

Photoelectrocatalytic properties and reactivity of Ti/Au-TiO₂ mesh electrodes^①

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[Abstract] A kind of photoelectrode was innovated by anodising titanium mesh in H₂SO₄ solution and photo-reduced in HAuCl₄ solution and named Ti/Au-TiO₂ mesh electrode. The structural and surface morphology of the Ti/Au-TiO₂ mesh was examined by X-ray diffraction, laser Raman spectra, scanning electronic microscopy(SEM) and X-ray photoelectron spectroscopy respectively. The results indicate that its crystal structure, morphology and the size of pore are affected greatly by gold deposition. XPS measurement shows that the valence band of Ti/TiO₂ has two peaks: a wide one at 4.97 eV and a narrow one at 6.61 eV, which correspond mainly to π (nonbonding) and σ (bonding) O 2p orbitals. The emission intensity of O 2p orbitals becomes stronger and the width of the valence band increases with the increase of Au content. And the emission of nonbonding shifts toward lower binding energy and that of bonding O 2p orbitals shifts toward higher binding energy. The photoelectrocatalytic (PEC) oxidation of humic acid(HA) was investigated in terms of TOC. The PEC oxidation efficiency of Ti/Au-TiO₂ mesh with optimal content of gold is higher than that of Ti/TiO₂ mesh. It is suggested that the recombination of electrons and holes is hindered owing to gold deposition. The investigation shows that PEC oxidation is a convenient way to mineralize organic matter for water treatment.

[Key words] photoelectrocatalytic oxidation; water treatment; humic acid; mesh electrode; gold

[CLC number] TB 39

[Document code] A

1 INTRODUCTION

It has been confirmed that the quality of raw water in Guangzhou was deteriorating during the last few years, owing to the rapid industrialization in the region. The main pollutants in the raw water include organics (such as humic acid) and pathogens. To ensure the drinking water quality, the local water supply treatment works have to raise the chlorine dosage in the disinfection processes. However, higher dosage of chlorine results in both a higher chlorine residual concentration and also a higher trihalomethanes (THMs) concentration in product water. The presence of humic acids (HA) has been a problem in the water industry because of their water-solubility, wide range of distribution in molecular mass and size, and their non-biodegradability. It has been confirmed that humic substances play a role as precursors of chlorinated by-product (DBPs) in chlorination processes^[1, 2]. It is essential to seek for new ways for water treatment.

Titanium dioxide (TiO₂)-based photo-catalysis has

been the focus of numerous investigations in recent years, particularly owing to its application for the complete mineralisation of undesirable organic contaminants to CO₂, H₂O and inorganic constituents^[3]. Some papers reported that humic substances could be effectively degraded in TiO₂ suspension system UV irradiation^[4-6]. However, it has been also known that this type of photo-oxidation has two typical drawbacks, i.e., difficulty to separate TiO₂ particles from its aqueous suspension and low quantum yield (less than 10%, generally)^[7]. To solve the problem of TiO₂ particle separation from wastewater, many researchers were trying to immobilize TiO₂ film on solid carriers such as sand, glass media or resins by coating, soaking, precipitating or spanning methods^[8]. However while these immobilized TiO₂ photo-oxidation processes made the TiO₂ separation from water phase much easier, they did not improve its low quantum efficiency very much in photo-oxidation reaction.

Recently, a number of studies demonstrated that photoelectrocatalytic(PEC) oxidation process achieved a

① **[Foundation item]** Project(01015) supported by Guangdong Natural Science Foundation; Project(C31507) supported by Guangdong Scientific and Technological development; Project(010873) supported by Guangdong Natural Science Foundation

[Received date] 2002 - 05 - 14; **[Accepted date]** 2002 - 8 - 08

higher efficiency by applying an electrical bias between a working electrode and a counter electrode for organic degradation, which can prevent carrier charge recombination and result in an extension of the active holes lifetime^[9, 10]. In their studies, the photo-anodes were prepared by coating TiO₂ on a conducting glass initially covered by an indium tin-oxide. It was found that electron mass transfer between TiO₂ films and supporting carriers was not very efficient owing to poor connection between the two materials. Recently, we successfully prepared an innovative Ti/TiO₂ mesh photoelectrode by anodising a TiO₂ film on titanium (Ti) mesh for PEC degradation of rose Bengal^[11], rhodamine B^[12], humic acid^[13] and methyl orange^[14, 15]. This electrode has a large surface area and its microporous surface structure achieves an excellent adsorption of substances, compared with other kinds of the immobilised TiO₂ film. And it has been proved that the PEC oxidation rate using Ti/TiO₂ mesh is higher than the photo-catalytic oxidation rate in a powder suspension solution. In order to evaluate the applicability of such Ti/TiO₂, more detail investigation is needed to seek for an available way to improve its PEC activity.

In this study, the Ti/TiO₂-mesh electrode as a new type of electrode was modified by gold deposited on the surface. Humic acid was used as a model chemical to conduct the PEC oxidation experiments using the Ti/Au-TiO₂ mesh photoelectrode. The objectives of this research are: 1) to seek for an available way for water treatment and to enhance the PEC reactivity of Ti/TiO₂ mesh by surface modification; 2) to investigate the influence of gold on the crystal structure, surface morphology, chemical and electronic structure of the Ti/TiO₂-mesh electrode.

2 EXPERIMENTAL

2.1 Materials

Titanium mesh (purity > 99.6%, nominal aperture 0.19 mm, wire diameter 0.23 mm, wires/inch 60×60, open area 20%, twill weave) was purchased from Goodfellow Cambridge Limited. Commercially available HA was supplied from Fluka with an ash content of about 20%. All other chemicals were used with AR grade.

2.2 Preparation of Ti/TiO₂ and Ti/Au-TiO₂ mesh electrode

A large piece of raw titanium mesh was cut into small rectangle pieces of 50 mm×10 mm×0.5 mm, which was then cleaned with alcohol and acetone solution successively. The treated Ti mesh and a copper plate

with the same size were submerged in 0.5 mol/L H₂SO₄ solution and an electrical current was applied by using a laboratory-made DC power supply. The anodisation process was conducted in two stages, in which the galvanostatic anodisation with a constant current density (0.110 A/m²) was first performed until a designated anode-to-cathode voltage (160 V) was reached. Then the constant voltage was maintained until the end of anodisation, while the current was decreasing gradually. The freshly generated TiO₂-mesh electrode was then rinsed by distilled water and dried in an oven at 378 K for 2 h. The preparation method had been described by Li et al^[11]. The photo-electrode was noted as 160 V-mesh (or No gold).

The anodized Ti/TiO₂ mesh electrode is modified by the photo-reduction method to deposit gold. The Ti/TiO₂ mesh is placed in a mixture of solution containing the required concentration of tetrachloroauric acid and 0.1 mol/L methanol. The solution was irradiated with a high-pressure mercury lamp. The irradiation lasted 20, 40, 60, 80 min respectively. The produced Ti/Au-TiO₂ mesh electrode is washed repeatedly with double distilled water and dried at 378 K for 2 h. The gold concentration on the surface of Ti/TiO₂ mesh was detected by XPS. The gold concentration on the surface of four pieces of Ti/Au-TiO₂ mesh electrodes was 2.01% (mole fraction), 3.34%, 6.71% and 17.3% respectively. And they were noted as Au1 #-mesh, Au2 #-mesh, Au3 #-mesh and Au4 #-mesh respectively.

2.3 Characterization of Ti/Au-TiO₂-mesh electrode

To determine the crystal phase composition of Ti/Au-TiO₂ mesh, X-ray diffraction (XRD) measurements was performed at room temperature with a Philips PW3710 diffractometer using Cu K_α radiation ($\lambda_1 = 0.154\ 060\ \text{nm}$; $\lambda_2 = 0.154\ 439\ \text{nm}$); accelerating voltage was 40 kV, and emission current was 30 mA. Laser Raman spectroscopy was used to determine the crystal pattern of TiO₂ (anatase and rutile) on the mesh electrode. At room temperature, Raman spectra were excited by 514.5 nm laser line from a CW argon laser (Coherent Innova 70). The laser power was kept at 250 mW to avoid laser annealing of the samples. A 55 mm f/1.8 lens was used for collecting the scattering light which was dispersed and detected using a double grating monochromator (Spex 1403) equipped with a cooled photomultiplier tube (PMT, Hamamatus R943-2). All spectra were recorded in a small angle scattering geometry. The resolution obtained was expected to be as good as 1 cm⁻¹. Scanning Electron Microscopy (SEM) (Leica Stereoscan

400i Series) was used to study the surface morphology, average pore size and pores distribution. High tension was selected at 15 kV. X-ray photoelectron spectroscopy (XPS) was recorded with a PHI Quantum ESCA Microprobe System, using the Mg K α line of a 250 W Mg X-ray tube as the radiation source with the energy of 1253.6 eV, 16 mA \times 12.5 kV. The working pressure was less than 1×10^{-8} N/m 2 . As an internal reference for the absolute binding energies, the C1s peak of hydrocarbon contamination was used. The fitting XPS curves were analyzed by use of Multipak 6.0 A.

2.4 Photo-reactor and PEC oxidation of HA

PEC oxidation experiment was performed in a batch photo-reactor system, which consisted of a cylindrical borosilicon glass reactor with an effective volume of 165 mL, a 125 W high-pressure mercury lamp, a potentiostat (ISO-TECH IPS 1810H) and electrodes. The 125 W high-pressure mercury lamp (Institute of Electric Light Source, Beijing) was positioned inside a cylindrical Pyrex vessel surrounded by a circulating water jacket (Pyrex) to cool the lamp. In the photo-reactor, two electrodes of Ti/Au-TiO $_2$ -mesh electrode as anode and copper plate electrode (50 mm \times 10 mm) as cathode were placed oppositely and connected with the potentiostat. 165 mL of HA solution was irradiated under constant aerating. At given irradiation time intervals, samples of the suspension were taken and analyzed.

In the study, a set of tests was carried out by using the Ti/Au-TiO $_2$ -mesh electrode to investigate the effect of gold on PEC efficiency. The TOC representing the PEC degradation degree of humic acid was determined with a Shimadzu TOC-5000A TOC analyzer. The light intensity was measured by using a Black-Ray UV meter (UVP Inc, Model No. J 221).

3 RESULTS AND DISCUSSION

3.1 Crystal structural analyses of Ti/Au-TiO $_2$ -mesh electrodes

The crystal phase composition of the Ti/Au-TiO $_2$ -mesh electrodes was examined by XRD. Their diffractions are shown in Fig. 1, where the peaks representing anatase and rutile forms of TiO $_2$ are labelled with A and R respectively. For 160 V-mesh, rutile phase presents at 27.405 $^\circ$ (110), 36.105 $^\circ$, 41.310 $^\circ$ and 56.685 $^\circ$. The 101 peak is very weak. For Ti/Au-TiO $_2$ mesh, the intensity of anatase and rutile peaks is influenced by gold and increases with the increase of the gold content. For all Ti/Au-TiO $_2$ mesh, 101 peak and 110 peak present significantly, compared with 160 V-mesh.

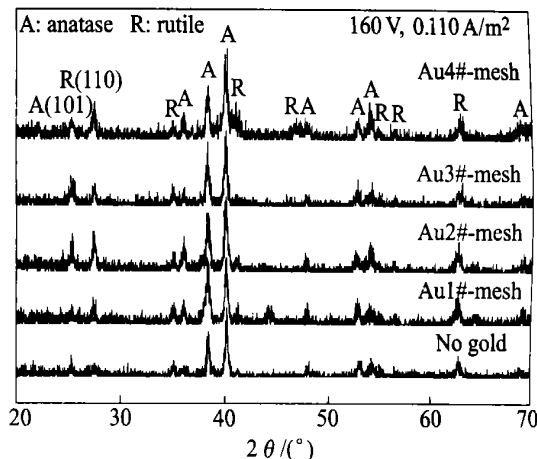


Fig. 1 X-ray diffraction results of Ti/Au-TiO $_2$ mesh electrodes

In this paper, the effect of gold deposition on the formation of crystal phase might be investigated in detail by means of laser Raman spectra shown in Fig. 2, in which the peaks representing the anatase and rutile forms of TiO $_2$ are labelled with A and R respectively. For 160 V-mesh, the anatase peaks are located at 102, 141, 222, 352, 519 and 638 cm $^{-1}$ while that of rutile are observed at 439 and 625 cm $^{-1}$. The strongest peak of anatase occurs at around 141 cm $^{-1}$ with a narrow line shape for 160 V-mesh. For all samples, the LR spectra also indicate that the anatase phase of TiO $_2$ is dominant. For Au1 #-mesh and Au2 #-mesh, the intensity of the phonon peak for anatase at 222 cm $^{-1}$ increases significantly whereas those of other peaks decrease slightly. For Au3 #-mesh and Au4 #-mesh, the intensity of the phonon peak for anatase decreases significantly and that of rutile does not appear, with the increase of the gold content. The surface morphology of Ti/Au-TiO $_2$ mesh as shown in Fig. 3 shows that gold clusters cover on the surface of Ti/Au-TiO $_2$ mesh.

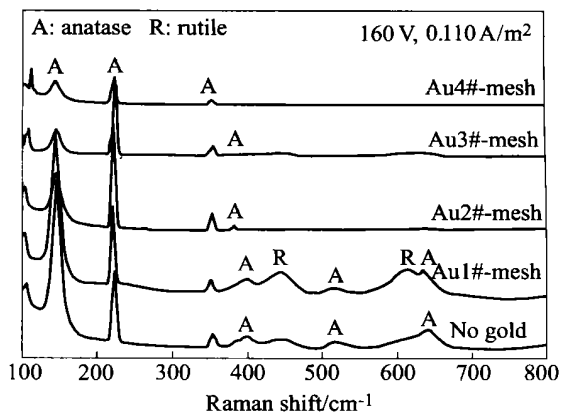


Fig. 2 Laser Raman spectroscopy of Ti/Au-TiO $_2$ mesh electrodes

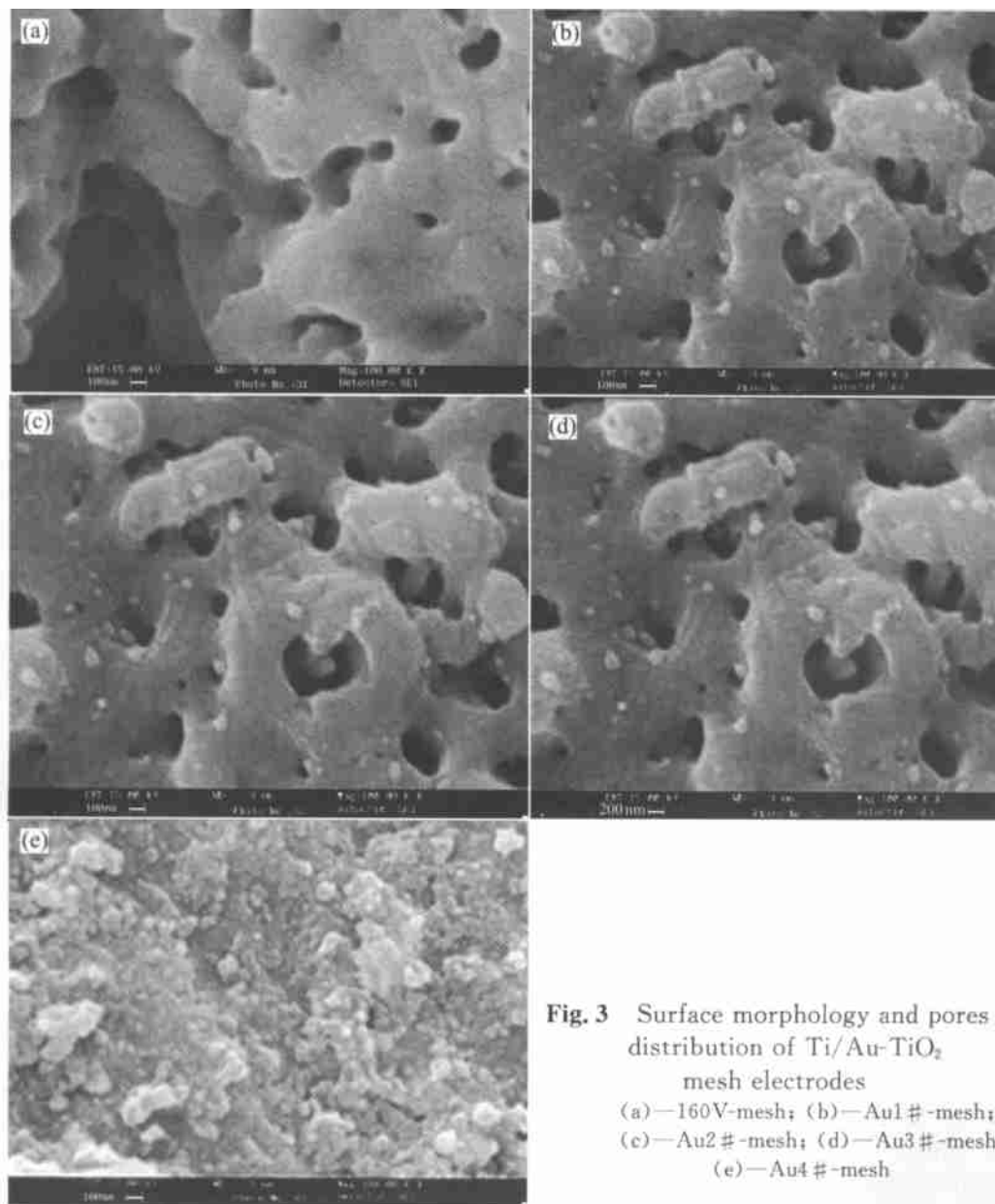


Fig. 3 Surface morphology and pores distribution of Ti/Au-TiO₂ mesh electrodes
 (a)—160V-mesh; (b)—Au1 #-mesh;
 (c)—Au2 #-mesh; (d)—Au3 #-mesh;
 (e)—Au4 #-mesh

3.2 Surface morphology and pore distribution of Ti/Au-TiO₂ mesh electrodes

The morphology of the Ti/Au-TiO₂-mesh electrodes is examined by means of SEM and their SEM photos are shown in Figs. 3(a) ~ 3(e). It is found that the surface of Ti/Au-TiO₂ film is microporous and rougher. The surface morphology of Ti/Au-TiO₂ mesh is influenced by the gold deposited on the surface, as shown in Figs. 3(b) ~ 3(e). For Au1 #-mesh, the size of gold clusters is about 15 ~ 30 nm and only a small part of surface is covered. However, for Au4 #-mesh, the size of gold clusters increases significantly and is about 15 ~ 70 nm. And the pores are almost filled up.

3.3 Electronic structure of Ti/Au-TiO₂ mesh electrodes

XPS analyses were carried out to determine the chemical and electronic structure of mesh surface and to

disclose the mechanism of the variation of PEC properties and PEC reactivity. The Ti 2p, O 1s, Au 4f and valence band XPS spectra of all mesh fitted by Multipak 6.0A software are listed in Table 1, Table 2 and shown in Figs. 4 ~ 7.

The O 1s spectra of Ti/Au-TiO₂ mesh are shown in Fig. 4, and the binding energy of O 1s is listed in Table 1. Sanjinés et al.^[19] reported that the O 1s spectra of reduced samples were proposed to consist of three components attributed to the oxide component (530.0 eV), hydroxyl groups or defective oxides (531.5 ± 0.5 eV), and oxygen chemisorbed or adsorbed water (533 ± 1 eV). In our investigation, the O 1s XPS spectra of Ti/Au-TiO₂ mesh shows a two-band structure. The peak at 529.59 ~ 529.83 eV is in agreement with O 1s electron binding energy for TiO₂ molecular with a shift toward to higher binding energy owing to gold deposition. Another O 1s band at 531 ~ 532 eV may be attributed to hydroxyl

Table 1 Molar percentage of gold and binding energy of O 1s, Ti 2p and Au 4f of Ti/ Au-TiO₂ mesh

Mesh	x(Au)/ %	Binding energy/ eV				
		O 1s	Ti 2p _{1/2}	Ti 2p _{3/2}	Au 4f _{5/2}	Au 4f _{7/2}
160 V-mesh	0	529.59	464.01	458.27		
Au1 #-mesh	2.2	529.63	464.23	458.35	87.62	83.94
Au2 #-mesh	3.4	529.81	464.44	458.54	87.49	83.86
Au3 #-mesh	6.7	529.83	464.54	458.59	87.39	83.71
Au4 #-mesh	17.3	529.94	464.37	458.62	87.30	83.63

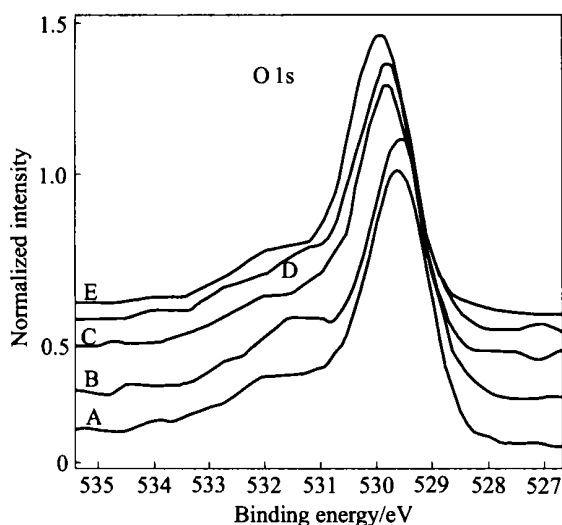


Fig. 4 O 1s XPS curves of Ti/ Au-TiO₂ mesh fitted by Multipak 6.0 A (Remarks A, B, C, D, E represent Ti/ TiO₂ mesh, Au 1 #, Au 2 #, Au 3 #, Au 4 # respectively, the same as following figures)

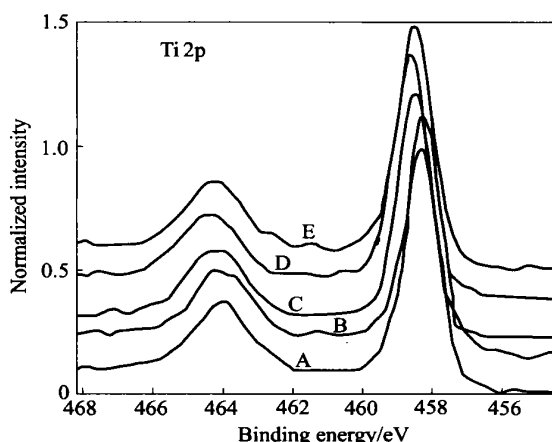


Fig. 5 Ti 2p XPS curves of Ti/ Au-TiO₂ mesh fitted by Multipak 6.0 A

groups. The intensities of Ti 2p and O 1s XPS peaks decrease with the increase of gold cluster covered on the surface of Ti/ Au-TiO₂ mesh.

The Ti 2p_{3/2} peak for Ti/ Au-TiO₂ mesh is narrow with slight asymmetry and has a binding energy of 458.27 ~ 458.62 eV, attributed to Ti⁴⁺, as shown in Fig.5. In the meantime, the Ti 2p_{1/2} peak is weaker and wider with a core level of 464.01 ~ 464.54 eV. The results show that the binding energies of Ti 2p_{3/2} and Ti 2p_{1/2} increase slightly owing to gold deposition. The intensity of Ti 2p XPS peaks decreases with the increase of gold cluster covered on the surface of Ti/ Au-TiO₂ mesh.

In Fig. 6, the Au 4f peak of Ti/ Au-TiO₂ mesh consists of two peaks, corresponding to Au 4f_{5/2} and Au 4f_{7/2} respectively. The binding energies of Au 4f_{5/2} and Au 4f_{7/2} slightly decrease with the increase of photo-reduction time and gold content. Ti/ Au-TiO₂ mesh was prepared by photo-reduction. At first, AuCl₄⁻ was absorbed on the surface of Ti/ TiO₂ mesh, then reduced into AuCl₂⁻ and Au⁰ or (Au⁰)_m at the end. Some nanopar-

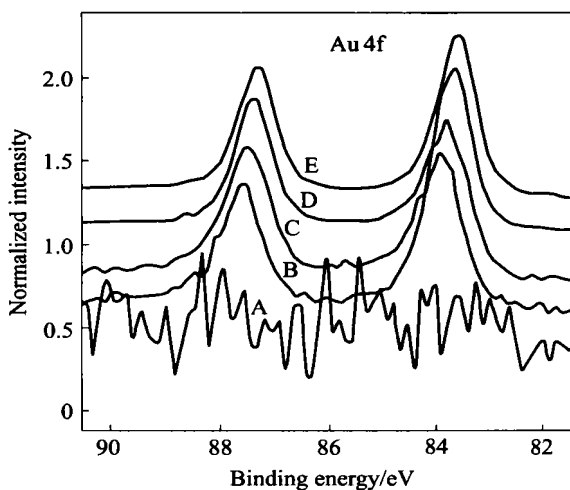


Fig. 6 Au 4f XPS curves of Ti/ Au-TiO₂ mesh fitted by Multipak 6.0 A

ticles of metals such as Au have strong optical absorption by surface plasmon resonance in the visible light wavelength^[17]. Metallic gold deposited on the surface of TiO₂ can significantly absorb visible light.

The valence-band spectra of Ti/ TiO₂ mesh and Ti/ Pt-TiO₂ mesh are shown in Fig.7. The valence band of Ti/ TiO₂ have two peaks: a board one at 4.97 eV and a narrow one at 6.61 eV, which correspond mainly to π (nonbonding) and σ (bonding) O 2p orbitals, as listed

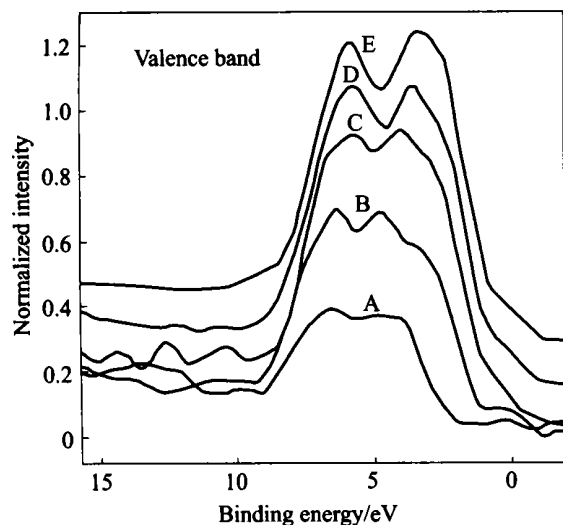


Fig. 7 Valence band spectra of XPS curves of Ti/Au-TiO₂ mesh fitted by Multipak 6.0 A

in Table 2. It deserves noting that the intensity emissions of O 2p nonbonding orbitals located at 4.84, 4.16, 3.66, 3.39 eV respectively for Au1 #, Au2 #, Au3 #, Au4 #, and became stronger and the width of the valence band and the emission intensity increased with the increase of Au content on the surface of mesh. In the meantime, the emission peaks of O 2p bonding orbitals are located at 5.56, 5.76, 5.89, 5.77 eV respectively. Compared to literature, Sanjinés et al.^[16] reported that the valence band of anatase TiO₂ was confined to binding energy between 3 and 9 eV, relative to Fermi level (E_F) showing two peaks: a board one at 6 eV and a narrow one at 8 eV; and a significant electron emission near the Fermi level indicated that Ti³⁺ defect states were located in the band gap, 0.7 ~ 0.9 eV below E_F . In our investigation, what was different from the XRD diagram of powder, the strongest peak of anatase for all mesh presented at 40.295°, not at 25.405° attributable to 101 peak. The intensities of anatase and rutile peaks increase with the increase of the gold content. The differences of XRD peaks position and intensity lead to the change of the valence band, band gap and surface states.

Table 2 Peaks of valence band spectra and hydroxyl group (eV)

Catalyst	Peak 1	Peak 2
TiO ₂	4.97	6.61
Au1 #	4.84	6.56
Au2 #	4.16	5.76
Au3 #	3.66	5.89
Au4 #	3.39	5.77

3.4 PEC reactivity of Ti/Au-TiO₂ mesh electrodes

To investigate the effect of gold deposition on the PEC reactivity of Ti/Au-TiO₂ mesh, a set of tests were conducted by using 160 V-mesh, Au1 #-mesh, Au2 #-mesh, Au3 #-mesh and Au4 #-mesh with the same light intensity of 4.38 mW·cm⁻² and without electrical bias voltage. The initial concentration of HA solution was 10 mg/L. The results show that the TOC removal would be greatly promoted when optimal gold was deposited on the surface of Ti/Au-TiO₂ mesh electrode as shown in Fig. 8. For 4 Ti/Au-TiO₂ mesh electrodes (Au1 #, Au2 #, Au3 #, Au4 #), the first-order kinetic constants were 0.014 9, 0.025 6, 0.018 8, 0.007 8 min⁻¹ respectively. And that of Ti/TiO₂ mesh was 0.010 4 min⁻¹. When the content of gold was 3.34%, the PEC efficiency was the highest, and the higher content of gold on the surface of Ti/TiO₂ mesh was detrimental to raise the PEC efficiency.

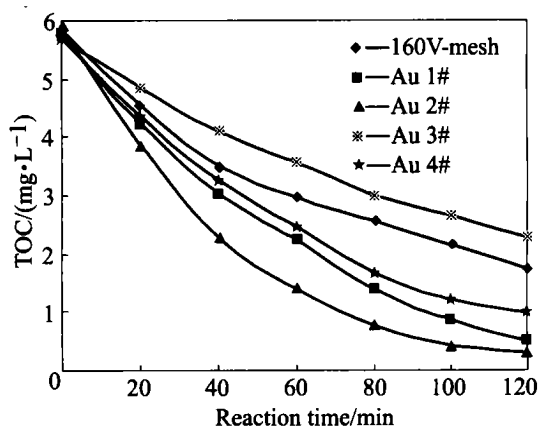
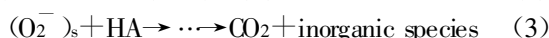
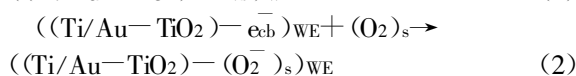
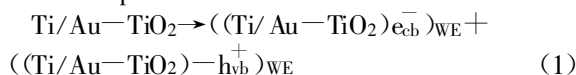


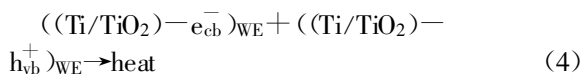
Fig. 8 TOC removal of HA solution with reaction time at pH=3.01

4 DISCUSSION

The PEC mechanism on the Ti/TiO₂ working electrode and the copper counter electrode has been proposed in detail in our previous works^[18]. For Ti/Au-TiO₂ mesh, the excited electrons from valence band to conduction band migrated to gold clusters, then migrated to O₂ molecule adsorbed on the surface of gold. Au on TiO₂ surface produces the highest Schottky barrier among the metals facilitating the electron capture^[19]. And O₂⁻ is an active species to oxidise the HA.



In addition to the redox reactions on the working and counter electrodes, there is an important reaction on the working electrode. The reaction is the recombination of excited electrons and holes.



In fact, the charge recombination process was a significant factor in decreasing the PEC efficiency on TiO_2 film^[20-21]. Therefore, reducing the recombination of electrons and holes is essential to promote PEC efficiency. In our investigation, optimal content of gold deposited on the surface of Ti/TiO_2 mesh can improve the PEC rate. Many investigations show that gold deposited on the TiO_2 surface could promote the excited electron transfer, immigration and separation and hinder the recombination of electrons and holes^[19-21, 22].

5 CONCLUSIONS

1) Photoelectrocatalysis using a new type of photo-electrode, $\text{Ti}/\text{Au}-\text{TiO}_2$ -mesh electrode that was successfully innovated by anodizing Ti Mesh in 0.5 mol/L H_2SO_4 solutions and by photo-reduction, can be used to degrade efficiently HA. TOC in HA solution was removed rapidly after irradiation for 120 min.

2) The XRD and laser Raman spectra indicated that the anatase form of TiO_2 was dominant on the surface of the electrode, and the intensities of anatase and rutile XRD peaks all increased with the increase of gold on the mesh surface, and the intensity of laser Raman peaks decreased owing to a higher gold deposition.

3) SEM measurement showed that gold cluster covered on the surface of $\text{Ti}/\text{Pt}-\text{TiO}_2$ mesh and filled up the pore of the mesh. XPS measurement showed that the binding energy of O 1s and Ti 2p increased slightly and that of Au 4f_{7/2} and Au 4f_{5/2} decreased slightly owing to gold deposition. The PEC activity of 160 V-mesh was promoted when an optimal content of gold was deposited.

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(Edited by PENG Chao-qun)