

1 **Cost evaluation and sensitivity analysis of the alkaline zinc-iron flow battery**
2 **system for large-scale energy storage applications**

3
4 Ziqi Chen^a, Yongfu Liu^{a,b}, Wentao Yu^a, Qijiao He^c, Meng Ni^c, Shuquan Yang^d,
5 Shuanglin Zhang^d, Peng Tan^{a*}

6
7 ^a Department of Thermal Science and Energy Engineering, University of Science
8 and Technology of China (USTC), Hefei 230026, Anhui, China.

9 ^b Shanghai JINGYI Electrical Apparatus Factory Co., Ltd. Shanghai 201700, China.

10 ^c Department of Building and Real Estate, Research Institute for Sustainable Urban
11 Development (RISUD) & Research Institute for Smart Energy (RISE), The Hong
12 Kong Polytechnic University, Hung Hom, Kowloon, Hong Kong, China.

13 ^d Anhui Conch New Energy Co., Ltd, Wuhu 241000, Anhui, China.

14 * Corresponding author:

15 Email: pengtan@ustc.edu.cn (Peng Tan)

16
17 **Abstract:** Alkaline zinc-iron flow batteries attract great interest for remarkable energy
18 density, high safety, environmentally benign. However, comprehensive cost evaluation
19 and sensitivity analysis of this technology are still absent. In this work, a cost model for
20 a 0.1 MW/0.8 MWh alkaline zinc-iron flow battery system is presented, and a capital
21 cost under the U.S. Department of Energy's target cost of 150 \$ per kWh is achieved.
22 Besides, the effects of electrode geometry, operating conditions, and membrane types
23 on the system cost are investigated. The results illustrate that a low flow rate and thin
24 electrodes with high porosity contribute to low capital costs under low current densities.
25 Furthermore, the porous polybenzimidazole (PBI) membrane is more cost-effective
26 than Nafion 212 membrane. This work provides an integrated estimation for the zinc-
27 iron flow battery system, demonstrating its tremendous potential for grid-level energy
28 storage applications.

29 **Keywords:** Zinc-iron redox flow battery; Large-scale energy storage; System cost;
30 Sensitivity analysis

31 **1. Introduction**

32 To address the pollution, rising energy costs, and climate change caused by the
33 intensive use of fossil fuels, tremendous attention has been paid to alternative,
34 renewable energy such as solar and wind energy [1–3]. For instance, 95% of new energy
35 consumption in Europe from 2005 to 2030 will come from renewable energy sources
36 [4]. However, the intermittent and unreliable characteristic of this kind of energy
37 triggers the pursuit of energy storage technologies with competitive cost and high
38 reliability [4–7].

39 Flow batteries, of which the energy and power can be designed independently,
40 combine excellent traits of great safety, high efficiency, and durable cycle life,
41 becoming a promising candidate for the back power of renewable energy sources [8–
42 10]. Since Thaller proposed the first iron-chromium (Fe-Cr) flow battery in 1975 [11],
43 various kinds of flow batteries have been invented and advanced, for instance, all
44 vanadium (V-V) flow batteries [12–20], zinc-based flow batteries [21–25], iron-based
45 flow batteries [26–30], and polymer-based flow batteries [31–35]. To balance the
46 variability of electricity load and meet the rapid growth of energy needs, energy storage
47 over GWh magnitude is pursued [36]. The U.S. Department of Energy (DOE) proposed
48 a long-term target for energy storage technologies of a system capital cost under 150
49 \$ kWh⁻¹ [37]. For this purpose, numerous works have been performed to give
50 comprehensive cost analyses on flow battery systems for large power capacity and low
51 capital cost. The Pacific Northwest National Laboratory (PNNL) developed a cost-
52 performance model for V-V and iron-vanadium (Fe-V) redox flow batteries and

53 determined the optimal chemistry and operating conditions for desired power or energy,
54 from which the V-V chemistry and the Fe-V chemistry are the optimum options for
55 energy-intensive and power-intensive applications, respectively [36]. Zeng et al.
56 conducted a comparative study for V-V and iron-chromium (Fe-Cr) redox flow batteries
57 on the cost and performance of the 1 MW-8 h system. It was found that the Fe-Cr system
58 shows a more serious capacity decline than the V-V system and is more cost-effective
59 at a higher range of power densities and capacities [38]. Singh et al. investigated the
60 levelized cost of a hydrogen-bromine (H-Br) flow battery system with different
61 configurations and performed sensitivity analyses on the system components. The
62 lowest system capital cost of 220 \$ kWh⁻¹ is achieved for a four-hour discharge system,
63 and extending the lifetime of electrocatalysts is needed [39]. Gong et al. presented a 1
64 MW/8 MWh zinc-iron (Zn-Fe) flow battery system utilizing twofold membranes with
65 threefold electrolytes, achieving a system cost lower than \$100 kWh⁻¹ [40].

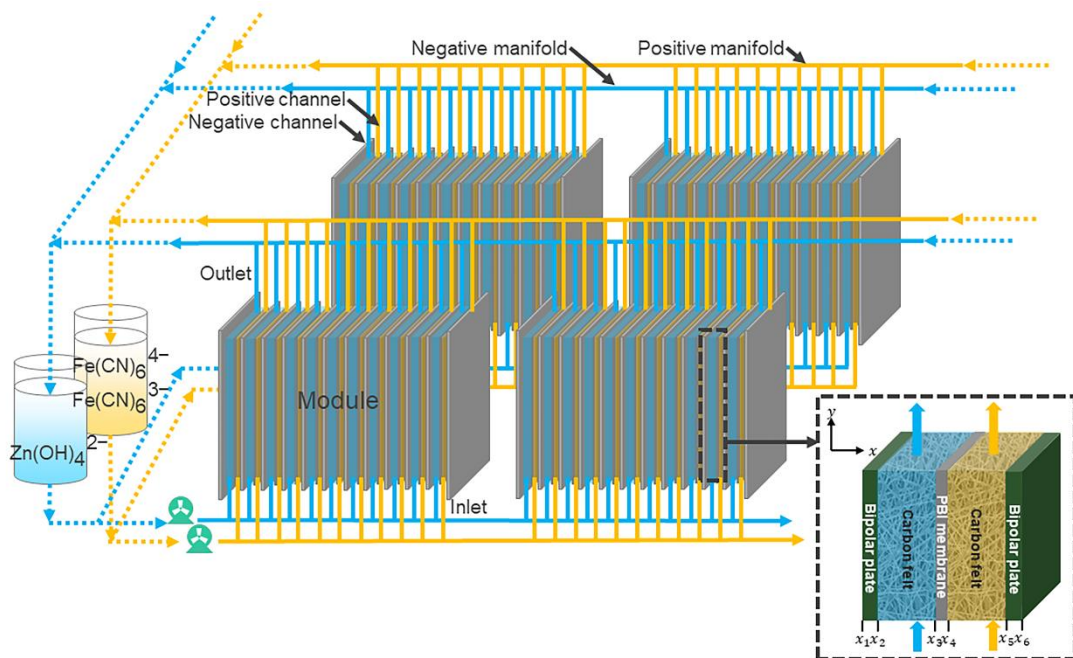
66 From the perspective of research and development, the dominated flow battery at
67 present is still the V-V redox flow battery. Plentiful experimental and numerical studies
68 have been carried out to improve its performance. For instance, Yoon et al. inserted an
69 additional electrode layer at the entrance and exit of a V-V redox flow battery to
70 manipulate the regional porosity, and proposed an empirical equation to optimize the
71 local electrode porosity distribution, through which, the energy efficiency of the cell
72 can be increased to 67.7% at 150 mA cm⁻² [2]. Sun et al. established a three-
73 dimensional, multi-physical model to explore the flow field design for large-scale V-V
74 redox flow batteries. The results suggested that the cell with a split-interdigitated flow

75 field achieved the highest overall energy efficiency, causing the increase of pressure
76 drop losses [18]. However, the expensive chemical cost, high demand for the purity of
77 vanadium, and temperature-constrained inferior solubility of redox species hamper the
78 commercialization of the V-V redox flow batteries [41]. Zn-Fe flow batteries, feature
79 the advantages of abundant zinc metal reserve, high energy density, and low price,
80 becoming a promising alternative to V-V flow batteries. Over the past 20 years,
81 enormous efforts have been devoted to zinc-iron flow batteries and huge progress has
82 been made. Yuan et al. proposed an alkaline Zn-Fe flow battery that achieved steady
83 performance for more than 500 cycles with a self-made porous PBI membrane and a
84 three-dimensional porous carbon felt electrode. Besides, a kilowatt cell stack with a
85 capital cost lower than 90 \$ kWh⁻¹ was constructed [42]. In their other work, a
86 nanoporous membrane with negative charges is developed for alkaline Zn-Fe flow
87 batteries, with which the battery can operate stably for 240 cycles at a current density
88 of 80 to 160 mA cm⁻² [43]. Chang et al. presented an alkaline Zn-Fe flow battery based
89 on a cost-effective membrane with highly anti-alkali microporous hollow spheres,
90 exhibiting 500 stable cycles with coulombic efficiency of 98.6% and energy efficiency
91 of 88.3% at 80 mA cm⁻² [44]. Xie et al. proposed a neutral Zn-Fe flow battery utilizing
92 highly soluble FeCl₂/ZnBr₂ species, exhibiting steady performance for over 100
93 cycles and a charge energy density of 56.3 Wh L⁻¹ [45]. However, few studies have
94 been performed on the cost prediction for this technology. Despite the cost of the
95 double-membrane triple-electrolyte design is attractive [40], its complex configuration
96 can be a huge barrier to further industrialization. Therefore, a comprehensive estimate

97 of the capital cost of a Zn-Fe flow battery system with a conventional two-electrolyte
98 structure is in great demand.

99 Herein, we provided a cost model for an alkaline Zn-Fe flow battery system
100 following our previous work [46]. First, a stationary, two-dimensional electrochemical
101 model for a Zn-Fe flow battery was established. After model validation, the capital cost
102 for a 0.1 MW/0.8 MWh Zn-Fe flow battery system was calculated considering the shunt
103 losses and pumping losses. Moreover, a sensitivity analysis for the battery system is
104 performed to deeply understand the influences of the operating conditions and cell
105 components on the system economy. This work presents a powerful approach to
106 estimate the feasibility of the Zn-Fe flow battery system as well as offering guidance
107 for practical applications.

108 2. Model development



109

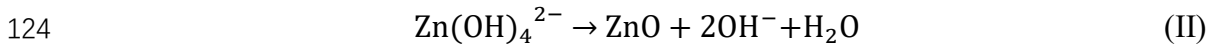
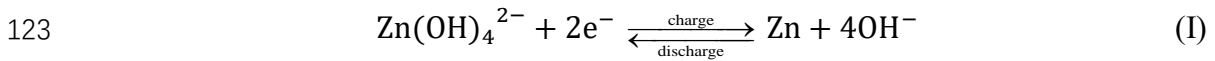
110

Fig. 1 Schematic of a zinc-iron flow battery stack.

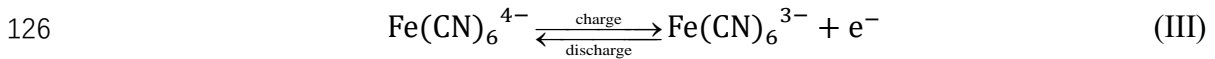
111

112 Fig. 1 displays the configuration of a Zn-Fe flow battery stack. The system is
 113 constructed of modules connected in 5 series and 5 parallels to meet the desired power
 114 output of 0.1 MW, in which each module contains 11 single cells connected by bipolar
 115 plates. The positive electrolyte consists of $\text{Fe}(\text{CN})_6^{3-}/\text{Fe}(\text{CN})_6^{4-}$ redox pair and a
 116 KOH solution, while the negative electrolyte contains $\text{Zn}(\text{OH})_4^{2-}/\text{Zn}$ redox pair and
 117 a NaOH solution. Porous carbon felts are adopted as the electrodes and sandwich a PBI
 118 membrane to separate the negative and positive electrolytes. Gaskets are used between
 119 the contiguous components to avoid electrolyte leakage. The electrolytes are contained
 120 in separate external reservoirs and circulated through a complex pipe system with the
 121 drive of pumps. The electrochemical reactions are as follows:

122 Negative electrode:



125 Positive electrode:



127 During charge, $\text{Zn}(\text{OH})_4^{2-}$ reduction reaction takes place on the negative
 128 electrode, forming the deposition of metallic zinc. In the discharging process, zinc gains
 129 two electrons oxidized to $\text{Zn}(\text{OH})_4^{2-}$ ions, then $\text{Zn}(\text{OH})_4^{2-}$ ions accumulate
 130 increasingly and convert to ZnO once saturated. In this work, the predominant reaction
 131 at the negative electrode is assumed to be the transformation between $\text{Zn}(\text{OH})_4^{2-}$ ions
 132 and zinc (Reaction I) with neglecting the existence of solid ZnO because the adopted
 133 concentration of $\text{Zn}(\text{OH})_4^{2-}$ is lower than its saturation value [47]. In the positive
 134 half-cell, the $\text{Fe}(\text{CN})_6^{4-}$ oxidation reaction and the $\text{Fe}(\text{CN})_6^{3-}$ reduction reaction
 135 take place during charge and discharge, respectively (Reaction III).

136 Several assumptions are made for simplification during simulation:

- 137 1. The whole cell is assumed to be isothermal.
- 138 2. Incompressible fluid is utilized for electrolyte flow.
- 139 3. All physical properties are considered isotropic and homogeneous.
- 140 4. Hydrogen evolution reaction (HER) and oxygen evolution reaction (OER) are
141 not considered.
- 142 5. Dilute solution hypothesis is adopted for mass transfer.
- 143 6. Changes of concentration, potential, and pressure along the width of the cell are
144 ignored.
- 145 7. The variation of electrolyte volume driven by water transfer through the
146 membrane is neglected.

147 This model combines an electrochemical model with a pumping loss model and a
148 shunt current model. For a specific current density, the relationship between the
149 electrochemical voltage and the current density is calculated with an assumed stack area.
150 Then, the pump loss and the shunt loss are calculated to get the effective voltage and
151 associated power. Adjustments for flow rates and flow channel geometry are made to
152 achieve as high a voltage as possible. When the calculated power is less than 0.1 MW,
153 the stack area is increased and this process repeats. Once the required power is reached,
154 the area is fixed to determine the stack component costs. Next, the electrolyte costs are
155 determined based on the electrolyte volume needed for the 0.8 MWh output energy. All
156 calculations above are performed at the 50% SOC for simplification.

157 **2.1 Electrochemical model**

158 The detailed electrochemical model with the dissolution-deposition process has
 159 been developed in our previous work [46]. Here, only the main governing equations
 160 and boundary conditions are presented.

161 2.1.1 Governing equations

162 The ionic and electronic current densities at all domains of the cell obey the charge
 163 conservation:

$$164 \quad \nabla \cdot \vec{j}_l^e + \nabla \cdot \vec{j}_s^e = 0 \quad (1)$$

165 where \vec{j}_l^e and \vec{j}_s^e are given by the following equations:

$$166 \quad \vec{j}_l^e = F \sum_i z_i \vec{N}_i^e \quad (2)$$

$$167 \quad \vec{j}_s^e = \sigma_s^e \nabla \phi_s^e \quad (3)$$

168 in which F denotes the Faraday constant, z_i and \vec{N}_i^e denote the valence and flux of
 169 species i , respectively. The Nernst-Planck equation is employed to describe the flux \vec{N}_i^e
 170 and consists of diffusion, migration and convection in three terms:

$$171 \quad \vec{N}_i^e = -D_i^{eff} \nabla c_i^e - z_i u_i^e c_i^e F \nabla \phi_l^e + \vec{v}_l^e c_i^e \quad (4)$$

172 where u_i^e represents the ionic mobility, c_i^e the concentration of species i , ϕ_l^e the
 173 electrolyte potential, and \vec{v}_l^e the velocity vector. The effective diffusion coefficient
 174 D_i^{eff} is corrected from the diffusion coefficient D_i using the Bruggemann relation:

$$175 \quad D_i^{eff} = \varepsilon^{3/2} D_i \quad (5)$$

176 where ε is the electrode porosity.

177 The Butler-Volmer equations are adopted to describe the reversible
 178 electrochemical reactions at the electrode surfaces:

$$179 \quad F D_{Fe^{2+}} \left(\frac{c_{Fe^{2+}}^e - c_{Fe^{2+}}^s}{r_p} \right) = F k_+ (c_{Fe^{2+}}^e)^{(1-\alpha_+)} (c_{Fe^{3+}}^e)^{\alpha_+} \left[\left(\frac{c_{Fe^{2+}}^s}{c_{Fe^{2+}}^e} \right) \exp \left(\frac{(1-\alpha_+) F \eta_-}{RT} \right) - \right.$$

$$180 \quad \left(\frac{c_{Fe^{3+}}^s}{c_{Fe^{3+}}^e} \right) \exp\left(\frac{-\alpha_+ F \eta_-}{RT}\right) \quad (6a)$$

$$181 \quad FD_{Fe^{3+}} \left(\frac{c_{Fe^{3+}}^e - c_{Fe^{3+}}^s}{r_p} \right) = Fk_+ (c_{Fe^{2+}}^e)^{(1-\alpha_+)} (c_{Fe^{3+}}^e)^{\alpha_+} \left[\left(\frac{c_{Fe^{2+}}^s}{c_{Fe^{2+}}^e} \right) \exp\left(\frac{(1-\alpha_+) F \eta_-}{RT}\right) - \right.$$

$$182 \quad \left. \left(\frac{c_{Fe^{3+}}^s}{c_{Fe^{3+}}^e} \right) \exp\left(\frac{-\alpha_+ F \eta_-}{RT}\right) \right] \quad (6b)$$

$$183 \quad j_- = a_- Fk_- (c_{OH^-})^{(1-\alpha_-)} (c_{Zn(OH_4)^{2-}})^{\alpha_-} \left[\left(\frac{c_{OH^-}}{c_{OH^-,ref}} \right)^4 \left(\frac{M_{Zn}}{M_{Zn,ref}} \right) \exp\left(\frac{2(1-\alpha_-) F \eta_-}{RT}\right) - \right.$$

$$184 \quad \left. \left(\frac{c_{Zn(OH_4)^{2-}}}{c_{Zn(OH_4)^{2-},ref}} \right) \exp\left(\frac{-2\alpha_- F \eta_-}{RT}\right) \right] \quad (6c)$$

185 where j is the local reaction current density, a and r_p are the specific surface area and
 186 average pore radius of the porous electrode, respectively. k and α are the standard
 187 reaction rate constant and charge transfer coefficient, respectively. c_i^s denotes the
 188 surface concentration of species i at the electrolyte-electrode interface. The
 189 overpotential η is defined by:

$$190 \quad \eta_j = \phi_l^e - \phi_s^e - E_{0,j} \quad (7)$$

191 where $E_{0,j}$ denotes the open-circuit voltage and is calculated using the Nernst
 192 equations:

$$193 \quad E_{0,-} = E'_{0,-} + \frac{RT}{nF} \ln\left(\frac{c_{Fe^{3+}}^e}{c_{Fe^{2+}}^e}\right) \quad (8)$$

$$194 \quad E_{0,+} = E'_{0,+} + \frac{RT}{nF} \ln\left(\frac{c_{Zn(OH_4)^{2-}}^e}{c_{OH^-}^e}\right) \quad (9)$$

195 The volume of the negative electrode varies ascribed to the zinc deposition during
 196 charging, which is given by:

$$197 \quad \frac{\partial \varepsilon}{\partial t} = \frac{1}{2F} \frac{MW^{Zn}}{\rho^{Zn}} a_- i_- \quad (10)$$

198 where MW and ρ denote the molecular weight and density, respectively.

199 The porosity and specific surface area also change when zinc deposits fill the pores
 200 of the electrode, which is described as:

$$201 \quad \frac{a}{a_0} = 1 - \left(\frac{\varepsilon_p}{\varepsilon}\right)^{0.5} \quad (11)$$

202 where a_0 and ε_p denote the initial specific surface area and the volume fraction

203 of the solid products, respectively.

204

205 **2.1.2 Boundary conditions**

206 At the inlet and outlet of both electrodes, the electrolyte fluxes are defined by the
207 following relations, flux driven by diffusion is supposed to be zero at the exits:

$$208 \quad y = 0(\text{inlet}) \quad -\vec{n} \cdot \vec{v} = \frac{Q}{\varepsilon w_g t_g} \quad (12)$$

$$209 \quad y = h_g(\text{outlet}) \quad -\vec{n} \cdot D_i^{eff} \nabla c_i^e = 0 \quad (13)$$

210 where w_g and t_g are the width and thickness of the electrode, respectively. Q
211 represents the volumetric flow rate.

212 At the bipolar plates, the boundary conditions in the charging process are
213 expressed as:

$$214 \quad -\vec{n} \cdot \vec{j}_s^{bp} = \begin{cases} -\frac{I}{h_g w_g} & x = x_2 \\ \frac{I}{h_g w_g} & x = x_5 \end{cases} \quad (14)$$

215 where the applied current I is set as constant, and the sign is reversed during discharge.

216 The grounding condition is applied to the negative bipolar plate:

$$217 \quad \phi_s^{bp} = 0 \quad x = x_2 \quad (15)$$

218 The boundary condition at the boundaries between the membrane and the porous
219 electrode domains is set up by the following relation:

$$220 \quad \phi_l^m = \phi_l^e - \frac{RT}{z_i F} \ln\left(\frac{c_i^m}{c_i^e}\right) \quad (16)$$

221 where c_i^m is the species concentration in the membrane. The potential shift caused by
222 the above equation is called the Donnan potential [48].

223 **2.2 Shunt current model**

224 An electrical circuit analog is adopted to describe the shunt losses caused by the

225 electrolyte bypassing flow through pipes. The cell is represented by an ideal voltage
226 source and a resistor in series. The channels and manifolds are represented by
227 equivalent resistors. During calculations, the electronic potentials of all cells are
228 assumed to be identical, so are the channels and manifolds. The electronic and ionic
229 potential inside each electrode is assumed to be uniform, and shunt losses between
230 modules are not considered.

231 The values of the equivalent resistors can be calculated through the following
232 equation [49]:

$$233 \quad r = \frac{1}{\sigma} \frac{l}{s} \quad (17)$$

234 in which l is the length of the flow path, s is the cross-sectional area, and σ is the
235 electrolyte conductivity.

236 The Kirchhoff's loop law is employed to calculate the values and directions of
237 current flow through each resistor. The total power losses are given as below:

$$238 \quad W_s = \sum I^2 r \quad (18)$$

239 **2.3 Pumping loss model**

240 The inlet electrolytes flow through the polyvinyl chloride (PVC) frame and across
241 the porous electrode. The pressure drop of the cell can therefore be defined by Darcy's
242 law [36]:

$$243 \quad \Delta P_e = \frac{L}{w_g} \frac{\mu Q}{t_g K} \quad (19)$$

244 in which L denotes the length of the flow path through the porous electrode, μ denotes
245 the electrolyte dynamic viscosity. The permeability coefficient of the porous electrode
246 (K) can be calculated using the Kozeny–Carman equation [50]:

247
$$K = \frac{d^2}{c} \frac{\varepsilon^{m+1}}{(1-\varepsilon)^m} \quad (20)$$

248 where d represents the diameter of the fiber, C and m are geometry coefficients taken
 249 as 0.07025 and 0.2282 obtained from experimental tests [50], respectively.

250 The pressure drop from each channel and manifold can be determined via the
 251 Hagen-Poiseuille equation [36]:

252
$$\Delta P = \frac{128\mu Q}{\pi d_H^4} \quad (21)$$

253 where d_H represents the hydraulic diameter of the flow channel.

254 Since the flow rates of the negative and positive sides are identical, the total power
 255 losses can be calculated by:

256
$$W_P = \frac{2\Delta P Q}{\eta} \quad (22)$$

257 where η is the pump efficiency and is taken as a constant of 67% [40].

258 It is worth noting that while the side reaction HER happens on the electrode
 259 porosity changes with the zinc dissolution-deposition, the pressure of the cell changes
 260 simultaneously, the neglect of which may contribute to the discrepancy between the
 261 model and experimental data. These factors can be amplified especially when it comes
 262 to a stack and should be paid attention to during practical engineering.

263 **2.4 Capital cost model**

264 The system cost is comprised of two parts: electrolyte cost and stack cost, which
 265 is given by the following equation [40]:

266
$$C_{sys} \approx C_e + C_s = \frac{U_e}{V_{eff}} + \frac{U_s}{tiV_{eff}} \quad (23)$$

267 where C_{sys} , C_e and C_s are the system, electrolyte, and stack costs, respectively. U_e
 268 denotes the unit price of the redox species and supporting electrolytes, U_s the unit

269 price of the stack components accounting for electrodes, membranes, and bipolar plates,
270 t the designed discharge duration, and i the current density. V_{eff} represents the
271 effective discharge voltage and is given as [36]:

$$272 \quad V_{eff} = V \left(\frac{VI - W_P - W_S}{VI} \right) \quad (24)$$

273 where V is the operating voltage at the SOC of 50%, which is determined from the
274 electrochemical model.

275 The system efficiency is defined as the energy efficiency of the battery system in
276 consideration of shunt loss and pump loss, which is determined by the following
277 equation:

$$278 \quad \eta_s = \frac{VI - W_P - W_S}{VI} \quad (25)$$

279 **2.5 Membrane Model**

280 The mechanisms of the ion exchange membrane and the porous membrane are
281 different. Ion exchange membranes transfer counter ions and isolate active species with
282 permanent fixed space charge, while porous membranes separate active species and
283 charge carrier ions by pore size exclusion [51]. In this model, the fixed space charge
284 concentration of Nafion 212 is 1990 mol m^{-3} and only proton is transferred, while the
285 charge carrier ion of PBI is OH^- . All properties mentioned above are set in the built-
286 in interface of COMSOL.

287 **3. Numerical conditions**

288 **3.1 Calculation details**

289 The parameters utilized for simulation are listed in **Table 1**, **Table 2**, and **Table 3**.

290 The costs for various components are given in **Table 4**.

291 The electrochemical model was simulated via COMSOL Multiphysics software
 292 utilizing the tertiary current distribution interface. The mesh including 4616 elements
 293 and the relative tolerance of 10^{-3} was adopted. The shunt loss, pumping loss, and cost
 294 models were solved using MATLAB software.

295 **Table 1** Parameters used in the simulation

Parameter	Symbol	Value	Unit	Ref.
Geometrical properties				
Thickness of the membrane	t_m	3.5×10^{-5}	m	[42]
Thickness of the electrode	t_g	0.007	m	-
Porosity of the electrode	ε	0.94	-	[42]
Fiber diameter of the electrode	d	19.2	μm	[50]
Specific surface area of the electrode	a	5.0×10^4	m^{-1}	-
Height of the cell	h_g	0.06	m	[42]
Width of the cell	w_g	0.08	m	[42]
Channel height	-	0.006	m	-
Channel width	-	0.01	m	-
Channel length	-	0.2	m	-
Manifold diameter	-	0.04	m	-
Manifold segment length	-	0.007	m	-
Electrolyte properties				
Diffusion coefficient of OH^-	D_{OH^-}	5.27×10^{-9}	$m^2 s^{-1}$	[52]
Diffusion coefficient of K^+	D_{K^+}	1.96×10^{-9}	$m^2 s^{-1}$	[52]
Diffusion coefficient of Na^+	D_{Na^+}	1.33×10^{-9}	$m^2 s^{-1}$	[52]
Diffusion coefficient of $Zn(OH)_4^{2-}$	$D_{Zn(OH)_4^{2-}}$	3.10×10^{-10}	$m^2 s^{-1}$	[52]
Diffusion coefficient of $Fe(CN)_6^{4-}$	$D_{Fe(CN)_6^{4-}}$	7.35×10^{-10}	$m^2 s^{-1}$	[52]
Diffusion coefficient of $Fe(CN)_6^{3-}$	$D_{Fe(CN)_6^{3-}}$	8.96×10^{-10}	$m^2 s^{-1}$	[52]
OH^- initial concentration in negative electrolyte	c_{-,OH^-}^0	4000	$mol m^{-3}$	[42]

OH ⁻ initial concentration in positive electrolyte	$c_{+,OH-}^0$	3000	mol m ⁻³	[42]
Zn(OH) ₄ ²⁻ initial concentration in negative electrolyte	$c_{Zn(OH)_4^{2-}}^0$	250	mol m ⁻³	-
Zn initial concentration in negative electrolyte	c_{Zn}^0	250	mol m ⁻³	-
Na ⁺ initial concentration in negative electrolyte	$c_{-,Na+}^0$	4000	mol m ⁻³	[42]
Na ⁺ initial concentration in positive electrolyte	$c_{+,Na+}^0$	1000	mol m ⁻³	[42]
Fe(CN) ₆ ⁴⁻ initial concentration in positive electrolyte	$c_{Fe(CN)_6^{4-}}^0$	500	mol m ⁻³	-
Fe(CN) ₆ ³⁻ initial concentration in positive electrolyte	$c_{Fe(CN)_6^{3-}}^0$	500	mol m ⁻³	-
K ⁺ initial concentration in positive electrolyte	c_{K+}^0	3000	mol m ⁻³	[42]

Table 2 Kinetic parameters used in the simulation

Parameter	Symbol	Value	Unit
Anodic transfer coefficients of negative reaction, charge	$\alpha_{-,cha}^a$	0.6	-
Cathodic transfer coefficients of negative reaction, charge	$\alpha_{-,cha}^c$	0.4	-
Anodic transfer coefficients of positive reaction, charge	$\alpha_{+,cha}^a$	0.6	-
Cathodic transfer coefficients of positive reaction, charge	$\alpha_{+,cha}^c$	0.4	-
Anodic transfer coefficients of negative reaction, discharge	$\alpha_{-,dis}^a$	0.5	-
Cathodic transfer coefficients of negative reaction, discharge	$\alpha_{-,dis}^c$	0.5	-
Anodic transfer coefficients of positive reaction, discharge	$\alpha_{+,dis}^a$	0.2	-
Cathodic transfer coefficients of positive reaction, discharge	$\alpha_{+,dis}^c$	0.8	-
Standard rate constant of negative reaction, charge	$k_{-,cha}$	4×10^{-8}	m s^{-1}
Standard rate constant of positive reaction, charge	$k_{+,cha}$	4×10^{-8}	m s^{-1}
Standard rate constant of negative reaction, discharge	$k_{-,dis}$	1×10^{-7}	m s^{-1}
Standard rate constant of positive reaction, discharge	$k_{+,dis}$	6×10^{-5}	m s^{-1}

Table 3 General parameters

Parameter	Symbol	Value	Unit	Ref.
Density of Zn	ρ_{Zn}	7,140	kg m ⁻³	-
Molecular weight of Zn	MW_{Zn}	65.38	g mol ⁻¹	-
Operating temperature	T	298.15	K	-
Conductivity of the electrode	σ	300	S m ⁻¹	-
Number of transferred electrons of negative reaction	n_-	2	-	-
Number of transferred electrons of positive reaction	n_+	1	-	-
Standard potential of negative reaction	$E'_{0,-}$	-1.41	V	[42]
Standard potential of positive reaction	$E'_{0,+}$	0.33	V	[42]

298

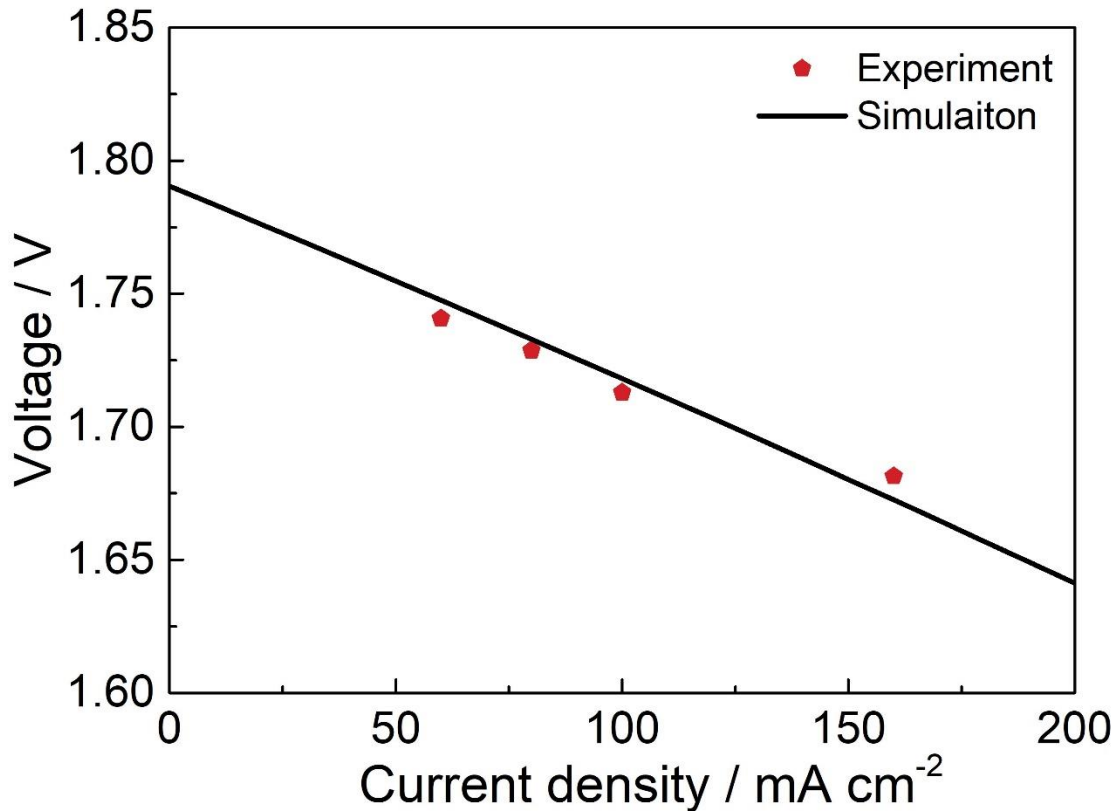
Table 4 Component cost used in the simulation

Component	Cost	Unit	Ref.
Carbon felt	70	\$ m ⁻²	[40]
Bipolar plate	55	\$ m ⁻²	[36]
PBI membrane	40	\$ m ⁻²	[42]
Valves	150	\$ per unit	[40]
Pipes	8	\$ m ⁻¹	[40]
Bolts	15	\$ per unit	[40]
Gaskets	2.5	\$ per unit	[40]
Collector plates	150	\$ per unit	[40]
End plates	193	\$ per unit	[40]
PVC pipe 1	8.6	\$ ft ⁻¹	[40]
PVC pipe 6	285	\$ ft ⁻¹	[40]
PVC frame	16.56	\$ m ⁻²	[40]
Pump cost estimate, base	8343.1	\$ per unit	[36]
Pump cost estimate	18	\$ GPM ⁻¹	[36]
PCS	210	\$ kW ⁻¹	[40]
Labor	1	\$ kWh ⁻¹	[40]
Tank	0.41	\$ gal ⁻¹	[40]
ZnO cost	1.3	\$ kg ⁻¹	[40]
Na ₄ Fe(CN) ₆ · 10H ₂ O cost	1.07	\$ kg ⁻¹	[42]
NaOH cost	0.4	\$ kg ⁻¹	[40]
KOH cost	0.84	\$ kg ⁻¹	[42]

299 3.2 Single-cell model validation

300 The electrochemical model is validated by comparing the experimental statistics
301 extracted from Yuan's research (points at the current density of 60, 80, 100, and 160
302 mA cm⁻²) at 50% SOC, as displayed in **Fig. 2** [42]. The polarization curve is well fitted
303 and shows the correct trend. The slight discrepancies are possibly the result of lacking
304 precise data like reaction rate constants and ignoring the side reactions like HER. Thus,
305 the proposed model can capture the voltage-current relationship of an alkaline Zn-Fe

306 flow battery.



307

308 **Fig. 2** Comparison of the simulated polarization curve of a zinc-iron flow battery at 50%

309 SOC with the experimental data.

310 **4. Sensitivity analysis and discussion**

311 **4.1 Capital cost**

312 The capital cost of a 0.1 MWh/0.8 MW Zn-Fe flow battery system is shown in **Fig.**

313 **3**. At a current density of 35 mA cm⁻², a capital cost of 150 \$ kWh⁻¹ is obtained with the

314 system energy efficiency of 68%, which satisfies the 2023 DOE's cost target of 150

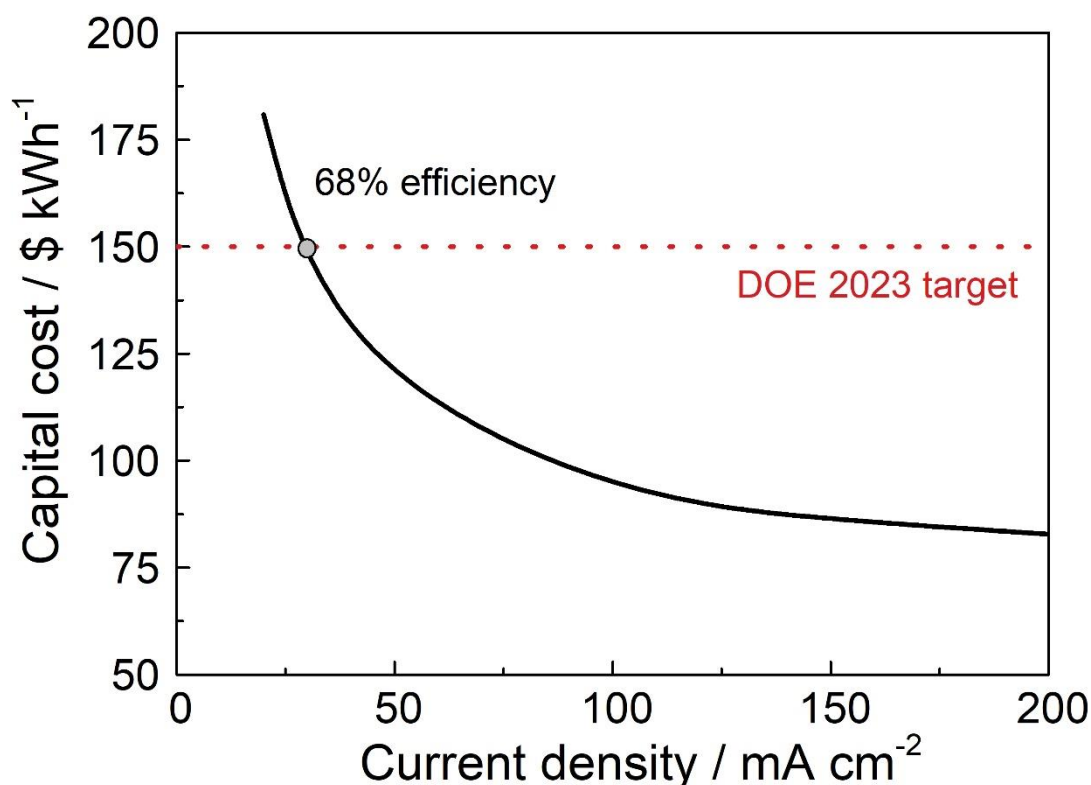
315 \$ kWh⁻¹. With an increase in the current density, the capital cost is reduced while the

316 system efficiency decreases at the same time. When the current density goes up to a

317 larger magnitude, the capital cost under 100 \$ kWh⁻¹ can be attained, demonstrating its

318 competitive economy. It is worth noting that the current density is ranging from 20 to

319 200 mA cm⁻², since few Zn-Fe flow batteries can work at a quite high current density,
320 at which severe polarization occurs, leading to very low energy efficiency [42].
321 Therefore, tremendous efforts should be made to improve the working current density,
322 such as increasing the specific surface area of electrodes, adopting membranes with
323 high ion conductivity, or improving the conductivity of supporting electrolytes [53].



324

325 **Fig. 3** Capital cost for 0.1MW/0.8MWh zinc-iron flow battery system.

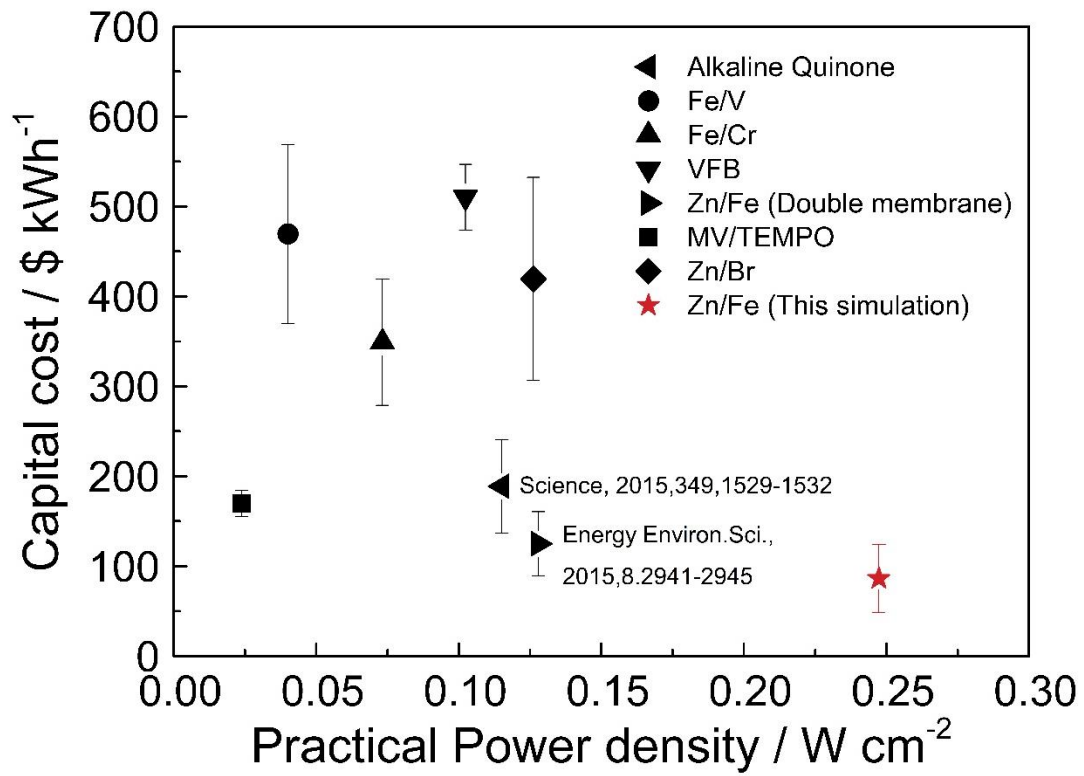
326 4.2 Cost comparisons

327 To further investigate the capital cost of the Zn-Fe flow battery system, the value
328 is compared with other flow battery systems presented of late, as shown in **Fig. 4** [42].
329 The Zn-Fe system owns the lowest cost with a relatively high power density of 0.247
330 W cm⁻², revealing its potential for energy storage applications.

331 **Fig. 5** demonstrates the comparison of the cost breakdown for the V-V, Fe-Cr, and

332 Zn-Fe systems [38]. The Zn-Fe system is the most cost-effective one with a capital cost
333 of 95 \$ kWh⁻¹ at the current density of 100 mA cm⁻². For both V-V and Zn-Fe systems,
334 the electrolyte is the dominant component and accounts for the cost of 62% and 50%,
335 respectively. While with a high total cost of 229 \$ kWh⁻¹, the electrolyte cost for the V-
336 V system is much higher than that of the Fe-Cr system. Besides, zinc production is as
337 high as 12 million tons per year, while vanadium production is 56000 tons per year [39],
338 making zinc-based batteries more suitable for large-scale production. Compared to the
339 high proportion of membrane cost in the Fe-Cr system (48%), the membrane cost
340 contributes only 9% to the Zn-Fe system, exhibiting the economic advantages of the
341 PBI membrane employed in the Zn-Fe system confronted with the Nafion membrane
342 used for other flow batteries. It should be noted that the system accessories like PVC
343 frame, bipolar plates, gaskets, bolts, and end plates cost totally account for 19% in the
344 Zn-Fe system, which is higher than those in the other two systems, indicating the
345 potential to further reduce the system cost.

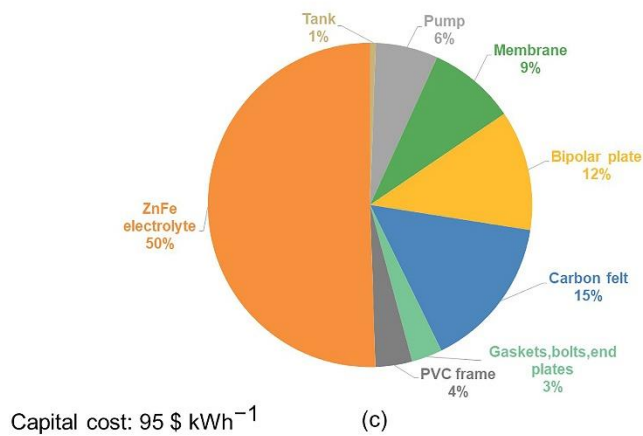
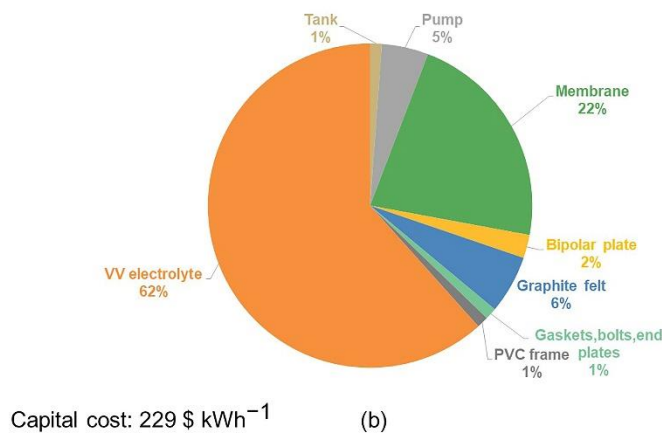
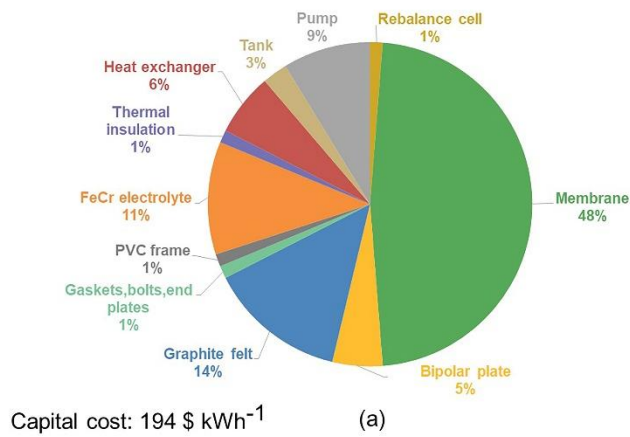
346



347

348 **Fig. 4** The cost and practical output power densities for different flow battery systems

349 (Except as marked on the figure, other data are from [42]).



350

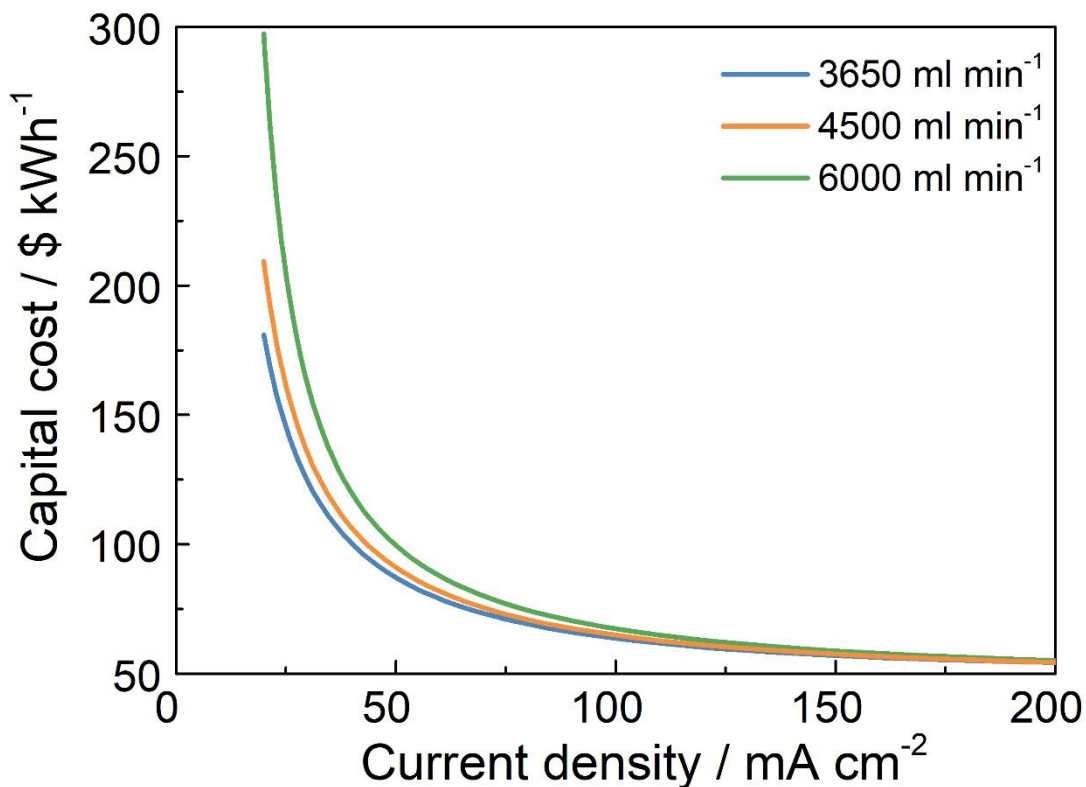
351 **Fig. 5** Capital cost breakdown for different flow battery systems: (a) iron-chromium
 352 system (b) all-vanadium system (c) zinc-iron system.

353 **4.3 Effects of flow rate**

354 The capital cost for the Zn-Fe system is plotted at various flow rates, as displayed

355 in Fig. 6. When the volumetric flow rate rises from 3650 to 6000 ml min⁻¹, the capital
356 cost shows a significant rise from 181 to 297 \$ kWh⁻¹ at the current density of 20 mA
357 cm⁻². Whereas at a higher current density of more than 100 mA cm⁻², the influence of
358 flow rate exhibits quite slight. According to previous research, higher flow rates
359 contribute to faster mass transfer, less concentration polarization, and higher discharge
360 voltages [46]. However, the pump losses also increase to overcome the resistance
361 caused by the increased flow, which reduces the effective voltage and results in a high
362 capital cost. It is worth noting that the working current density of alkaline Zn-Fe flow
363 batteries is ranging from 35 to 160 mA cm⁻² [53]. In this range, the capital costs of all
364 flow rates are under 150 \$ kWh⁻¹, which meets the DOE's target cost for energy storage
365 technologies. Besides, according to **Fig. 5 (c)**, the pump cost contributes only 6% to the
366 Zn-Fe system, thereby the increase of system cost causing by higher flow rates to
367 achieve better performance would be small.

368



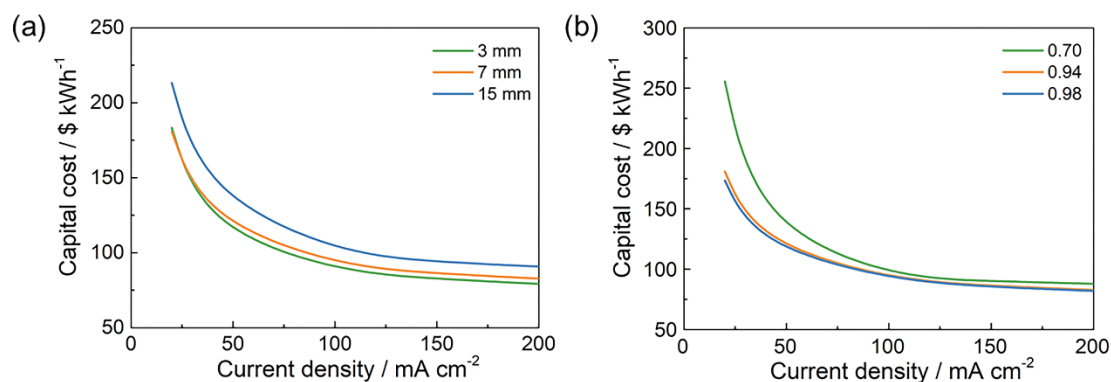
369
370 **Fig. 6** Capital cost of the system at different flow rates.

371 **4.4 Effects of electrode geometry**

372 **4.4.1 Electrode thickness**

373 **Fig. 7 (a)** displays the influences of electrode thickness on the capital cost. It is
 374 evident that the thinner the electrode, the lower the capital cost. This can be explained
 375 by that the thicker electrode leads to a longer mass transfer pathway and worse
 376 conductivity, which further decreases the discharge voltage. Meanwhile, with a fixed
 377 volumetric flow rate, larger electrode thickness results in lower velocity and smaller
 378 pump losses, yet this contribution is very slight and offset by the voltage decrease.
 379 Especially, at lower current densities of below 25 mA cm⁻², the capital costs of the
 380 systems with the 3 mm and 7 mm electrodes are approximately identical, indicating
 381 limited impacts on the system cost when the electrode thickness is small. It is worth

382 noting that during the calculation, the relationship between the electrode thickness and
 383 the unit price of carbon felt is assumed to be linear, which may be more complicated
 384 (e.g., the price of a thicker electrode is much higher) in practical. In addition, only the
 385 present-day cost of electrodes and other components are considered
 386 during calculation, which is based on low volume demand of 50 MW/100
 387 MWh [36]. With further industrialization, the electrode price would be lower
 388 for rising annual production rates and large-scale purchases, driving
 389 the capital cost down.



390

391 **Fig. 7** Capital cost of the system with different electrode geometry: (a) electrode
 392 thicknesses (b) porosity.

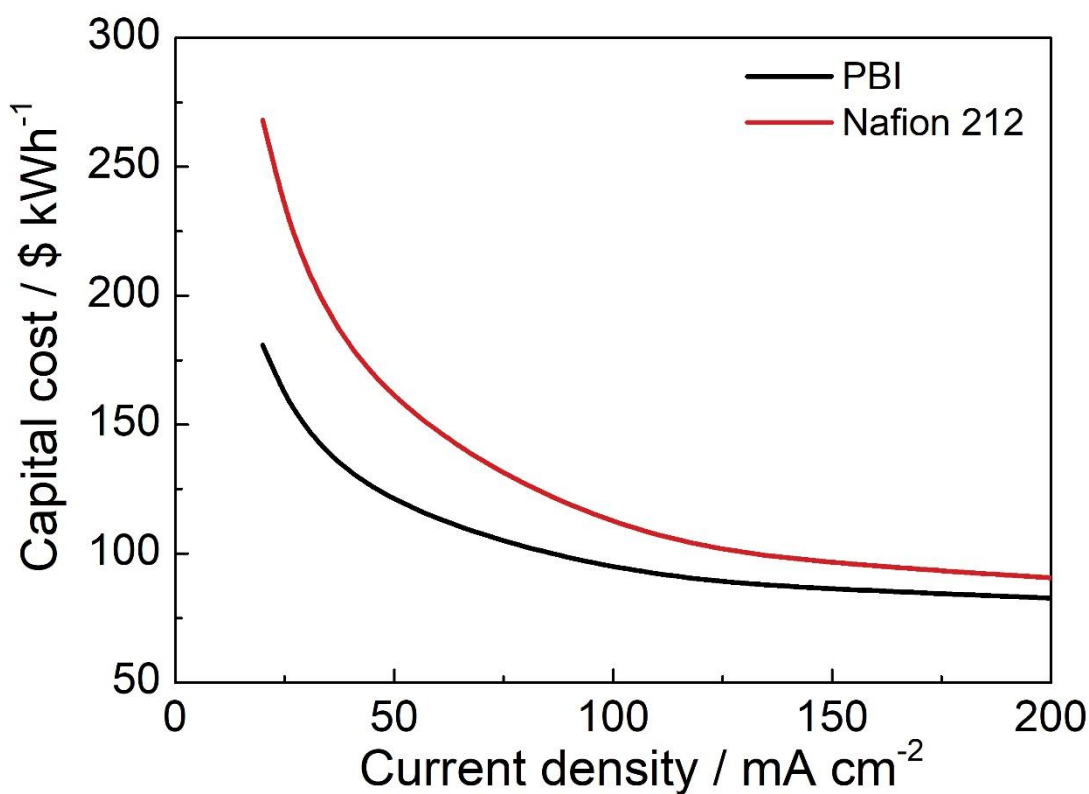
393 4.4.2 Electrode porosity

394 **Fig. 7** (b) displays the influences of various porosity on the capital cost with
 395 neglecting the variety of carbon felt unit price with the change of porosity. With the
 396 electrode porosity increasing from 0.7 to 0.94, a great decline in capital cost is observed
 397 among the whole range of the current density. This is because the electrode with higher
 398 porosity provides more reaction areas to sufficiently utilize the active species, leading
 399 to lower concentration polarization and a higher discharge voltage. In addition, high

400 porosity can reduce the pressure drop of the cell, which further decreases the pumping
401 losses. For the system with the electrode porosity of 0.98, the profile is very similar to
402 the system with 0.94 porosity, indicating the slight scope for improvement when the
403 porosity is high enough.

404 4.5 Effects of membrane

405 The capital costs of the systems with different membranes are displayed in **Fig. 8**.
406 The total cost of the system with the PBI membrane is lower than that with the Nafion
407 212 membrane among the whole range of the current density, which largely benefits
408 from the low unit price of the PBI membrane ($40 \text{ \$ m}^{-2}$) compared to the Nafion 212
409 membrane ($225 \text{ \$ m}^{-2}$). Besides, with the current density increasing, the gap between
410 the system costs with two membranes diminishes gradually, which demonstrates that
411 the impact of membranes on the system cost is higher under lower current densities.



412

413 **Fig. 8** Capital cost of the system with different membranes.

414 **5. Conclusions**

415 This work reported a cost-performance model for a 0.1 MW/0.8 MWh alkaline
416 zinc-iron flow battery system, including a two-dimensional electrochemical model, a
417 shunt losses model, and a pump losses model. The model validation verifies its ability
418 to predict the correct trend of the voltage-current profiles. A capital cost under 2023
419 DOE's cost target of 150 \$ kWh⁻¹ is obtained at the current density of 35 mA cm⁻².
420 Compared with other flow battery systems such as all vanadium and iron-chromium
421 flow batteries, the zinc-iron system owns the superiority in cost. Moreover, the
422 influences of the operating conditions, electrode geometry, and cell component on the
423 system cost are investigated. The results indicate that the low current density, low flow
424 rate, high porosity, and thin electrode lead to a low capital cost, while at high current
425 densities, the impacts become smaller. Moreover, the PBI membrane shows better
426 performance on cost than the Nafion 212 membrane. This research can guide the
427 selection of components when constructing a zinc-iron system and design the proper
428 operating conditions, and also be favorable to further reduce the system cost for large-
429 scale energy storage applications.

430 **Declaration of interests**

431 The authors declare no competing interests.

432 **CRedit authorship contribution statement**

433 **Ziqi Chen:** Conceptualization, Investigation, Methodology, Writing - original
434 draft. **Yongfu Liu:** Investigation. **Wentao Yu:** Investigation, Methodology. **Qijiao He:**

435 Methodology, Writing-review & editing. **Meng Ni**: Supervision, Funding acquisition,
436 Writing - review & editing. **Shuquan Yang**: Investigation, Writing-review & editing.
437 **Shuanglin Zhang**: Investigation, Writing-review & editing. **Peng Tan**: Supervision,
438 Project administration, Funding acquisition, Writing - review & editing.

439 **Acknowledgments**

440 P. Tan thanks the funding support from Chinese Academy of Sciences (CAS)
441 Program (KJ2090130001), Shanghai JINGYI Electrical Apparatus Factory Co., Ltd.
442 (ES2090130106), and USTC Tang Scholar (KY2090000065). M. Ni thanks the funding
443 support from The Hong Kong Polytechnic University (G-YW2D) and a grant (Project
444 Number: PolyU 152214/17E and PolyU 152064/18E) from Research Grant Council,
445 University Grants Committee, Hong Kong SAR.

446 **Nomenclature**

<i>a</i>	specific electroactive area ($\text{m}^2 \text{m}^{-3}$)
<i>c</i>	concentration (mol m^{-3})
<i>C</i>	special coefficient
<i>d</i>	diameter
<i>D</i>	diffusion coefficient ($\text{m}^2 \text{s}^{-1}$)
<i>E</i> ₀	equilibrium potential (V)
<i>E'</i> ₀	standard reaction potential (V)
<i>F</i>	Faraday constant ($96,485 \text{ C mol}^{-1}$)
<i>h</i>	height
<i>i</i>	current density (A m^{-2})
<i>I</i>	current (A)
<i>j</i>	reaction current density (A m^{-2})
\vec{j}	current density (vector)
<i>k</i>	reaction rate
<i>K</i>	permeability coefficient
<i>L</i>	length (m)
<i>m</i>	special coefficient

n	transferred electron
\vec{n}	outward normal vector
\vec{N}	molar species flux ($\text{mol m}^{-2} \text{s}^{-1}$)
P	pressure
ΔP	pressure drop
Q	volumetric flow rate ($\text{m}^3 \text{s}^{-1}$)
r	resistance (Ω)
R	universal gas constant ($8.314 \text{ J mol}^{-1} \text{ K}^{-1}$)
s	area (m^2)
S	source ($\text{mol m}^{-3} \text{s}^{-1}$)
T	temperature (K)
t	component thickness (m)
u	ion mobility
\vec{v}	velocity (m s^{-1})
V	voltage (V)
w	component width (m)
W	power
x	distance along cell width (m)
y	distance along cell height (m)
z	valence

447 *Greek*

α	transfer coefficient
η	overpotential (V)/ efficiency
ε	porosity
ρ	density (kg m^{-3})
σ	conductivity (S m^{-1})
μ	dynamic viscosity
ϕ	electric potential (V)

448 *Subscripts*

<i>cell</i>	property of cell
<i>cha</i>	charge
<i>dis</i>	discharge
<i>diff</i>	diffusion
<i>e</i>	electrolyte
<i>eff</i>	effective, corrected for tortuosity
<i>g</i>	electrode
<i>l</i>	liquid or ionic
<i>j</i>	reaction
<i>ref</i>	reference

<i>s</i>	solid or electronic
<i>sys</i>	system
<i>i</i>	specie
+	positive half-cell
–	negative half-cell

449 *Superscripts*

<i>e</i>	electrode or electrolyte domain
<i>eff</i>	effective value
<i>m</i>	membrane domain
<i>out</i>	outlet
<i>s</i>	surface

450

451 **References**

- 452 [1] Dovì VG, Friedler F, Huisingh D, Klemeš JJ. Cleaner energy for sustainable
453 future. *Journal of Cleaner Production* 2009;17:889–95.
454 <https://doi.org/10.1016/j.jclepro.2009.02.001>.
- 455 [2] Yoon SJ, Kim S, Kim DK. Optimization of local porosity in the electrode as an
456 advanced channel for all-vanadium redox flow battery. *Energy* 2019;172:26–35.
457 <https://doi.org/10.1016/j.energy.2019.01.101>.
- 458 [3] Virulkar V, Aware M, Kolhe M. Integrated battery controller for distributed
459 energy system. *Energy* 2011;36:2392–8.
460 <https://doi.org/10.1016/j.energy.2011.01.019>.
- 461 [4] Ferreira HL, Garde R, Fulli G, Kling W, Lopes JP. Characterisation of electrical
462 energy storage technologies. *Energy* 2013;53:288–98.
463 <https://doi.org/10.1016/j.energy.2013.02.037>.
- 464 [5] Zhang Y, Wu Y, You W, Tian M, Huang PW, Zhang Y, et al. Deeply rechargeable
465 and hydrogen-evolution-suppressing zinc anode in alkaline aqueous electrolyte.
466 *Nano Letters* 2020;20:4700–7. <https://doi.org/10.1021/acs.nanolett.0c01776>.
- 467 [6] Chao D, Zhou W, Xie F, Ye C, Li H, Jaroniec M, et al. Roadmap for advanced
468 aqueous batteries: From design of materials to applications. *Science Advances*
469 2020;6. <https://doi.org/10.1126/sciadv.aba4098>.
- 470 [7] Sun H, Yu M, Li Q, Zhuang K, Li J, Almheiri S, et al. Characteristics of
471 charge/discharge and alternating current impedance in all-vanadium redox flow
472 batteries. *Energy* 2019;168:693–701.
473 <https://doi.org/10.1016/j.energy.2018.11.130>.
- 474 [8] Yuan Z, Zhang H, Li X. Ion conducting membranes for aqueous flow battery
475 systems. *Chemical Communications* 2018;54:7570–88.
476 <https://doi.org/10.1039/C8CC03058H>.
- 477 [9] Chen W, Kang J, Shu Q, Zhang Y. Analysis of storage capacity and energy
478 conversion on the performance of gradient and double-layered porous electrode
479 in all-vanadium redox flow batteries. *Energy* 2019;180:341–55.
480 <https://doi.org/10.1016/j.energy.2019.05.037>.

- 481 [10] Badrinarayanan R, Tseng KJ, Soong BH, Wei Z. Modelling and control of
482 vanadium redox flow battery for profile based charging applications. *Energy*
483 2017;141:1479–88. <https://doi.org/10.1016/j.energy.2017.11.082>.
- 484 [11] Thaller LH. Electrically rechargeable redox flow cells. NASA TM X-71540,
485 Lewis Research Centre, 1974.
- 486 [12] Huamin Z, Yu Z, Zonghao L, Xiaoli W. Redox flow battery technology. *Progress*
487 *in Chemistry* 2009;21:2333–40. <https://doi.org/10.3990/1.9789036532259>.
- 488 [13] Rychcik M, Skylas-Kazacos M. Characteristics of a new all-vanadium redox
489 flow battery. *Journal of Power Sources* 1988;22:59–67.
490 [https://doi.org/10.1016/0378-7753\(88\)80005-3](https://doi.org/10.1016/0378-7753(88)80005-3).
- 491 [14] Zhou XL, Zeng YK, Zhu XB, Wei L, Zhao TS. A high-performance dual-scale
492 porous electrode for vanadium redox flow batteries. *Journal of Power Sources*
493 2016;325:329–36. <https://doi.org/10.1016/j.jpowsour.2016.06.048>.
- 494 [15] Wu Q, Lv Y, Lin L, Zhang X, Liu Y, Zhou X. An improved thin-film electrode
495 for vanadium redox flow batteries enabled by a dual layered structure. *Journal*
496 *of Power Sources* 2019;410–411:152–61.
497 <https://doi.org/10.1016/j.jpowsour.2018.11.020>.
- 498 [16] Wei L, Zhao T, Zeng L, Zhou X, Zeng Y. Titanium carbide nanoparticle-
499 decorated electrode enables significant enhancement in performance of all-
500 vanadium redox flow batteries. *Energy Technology* 2016;4:990–6.
501 <https://doi.org/10.1002/ente.201600016>.
- 502 [17] Oh K, Yoo H, Ko J, Won S, Ju H. Three-dimensional, transient, nonisothermal
503 model of all-vanadium redox flow batteries. *Energy* 2015;81:3–14.
504 <https://doi.org/10.1016/j.energy.2014.05.020>.
- 505 [18] Sun J, Zheng M, Yang Z, Yu Z. Flow field design pathways from lab-scale toward
506 large-scale flow batteries. *Energy* 2019;173:637–46.
507 <https://doi.org/10.1016/j.energy.2019.02.107>.
- 508 [19] Yin C, Gao Y, Guo S, Tang H. A coupled three dimensional model of vanadium
509 redox flow battery for flow field designs. *Energy* 2014;74:886–95.
510 <https://doi.org/10.1016/j.energy.2014.07.066>.
- 511 [20] Vynnycky M. Analysis of a model for the operation of a vanadium redox battery.
512 *Energy* 2011;36:2242–56. <https://doi.org/10.1016/j.energy.2010.03.060>.
- 513 [21] Leung PK, Ponce-De-León C, Low CTJ, Shah AA, Walsh FC. Characterization
514 of a zinc-cerium flow battery. *Journal of Power Sources* 2011;196:5174–85.
515 <https://doi.org/10.1016/j.jpowsour.2011.01.095>.
- 516 [22] Cheng J, Zhang L, Yang YS, Wen YH, Cao GP, Wang XD. Preliminary study of
517 single flow zinc-nickel battery. *Electrochemistry Communications*
518 2007;9:2639–42. <https://doi.org/10.1016/j.elecom.2007.08.016>.
- 519 [23] Lai Q, Zhang H, Li X, Zhang L, Cheng Y. A novel single flow zinc-bromine
520 battery with improved energy density. *Journal of Power Sources* 2013;235:1–4.
521 <https://doi.org/10.1016/j.jpowsour.2013.01.193>.
- 522 [24] Yang H, Qiao Y, Chang Z, Deng H, He P, Zhou H. A metal–organic framework
523 as a multifunctional ionic sieve membrane for long-life aqueous zinc–iodide
524 batteries. *Advanced Materials* 2020;32:1–8.

- 525 <https://doi.org/10.1002/adma.202004240>.
- 526 [25] Selverston S, Savinell RF, Wainright JS. Zinc-iron flow batteries with common
527 electrolyte. *Journal of The Electrochemical Society* 2017;164:A1069–75.
528 <https://doi.org/10.1149/2.0591706jes>.
- 529 [26] Zhou X, Lin L, Lv Y, Zhang X, Wu Q. A Sn-Fe flow battery with excellent rate
530 and cycle performance. *Journal of Power Sources* 2018;404:89–95.
531 <https://doi.org/10.1016/j.jpowsour.2018.10.011>.
- 532 [27] Zeng YK, Zhao TS, Zhou XL, Wei L, Ren YX. A novel iron-lead redox flow
533 battery for large-scale energy storage. *Journal of Power Sources* 2017;346:97–
534 102. <https://doi.org/10.1016/j.jpowsour.2017.02.018>.
- 535 [28] Zeng YK, Zhao TS, Zhou XL, Wei L, Jiang HR. A low-cost iron-cadmium redox
536 flow battery for large-scale energy storage. *Journal of Power Sources*
537 2016;330:55–60. <https://doi.org/10.1016/j.jpowsour.2016.08.107>.
- 538 [29] Zeng YK, Zhao TS, Zhou XL, Zou J, Ren YX. A hydrogen-ferric ion rebalance
539 cell operating at low hydrogen concentrations for capacity restoration of iron-
540 chromium redox flow batteries. *Journal of Power Sources* 2017;352:77–82.
541 <https://doi.org/10.1016/j.jpowsour.2017.03.125>.
- 542 [30] Gong K, Xu F, Grunewald JB, Ma X, Zhao Y, Gu S, et al. All-soluble all-iron
543 aqueous redox-flow battery. *ACS Energy Letters* 2016;1:89–93.
544 <https://doi.org/10.1021/acsenerylett.6b00049>.
- 545 [31] Winsberg J, Janoschka T, Morgenstern S, Hagemann T, Muench S, Hauffman G,
546 et al. Poly(TEMPO)/zinc hybrid-flow battery: a novel, “green,” high voltage, and
547 safe energy storage system. *Advanced Materials* 2016;28:2238–43.
548 <https://doi.org/10.1002/adma.201505000>.
- 549 [32] Janoschka T, Martin N, Martin U, Friebe C, Morgenstern S, Hiller H, et al. An
550 aqueous, polymer-based redox-flow battery using non-corrosive, safe, and low-
551 cost materials. *Nature* 2015;527:78–81. <https://doi.org/10.1038/nature15746>.
- 552 [33] Winsberg J, Hagemann T, Janoschka T, Hager MD, Schubert US. Redox-flow
553 batteries: From metals to organic redox-active materials. *Angewandte Chemie -
554 International Edition* 2017;56:686–711. <https://doi.org/10.1002/anie.201604925>.
- 555 [34] Yang B, Hooper-Burkhardt L, Wang F, Surya Prakash GK, Narayanan SR. An
556 inexpensive aqueous flow battery for large-scale electrical energy storage based
557 on water-soluble organic redox couples. *Journal of The Electrochemical Society*
558 2014;161:A1371–80. <https://doi.org/10.1149/2.1001409jes>.
- 559 [35] Huskinson B, Marshak MP, Suh C, Er S, Gerhardt MR, Galvin CJ, et al. A metal-
560 free organic-inorganic aqueous flow battery. *Nature* 2014;505:195–8.
561 <https://doi.org/10.1038/nature12909>.
- 562 [36] Viswanathan V, Crawford A, Stephenson D, Kim S, Wang W, Li B, et al. Cost
563 and performance model for redox flow batteries. *Journal of Power Sources*
564 2014;247:1040–51. <https://doi.org/10.1016/j.jpowsour.2012.12.023>.
- 565 [37] Grid Energy Storage, U.S. Department of Energy 2013.
- 566 [38] Zeng YK, Zhao TS, An L, Zhou XL, Wei L. A comparative study of all-vanadium
567 and iron-chromium redox flow batteries for large-scale energy storage. *Journal
568 of Power Sources* 2015;300:438–43.

- 569 <https://doi.org/10.1016/j.jpowsour.2015.09.100>.
- 570 [39] Singh N, Mcfarland EW. Levelized cost of energy and sensitivity analysis for the
571 hydrogen bromine flow battery. *Journal of Power Sources* 2015;288:187–98.
572 <https://doi.org/10.1016/j.jpowsour.2015.04.114>.
- 573 [40] Gong K, Ma X, Conforti KM, Kuttler KJ, Grunewald JB, Yeager KL, et al. A
574 zinc-iron redox-flow battery under \$100 per kW h of system capital cost. *Energy*
575 and *Environmental Science* 2015;8:2941–5. <https://doi.org/10.1039/c5ee02315g>.
- 576 [41] Soloveichik GL. Flow batteries: current status and trends. *Chemical Reviews*
577 2015;115:11533–58. <https://doi.org/10.1021/cr500720t>.
- 578 [42] Yuan Z, Duan Y, Liu T, Zhang H, Li X. Toward a low-cost alkaline zinc-iron
579 flow battery with a polybenzimidazole custom membrane for stationary energy
580 storage. *iScience* 2018;3:40–9. <https://doi.org/10.1016/j.isci.2018.04.006>.
- 581 [43] Yuan Z, Liu X, Xu W, Duan Y, Zhang H, Li X. Negatively charged nanoporous
582 membrane for a dendrite-free alkaline zinc-based flow battery with long cycle
583 life. *Nature Communications* 2018;9:3731. [https://doi.org/10.1038/s41467-018-](https://doi.org/10.1038/s41467-018-06209-x)
584 [06209-x](https://doi.org/10.1038/s41467-018-06209-x).
- 585 [44] Chang N, Yin Y, Yue M, Yuan Z, Zhang H, Lai Q, et al. A cost-effective mixed
586 matrix polyethylene porous membrane for long-cycle high power density
587 alkaline zinc-based flow batteries. *Advanced Functional Materials* 2019;29:1–
588 11. <https://doi.org/10.1002/adfm.201901674>.
- 589 [45] Xie C, Duan Y, Xu W, Zhang H, Li X. A low-cost neutral zinc-iron flow battery
590 with high energy density for stationary energy storage. *Angewandte Chemie*
591 2017;129:15149–53. <https://doi.org/10.1002/ange.201708664>.
- 592 [46] Chen Z, Yu W, Liu Y, Zeng Y, He Q, Tan P, et al. Mathematical modeling and
593 numerical analysis of alkaline zinc-iron flow batteries for energy storage
594 applications. *Chemical Engineering Journal* 2021;405:126684.
595 <https://doi.org/10.1016/j.cej.2020.126684>.
- 596 [47] Dietrich HG, Johnston J. Equilibrium between crystalline zinc hydroxide and
597 aqueous solutions of ammonium hydroxide and of sodium hydroxide. *Journal of*
598 *the American Chemical Society* 1927;49:1419–31.
599 <https://doi.org/10.1021/ja01405a005>.
- 600 [48] Knehr KW, Kumbur EC. Open circuit voltage of vanadium redox flow batteries:
601 Discrepancy between models and experiments. *Electrochemistry*
602 *Communications* 2011;13:342–5. <https://doi.org/10.1016/j.elecom.2011.01.020>.
- 603 [49] Ratcliff GA. Unit operations of chemical engineering. *Chemical Engineering*
604 *Science* 1957;6:287. [https://doi.org/10.1016/0009-2509\(57\)85034-9](https://doi.org/10.1016/0009-2509(57)85034-9).
- 605 [50] Kossenko A, Lugovskoy S, Averbukh M. Electric and hydraulic properties of
606 carbon felt immersed in different dielectric liquids. *Materials* 2018;11:1–12.
607 <https://doi.org/10.3390/MA11040650>.
- 608 [51] Wu J, Dai Q, Zhang H, Li X. Recent development in composite membranes for
609 flow batteries. *ChemSusChem* 2020;13:3805–19.
610 <https://doi.org/10.1002/cssc.202000633>.
- 611 [52] J.R. Rumble, Ionic conductivity and diffusion at infinite dilution, CRC

612 Handbook of Chemistry and Physics, 101st ed. (Internet Version 2020), CRC
613 Press/Taylor & Francis, Boca Raton, FL, 2020.
614 [53] Yuan Z, Yin Y, Xie C, Zhang H, Yao Y, Li X. Advanced materials for zinc-based
615 flow battery: development and challenge. *Advanced Materials* 2019;31:1–27.
616 <https://doi.org/10.1002/adma.201902025>.
617