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1 **Rapid hybrid microwave cladding of SiO<sub>2</sub>/TiO<sub>2</sub> sol-gel derived composite coatings**

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19

20 **Highlights**

- 21 - A hybrid microwave cladding process is developed for the fabrication of composite coating
- 22 systems with disparate dielectric properties
- 23 - The difficulty of processing low microwave absorbing and microwave transparent materials has
- 24 been overcome
- 25 - Fabrication of crack-free silicon dioxide (SiO<sub>2</sub>) / titanium dioxide (TiO<sub>2</sub>) composite coatings on
- 26 polycarbonate substrates with enhanced surface mechanical properties has been demonstrated
- 27 - This process greatly reduces the processing time from hours to within a minute compared to
- 28 conventional methods

29 **Abstract**

30 UV protection and coatings for plastics are important for various applications. Cladding of low microwave  
31 (MW) absorption composite coatings on high MW transparent plastic substrates is a challenge due to their  
32 disparate dielectric properties. Moreover, an uneven heat energy conversion within the composite creates  
33 an additional hurdle to produce a coating with good surface integrity. In this study, a protocol was  
34 developed to overcome these difficulties based on a hybrid approach. The adverse effect of temperature  
35 mismatch between the coating and substrate was reduced through a two-way susceptor-aided heating  
36 mechanism. Low MW absorbing sol-gel derived composite coatings consisting of silicon dioxide (SiO<sub>2</sub>)  
37 and titanium dioxide (TiO<sub>2</sub>) were successfully cladded on the surface of MW and visual light transparent  
38 polycarbonate to produce a clear protective coating with UV-resistance. Nanoindentation tests were  
39 conducted to assess the effectiveness of the proposed protocol. Significant enhancement in the surface  
40 elastic modulus and hardness were achieved.

41 **Keywords:** microwave; susceptor heating; composite coating; transparent substrate; scratch resistance; UV  
42 protection

## 43 **Introduction**

44 Engineering plastics, such as polycarbonate (PC), have been widely used to produce optical lenses due  
45 to their low cost, high clarity, impact resistance and chemical inertness. However, they are susceptible to  
46 abrasion damage and photodegradation under UV light, which alters their optical transparency and  
47 mechanical properties [1]. Therefore, cladding protective coatings on the surface is crucial to extend the  
48 lifespan of a plastic product. Inorganic metal oxide particles, such as silicon dioxide ( $\text{SiO}_2$ ), are commonly  
49 used in coating materials because they possess high visible transmittance and can improve the scratch  
50 resistance of the surface [2-6]. In addition, titanium dioxide ( $\text{TiO}_2$ ), with an absorption band in the  
51 ultraviolet region, has been widely used as a functional additive in, or as a coating material on, polymers to  
52 enhance photo-stability [7] and reduce yellowing of polymer substrates [8,9]. UV protection can also  
53 provide a long-term adhesive strength of the coating as stress may occur at the coating-substrate interface  
54 due to the photodegradation of the polymer substrates [10]. Numerous techniques, for instance, physical  
55 and chemical vapor deposition methods, have been developed to apply such coatings [11,12]. However,  
56 most of the methods are expensive and require high energy input. Alternatively,  $\text{SiO}_2/\text{TiO}_2$  composite film  
57 can be synthesized on a substrate by bottom-up approaches such as sol-gel method to reduce the production  
58 cost [13]. The consolidation of  $\text{SiO}_2/\text{TiO}_2$  precursors can lead to the formation of inorganic composite film  
59 with a three-dimensional atomic structure through a thermal-induced simultaneous hydrolysis and  
60 condensation [1].

61 Microwave (MW) technology has been widely applied in the synthesis of nanomaterials. It has been  
62 used in synthetic chemistry due to its high efficiency and accelerated reaction rates [14,15]. During sol-gel  
63 synthesis, MW processing may promote further condensation and offers a higher yield of the product.  
64 However, direct MW processing of low MW absorption composite coatings on MW transparent plastic  
65 substrate remains challenging due to different dielectric properties and poor microwave absorptivity.  
66 Several attempts have been made to synthesize sol-gel based  $\text{SiO}_2/\text{TiO}_2$  coating on MW transparent  
67 polymer substrates by direct MW irradiation [16-18]. As the  $\text{SiO}_2$  and  $\text{TiO}_2$  coating materials coupled  
68 weakly with MW irradiation due to their low dielectric loss properties, heat generation was limited. This  
69 led to incomplete hydrolysis and condensation of the sol-gel precursors at a relatively low temperature [19].  
70 Therefore, no improvement of surface properties could be made. Another major difficulty was the  
71 temperature inhomogeneity during the direct MW heating process. Due to different MW absorption rates

72 of the coating and MW transparent substrate, thermal stress may be generated at the interface, subsequently  
73 leading to coating failures, such as cracking and coating delamination [20].

74 In this study, a hybrid MW cladding protocol has been developed to overcome these challenges. A kiln  
75 made of silicon carbide (SiC) susceptor was introduced into the heating system to investigate a two-way  
76 susceptor-aided heating mechanism. During the heating process, the sample was heated volumetrically by  
77 means of direct coupling with the MW irradiation, and from the surface by means of the conventional mode  
78 of heat transfer from the susceptor. Using the proposed method, sol-gel-derived SiO<sub>2</sub>/TiO<sub>2</sub> composite  
79 coatings were successfully fabricated in a short period of time. This hybrid heating approach could provide  
80 a faster heating rate and better thermal homogeneity between the coating and substrate. The coatings  
81 exhibited improved mechanical properties and good UV absorbability with a crack-free surface.

## 82 **Experimental section**

### 83 *Materials*

84 PC substrates (Lexan XL10) with dimensions of 10 mm (W) × 20 mm (L) × 3 mm (H) were used as the  
85 starting substrate material. Potassium dichromate (K<sub>2</sub>Cr<sub>2</sub>O<sub>7</sub>, ≥98%, Aladdin), sulfuric acid (H<sub>2</sub>SO<sub>4</sub>, 95%,  
86 AnalaR NORMAPUR) and 3-aminopropyl triethoxysilane (Si(OC<sub>2</sub>H<sub>5</sub>)<sub>3</sub>(CH<sub>2</sub>)<sub>3</sub>NH<sub>3</sub> or 3-APS, ≥98%,  
87 Sigma-Aldrich) were used for surface functionalization of the PC. Tetraethyl orthosilicate (Si(OC<sub>2</sub>H<sub>5</sub>)<sub>4</sub>,  
88 TEOS, ≥98%) and titanium tetrakisopropoxide (Ti(OCH(CH<sub>3</sub>)<sub>2</sub>)<sub>4</sub>, TTIP, ≥99.5%), purchased from  
89 International Laboratory, were used as the precursor for SiO<sub>2</sub> and TiO<sub>2</sub> sol-gel. Ethanol (C<sub>2</sub>H<sub>5</sub>OH, EtOH,  
90 ≥99.8%, Sigma-Aldrich) and 2-propanol ((CH<sub>3</sub>)<sub>2</sub>CHOH, 2-IPA, ≥99.5%, anhydrous, International  
91 Laboratory) were used as organic solvents during the sol-gel preparation. 0.1 M hydrochloric acid (HCl,  
92 Sigma-Aldrich) and glacial acetic acid (CH<sub>3</sub>COOH, ≥99.8%, Sigma-Aldrich) were used as reagents to  
93 produce a stable sol for the coating process.

### 94 *Two-step surface functionalization of PC specimen*

95 Polycarbonate (PC) features low chemical affinity and surface wettability with most coating materials  
96 [1]. To improve the adhesion between the composite coating and the PC substrate, a two-step surface  
97 functionalization method was developed. All PC substrates were firstly cleaned by compressed air to  
98 remove debris on the surface, and then cleaned by detergent and rinsed with distilled water. The substrates  
99 were ultrasonically cleaned using 2-IPA for 10 mins, followed by drying under room temperature. After

100 drying, the PC substrates were immersed in a sulfochromic acid solution for 30 mins. The sulfochromic  
 101 acid solution was prepared by mixing 37.5 ml H<sub>2</sub>SO<sub>4</sub> and 4 g K<sub>2</sub>Cr<sub>2</sub>O<sub>7</sub> with 12.5 ml DI water. The acid  
 102 treated substrates then underwent a second treatment with a silanization solution, which was prepared by  
 103 mixing 1 g 3-APS with 100 ml of 2-IPA, for 5 mins. The silanized substrates were dried at room temperature  
 104 prior to use.

### 105 *Sol-gel preparation and film deposition*

106 The SiO<sub>2</sub> and TiO<sub>2</sub> sol-gels were prepared separately, using TEOS and TTIP as precursors, EtOH and  
 107 2-IPA as solvents, respectively. To prepare the SiO<sub>2</sub> sol, TEOS was mixed with EtOH in a molar ratio of  
 108 1:10. Subsequently, 0.1 M HCl was added to the mixture until the pH value reached 2. TTIP was mixed  
 109 with 2-IPA and CH<sub>3</sub>COOH in a ratio of 1:35:7 to prepare the TiO<sub>2</sub> sol. The solution was adjusted to pH 4  
 110 by adding 0.1 M HCl. Water was added to the solution dropwise to ensure the molar ratio of TEOS:H<sub>2</sub>O  
 111 and TTIP:H<sub>2</sub>O was 1:4. The hydrolysis and condensation reactions started once the water was added. Both  
 112 sols were then aged for 4 hours before depositing onto the substrate.

113 To study the effect of TiO<sub>2</sub> content on the anti-scratching and UV-blocking capabilities, samples with  
 114 different coating compositions were prepared, as shown in Table 1. The pretreated PC substrates were dip-  
 115 coated with the SiO<sub>2</sub>/TiO<sub>2</sub> composite sol using the PTL-MM01 dip coater (MTI Corporation) at a  
 116 withdrawal speed of 3 mm/s. The coated samples were left to dry at room temperature to remove excessive  
 117 solvent. The process was repeated three times so that three layers of coating were deposited on the PC  
 118 surface.

119 **Table 1** Formulation of SiO<sub>2</sub>/TiO<sub>2</sub> composites

Samples	Coating compositions	
	SiO <sub>2</sub> sol (vol. %)	TiO <sub>2</sub> Sol (vol. %)
S0 (untreated PC)	---	---
S0-1 (1 step surface treatment)	---	---
S0-2 (2 steps surface treatment)	---	---
S1-0T	100	0
S2-5T	95	5
S3-10T	90	10
S4-30T	70	30
S5-50T	50	50
S6-100T	0	100

### 120 *Hybrid microwave cladding process*

121 A 2.45 GHz MW furnace (HAMiLab-HV3, SYNOTHERM Corporation) was used for the cladding  
 122 process. A cylindrical alumina (Al<sub>2</sub>O<sub>3</sub>) kiln with an inner layer of silicon carbide (SiC) was used as the

123 MW susceptor. Before the cladding process, the MW furnace was preheated with the kiln for 2 mins using  
124 a MW power of 500 W. The temperature was monitored by an infrared sensor. The samples were then put  
125 into the kiln and heated under a MW power of 100 W for 30 s at 120 °C.

## 126 **Characterization**

127 The surface morphology of the coatings was studied using an optical microscope (Leica DM 4000M)  
128 and a scanning electron microscope (SEM, JEOL JSM-6490). The coating-substrate interfaces were  
129 investigated by the SEM cross-sectional image. The samples were cut in half and the cut surfaces were  
130 ground using 400 and 600 grit sand paper. All SEM samples were sputter coated with a thin layer of gold  
131 to enhance their conductivity.

132 The characteristics of the coatings were investigated with respect to their adhesion strength and scratch  
133 resistance. A crosscut tape test, in compliance with ASTM D3359-17, was performed to study the adhesion  
134 strength between the PC substrate and the coatings. A crosscut tester (QFHD60D, GYX International  
135 Instrument Co., Ltd) with a 1 mm cutting space was used to create a 5 × 5 grid for further inspection. The  
136 scratch resistance was determined by a pencil hardness tester (Scratch Hardness Tester Model QHQ-A,  
137 GYX International Instrument Co., Ltd.) which also conformed to the ASTM standard D3363-05 method.

138 UV absorption property of the composite coatings was investigated by using an ultraviolet-visible (UV-  
139 Vis) spectrophotometer (HR2000, Ocean Optic, USA) in a wavelength range of 300 to 700 nm. The UV  
140 absorbability for the aged sol of the corresponding composition was measured.

141 Nanomechanical performance of the coatings was studied through nanoindentation test by using a  
142 Hysitron Inc. TriboScope® nanomechanical test instrument. A three-sided pyramidal diamond cube corner  
143 indenter with a 150 nm tip radius was used in the test with a maximum loading force of 400 μN. The  
144 instrument was calibrated using fused silica samples prior to the testing. In all tests, a total of 5 indents  
145 were made, and the mean hardness ( $H$ ) and elastic modulus ( $E$ ) were evaluated accordingly. Nano-scratch  
146 testing was performed using a conical indenter with a tip radius of 1 μm. A progressive normal force from  
147 0 μN to 1000 μN in 5 s was applied to the coating surface with a dwell time of 15 s. Scratches with a length  
148 of 6 μm were created. The friction coefficient of the samples was evaluated by taking the ratio of tangential  
149 force to the normal load (LF/NF). Three measurements were performed, and the average friction coefficient  
150 was calculated accordingly.

## 151 **Results and discussion**

### 152 *Mechanism of hybrid microwave cladding of SiO<sub>2</sub>/TiO<sub>2</sub> composite*

153 During the cladding process, as depicted in Fig. 1a, MW energy was partly absorbed by the SiC when  
154 passing through the kiln. As SiC was a high loss material which absorbed MW energy efficiently, a large  
155 amount of heat was generated within a few minutes so that a rapid increase of temperature could be achieved  
156 [21]. The transmitted MW energy was absorbed by the SiO<sub>2</sub>/TiO<sub>2</sub> coating simultaneously. The heat energy  
157 transferred to the sample was a combination of (1) the heat generated through MW coupling of the  
158 SiO<sub>2</sub>/TiO<sub>2</sub> coating and (2) the heat transfer by conduction, convection and radiation from the heated  
159 susceptor to the coating and MW transparent PC substrate (Fig. 1b). Therefore, this setup provided more  
160 homogeneous heating to the sample. Compared to direct MW irradiation, where energy was partially  
161 absorbed by the coating (Fig. 1c), hybrid MW heating provided a uniform temperature distribution between  
162 the coating and the MW transparent PC substrate [14,21]. Moreover, the samples could be heated up more  
163 rapidly to the target temperature. MW irradiation could also promote hydrolysis and condensation reactions,  
164 leading to fast densification of the coating. Under MW irradiation at high temperature, unreacted Si-OH  
165 and Ti-OH underwent further condensation [14], enabling enhancement of the mechanical properties of the  
166 coating after the cladding process.

### 167 *Surface morphology and coating adhesion*

168 Morphology of the coating surfaces with different sol-gel compositions is shown in Fig. 2. Samples S1-  
169 0T, S2-5T and S3-10T show a uniform crack-free surface. However, cracks can be observed in sample S4-  
170 30T. Numerous micro-cracks can be seen on the surface of sample S6-100T. Cracking could be primarily  
171 attributed to the fast hydrolysis reaction of the TiO<sub>2</sub> sol during the cladding process. This produced large  
172 stresses in the coating [22], and the severity of cracking increases with increasing TiO<sub>2</sub> sol concentration.  
173 Therefore, the coating composition has a significant effect on the coating integrity. However, although  
174 abundant cracks were found on the surface of sample S6-100T, no detachment of the coating was observed,  
175 owing to the enhanced coating adhesion by the two-step functionalization of the PC substrate. The adhesion  
176 rating and the surface morphology after the tests are shown in Table 2 and Fig. 3, respectively. The highest  
177 adhesion strength rating was recorded for the samples S1-0T, S2-5T, S3-10T and S4-30T with no coating  
178 removal. Good coating adhesion was observed as confirmed by the cross-cut test. This was due to the  
179 improved surface wettability and the surface energy of the substrate [23] by the two-step functionalization

180 method. This process enhanced the hydrophilicity of the surface by providing active sites for chemical  
 181 reaction with the coating sol during the cladding process [24,25]. Moreover, the MW cladding process may  
 182 also promote the reaction between the coating sol and the surface functional group. The adhesion rating for  
 183 sample S5-50T (Fig. 3b) and sample S6-100T (Fig. 3c) were specified as 4B (<5% area of removal) and  
 184 3B (5-15 % area of removal) respectively. The reduction of the adhesion rating was mainly due to the crack  
 185 formation of the coating with the increase of TiO<sub>2</sub> content [1,16,22].

186 **Table 2** Test result of adhesion rating.

Sample	S0	S0-2	S1-0T	S2-5T	S3-10T	S4-30T	S5-50T	S6-100T
Adhesion rating	-	-	5B	5B	5B	5B	4B	3B

187

188 The interface between the coating and substrate was investigated by SEM cross-sectional images of the  
 189 samples, as shown in Fig. 4. Due to minimized temperature discrepancy between the coating and substrate,  
 190 no coating delamination was observed at the interface for all the samples containing TiO<sub>2</sub>, from S3-10T to  
 191 S6-100T. Therefore, good coating adhesion could be achieved. These results showed the advantage of the  
 192 hybrid MW method over the direct MW processing [17] in handling materials with different dielectric  
 193 properties.

194 ***UV-VIS spectra***

195 Optical performance and UV protection are important for a transparent substrate such as PC. To prevent  
 196 photo-degradation by UV irradiation, the coating should have high optical transparency with UV  
 197 absorptivity. The results of UV-visible spectra analysis in both the visible light and UV ranges for different  
 198 coating compositions is shown in Fig. 5. The S1-0T coating sol shows no absorption of light within the  
 199 scanning wavelength range. For all sol-gel compositions containing TiO<sub>2</sub>, the light transmission drops  
 200 drastically and reaches zero within the UV range, starting from 380 nm. The presence of TiO<sub>2</sub> within the  
 201 coatings demonstrated good UV absorbing properties even at a low concentration of TiO<sub>2</sub>, as in S2-5T  
 202 coating sol (i.e. 5 vol.% of TiO<sub>2</sub>). However, a further increase of TiO<sub>2</sub> reduces the transparency of the  
 203 visible range from 380 to 700 nm, leading to an adverse effect on optical clarity to the PC substrate.

204 ***Nanomechanical performance***

205 Sample S3-10T was used to study the nanomechanical properties of the coatings because of its crack-  
 206 free surface with exceptional adherence, good optical transmittance, and UV-blocking capability compared

207 with other compositions. Nanoindentation was conducted on the samples with 1, 3 and 5 coating layers to  
208 investigate the effect of the number of coating layers on their hardness and modulus (Fig. 6). As compared  
209 to that of the pristine PC substrate, the elastic modulus of the S3-10T samples with different number of  
210 coating layers almost doubles, increasing from  $3.3 \pm 0.04$  GPa to  $5.9 \pm 0.06$  GPa, while the hardness  
211 improves by over threefold, from  $277.4 \pm 1.5$  MPa to  $897.0 \pm 3.5$  MPa. The enhancement of the mechanical  
212 properties was mainly due to the increase in the coating thickness associated with the increasing number  
213 of layers. This result demonstrated the remarkable improvement of mechanical performance brought by the  
214 deposition of such composite coating on the PC.

215 The average friction coefficients obtained during the steady nano-scratching process of the samples are  
216 summarized in Fig. 7a, with the highest friction coefficient found for the pristine PC sample, which has a  
217 value of  $0.74 \pm 0.03$ . All the three coated S3-10T samples show a significant decrease in friction coefficient  
218 compared to that of the pristine PC sample. This may be due to the rolling effect of the nanoparticles  
219 detached from the coating as proposed by Chang and Zhang [26]. The samples with 3-layer coating possess  
220 the lowest friction coefficient with a value of  $0.13 \pm 0.02$ , which implies the highest scratch resistance. This  
221 was also verified by the penetration depth in the nano-scratch test. As shown in Fig. 7b, the samples with  
222 3-layer coating show the lowest penetration depth of  $0.22 \pm 0.02$   $\mu\text{m}$ , i.e. better resistance against scratching  
223 force. Although the coatings with 5 layers have the highest elastic modulus and hardness. During the  
224 scratching process, the stress tended to accumulate at the tip front, leading to a scratch fracture on the  
225 coating [27]. This phenomenon could be confirmed by the presence of “kinks” in the plot of the normal  
226 force against penetration depth for 5-layer coating, as compared to the plot for 3-layer coating (Fig. 8).

227 A comparison with a recently reported  $\text{SiO}_2$  based protective coatings on polymer substrates is  
228 summarized in Table 3. By using the hybrid MW approach, similar mechanical properties of the coating  
229 have been achieved as compared to the findings by Le Bail *et al.* [28]. The post thermal treatment time has  
230 been significantly reduced from 1 hour to 30 s in this study. Compared with the coating fabricated through  
231 UV irradiation [29], the coating hardness was also significantly improved (about 2.5 times improvement).  
232 The results showed that the proposed protocol using hybrid microwave approach would offer a more  
233 efficient and effective method for coating fabrication.

234

235

236

237 **Table 3** Comparison for recently reported hard coatings on polymer substrates.

Post-Treatment method	Post-treatment time	Hard coating materials	Nanoindentation		Ref.
			Hardness (MPa)	Elastic Modulus (GPa)	
Atmospheric plasma deposition	> 2 h	GPTMS, ZrO <sub>2</sub> , SiO <sub>2</sub>	670	10.6	[29]
Ventilated oven	1 h	GPTMS, ZrO <sub>2</sub> , SiO <sub>2</sub>	900	5.8	[22]
UV	-	urethane acrylate, SiO <sub>2</sub>	362	5.6	[28]
MW and UV	6 min	SiO <sub>2</sub> , TiO <sub>2</sub>	-	-	[17]
Hybrid MW	30 s	SiO <sub>2</sub> , TiO <sub>2</sub>	897	5.9	This work

238 **Conclusion**

239 A protocol for cladding of sol-gel derived SiO<sub>2</sub>/TiO<sub>2</sub> composite coatings has have been developed based  
 240 on the hybrid microwave (MW) approach. It has offered homogenous heating by exploiting direct  
 241 microwave and susceptor-aided heating simultaneously. The challenges of MW processing of low MW  
 242 absorption composite coating on MW transparent substrate have been resolved. Good coating adhesion and  
 243 surface integrity have been achieved. Scratch resistant and UV protective coatings were successfully  
 244 demonstrated, and their mechanical and optical performances were verified. UV absorbability could be  
 245 realized by incorporating TiO<sub>2</sub> into the coatings, however, increasing the content has been shown to have  
 246 an adverse effect on its adhesion and scratch resistance. Coating with 10 vol. % of TiO<sub>2</sub> was found to be  
 247 the optimal composition for the composite, resulting in a crack-free surface, as well as good clarity. With  
 248 increasing numbers of coating layers, the elastic modulus and hardness drastically increased to 5.9 GPa and  
 249 897 MPa, respectively, as compared to those of the pristine PC substrate. By utilizing a two-way susceptor-  
 250 aided MW heating mechanism, deposition of composite coatings with different dielectric properties have  
 251 been successfully demonstrated. The hybrid MW cladding protocol would offer a fast and effective process  
 252 for composite coating fabrications.

253 **Declarations**

254 Conflict of Interest: The authors declare that they have no conflict of interest.

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- 325
- 326

327 **List of figures**

328 **Fig. 1** (a) Schematic of hybrid MW cladding process, and comparison between (b) hybrid MW cladding  
329 method and (c) direct MW curing.

330 **Fig. 2** Optical microscopy images and SEM images (inset) of (a) S1-0T, (b) S2-5T, (c) S3-10T, (d) S4-30T  
331 and (e) S6-100T samples.

332 **Fig. 3** Optical microscopy image of (a) S3-10T, (b) S5-50T and (c) S6-100T samples after cross-cut tape  
333 test.

334 **Fig. 4** SEM image of cross-sections (a) Sample S3-10T and (b) Sample S6-100T.

335 **Fig. 5** UV-vis transmission spectra of different sol-gel compositions.

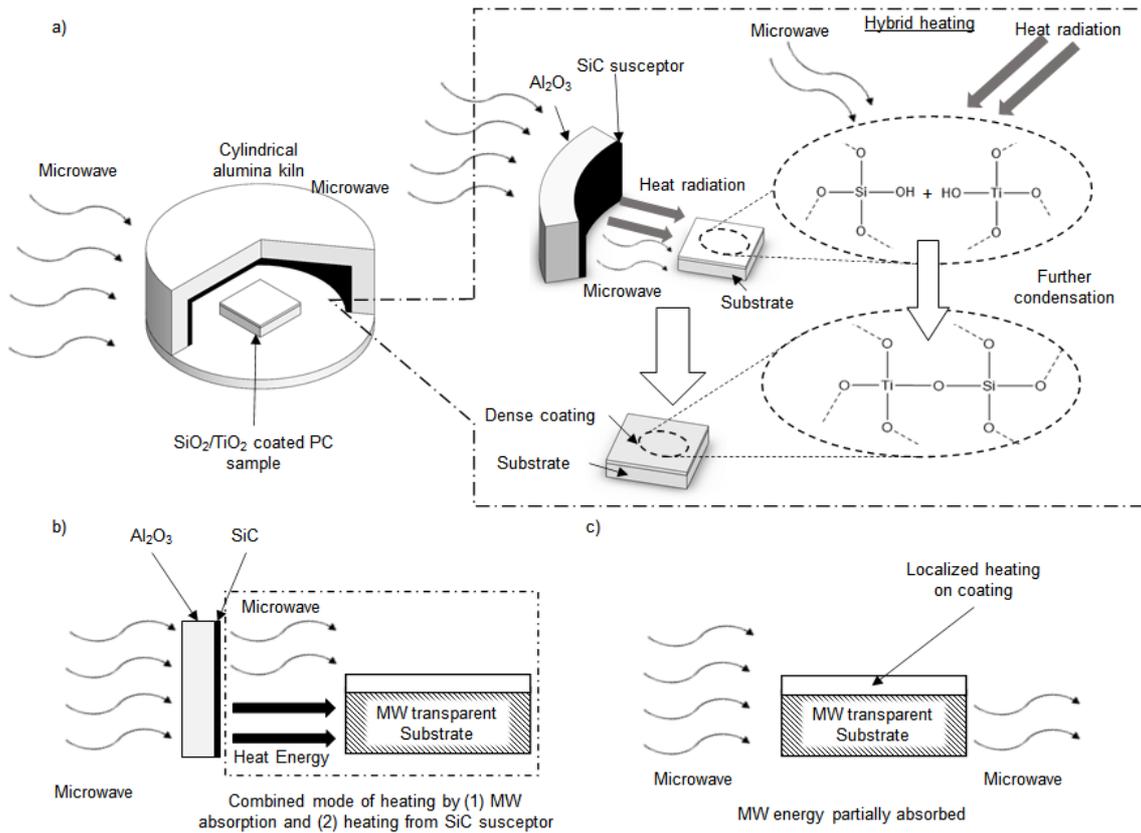
336 **Fig. 6** Elastic modulus ( $E$ ) and hardness ( $H$ ) for pristine PC, 1 layer, 3 layers, and 5 layers of coating (S3-  
337 10T).

338 **Fig. 7** Variation of (a) average frictional coefficient, and (b) penetration depth of pristine PC, 1 layer, 3  
339 layers and 5 layers of coating (S3-10T) during nanoscratch test.

340 **Fig. 8** Normal force against penetration depth for a) 3 layers and b) 5 layers of coating during nanoscratch  
341 test.

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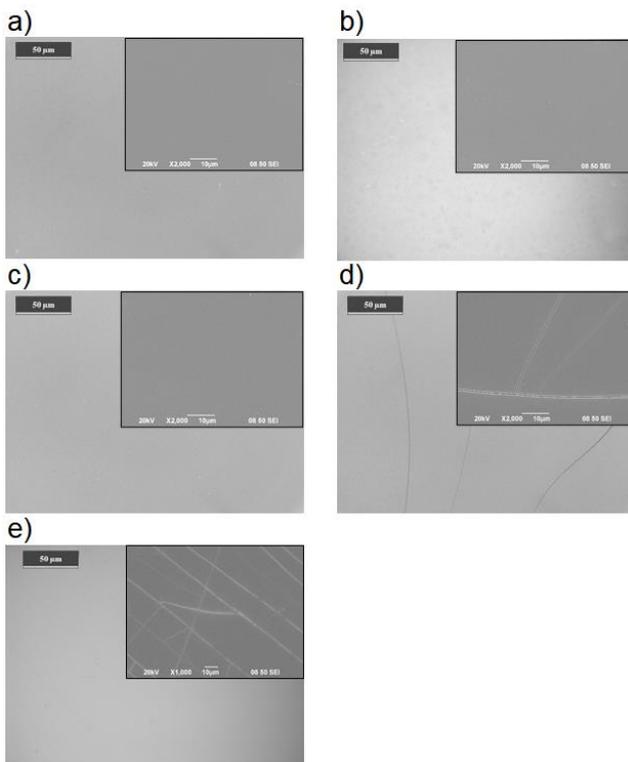


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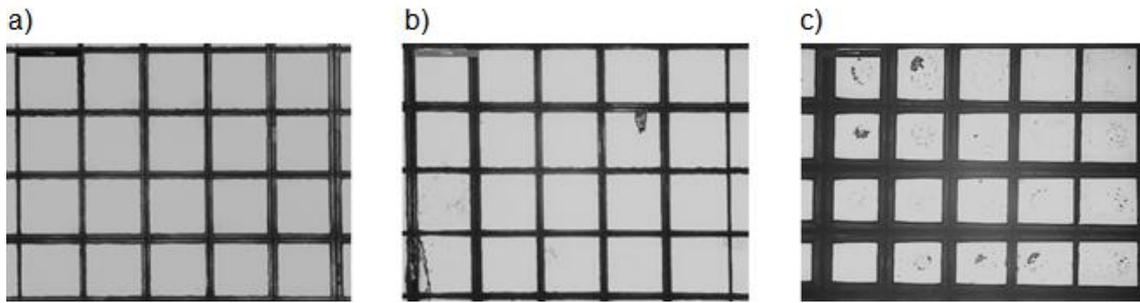
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**Fig. 2** Optical microscopy images and SEM images (inset) of (a) S1-0T, (b) S2-5T, (c) S3-10T, (d) S4-30T and (e) S6-100T samples.

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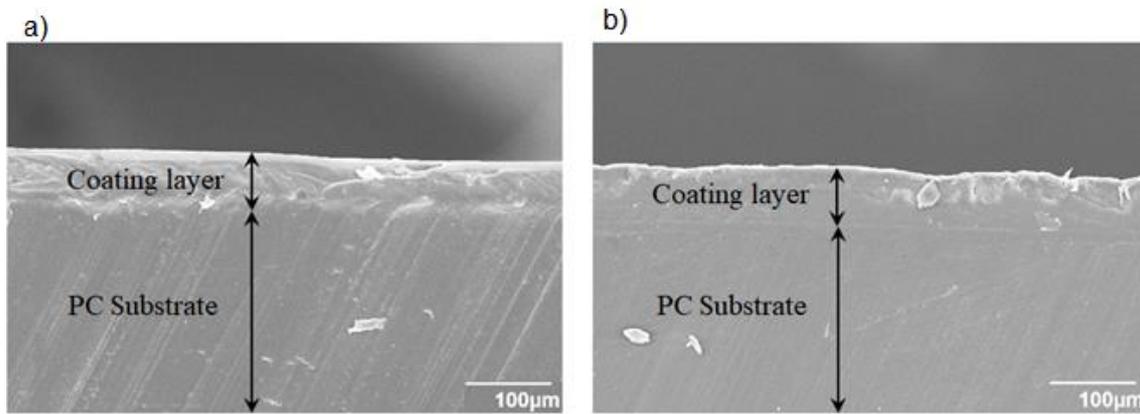


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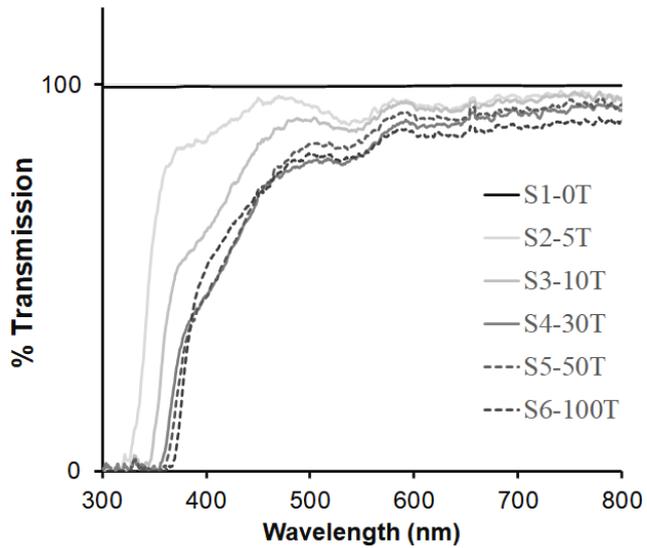
**Fig. 3** Optical microscopy image of (a) S3-10T, (b) S5-50T and (c) S6-100T samples after cross-cut tape test.



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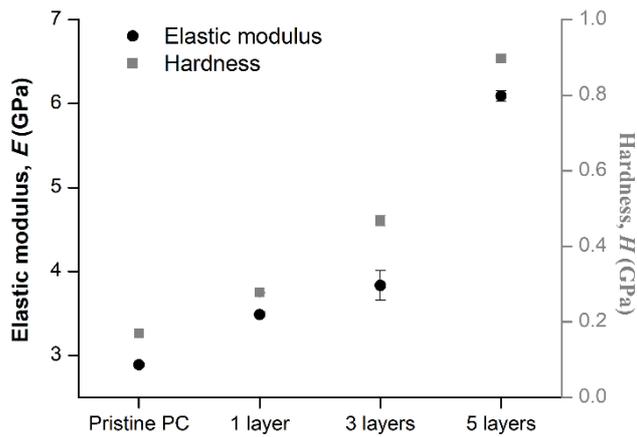
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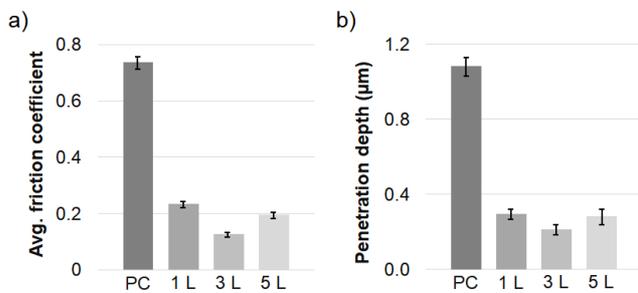
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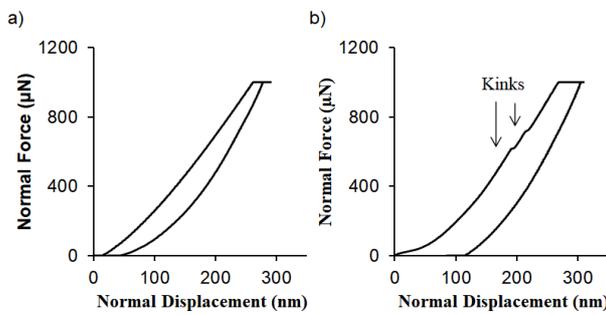
**Fig. 5** UV-vis transmission spectra of different sol-gel compositions.



358  
 359 **Fig. 6** Elastic modulus ( $E$ ) and hardness ( $H$ ) for pristine PC, and S3-10T samples with 1 layer, 3 layers,  
 360 and 5 layers of the coating.  
 361



362  
 363 **Fig. 7** Variation of (a) average frictional coefficient, and (b) penetration depth of pristine PC, and S3-10T  
 364 samples with 1 layer, 3 layers and 5 layers of the coating for nano-scratch test.



365  
 366 **Fig. 8** Normal force against penetration depth for the S3-10T sample with a) 3 layers and b) 5 layers of  
 367 the coating for nano-scratch test.  
 368