In the charging progress, the produced heat is totally released to the hot reservoir. In the discharging process, the produced heat is totally released to the cold reservoir. The heat absorbed from the hot reservoir at temperature T_2 and the heat released to the cold reservoir at temperature

233 T_1 during the isothermal processes can be, respectively, expressed as

$$234 \qquad q_2^* = \alpha_c T_2 C_q \tag{23}$$

$$235 \qquad q_1^* = \alpha_c T_1 C_q \tag{24}$$

236 where α_c is the temperature coefficient; C_q is the charge capacity.

The heat released to the hot reservoir and the heat dissipated into the cold reservoir during thetwo isothermal processes can be, respectively, expressed as

239
$$q_{J1}^* = I_{ch}^2 R t_{ch}$$
 (25)

240
$$q_{J2}^* = I_{dis}^2 R t_{dis}$$
 (26)

where I_{ch} and I_{dis} are the electric currents of the charging process and discharging process, respectively; *R* is the electric resistance; t_{ch} is the time of charging process; t_{dis} is the time of discharging process.

244 The regenerative losses in a cycle is given by [30]

245
$$q_{RL}^* = (1 - \eta_{RL}) C_p (T_2 - T_1)$$
 (27)

246 where η_{RL} is the regenerative efficiency; C_p is the heat capacity.

Thus, the heat absorbed from the hot reservoir [32] and the heat released to the cold reservoir during the cycle time for a TREC element can be, respectively, expressed as

249
$$Q_2^* = \alpha_c T_2 C_q - I_{ch}^2 R t_{ch} + (1 - \eta_{RL}) C_p (T_2 - T_1)$$
(28)

250
$$Q_2^* = \alpha_c T_1 C_q - I_{dis}^2 R t_{dis} + (1 - \eta_{RL}) C_p (T_2 - T_1)$$
 (29)

251 The cycle time of the TREC, τ , satisfies the following relations

253
$$n t_{ch} = m \tau$$

where $t_{ch} = C_q / I_{ch}$; *n* is the number of TRECs; *m* is the number of the cells charging 254 255 simultaneously [57].

256 When $m \ge 2$, both m and n should satisfy two relations: (1) $n/m \ge 2$, and (2) $n/m \ne N$ (N is an integer). The simplest case is: n = 2 and m = 1 [32]. Assuming the charging time is 257 258 equal to the discharging time [56], one has

$$259 t_{ch} = t_{dis} = t (32)$$

260 and consequently, the charging electric current equals the discharging electric current, i.e.,

$$261 I_{ch} = I_{dis} = I (33)$$

262 As the regenerative efficiency is closely related to the regenerative time, the regenerative 263 efficiency of the TREC can be calculated as

264
$$\eta_{RL} = \frac{2t_r}{\tau} = 1 - \frac{2m}{n}$$
 (34)

For the TREC consisting of *n* TREC elements, the heat absorbed from the hot reservoir and the 265 266 heat released to the cold reservoir can be, respectively, expressed as

$$Q_{2n}^{*} = n \left(\alpha_{c} T_{2} C_{q} - I^{2} R t + \frac{2m}{n} C_{p} (T_{2} - T_{1}) \right)$$

$$= m \tau \left(\alpha_{c} T_{2} I - I^{2} R + \frac{2m}{n} \frac{C_{p}}{C_{q}} I (T_{2} - T_{1}) \right)$$

$$Q_{1n}^{*} = n \left(\alpha_{c} T_{1} C_{q} + I^{2} R t + \frac{2m}{n} C_{p} (T_{2} - T_{1}) \right)$$

$$268 \qquad = m \tau \left(\alpha_{c} T_{1} I + I^{2} R + \frac{2m}{n} \frac{C_{p}}{C_{q}} I (T_{2} - T_{1}) \right)$$

$$(35)$$

$$(35)$$

$$(36)$$

Therefore, the heat fluxes in the charging and discharging processes can be, respectively, 269 calculated by 270

271
$$Q_{2n} = Q_{2n}^* / \tau = m \left(\alpha_c T_2 I - I^2 R + \frac{2m}{n} \frac{C_p}{C_q} I (T_2 - T_1) \right)$$
(37)

272
$$Q_{1n} = Q_{1n}^* / \tau = m \left(\alpha_c T_1 I + I^2 R + \frac{2m}{n} \frac{C_p}{C_q} I (T_2 - T_1) \right)$$
(38)

273 Hence, the power output and efficiency of the TREC can be, respectively, written as

274
$$P_T = Q_{2n} - Q_{1n}$$
(39)

275
$$\eta_T = \frac{P_T}{Q_{2n}} = \frac{Q_{2n} - Q_{1n}}{Q_{2n}}$$
(40)

276 2.3. Photovoltaic-electrochemical hybrid system

277 The total solar energy irradiated on the DSSC can be calculated as

$$278 \qquad Q_{ALL} = \eta_{opt} G A_D \tag{41}$$

279 where η_{opt} is the optical efficiency of DSSC.

280 The part of heat directly released to the environment via convection can be described as [58]

281
$$Q_L = h^* A_D \left(T - T_e \right)$$
 (42)

where T_e is the environment temperature; h^* is the heat leak coefficient. Assuming the heat transfer is governed by the forced convection from the front surface, h^* is taken as 2.8 W m⁻² K⁻¹ [44].

Based on the energy conservation law, the heat transferred to the TREC can be calculated as

$$286 \qquad Q_{2n} = Q_{ALL} - P_D - Q_L \tag{43}$$

287 Combining Eq. (35) and Eq. (41), one may obtain the mathematical relationship between the 288 electric current density of DSSC and the electric current of the TREC

289
$$A_1 i^2 + B_1 i + C_1 = 0$$
 (44)

290 where
$$A_1 = -mR$$
, $B_1 = m\alpha_c T_2 + \frac{2m^2 C_p (T_2 - T_1)}{nC_q}$, $C_1 = -\eta_{opt} GA_D + VJA_D - \lambda (T_2 - T_1) + h^* A_D (T_2 - T_1)$.

It can be seen from Eq. (44) that the bottoming TREC could deliver electricity only when the following condition is fulfilled:

$$293 \qquad B_1^2 - 4A_1C_1 \ge 0 \tag{45}$$

Based on the parameters list in Table 1 and Table 2 and Eq. (42) and Eq. (43), one may plot the TREC output electric current varying with the DSSC current density, as shown in Fig. 2. It is observed from Fig. 2 that the TREC electric current first decreases and then increases as the DSSC current density increases. Furthermore, the TREC electric current increases with the increasing solar irradiance.

299 Considering the electric power generated by the TREC, the power output and efficiency of the 300 hybrid system are, respectively, described by

$$301 \qquad P = P_D + P_T \tag{46}$$

$$302 \qquad \eta = \frac{P_{\rm D} + P_T}{Q_{ALL}} \tag{47}$$

303 **3. Results and discussion**

The above mathematical model is solved by commercial software Matlab[®]. In this section, whether the DSSC model is accurate or not will be checked, and the generic performance characteristics of the hybrid system will be revealed. Moreover, parametric studies will be conducted to examine how the hybrid system performance is affected by various parameters, such as porosity of the electrode material, photoelectron absorption coefficient, Schottky barrier, film thickness and TREC internal resistance.

310 **3.1. Model validation**

311 Since there is no experimental study reported on this specific hybrid system yet, the topping 312 DSSC will be validated by comparing with both theoretical results and experimental data in the 313 literatures. Fig. 3 (a) compares the $V \sim J$ characteristic curves between the present model and the 314 theoretical model from Ref. [59]. It is seen that both models predict that the photovoltages are 315 monotonic decreasing functions of current density, and $V \sim J$ curves change moderately in the middle 316 part while change rapidly at the two ends. In addition, the photovoltage become larger as the 317 Schottky barrier increases. However, there exists obvious discrepancy between the present model 318 and the model in Ref. [59]. The photovoltage predicted in the present model is smaller than that predicted by the model from Ref. [59]. This is because the electron diffusion coefficient and light 319 320 absorption coefficient depend on the porosity electrode material in the present model, while they are 321 regarded as porosity-independent constants in the model in Ref. [59]. Another model validation 322 between the present modeling results and experimental data from Ref. [62] is conducted, as shown in 323 Fig. 3 (b). It is seen that the modeling results are in good agreement with the experimental data. It 324 indicates that the present theoretical model could well predict the performance of an actual DSSC 325 and the present model is more accurate than the one developed by Ni et al. [59]. Moreover, it is also 326 found from Fig. 3 (b) that the DSSC output voltage decreases with the increasing film thickness. This 327 is because a thicker film leads to a larger electron transport resistance, which significantly drops the 328 DSSC output voltage.

329 **3.2.** Generic performance characteristic of the hybrid system

330 The power densities and energy efficiencies of the DSSC, TREC and hybrid system varying with the DSSC operating current density is shown in Fig. 4, where $P_D^* = P_D / A_D$, $P_T^* = P_T / A_D$ and 331 $P^* = P/A_D$ are, respectively, the corresponding power densities; J_D , J_T and J_H are, 332 respectively, the operating current densities at maximum power densities $P_{D,\max}^*$, $P_{T,\max}^*$ and P_{\max}^* , 333 $\eta_{D,\text{max}}$, $\eta_{T,\text{max}}$ and η_{max} are maximum energy efficiencies for the DSSC, TREC and hybrid system. 334 335 It is found that the power output densities and energy efficiencies of the DSSC, TREC and hybrid 336 system share the same trend (i.e., first increasing and then decreasing) as the current density Jincreases. Both power density and efficiency of the hybrid system are greater than that of the single 337 338 DSSC. For the parameters in Table 1 and Table 2, the maximum power density and maximum efficiency for the DSSC are, respectively, 77.82 W m⁻² and 4.94 %, and the maximum power density 339 and maximum efficiency of the hybrid system are 102.76 W m⁻² and 6.53%, respectively. By 340 341 mounting a bottoming TREC, the power density and efficiency of the DSSC are improved by 342 32.04% and 32.18% compared with the sole DSSC, respectively. It clearly shows that the proposed 343 method in this paper is feasible and effective, and the TREC can be treated as a potential waste heat 344 recovery candidate for DSSC.

345 3.3. Effect of the electrode porosity

The porosity of the electrode has a decisive influence on the light absorption coefficient and electron diffusion coefficient of the DSSC [63], and therefore, the porosity greatly affects the overall hybrid system performance.

Based on both experimental and modeling results [64], the effects of electrode porosity on the
light absorption coefficient [65] and the electron diffusion coefficient can be expressed as [66]

$$351 \qquad \alpha = 2586 \left(1 - \varepsilon\right) \left(\varepsilon + 2.89\right) \tag{48}$$

352
$$D = a \left| \varepsilon - \varepsilon_c \right|^{\mu} \qquad \varepsilon \le 0.41$$
 (49)

$$353 \qquad \alpha = 2.97 \times 10^4 \varepsilon^2 \tag{50}$$

354
$$D = 1.69 \times 10^{-4} \left(-14.78\varepsilon^3 + 7.39\varepsilon^2 - 2.89\varepsilon + 2.15 \right) \qquad \varepsilon \ge 0.41$$
 (51)

355 where ε is porosity of DSSC electrode material, a, μ and the critical porosity ε_c are 4×10^{-4} 356 cm² s⁻¹, 0.82 and 0.76, respectively [25].

As shown in Fig. 5 (a), the light absorption coefficient and electron diffusion coefficient display different features in the whole range of electrode porosity. The electron diffusion coefficient approximately decreases from $3.58 \times 10^{-4} \text{ m}^2 \text{ s}^{-1}$ to $5.89 \times 10^{-5} \text{ m}^2 \text{ s}^{-1}$ as the electrode porosity increases from 0 to 0.41. The electron diffusion coefficient first decreases from $2.54 \times 10^{-4} \text{ m}^2 \text{ s}^{-1}$ to 0 when the porosity increases from 0.41 to 0.76, and then it grows to $1.86 \times 10^{-4} \text{ m}^2 \text{ s}^{-1}$ as the electrode porosity approaches 1.0. The light absorption coefficient first grows and then decreases as the electrode porosity increases, and its maximum value is about 5000 cm⁻¹ when the porosity is 0.41.

It is well-known that the short-circuit current density of DSSC is a key factor affecting the 364 365 DSSC performance and thus the hybrid system performance. As shown in Fig. 5 (b), the short-circuit current density increases to attain its maximum 7.81 A m⁻² and then begins to drop as the electrode 366 367 porosity increases. As the electrode porosity gradually increases, the ability of the electrode material to absorb photoelectrons gradually increases while the electron diffusion coefficient gradually 368 369 decreases. Therefore, the compound effect is that the number of electrons entering the circulation 370 state gradually increases. However, when the electrode porosity continues to increase, the mass of 371 the TiO₂ film per unit area will decrease, so the surface area of the TiO₂ film per unit area decreases and the number of dye molecules adsorbed by the thin film per unit area become less. Finally, thecurrent density of the short-circuit eventually decreases with the increase of electrode porosity.

374

3.4. Effect of the photoelectron absorption coefficient

375 The $\eta \sim J$ curves for different photoelectron absorption coefficients [66] are shown in Fig. 6. It 376 is seen that both efficiency and power output density of the hybrid system increase with an increase 377 in the photoelectron absorption coefficient, and the current density corresponding to the maximum 378 efficiency or maximum power density also shifts rightward with the increasing photoelectron 379 absorption coefficient. Meanwhile, the limiting current density for either efficiency or power density 380 of the hybrid system is enlarged as the photoelectron absorption coefficient increases. It should be 381 noted that both efficiency and power density are less sensitive to the change of photoelectron absorption coefficient when the photoelectron absorption coefficient exceeds 5.0×10^5 m⁻¹. This is 382 383 because the power output of the hybrid system first sharply and then slowly increases as the photoelectron absorption coefficient increases to 5.0×10^5 m⁻¹, as shown in Fig. 6 (c). As the 384 photoelectron absorption coefficient increases from 5.0×10^5 m⁻¹, the power density of the hybrid 385 386 system is almost unchanged. Hence, an appropriate photoelectron absorption coefficient should be 387 carefully selected since high photoelectron absorption coefficient may tremendously increase the 388 manufacture cost for DSSCs.

389 **3.5. Effect of the Schottky barrier**

As shown by Eq. (1), the Schottky barrier of combination interface has a significant impact on the voltage loss. With the help of the Schottky barrier model [58], the voltage loss V_1 varying with the Schottky barrier Φ_b can be obtained, as shown in Fig. 7 (a). The voltage loss maintains 393 negligibly small when the Schottky barrier is smaller than 0.5 eV. As the Schottky barrier increases from 0.5 eV, the voltage loss sharply increases. The effects of the operating temperature on the 394 395 voltage loss are negligible when the Schottky barrier is below 0.5 eV and becoming obvious with a 396 further increase in the Schottky barrier. In addition, the voltage loss decreases as the DSSC operating 397 temperature is elevated. This is because it is easier for the electrons at higher temperature to 398 crossover the combination interface [67] with a smaller voltage loss. Therefore, the performance of 399 DSSC can be optimized by adjusting the Schottky barrier to reduce the voltage loss at the composite 400 interface [68].

401 Fig. 7 (b) shows that both efficiency and power output of the DSSC are almost unchanged when 402 the Schottky barrier is below 0.5 eV and improved as the operating temperature increases due to the 403 reduced voltage losses. Fig. 7 (c) and Fig. 7 (d) show that the efficiency and power output of the 404 hybrid system varying with the operating current density for different Schottky barriers. It is seen 405 that both efficiency and power output of the hybrid system are improved as the Schottky barrier is 406 reduced, and the current densities corresponding to the maximum power density and maximum 407 efficiency are also enlarged. However, the effects of Schottky barrier on the hybrid system 408 performance become insensitive as the Schottky barrier is smaller than 0.5 eV.

409 **3.6. Effect of the film thickness**

A larger film thickness means it incorporates more dye molecules to produce electrons, and thus the photoelectron current density will be increased [69]. When the film thickness exceeds a certain length and continues to grow, the transport distance of the produced electrons will be also simultaneously increased, which in return drops the photoelectron current density. Fig. 8 shows the effects of film thickness on the efficiency and power density of the hybrid system. Both efficiency and power density of the hybrid system first increase and then decrease as the film thickness is increased. The limiting current density for the hybrid system is also first increased and then decreased with an increase in the film thickness. For the parameters given in Table 1 and Table 2, the optimum film thickness is found to be between 1.0×10^{-2} and 4.0×10^{-2} mm.

419 **3.7. Effect of the internal resistance of thermally regenerative electrochemical cycle**

420 The TREC internal resistance is an important factor affecting the performance of TREC and 421 thus affecting the overall hybrid system performance as well. As shown in Fig. 9, both efficiency and 422 power density of the hybrid system are improved as the TREC internal resistance is reduced. This is because a smaller TREC internal resistance indicates that less heat is wasted per unit time in the 423 424 TREC. It is also seen from Fig. 9 that the working current density range of TREC becomes narrower 425 as the TREC internal resistance increases. Fig. 9 (c) further shows that the maximum power density 426 and maximum efficiency of the hybrid system continuously decline as the TREC internal resistance 427 increases. Therefore, reducing the TREC internal resistance is an important mean to effectively 428 improve the hybrid system performance.

429 4. Conclusions

A novel photovoltaic-electrochemical hybrid system composed of a DSSC, an SSA and a TREC is proposed with the aim to harvest the relatively high wavelength sunlight that not utilized by the DSSC. The models of DSSC and TREC are mathematically described in detail, and the condition that the bottoming TREC participates in the additional electricity generation is derived. The DSSC model is demonstrated to be accurate by comparing with both modeling results and experimental 435 data in the current literatures. It is feasible and effective to use TREC as an alternative waste heat recovery technology for DSSC. Through mounting a bottoming TREC, the maximum power density 436 and maximum efficiency of the hybrid system gain increases by 32.04% and 32.18% in comparison 437 438 with the single DSSC, respectively. Extensive parametric studies are carried out to examine how the 439 hybrid system performance is impacted by some microstructure parameters and operating conditions. 440 The light absorption coefficient, electron diffusion coefficient and short-circuit current density 441 display different trends varying with the electrode porosity. An increase in photoelectron absorption coefficient improves the efficiency and power output density of the hybrid system, while an increase 442 443 in Schottky barrier or internal resistance of TREC degrades the hybrid system performance. It is also 444 found that there exists an optimum film thickness to maximize the efficiency and power output 445 density of the hybrid system. The obtained results offer some new insights into further improving the 446 performance of photovoltaic cells through hybridizing feasible TRECs.

448 Nomenclature

449	Α	Active area (m ²)
450	A^*	Richardson constant (A m ⁻² K ⁻²)
451	C_p	Heat capacity (J kg ⁻¹ K ⁻¹)
452	C_q	Charge capacity (A s ⁻¹ kg ⁻¹)
453	d	Thickness of the thin film (m)
454	D	Electron diffusion coefficient (m ² s ⁻¹)
455	G	Solar spectrum intensity (W m ⁻²)
456	h	Planck constant (J s ⁻¹)
457	h^*	Heat leak coefficient (W $m^{-2} K^{-1}$)
458	Ι	Current of TREC (A)
459	J	Current density of the DSSC (A m ⁻²)
460	k	Boltzmann constant (J K ⁻¹)
461	L	Electron diffusion length (m)
462	<i>m</i> *	Equal to 5.6 times the electron mass (g)
463	<i>m</i> [#]	Ideality factor
464	m	Number of the cells charged simultaneously
465	n	Number of TREC
466	n(x)	Excessive electron concentration (cm ⁻³)
467	n_0	Electron concentration under a dark condition (cm ⁻³)
468	Р	Power output (W)

469	P^*	Power output density (W m ⁻²)
470	Q	Heats for a TREC (J)
471	Q^{*}	Heats for a TREC (J)
472	q	Electron charge (C)
473	q^*	Heats (J)
474	R	Electric resistance (Ω)
475	Т	Operating temperature (K)
476	t	Time (s)
477	V_0	Potential difference of the redox potential (V)
478	V_{I}	Voltage loss at the TiO ₂ /TCO interface (V)
479	Greek symbols	
480	${\Phi}$	Light intensity $(m^{-2} s^{-1})$
481	$arPhi_0$	Light intensity at 1 sun condition (m ⁻² s ⁻¹)
482	$arPsi_b$	Schottky barrier height (eV)
483	τ	Cycle time (s)
484	τ^*	Electron lifetime (s)
485	η	Efficiency
486	λ	Decreasing rate of the power output density (W K ⁻¹)
487	β	Decreasing rate of efficiency (K ⁻¹)
488	α_c	Temperature coefficient (V K ⁻¹)
489	α	Light absorption coefficient of the electrode (m ⁻¹)

490	Superscripts and	subscripts
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491	1	Cold source
492	2	Heat source
493	All	Gross
494	D	DSSC
495	e	Environment
496	Т	TREC
497	L	Heat release
498	dev	Deviations
499	RL	Regenerative losses
500	ref	Reference
501	opt	Optimal state
502	r	Recombination process
503	dis	Discharge process
504	ch	Charge process
505	n	Number of TREC
506	J	Joule
507	SC	Short-circuit current density
508		

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- 514
- 515

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- 682

683 Table captions:
684
685 Table 1. Parameters used in the DSSC modeling.

Table 2. Parameters used in the TREC modeling.

687

688 **Figure captions:**

- 689
- 690 Fig. 1. Conceptual diagram of the photovoltaic-electrochemical hybrid system.
- Fig. 2. TREC output electric current varying with the DSSC current density.
- Fig. 3. Comparisons between the present modeling results and (a) the modeling results from Ref.
- 693 [59], where $m^{\#}$ equals 4.5; and (b) the experimental data from Ref. [62].
- Fig. 4. (a) Efficiency and (b) power density comparisons between the DSSC, TREC and hybrid
- 695 system, where J_D , J_T and J_H are, respectively, the operating current densities at maximum 696 power densities $P_{D,\text{max}}^*$, $P_{T,\text{max}}^*$ and P_{max}^* ; $\eta_{D,\text{max}}$, $\eta_{T,\text{max}}$ and η_{max} are maximum energy 697 efficiencies for the DSSC, TREC and hybrid system.
- 698 Fig. 5. (a) electron diffusion coefficient and light absorption coefficient and (b) short-circuit current
- 699 density varying with the electrode porosity.
- Fig. 6. Effects of the photoelectron absorption coefficient on (a) $\eta \sim J$, (b) $P^* \sim J$ and (c) P^* .
- Fig. 7. Effects of the Schottky barrier on (a) voltage loss, (b) power density and efficiency of DSSC,
- 702 (c) efficiency of the hybrid system, and (d) power density of the hybrid system.
- Fig. 8. Effects of the film thickness on (a) efficiency and (b) power density of the hybrid system.

- Fig. 9. Effects of the TREC internal resistance on (a) efficiency, (b) power density, and (c) maximum
- 705 power density and maximum efficiency of the hybrid system.

708	Parameters	Value
709	$n_0 ({ m m}^{-3})$	1.0×10^{22}
710	$\Phi_0 ({ m m}^{-2}{ m s}^{-1})$	1.0×10^{21}
711	$\Phi_{b}\left(\mathrm{J} ight)$ [59]	1.12×10^{-19}
712	α (m ⁻¹) [60]	5.0×10 ⁵
713	$D (\mathrm{m}^2\mathrm{s}^{-1})$ [60]	2.0×10 ⁻⁸
714	τ (s)	10×10 ⁻²
715	$A^{*}(A \text{ m}^{-2} \text{ K}^{-2})$ [60]	6.71×10^{6}
716	<i>d</i> (m) [60]	10×10 ⁻⁶
717	<i>T</i> (K)	330
718	$T_{ref}(\mathbf{K})$	298
719	$T_{e}(\mathbf{K})$	298
720	<i>q</i> (C)	1.60218×10^{-19}
721	k	1.38006×10 ⁻²³
722	$G (W m^{-2}) [61]$	1000
723	h	6.626×10 ⁻³⁴
724	η_{opt} [53]	0.970
725	h^* (W m ⁻² K ⁻¹) [53]	2.8
726	λ (W K ⁻¹) [53]	0.00506×10 ⁻³
727	β (K ⁻¹)	0.0114
728	<i>m</i> [#] [62]	4.5
729		

Table 1. Parameters used in the DSSC modeling.

Parameters	Value
 $R\left(\Omega ight)$	0.004
$\alpha_c (V K^{-1})$	0.027
<i>m</i> [32]	1.0
n [32]	2.0
$T_2(\mathbf{K})$	330
$T_{l}(\mathbf{K})$	298
$C_p (J \text{ kg}^{-1} \text{K}^{-1}) [32]$	2.408
C_a (A h kg ⁻¹) [32]	32.43

Table 2. Parameters used in the TREC modeling.

744 Fig. 1.





748 Fig. 2.



749



Fig. 2. TREC output electric current varying with the DSSC current density.





Fig. 3. Comparisons between the present modeling results and (a) the modeling results from Ref. [59], where $m^{\#}$ equals 4.5; and (b) the experimental data from Ref. [62].

759 Fig. 4.



762

760

Fig. 4. (a) Efficiency and (b) power density comparisons between the DSSC, TREC and hybrid system, where J_D , J_T and J_H are, respectively, the operating current densities at maximum power densities $P_{D,\text{max}}^*$, $P_{T,\text{max}}^*$ and P_{max}^* ; $\eta_{D,\text{max}}$, $\eta_{T,\text{max}}$ and η_{max} are maximum energy efficiencies for the DSSC, TREC and hybrid system.

768 Fig. 5.



Fig. 5. (a) electron diffusion coefficient and light absorption coefficient and (b) short-circuit current













Fig. 6. Effects of the photoelectron absorption coefficient on (a) $\eta \sim J$, (b) $P^* \sim J$ and (c) P^* .







Fig. 7. Effects of the Schottky barrier on (a) voltage loss, (b) power density and efficiency of DSSC,









Fig. 8. Effects of the film thickness on (a) efficiency and (b) power density of the hybrid system.

795 Fig. 9.







800 Fig. 9. Effects of the TREC internal resistance on (a) efficiency, (b) power density, and (c) maximum

801 power density and maximum efficiency of the hybrid system.