

1 **Development of poly(vinyl alcohol) /starch/ethyl lauroyl arginate**  
2 **blend films with enhanced antimicrobial and physical properties for**  
3 **active packaging**

4 Fuqi Wu<sup>a</sup>, Zeguang Zhou<sup>a</sup>, Nannan Li<sup>a</sup>, Yang Chen<sup>a</sup>, Lei Zhong<sup>a\*</sup>, Wing Cheung Law<sup>b</sup>,  
5 Chak Yin Tang<sup>b\*</sup>

6 <sup>a</sup> Guangxi Key Laboratory for Polysaccharide Materials and Modifications, School of  
7 Chemical Engineering, Guangxi University for Nationalities, Nanning, Guangxi,  
8 530006 China.

9 <sup>b</sup> Department of Industrial and Systems Engineering, The Hong Kong Polytechnic  
10 University, Hung Hom, Kowloon, Hong Kong, China.

11

12 **Abstract:** Active packaging films have emerged as alternatives to replace petroleum-  
13 based packaging materials. In this work, a blend of poly(vinyl alcohol) (PVA)  
14 /starch/ethyl lauroyl arginate (LAE) film was prepared to form a polymer composite  
15 structure that possesses enhanced properties. Scanning electron microscopy (SEM)  
16 showed that PVA and starch were compatible, the concentrations of LAE greatly  
17 affected the structural integrity of the films. Fourier transform infrared spectroscopy  
18 (FTIR) and X-ray diffraction showed that the inclusion of LAE did not significantly  
19 affect the intermolecular interactions and crystal structures of the polymer matrix. With  
20 an increase of the LAE content, the tensile strength (TS) was slightly decreased due to  
21 the altered microstructures, the elongation at break (EB) significantly increased  
22 ascribed to the synergistic effect of acetic acid, glycerol and LAE. The values of TS and

23 EB were 17.25 MPa and 586.08% when LAE was 10%. Active films showed good  
24 barrier properties from UV while retaining the transmittance in the visible light region.  
25 The films containing 1% of LAE exhibited antibacterial activity against *Escherichia*  
26 *coli* (*E. coli*) and *Staphylococcus aureus* (*S. aureus*), the inhibition zone of bacterial  
27 growth gradually expanded with increasing LAE content. This study demonstrates the  
28 potential of using LAE as the antibacterial agent for synthesizing natural-based  
29 polymeric films for active packaging applications.

30

31 **Keywords:** Poly(vinyl alcohol) (PVA), cassava starch, ethyl lauroyl arginate (LAE)

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### 33 **1. Introduction**

34 In recent years, food packaging materials based on petroleum-based polymers  
35 have caused serious environmental issues, such as natural resource depletion, waste  
36 production and energy crisis contribution [1]. In response to the demand of  
37 environmental sustainability and the market trends, a lot of research has been devoted  
38 to the development of novel packaging materials based on biopolymers that are  
39 biodegradable and eco-friendly [2-4].

40 Poly(vinyl alcohol) (PVA) is one of the most common types of synthetic  
41 biopolymers that has been widely used in the food packaging industry. PVA is non-toxic  
42 and biodegradable, it was approved in 2003 by the Joint FAO/WHO Expert Committee  
43 on Food Additives and permitted to be used as food packaging material in the United  
44 States [1,5-8]. However, compared with conventional synthetic plastics, PVA has a

45 relatively high price. Considerable efforts have been made to blend PVA with other  
46 renewable materials to fabricate composite materials with lower manufacturing cost  
47 [2,6]. Furthermore, novel PVA composites often possess superior performance and  
48 multiple functions to meet the new requirements of customers [9,10].

49 Starch is one of the most abundant polysaccharides on earth, and plays an  
50 important role in the food packaging industry because of its low cost, non-toxicity and  
51 biodegradability [1,7,11,12]. Moreover, gelatinized starch has good film-forming  
52 ability [13]. The cast starch films are usually tasteless, odorless and transparent,  
53 showing good oxygen barrier properties for food preservation [14]. Among starches  
54 from various botanical origins, cassava starch is suitable to prepare packaging films by  
55 its high paste clarity, high gel stability and low gelatinization temperature [15]. The  
56 primary deficiencies of starch-based films are their poor mechanical properties and  
57 inferior water barrier properties [16]. Blending starch with other biopolymers can form  
58 a new class of composite materials with enhanced physicochemical properties [6,17,18].

59 It has been known that blending PVA with starch can result in a high performance  
60 PVA/starch composite [6,7,19,20]. PVA and starch are compatible since both have  
61 adequate hydroxyl groups in their molecular chains to form strong intermolecular  
62 hydrogen bonds. Therefore, the obtained PVA/starch composite films can achieve  
63 homogeneous structures and improved properties [21]. Cano et al. [6] prepared  
64 composite films by blending PVA with pea starch, in which the tensile strength and  
65 elastic modulus of the obtained films were 32.30 MPa and 689 MPa, respectively,  
66 showing better mechanical properties than both pure PVA and pure starch films. Negim

67 et al. [7] blended PVA with corn starch to fabricate films, and the thermal stability and  
68 biodegradability of the obtained films were superior to those of pure PVA films.

69 Antimicrobial packaging is an important part of active packaging systems. The  
70 incorporation of antimicrobial agents into the packaging films can reduce the microbial  
71 contamination, which helps to extend the shelf life, guarantee the safety and maintain  
72 the quality of packaging food products [22,23,24]. LAE is an antibacterial agent derived  
73 from natural compounds, including lauric acid, L-arginine and ethanol [25,26]. It can  
74 effectively inhibit the growth of a wide range of spoilage microorganisms and food  
75 pathogens [27]. It is a surfactant that is capable of damaging the cell membrane with a  
76 low dose, subsequently altering the cell permeability and inhibiting the growth of  
77 microorganisms [28,29]. The Food and Drug Administration (FDA) classified LAE as  
78 generally recognized as safe (GRAS) in 2005, and the European Food Safety Authority  
79 (EFSA) approved its application as a food preservative in 2007 [28]. Moreover, LAE is  
80 flavorless, colorless, resistant to high temperature and stable at pH 3-7. These  
81 characteristics make it a promising candidate for the development of new antimicrobial  
82 packaging films. The physicochemical properties and antibacterial performances of  
83 active packaging films incorporated with LAE has been investigated in previous studies  
84 [26,30,31]. Theinsathid et al. [4] fabricated PLA films and coated them with LAE, and  
85 found that a small amount of LAE coating (0.07%, w/w) efficiently inhibited the growth  
86 of *Listeria monocytogenes* and *Salmonella typhimurium*. However, while increasing the  
87 amount of LAE coating could improve the antibacterial activity, it compromised the  
88 transparency of the films. Haghighi et al. [32] prepared chitosan / PVA / LAE films and

89 found that the addition of LAE increased the water vapor permeability and water  
90 solubility of the films while decreasing their mechanical properties and transparency.  
91 All films containing different concentrations of LAE (1%, 2.5%, 5% and 10%) showed  
92 antibacterial ability to against *Campylobacter jejuni*, *Escherichia coli*, *Listeria*  
93 *monocytogenes* and *Salmonella typhimurium*.

94 LAE has been previously introduced as the antimicrobial agent into various food  
95 packaging materials [33-37]. However, the effects of LAE on the properties of  
96 PVA/starch blend films are not fully understood. As both PVA and starch are not able  
97 to inhibit the growth of pathogens and microorganisms in food, in this research, we  
98 investigate the effects of incorporated LAE on mechanical and antimicrobial properties  
99 of PVA/starch blend films.

100 In this work, a film based on PVA/starch blends enriched with different  
101 concentrations of LAE was fabricated by a solution casting process. This blend film is  
102 designed for fresh food packaging such as meat and seafood to prolong their shelf life.  
103 It is safer and has longer-lasting protective abilities than conventional packaging  
104 systems because the LAE content in the polymer matrix is progressively released to the  
105 surface of packaged foods rather than directly added to the food formulation. The  
106 microstructure of the obtained blend films was characterized and the mechanical,  
107 optical and water barrier properties were evaluated. Moreover, the antimicrobial  
108 performances of blend films against two major food pathogens, *E. coli* and *S. aureus*  
109 were assessed in vitro. This study aims to develop a novel blend film with effective  
110 antimicrobial properties and enhanced physical properties, which can be potentially

111 used in active food packaging applications.

## 112 **2. Materials and methods**

### 113 **2.1 Materials**

114 Cassava starch (food grade, the amylose-amylopectin ratio was 19:81 determined  
115 by the iodine method) was obtained from Guangxi Hongfeng Starch Co., Ltd. (Nanning,  
116 China). PVA (polymerization degree  $1750\pm 50$ ) was purchased from Sinopharm  
117 Chemical Reagent Co., Ltd. (Shanghai, China). LAE was obtained from Shanghai Bide  
118 Pharmatech Co., Ltd. (Shanghai, China). Glycerol ( $\geq 99\%$ ) and Glacial acetic acid (AR)  
119 were purchased from Guangdong Guanghua Sci-Tech Co., Ltd. (Guangzhou, China).  
120 *E. coli* (CMCC(B)26003) and *S. aureus* (ATCC25922) were obtained from Shanghai  
121 Luwei Technology Co., Ltd. (Shanghai, China). Casein soya bean digest agar (TSA)  
122 and Casein soya bean digest broth (TSB) were purchased from Guangdong Huankai  
123 Microbial Sci-Tech Co., Ltd. (Guangzhou, China).

### 124 **2.2 Preparation of blend films**

125 A starch solution (10%, w/w) was prepared by dissolving the cassava starch  
126 granules in distilled water. Then, the starch solution was gelatinized by continuous  
127 stirring at 80°C for 60 min (430 rpm). Similarly, the PVA solution (10%, w/w) was  
128 prepared by adding PVA powder to distilled water, and continuously stirred at 80°C for  
129 90 minutes (430 rpm). Afterwards, gelatinized starch was mixed with PVA solution at  
130 a ratio of 3:7 (w/w). This ratio was pre-determined by a preliminary experiment, in  
131 which five different starch/PVA ratios (1:9, 3:7, 5:5, 7:3 and 9:1) were prepared. The  
132 sample with ratio 3:7 demonstrated better clarity and potential for further study.

133 Glycerin (20% of starch and PVA, w/w) and glacial acetic acid (20% of starch and PVA,  
134 w/w) were then added to the mixture and stirred at 80°C for 30 min (430 rpm) to obtain  
135 the film-forming solution (FFS). Subsequently, different contents of LAE (1%, 2.5%,  
136 5% and 10% of starch and PVA, w/w) were added to the FFS and stirred at 80°C for  
137 additional 30 min (430 rpm) to reach complete dissolution. After cooling and  
138 eliminating the bubbles in the solution at room temperature, 20 g FFS was cast on a  
139 polymethyl methacrylate plate (120 × 120 mm) and dried at 25°C in an oven for 24 h  
140 to obtain the blend films. Control films were prepared following the same procedure  
141 without adding LAE. All the film samples were stored in a desiccator at room  
142 temperature for at least 2 days before the measurement, and the relative humidity (RH)  
143 of the desiccator was adjusted to 53% by using saturated Mg(NO<sub>3</sub>)<sub>2</sub>.

### 144 **2.3 Morphology and structural characterizations**

145 The morphology and microstructure of the obtained blend films were  
146 characterized by scanning electron microscopy (SEM), Fourier transform infrared  
147 spectroscopy (FTIR) and X-ray diffraction (XRD).

148 The morphology of a cross-section of the blend film was observed by a scanning  
149 electron microscope (Supra 55, Zeiss, Oberkochen, Germany) operated at an  
150 acceleration voltage of 2 kV and 2000× magnification. Fractured samples were  
151 collected from the tensile test, fixed with a double-sided adhesive and sputter-coated  
152 with gold for the observation.

153 A Fourier transform infrared spectroscopy (Nicolette Magna 550II, GMI, Ramsey,  
154 MN, USA) was used to characterize the interactions between cassava starch, PVA and

155 LAE in the blend films. Samples were ground and mixed with KBr powder, then the  
156 mixtures were compressed to form the testing disc. Spectra were recorded in the 4000-  
157 400  $\text{cm}^{-1}$  wavenumber range, with a spectral resolution of 8  $\text{cm}^{-1}$ .

158 The X-ray diffraction patterns were collected by an X-ray diffractometer  
159 (MiniFlex 600, Rigaku, Tokyo, Japan) equipped with a  $\text{K}\alpha$  Cu radiation source operated  
160 at 45 kV and 15 mA. The films were cut into squares (20 × 20 mm) and fixed on a  
161 sample stage (glass plate). Samples were scanned between  $2\theta=3^\circ$  to  $40^\circ$  with a step size  
162 of  $0.02^\circ$ . The crystallinity index (CI) of blend films was calculated by Equation (1) [38].

$$163 \text{ CI (\%)} = A_c / (A_c + A_a) \times 100\% \quad (1)$$

164 where  $A_c$  represents the crystalline area, and  $A_a$  represents the amorphous area.

## 165 **2.4 Thickness and mechanical properties**

166 The film thickness was measured using a spiral micrometer (211-101, Jingyou  
167 Mould Hardware Co., Ltd. Dongguan, China). Six random positions were selected and  
168 measured on the films, one at the center and the other five at the edges. The thickness  
169 values were obtained based on the means of these six positions [39].

170 The tensile strength (TS) and elongation at break (EB) of blend films were  
171 determined by a universal testing machine (JDL-1000N, Tianfa Instruments Co., Ltd.  
172 Yangzhou, China) following the method used by Liu [40] with modifications.  
173 Rectangular strips (20 × 60 mm) were cut out from the sample film. Initial grip  
174 separation and overhead speed were set as 50 mm and 10 mm/min, respectively. TS was  
175 expressed in MPa and calculated by dividing the maximum load by the cross-sectional  
176 area of the sample. EB was expressed in percentage (%) and calculated by dividing the



177 film elongation at break by the initial distance of the grips. The calculations were  
178 undertaken by the built-in software. All tests were repeated at least five times for each  
179 sample.

## 180 **2.5 UV barrier properties and opacity**

181 UV barrier properties and the opacity of blend films were investigated by an UV-  
182 VIS spectrophotometer (UV-2600, Shimadzu, Kyoto, Japan). The film samples were  
183 cut into strips (10 × 40 mm) for the measurement. Their optical properties were  
184 evaluated at the UV wavelength range from 200 to 400 nm, and the visible light  
185 wavelength range from 400 to 800 nm. The opacity values were calculated according  
186 to Equation (2) [32].

$$187 \text{ Opacity value} = (-\log T_{600}) / H \quad (2)$$

188 where  $T_{600}$  is the fractional transmittance of the blend film at a wavelength of 600 nm,  
189 and  $H$  is the thickness of the blend film. All measurements of optical properties were  
190 performed in triplicate.

## 191 **2.6 Moisture absorption (MA) and water vapor permeability (WVP)**

192 MA of the blend films was measured by using Abral's method with some  
193 modifications [41]. The square film strips (20 × 20 mm) were cut from the film samples  
194 and dried in a vacuum oven at 50°C for 24 h to eliminate moisture. Afterwards, the  
195 dried film strips were stored in an airtight desiccator with 75% RH (using saturated  
196 NaCl) for 24 h to reach equilibrium and taken out for weighting. MA values were  
197 calculated using Equation (3) and averaged based on three replicates.

$$198 \text{ MA (\%)} = (M_f - M_i) / M_i \times 100\% \quad (3)$$

199 where  $M_f$  represents the final weight (g) and  $M_i$  represents the initial weight (g) of film  
200 strips. All measurements of moisture absorption were performed in triplicate.

201 WVP of the blend films was examined using the method in ASTM E96-00 with  
202 some modifications [41]. A plastic cup (50 mm diameter) containing distilled water (30  
203 mL, 100% RH) was covered by the sample film (60 mm) and completely sealed with  
204 Vaseline for avoiding moisture leakage. Then, the initial weight of the cup was weighed  
205 and placed into an airtight desiccator containing 1000g silica. An incubator was used to  
206 maintain the desiccator at 25 °C. The cup was weighed every 3 h until its weight  
207 stabilized. WVP was determined by the relation between the weight loss and storage  
208 time and calculated using Equation (4).

$$209 \text{ WVP (g}\cdot\text{m}^{-1}\cdot\text{s}^{-1}\cdot\text{Pa}^{-1}) = (dW \times H) / (dt \times dP \times S) \quad (4)$$

210 where  $dW$  represents the weight loss of the plastic cup system (g),  $H$  represents the  
211 mean film thickness (m),  $dt$  represents the time change (s) under the partial water vapor  
212 pressure gradient ( $dP = 2533$  Pa),  $S$  represents the permeation area of the film sample  
213 ( $\text{m}^2$ ). Tests were repeated four times and averaged to obtain WVP values.

## 214 **2.7 In vitro antimicrobial activity (Agar diffusion assay)**

215 In vitro antibacterial activity of the control films and active films with different  
216 concentrations of LAE against *E. coli* and *S. aureus* was assessed according to the agar  
217 diffusion method used by Theinsathid et al., with some modifications [4]. The films  
218 were cut into a disc shape with a diameter of 8 mm. Subsequently, the disc shape sample  
219 and the TSA plate were each sterilized with UV for 2 h. Then the TSA agar plates were  
220 streaked with 0.1 mL TSB inoculant containing  $10^8$  CFU/mL of the tested bacteria (*E.*

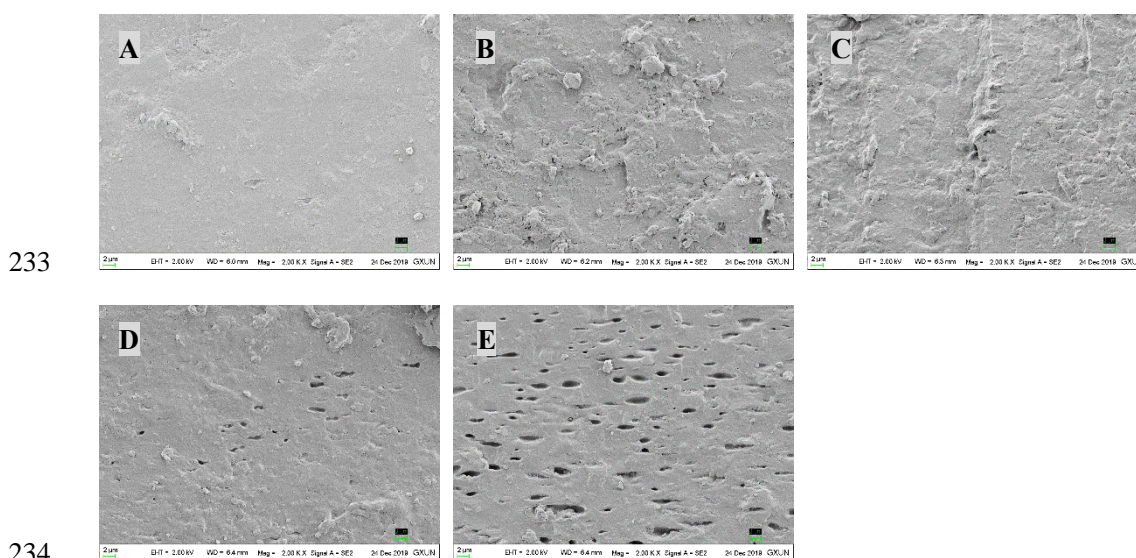
221 *coli* and *S. aureus*). Finally, the sample was placed on the surface of the TSA agar plate  
222 and incubated at 37°C for 24 h. The antibacterial activity was evaluated by the average  
223 diameter (mm) of the inhibition zone around the disc shape sample. All tests were  
224 repeated three times.

## 225 2.8 Statistical analysis

226 The experimental data were expressed as the mean  $\pm$  SD (standard deviation).  
227 Analysis of variance (ANOVA) was performed using SPSS statistical analysis software  
228 (SPSS Statistics 26 for Windows, IBM, New York, USA). Differences between the  
229 mean values of the properties of blend films were detected by Tukey's multiple range  
230 test ( $p \leq 0.05$ ).

## 231 3. Results and Discussions

### 232 3.1 Morphology

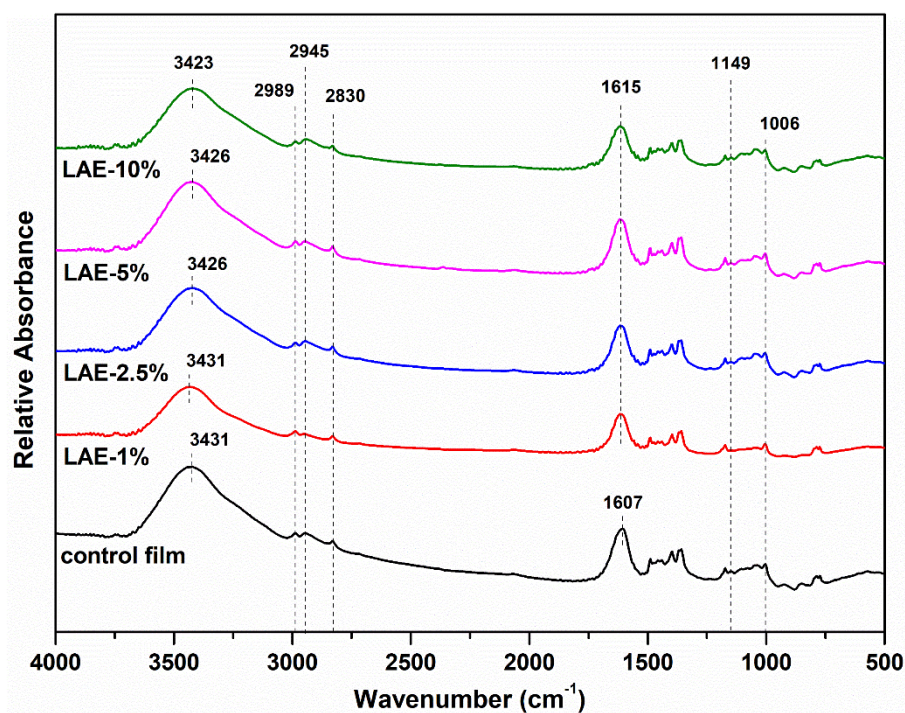


235 Figure 1. SEM images of films with different LAE content. (A without LAE, B-E LAE content  
236 was 1%, 2.5%, 5% and 10% of polymer matrix, w/w )

237 The cross-section images of the control film (LAE-0%) and the active films with

238 different concentrations of LAE are shown in Figure 1. It can be seen that the addition  
239 of LAE greatly affected the surface morphology of the blend films. The cross-section  
240 of the control film without LAE (Figure 1 A) is relatively smooth and homogeneous as  
241 compared with other LAE containing samples. It indicates that PVA and cassava starch  
242 have good compatibility in blending, which could be due to the formation of the  
243 hydrogen bonding between the PVA and cassava starch [21]. Small aggregations and  
244 rough surface were observed at the cross-section of active films incorporated with LAE  
245 (Figure 1B-1E), when the concentrations of LAE were 1% and 2.5%. It might be due  
246 to the aggregation of LAE in the film matrix, and the interaction between PVA and  
247 starch being disrupted by LAE, leading to the destruction of the compact structure of  
248 the polymer blends [30,32,42]. When the concentration of LAE was 5%, pores appeared  
249 on the film cross-section, and when the concentration of LAE was 10%, the number of  
250 pores clearly increased. This can be explained by the enhanced surfactant effect of LAE.  
251 Due to the high surface tension, small air bubbles were generated and sustained during  
252 the blending process in the film-forming solution, and the high viscosity of the mixture  
253 solution made it difficult to eliminate them, thereby causing the formation of pores in  
254 the film matrix [30].

### 255 **3.2 FTIR**



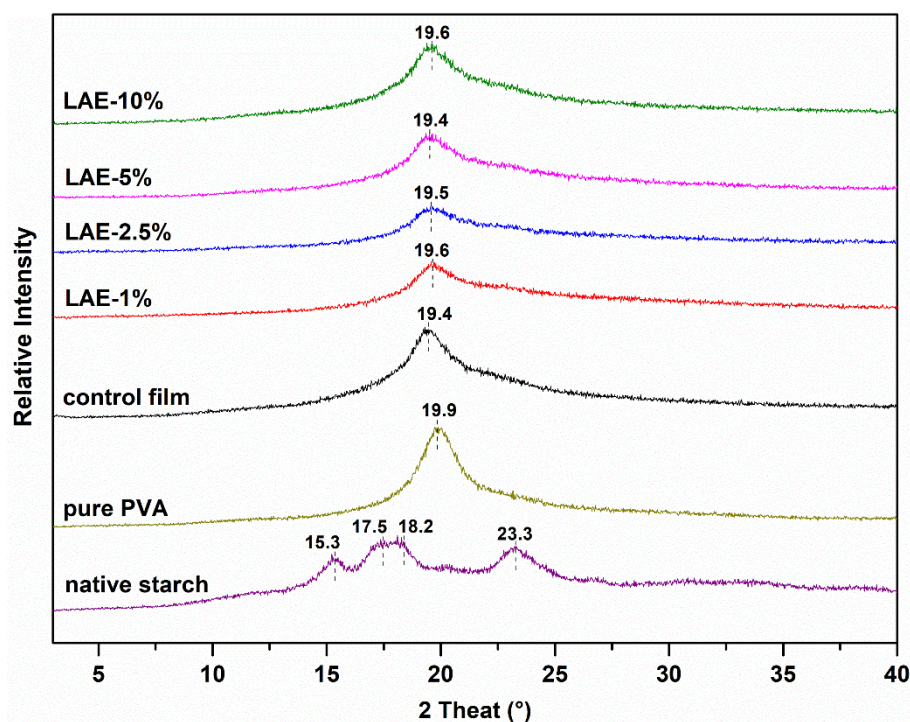
256

257 Figure 2. FTIR spectra of films with different LAE content (0, 1%, 2.5%, 5% and 10% of  
 258 polymer matrix, w/w).

259 Figure 2 shows the FTIR spectra of the control film and active films incorporated  
 260 with different concentrations of LAE. The broad adsorption band between 3200 and  
 261 3600  $\text{cm}^{-1}$  is attributed to the O-H stretching vibrations, indicating the intermolecular  
 262 hydrogen bonds form between PVA and cassava starch [7]. It is noted that there is a  
 263 slightly peak shift of the O-H stretching vibrations of different PVA/starch blends  
 264 appeared at this band. With the increase of LAE content, the peak shifted to a lower  
 265 wavenumber, which could be due to the hydrogen bond interaction between PVA and  
 266 starch. The characteristics peaks at 2945  $\text{cm}^{-1}$ , 2830  $\text{cm}^{-1}$  and 2989  $\text{cm}^{-1}$  are related to  
 267 the C-H stretching, symmetric and antisymmetric  $-\text{CH}_2$  stretching, respectively [7,32].  
 268 The peak observed around 1615  $\text{cm}^{-1}$  is associated to the starch with bound water  
 269 molecules, and the peak at 1006 $\text{cm}^{-1}$  is assigned to the C-O stretching vibrations of C-  
 270 O-C groups in the glucose ring of starch. The C-O stretching of acetate groups of PVA

271 is located at  $1149\text{ cm}^{-1}$ . The shapes of these five curves are very similar, suggesting the  
272 addition of LAE has no significant influence on the molecular structure and the  
273 intermolecular interactions of the blend films.

### 274 3.3 XRD



275  
276 Figure 3. XRD patterns of starch, PVA and films with different LAE content (0, 1%, 2.5%, 5%  
277 and 10% of polymer matrix, w/w).

278 The XRD patterns of the starch, PVA, PVA/starch control film and active films  
279 incorporated with different concentrations of LAE are presented in Figure 3. The  
280 diffraction peaks of starch granules were located at  $2\theta=15.3^\circ$ ,  $17.5^\circ$ ,  $18.2^\circ$  and  $23.3^\circ$   
281 [43], the diffraction peak of PVA were located at  $19.9^\circ$ . After blending starch with PVA,  
282 the peaks of the starch disappeared and only the main diffraction peak at  $19.4^\circ$  to  $19.6^\circ$   
283 of PVA remained, indicating the addition of starch did not affect the crystalline structure  
284 of PVA. The reason might be that the starch was gelatinized and the crystalline

285 structures were destroyed during the blending process, resulting in the diffraction  
 286 patterns of the blends were being dominated by the crystalline structures of PVA.  
 287 Furthermore, there were no major changes of the diffraction patterns of the blend films  
 288 in the presence of LAE. The CI of the control film and blend films with different  
 289 additions (1%, 2.5%, 5%, 10%) of LAE were 12.09%, 11.87%, 12.37%, 12.45% and  
 290 11.44%, respectively. The CI values were close, indicating the crystalline structure of  
 291 the blends were not significantly affected by the addition of LAE contents.

### 292 **3.4 Thickness and mechanical properties**

293 Thickness is one of the crucial parameters to determine the mechanical, optical  
 294 and barrier properties of blend films. Table 1 lists the thickness values of the control  
 295 and active films. It was found that thickness values ranged from 0.131 mm to 0.141  
 296 mm, and the thickness of the blend films was not significantly affected by the inclusion  
 297 of LAE. This was in agreement with the results of Rubilar et al. [44]. In their work,  
 298 LAE had a high solubility in the PVA/starch film matrix. In contrast, Haghghi et al.  
 299 [32] reported that LAE might loosen the microstructure in the chitosan/PVA films,  
 300 leading to an increase in the film thickness when using high concentrations of LAE.  
 301 This inconsistency in different studies suggests that the fabrication process might  
 302 strongly affect the thickness of blend films, despite the type of film matrix.

303

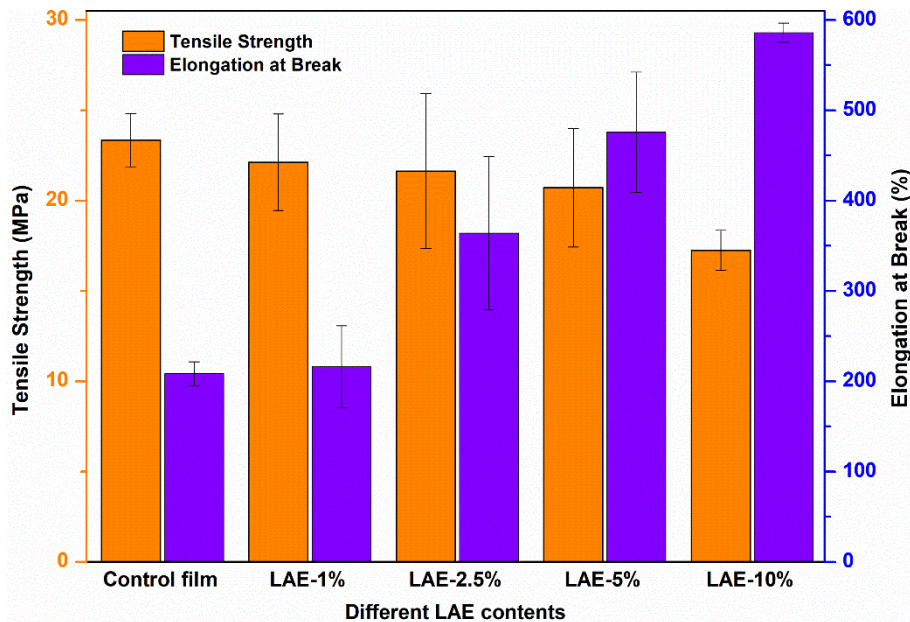
304 Table 1. Thickness of films based on control film and films with different LAE content (1%,  
 305 2.5%, 5% and 10% of polymer matrix, w/w).

Film sample	Control film	LAE-1%	LAE-2.5%	LAE-5%	LAE-10%
-------------	--------------	--------	----------	--------	---------

Thickness (mm)	0.14±0.03	0.14±0.02	0.14±0.01	0.13±0.02	0.14±0.02
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306 Values are shown as mean ± SD (n=3), and there is no statistical difference.

307



308

309 Figure 4. Light transmittance of the blend films with different LAE content (0, 1%, 2.5%, 5%  
310 and 10% of polymer matrix, w/w).

311 Mechanical properties are important for evaluating the applicability of blend films  
312 as packaging materials. In addition, adequate tensile strength and extensibility are  
313 necessary for most food packaging applications [39]. The TS and EB of blend films  
314 with different LAE contents are shown in Figure 4. It is evident that the inclusion of  
315 LAE strongly influences the mechanical properties of blend films, with an increase in  
316 the concentration of LAE, while the TS ( $p \leq 0.05$ ) of the film decreased while the EB  
317 ( $p \leq 0.05$ ) increased.

318 The presence of LAE slightly weakened the tensile strength of the blend films.  
319 The control film had the highest TS value, and decreased from 23.34 MPa to 17.25 MPa  
320 when incorporating 10% of LAE. This tendency of the tensile properties was consistent



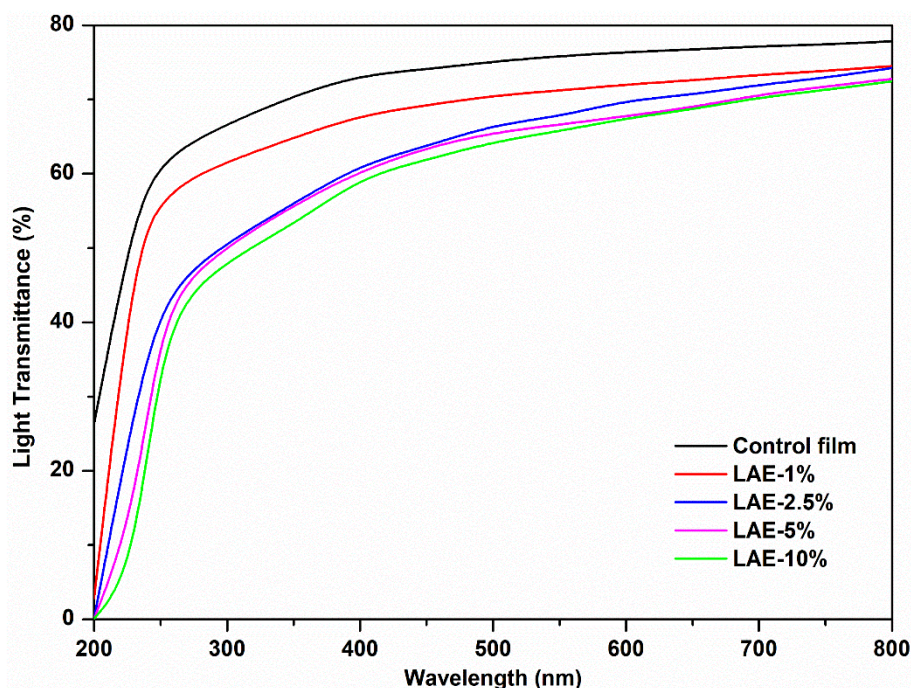
321 with the SEM analysis. The integrity and continuity of film matrix were destroyed by  
322 the incorporation of LAE, and small aggregates and pores emerged within the structure,  
323 thereby resulting in a decrease in the tensile strength [45]. Rubilar et al. [44] also  
324 reported that the inclusion of LAE reduced the tensile performances of chitosan-PVA  
325 films.

326 The introduction of LAE significantly improved the extensibility of blend films.  
327 The EB value increased from 208.45% to 586.08% when adding up to 10% LAE in this  
328 work. It might be ascribed to the synergistic effect of acetic acid, glycerol and LAE. It  
329 has been reported that the amylose and amylopectin chains in starch molecules can be  
330 hydrolyzed by the acetic acid, leading to the higher solubility of starch and reduced  
331 viscosity of the suspension. Therefore, both film-forming abilities and cohesion of the  
332 gel structure can be improved. [46,47]. Furthermore, the hydrogen bonds generated  
333 among glycerol, LAE and polymers substituted the intermolecular bonds in the polymer  
334 matrix, which promoted the sliding of long polymer chains during extension, and thus  
335 led to better flexibility of blend films [48]. The elongation properties of our active films  
336 were superior to some previous studies with similar formulas. The highest EB value of  
337 chitosan/PVA/LAE films prepared by Haghghi et al. was 39% [32], and the highest EB  
338 value of starch/LAE films prepared by Motta et al. was 119.63% [48]. It was also noted  
339 that the mechanical properties of our active films with LAE were comparable with the  
340 high density polyethylene (HDPE, 22 MPa, 500 %) films widely used in the market.

341 In previous studies, Haghghi et al. [39] reported that low content of LAE (0.1%,  
342 v/v) did not influence the mechanical properties of chitosan–gelatin active films.

343 Rubilar et al. [44] found that the inclusion of LAE clearly increased both the TS and  
344 EB values of chitosan films. Moreno et al. [45] observed that the addition of LAE  
345 decreased the TS values and notably reduced the EB values when preparing films by  
346 compress molding. These controversial results in the literature indicate that the  
347 mechanical properties of active films containing LAE are determined by multiple  
348 factors, including the type of polymer matrix, the amount and conditions of LAE and  
349 other additives, as well as the fabrication process.

### 350 3.5 UV barrier properties and opacity



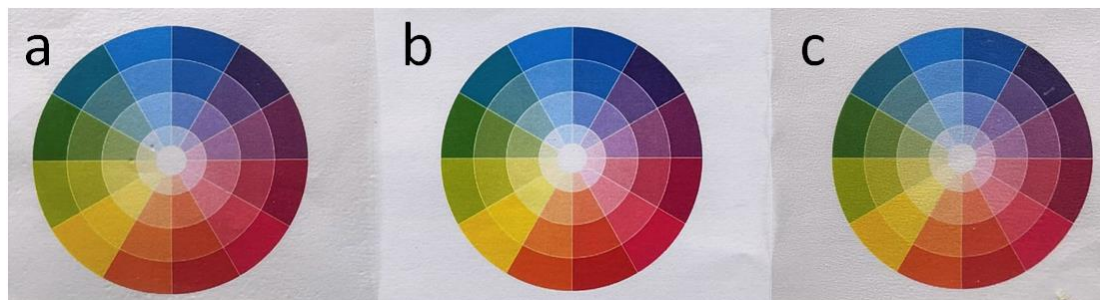
351  
352 Figure 5. Light transmittance of the blend films with different LAE content (0, 1%, 2.5%, 5%  
353 and 10% of polymer matrix, w/w).

354 The light transmittances of the control and active films with different  
355 concentrations of LAE in the wavelength range between 200 and 800 nm are presented  
356 in Figure 5. It was found that the active films possess lower UV light transmittances  
357 (200-400 nm) as compared to the control film. At a wavelength of 200 nm, the light

358 transmittance of active films was close to 0%, which means that active films performed  
359 as effective UV barriers. In food packaging applications, the excellent UV barrier  
360 properties of active films can help to prolong the shelf life of food products, and can  
361 reduce UV-induced oxidations, and preserve the nutrition, flavors and colors of foods  
362 [49,50].

363 In the VIS region (400-800 nm), the light transmittances of active films were  
364 slightly lower than that of the control film. At a wavelength of 800nm, the light  
365 transmittance of the control film was 77.84%, and the light transmittances of active  
366 films (containing 1%, 2.5%, 5% and 10% LAE) were 74.53%, 74.25%, 72.80% and  
367 72.45%, respectively. At the wavelength of 600 nm, the opacity value of our active  
368 films was 0.99, 1.10, 1.19 and 1.21, respectively, similar to the values of  
369 chitosan/PVA/LAE films and chitosan/gelatin/LAE films reported in previous studies  
370 [39].

371 Packaging materials were often used for wrapping different colors of food  
372 products. To evaluate the transparency of active films on different colors, color charts  
373 were printed on a white paper, pictures were taken when the color charts were covered  
374 by an active film, no film and a control film, respectively. Figure 6 shows that all three  
375 color charts were clear, the clarity was slightly decreased when the charts were covered  
376 by the films. There was no significant difference of color, and the difference between  
377 color charts covered by the active film (a) and the control film (c) in the picture was  
378 difficult to tell. This phenomena observed in the picture was in agreement with  
379 transmittance results shown in Figure 5.



380

381 Figure 6. To evaluate the transparency and clarity of the blend films, a color charts  
 382 covered by (a) LAE-10% blend film, (b) no film (b) and (c) control film (i.e. LAE-0%),  
 383 respectively.

384 In general, with an increase in the LAE concentration, light transmittances of  
 385 active films slightly decrease [39], likely due to the aggregation of LAE. Larger  
 386 aggregates formed by more LAE inhibits light transmission. Transparency also plays  
 387 an important role in food packaging, showing the original appearance of food to the  
 388 consumers. Since the maximum transmittance of all active films in the VIS are higher  
 389 than 70%, and their opacity values of them at 600 nm are less than 5, as shown in Table  
 390 2, all active films obtained in our study can be considered as transparent [32].

391

392 Table 2. Opacity value of films based on control film and films with different LAE content (1%,  
 393 2.5%, 5% and 10% of polymer matrix, w/w).

Film sample	Opacity value* (600 nm)
Control film	0.84±0.01 <sup>a</sup>
LAE-1%	0.99±0.05 <sup>b</sup>
LAE-2.5	1.10±0.07 <sup>bc</sup>
LAE-5%	1.19±0.02 <sup>c</sup>
LAE-10%	1.21±0.05 <sup>c</sup>

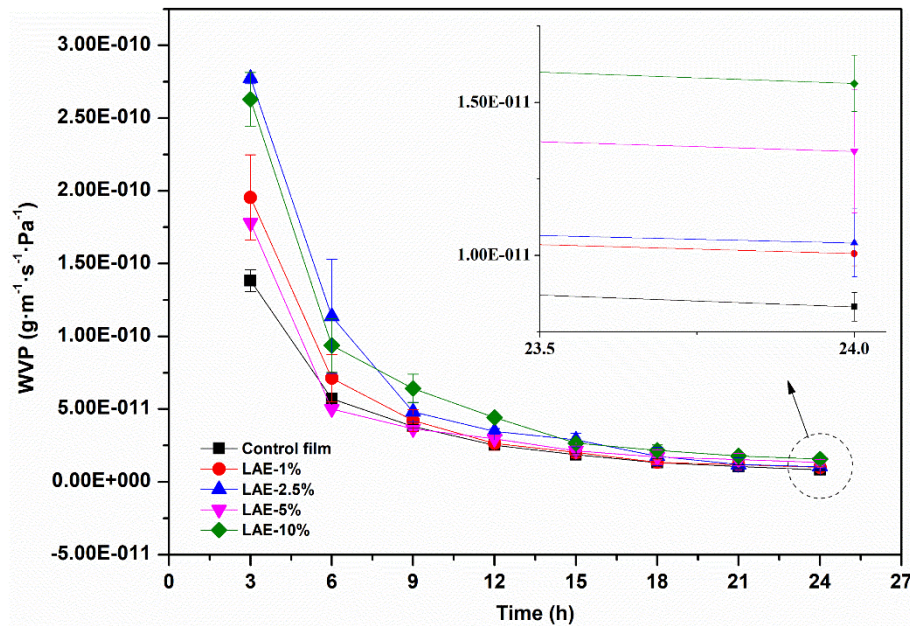
394 Values are shown as mean  $\pm$  SD (n=3).

395 \*The mean values in the same column of the table with different superscript letters (a, b, c, d)

396 indicate significant differences by Turkey's test at  $p \leq 0.05$ .

397

### 398 3.6 Moisture adsorption (MA) and water vapor permeability (WVP)



399

400 Figure 7. WVP of films with different LAE content (0, 1%, 2.5%, 5% and 10% of polymer

401 matrix, w/w).

402 Moisture adsorption and water vapor permeability are two significant factors that

403 affect the performances of food packaging materials [44,51], so it is crucial to

404 investigate the water barrier property of active films.

405

406 Table 3. MA and WVP of films based on control film and films with different LAE content (1%,

407 2.5%, 5% and 10% of polymer matrix, w/w).

Film sample	MA* (%)	WVP* ( $\text{g}\cdot\text{m}^{-1}\cdot\text{s}^{-1}\cdot\text{Pa}^{-1}$ )
-------------	------------	---

Control film	17.54±1.15 <sup>a</sup>	8.32×10 <sup>-12</sup> ±4.70×10 <sup>-13</sup> <sup>a</sup>
LAE-1%	22.92±0.77 <sup>b</sup>	1.01×10 <sup>-11</sup> ±4.07×10 <sup>-13</sup> <sup>b</sup>
LAE-2.5	23.44±0.94 <sup>b</sup>	1.04×10 <sup>-11</sup> ±1.11×10 <sup>-12</sup> <sup>b</sup>
LAE-5%	24.49±0.87 <sup>c</sup>	1.34×10 <sup>-11</sup> ±2.02×10 <sup>-12</sup> <sup>c</sup>
LAE-10%	25.46±1.12 <sup>d</sup>	1.56×10 <sup>-11</sup> ±9.21×10 <sup>-13</sup> <sup>d</sup>

408 Values are shown as mean ± SD (n=3).

409 \*The mean values in the same column of the table with different superscript letters (a, b, c, d)

410 indicate significant differences by Turkey's test at  $p \leq 0.05$ .

411

412 Table 3 presents the MA values of the control film and active films with different  
 413 concentrations of LAE. The addition of LAE led to an increase in MA values. The MA  
 414 value of the control film was 17.54%, and that of the active film with 10% LAE content  
 415 reached to 25.46%. This might be due to the high water affinity of LAE. The oil-water  
 416 equilibrium partition coefficient of LAE was low (less than 0.1), which made it easy to  
 417 capture the water molecules [26,44].

418 Figure 7 shows the WVP of control films and active films with different  
 419 concentrations of LAE. The incorporation of LAE led to higher WVP values. After 24  
 420 h, the WVP value of the control film was  $8.32 \times 10^{-12} \text{ g} \cdot \text{m}^{-1} \cdot \text{s}^{-1} \cdot \text{Pa}^{-1}$ , while that of the  
 421 active film with 10% LAE content was  $1.56 \times 10^{-11} \text{ g} \cdot \text{m}^{-1} \cdot \text{s}^{-1} \cdot \text{Pa}^{-1}$ . Besides the  
 422 hydrophilic properties of LAE, the water absorbed by starch and PVA during the  
 423 swelling might break the hydrogen bonding and destroy the crystal structure of PVA  
 424 matrix, thereby accelerating the movement of the polymer chains, resulting in an  
 425 increase in the passage of water molecules [32,44]. Moreover, porous structures

426 emerging in the active films with high LAE content might also promote the transfer of  
 427 water molecules. We have compared the LAE-10% sample with reported composite  
 428 based packaging materials in the literature (Table 4).

429 Table 4. Comparison of physical properties with recently reported packaging films.

Composition	TS (MPa)	EB (%)	MA (%)	WVP	Opacity value	Ref.
Chitosan/ polyvinyl alcohol/ 2.5% LAE	34.5±2.4	39.0±2.6	--	21±0.02 (g mm/ kPa day m <sup>2</sup> )	1.6±0.2	[1]
Modified starches/ 0.05% LAE	2.65±0.33	119.63±0.21	--	425.56±13.05 (g water m <sup>-2</sup> / day)	7.98±0.09	[2]
Chitosan/ gelatin/ 0.1% LAE	31.86±1.66	28.69±0.54	--	0.018±0.00031 (g mm/ kPa day m <sup>2</sup> )	1.43±0.22	[3]
PVA/ starch/ coconut shell extract/ sepiolite clay	32.77±4.98	15.57±1.46	--	0.90±0.06 (g mm/ m <sup>2</sup> day kPa)	--	[4]
PVA/ starch/ ZnO	7.32±0.45	210.3±12.75	5.49±0.11	--	--	[5]
PVA/starch/TiO <sub>2</sub>	8.85±0.85	221.7±15.63	7.78±0.19	--	--	[5]
PVA/starch/gelatin protein hydrolysate	4.27±0.33	208.5±17.62	6.52±0.23	--	--	[5]
Starch/ PVA/ cinnamon essential oil	52.6±4.1	241.8±22.9	--	0.25±0.02 (g mm kPa <sup>-1</sup> h <sup>-1</sup> m <sup>-2</sup> )	23.8±0.5 (AUmm)	[6]
Pineapple peel extract/ PVA/ starch	29.09±3.53	33.39±2.59	--	0.75 (mm/m <sup>2</sup> day kPa)	--	[7]
PVA/ starch	42.41	209.33	193	--	--	[8]
PVA /starch/10%LAE	17.25	586.08	25.46±1.12	1.56×10 <sup>-11</sup> ±9.21×10 <sup>-13</sup> (g·m <sup>-1</sup> ·s <sup>-1</sup> ·Pa <sup>-1</sup> )	1.21±0.05 (@ 600 nm)	This work

430

### 431 3.7 In vitro antimicrobial activity

432 The antimicrobial properties of the control and active films with different  
 433 concentrations of LAE against *E. coli* and *S. aureus* were assessed by disk diffusion  
 434 assay. As shown in Table 5, the control film did not exhibit inhibitory effects against  
 435 the tested bacteria. No inhibition zone appeared in the TSA plate of the control film

436 overgrown with *E. coli* and *S. aureus*, indicating that starch and PVA are not able to  
437 inhibit the growth of bacteria, as has been reported previously [52,53]. All active films  
438 containing LAE show antimicrobial effects. Moreover, the inhibition zones of tested  
439 bacteria gradually expanded with an increase in the LAE concentration. When the LAE  
440 concentration increased from 1% to 10%, the diameter of the inhibition zone increased  
441 from 12.2 mm to 29.07 mm for *E. coli*, and from 19.23 mm to 43.37 mm for *S. aureus*,  
442 respectively. The high antimicrobial efficiency of LAE could be ascribed to its role as  
443 an effective cationic surfactant on the cell membrane. Even at a low concentration, LAE  
444 can change the membrane potential and prevent the microorganisms from growing.  
445 When the concentration of LAE increased, the membrane structures were seriously  
446 destroyed, resulting in rupturing the entire cellular structures [28,29].


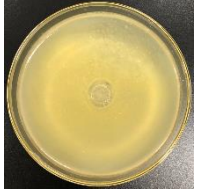
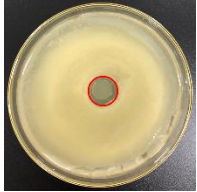

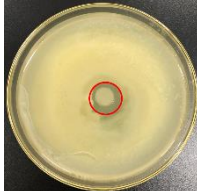
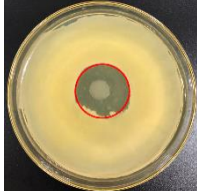

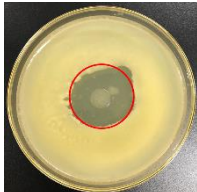
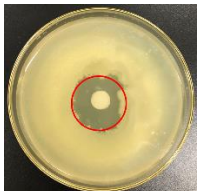
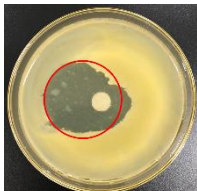
447 It was also found that *E. coli* showed better resistance to LAE than *S. aureus* in  
448 our study, which can be explained by the structural differences between these two  
449 bacteria. *E. coli* is a gram-negative bacteria with a thick layer lipopolysaccharide  
450 membrane outside of the cell, while *S. aureus* is a gram-positive bacteria only  
451 possessing a single peptidoglycan layer cell structure. It was more difficult for LAE to  
452 damage the membrane of *E. coli* [54]. The antibacterial performances of the active films  
453 in this work were superior to those previous studies using a hydrophobic polymer  
454 matrix [4,32], since the hydrophilicity of our film matrix can promote the diffusion of  
455 LAE in TSA plates [55,56].

456

457 Table 5. The inhibition zone of films with different LAE contents (0, 1%, 2.5%, 5% and 10%



458 of polymer matrix, w/w) on *E. coli* and *S. aureus*.

Film sample	inhibition zone of <i>E. coli</i> * (mm)	inhibition zone of <i>S. aureus</i> * (mm)
Control film	 no inhibition zone	 no inhibition zone
LAE-1%	 12.20±0.10 <sup>a</sup>	 19.23±0.21 <sup>a</sup>
LAE-2.5%	 16.33±0.25 <sup>b</sup>	 29.10±0.20 <sup>b</sup>
LAE-5%	 20.57±0.21 <sup>c</sup>	 37.30±0.20 <sup>c</sup>
LAE-10%	 29.07±0.25 <sup>d</sup>	 43.37±0.21 <sup>d</sup>

459 \*The mean values in the same column of the table with different superscript letters (a, b, c, d)

460 indicate significant differences by Turkey's test at  $p \leq 0.05$ .

461

#### 462 4. Conclusions

463 In this study, blend films based on PVA/starch enriched with different  
 464 concentrations of LAE (1%, 2.5%, 5% and 10%, w/w) were prepared by a casting  
 465 process. SEM results showed that PVA and cassava starch were compatible, the addition

466 of LAE changed the film morphology, and high content of LAE (5% and 10%) led to  
467 the generation of pores on the cross-section due to the surfactant effects. The molecular  
468 interaction, crystalline structure and film thickness were not significantly affected by  
469 the introduction of LAE. Incorporating LAE slightly decreased TS but significantly  
470 increased EB of the blend films, which might be ascribed to the aggregation and  
471 synergistic plasticizing effects of acetic acid, glycerol and LAE, respectively. The light  
472 transmittance values of active films with LAE were lower than those of control films  
473 in both the UV and visible light regions, suggesting that the active films had better UV  
474 and light barrier properties, which could help to prevent food products from degradation  
475 and oxidation. The incorporation of LAE resulted in higher MA and WVP values  
476 because of its hydrophilic nature and the increasing mobility of polymer chains. The  
477 active films showed antibacterial activities against both *E. coli* and *S. aureus* when  
478 including only 1% LAE, and they became more effective when the concentrations of  
479 LAE increased. The PVA/starch films containing LAE demonstrated potential as an  
480 active packaging material to effectively maintain the food quality and extend the shelf  
481 life. They are a promising alternative to replace synthetic plastics in certain food  
482 applications.

483

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489

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