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High-performance organic thin-film transistor by using LaNbO as gate dielectric

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Pentacene organic thin-film transistors (OTFTs) using $La_xNb_{(1-x)}O_y$ as gate dielectric with different La contents ($x = 0.347, 0.648$) have been fabricated and compared with those using Nb oxide or La oxide. The OTFT with $La_{0.648}Nb_{0.352}O_y$ as gate dielectric can achieve a high carrier mobility of $1.14\,cm^2V^{-1}s^{-1}$ (about 1000 times and 2 times those of the devices using Nb oxide and La oxide, respectively), and has negligible hysteresis of $-0.130\,V$, small sub-threshold swing of $0.280\,V/$dec, and low threshold voltage of $-1.35\,V$. AFM and XPS reveal that La can suppress the formation of oxygen vacancies in Nb oxide while Nb can alleviate the hygroscopicity of La oxide, which results in a more passivated and smoother dielectric surface, leading to larger pentacene grains grown and thus higher carrier mobility. The OTFT with Nb oxide has an anticlockwise hysteresis but the device with La oxide shows an opposite direction. This can be explained in terms of donor-like traps due to oxygen vacancies and acceptor-like traps originated from hydroxyl ions formed after La_2O_3 absorbing water moisture. © 2015 AIP Publishing LLC.
[http://dx.doi.org/10.1063/1.4927098]

Pentacene organic thin-film transistors (OTFTs) have been intensively studied in the past three decades due to their potential applications in radio-frequency identification tags, sensors, and large-area flexible displays. Although some OTFTs have shown high carrier mobility comparable to that of amorphous silicon devices,[1–3] the operating voltage of the OTFTs still needs to be reduced as many electronic products work with battery at just a few volts. OTFTs fabricated with traditional SiO_2, PMMA, or polyvidone (PVP) as gate dielectric require a high operating voltage due to large threshold voltage ($> 5\,V$).[4–6] There are several ways to reduce the threshold voltage: (1) Use cross-linking agents to modify the low-κ polymer gate dielectrics;[7,8] (2) use hybrid organic-inorganic dielectrics with smaller thickness;[9,10] (3) use high-κ materials as gate dielectric. Various high-κ materials have been employed as the gate dielectric of pentacene OTFTs, and their carrier mobility varies from 2×10^{-2} to $1.4\,cm^2V^{-1}s^{-1}$ with low threshold voltage.[11–13] The performance of the OTFT highly depends on the quality of its high-κ gate dielectric, such as surface roughness and trap density. Recently, the addition of La has been reported to suppress the oxygen vacancies in Hf-based high-κ materials,[14,15] and HfLaO was used as gate dielectric for high-performance pentacene OTFTs, showing high carrier mobility, small threshold voltage, low hysteresis, and small sub-threshold swing (SS).[16,17]

Nb_2O_5 has high κ value ($\sim$29), but it has small band gap ($\sim$4.4 eV) and suffers from a high concentration of oxygen vacancies,[18] whose gap states lead to charge trapping. On the other hand, La_2O_3 has larger band gap ($\sim$6 eV), but it is very hygroscopic and unstable in air, easily forming hydroxide.[19] In this work, $La_xNb_{(1-x)}O_y$ films with different La contents

are prepared by RF sputtering and proposed as the gate dielectric of pentacene OTFT in an attempt to combine the advantages of La_2O_3 and Nb_2O_5 while suppressing their undesirable properties. By using $La_xNb_{(1-x)}O_y$ with an appropriate La content (64.8% La) as gate dielectric, high-performance OTFT with high carrier mobility and low threshold voltage can be obtained.

Initially, the standard RCA method was employed to clean silicon wafers (n-type, $\langle 100 \rangle$, resistivity of $0.5 \sim 1.0$ Ω·cm). After removing the native oxide by dipping the wafers in 2% hydrofluoric acid, $La_xNb_{(1-x)}O_y$ (with $x = 0$, 0.347, 0.648 and 1) films were deposited on the wafers as gate dielectric by sputtering in an Ar/O_2 (24:6 sccm) ambient with a radio-frequency sputterer (Denton Vacuum LLC Discovery 635) using Nb and La_2O_3 targets. After that, the samples were annealed in N_2 at 400 °C for 20 min, with a flow rate of 1000 sccm. Next, an evaporator (Edwards Auto 306) was used to deposit 30-nm pentacene ($>99\%$, purchased from Lumtech) on the dielectrics in high vacuum (3.8×10^{-6} torr) at a deposition rate of 1.2 nm/min. Finally, the drain and source electrodes of the OTFTs were evaporated through a shadow mask with a channel of 200 μm in width and 30 μm in length.

The electrical characteristics of the OTFTs were characterized by HP 4145B semiconductor parameter analyzer. Al/high-κ dielectric/heavily doped Si MOS structure was also fabricated by lithography and measured by HP 4284A precision LCR meter to extract the capacitance per unit area (C_{ox}). All the measurements were conducted at room temperature, in air, and under shielded environment. The thickness of the dielectric films was measured by a Wvase 32 ellipsometer. A Veeco multi-mode scanning probe microscope was employed to record the surface morphologies of the dielectrics and pentacene films, and X-ray photoelectron spectroscopy

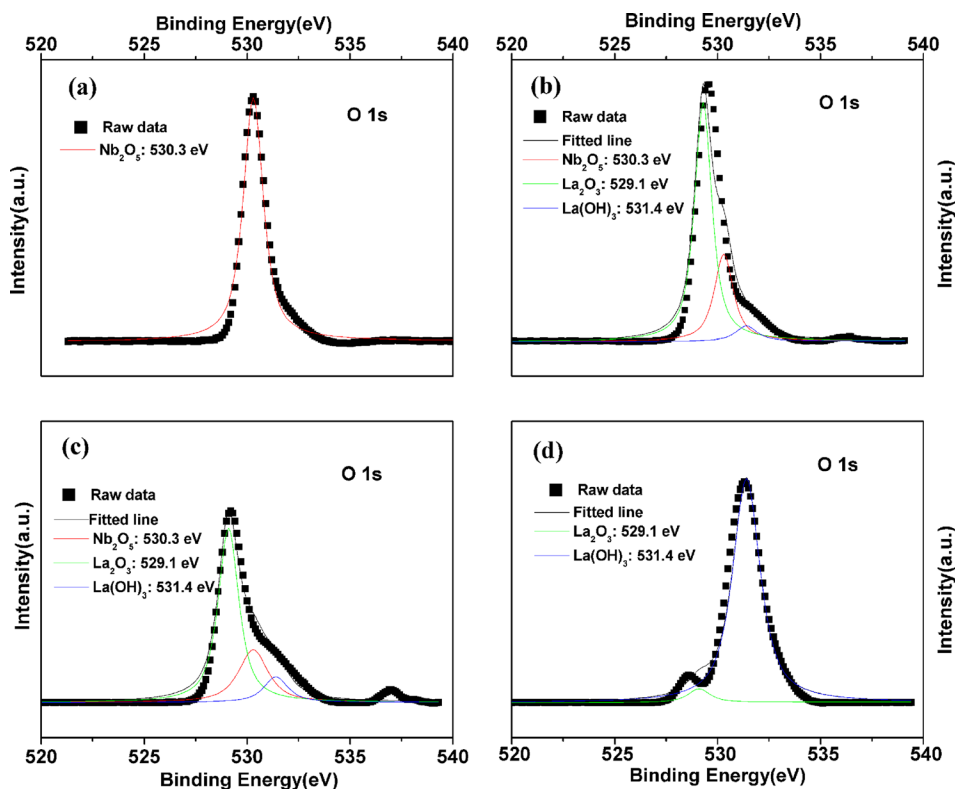a)Author to whom correspondence should be addressed. Electronic mail: laip@eee.hku.hk

FIG. 1. XPS spectrum of O 1s for $La_xNb_{(1-x)}O_y$ with $x = 0$ (a), 0.347 (b), 0.648 (c), and 1 (d).

(XPS) was employed to analyze the physical characteristics of the high-κ dielectric films. The binding energies were corrected based on C 1s peak at 284.5 eV.

According to XPS results, the atomic ratio of La/(Nb+La) is 0%, 34.7%, 64.8%, and 100% for the 4 samples, denoted as A, B, C, and D, respectively. Fig. 1 shows the XPS spectrum of O 1s core levels for the dielectric films. It is clear that when the La content increases, more La-OH bonds are formed in the $La_xNb_{(1-x)}O_y$ film due to the hygroscopicity of the La oxide. Sample D (with $x = 1$) becomes almost $La(OH)_3$ due to the absorption of the moisture in the air.

Fig. 2 shows the XPS spectra of Nb 3d and La 3d. As shown in Fig. 2(a), the binding energy of Nb 3d shifts to lower value with the increase of the La content, which should be caused by the La incorporation suppressing the oxygen vacancy in the Nb oxide. It is widely accepted that the oxygen vacancy is positively charged and generates an electric field,[20–22] thus leading to binding-energy shift.[23] The binding energy increases when an atom is placed near a positive charge.[23] Hence, reducing the oxygen vacancies in the Nb oxide and thus the positive charges around the Nb atoms by the addition of La can result in a shift of Nb 3d peak toward lower binding energy by 1.1 eV when La content reaches 64.8% in sample C. On the other hand, incorporating Nb in La oxide can generate more positive oxygen vacancies, resulting in higher binding energy for La 3d. However, Fig. 2(b) shows that the binding energy of La first decreases but then increases with increasing Nb content. La-OH bond has higher binding energy than La-O bond. The La 3d peak first shifts to lower binding energy because Nb incorporation enhances the moisture resistance of La oxide, thus reducing the La-OH bond in the oxide. Further increasing the Nb content can generate more oxygen vacancies, leading to a shift of the binding energy to a higher value.

The surface roughness of the dielectric is an important parameter in deciding the pentacene growth and thus device performance. Reducing the surface roughness (thus nucleation sites for pentacene growth) can cause increases in
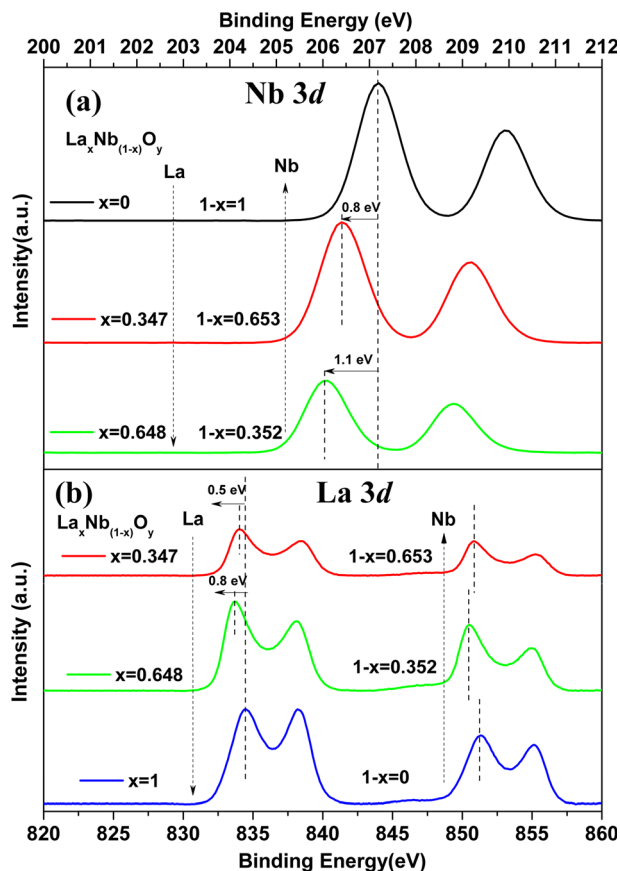

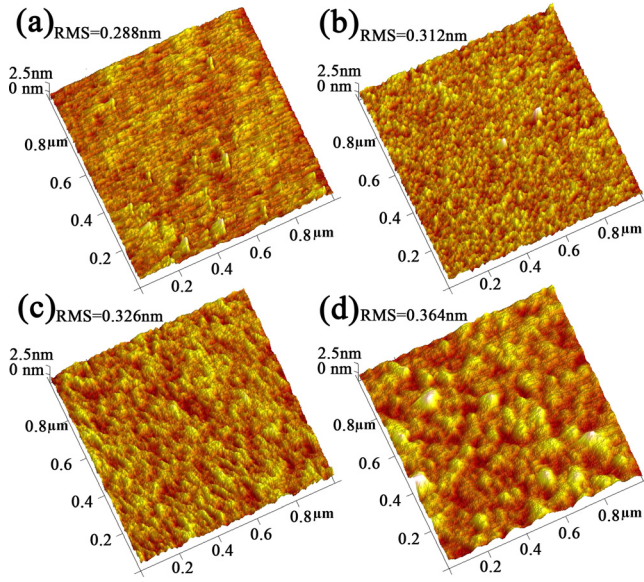FIG. 2. XPS spectra of Nb 3d (a) and La 3d (b).

FIG. 3. AFM image of the gate dielectric films: (a) for sample A with RMS = 0.288 nm; (b) for sample B with RMS = 0.312 nm; (c) for sample C with RMS = 0.326 nm, and (d) for sample D with RMS = 0.364 nm.
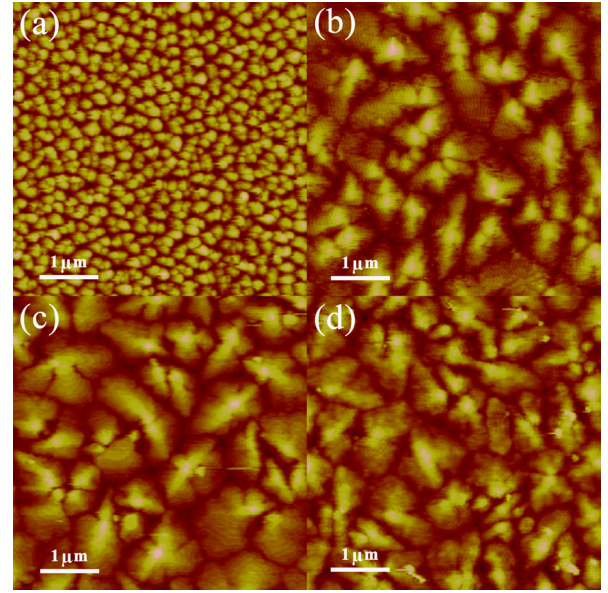

FIG. 4. AFM image (5×5 μm^2) of pentacene film (30 nm) on $La_xNb_{(1-x)}O_y$ with x =: (a) 0 for sample A, (b) 0.347 for sample B, (c) 0.648 for sample C, and (d) 1 for sample D.

pentacene grain size and carrier mobility.[24,25] La oxide is unstable in the air, and its surface roughness is increased after absorbing the moisture in the air. As depicted in Fig. 3, incorporation of Nb in La_2O_3 reduces the surface roughness of the gate dielectric from 0.364 nm to 0.288 nm because the addition of Nb can alleviate the hygroscopicity of La oxide.

The surface energy (γ_S) of the samples is calculated by measuring the contact angles (θ) of three liquids (water, diiodomethane and glycerol) in static mode. The relation between the surface energy and contact angle is expressed by[26]

$$1 + \cos(\theta) = \frac{2(\gamma^d)^{1/2}(\gamma_L^d)^{1/2}}{\gamma_L} + \frac{2(\gamma^p)^{1/2}(\gamma_L^p)^{1/2}}{\gamma_L}, \quad (1)$$

where γ^d and γ^p are the dispersion and polar components of the surface energy, respectively, and the total surface energy γ_S is calculated by summing the two components. The surface energy (γ_L), dispersion component (γ_L^d) and polar component (γ_L^p) used to solve Eq. (1) for γ^d and γ^p are, respectively, 77.2, 22.0, and 50.2 mJm^{-2} for water, 64, 34, and 30 mJm^{-2} for glycerol, and 50.8, 48.5, and 2.3 mJm^{-2} for diiodomethane. As shown in Table I, for increasing La content, the total surface energy and the polar component increase, but the dispersion component decreases.

Fig. 4 shows the AFM image of 30-nm pentacene film deposited on different dielectric films. Higher surface energy was reported to cause larger pentacene grains.[27–29] So, by comparing samples B, C, and D with sample A, it is highly possible that their higher surface energies (see Table I) cause the larger grains. Sample C has nearly the same surface energy as sample B but lower surface energy than sample D. However, sample C has the largest grains among the samples. This indicates that surface roughness and oxygen vacancies (both can act as nucleation centers) can also impact the surface morphology of the pentacene film. The pentacene grains on the La-doped Nb oxide or La oxide (see Figs. 4(b)–4(d)) are much larger than those on the Nb oxide (see Fig. 4(a)). Although the Nb oxide has the smoothest surface among the samples, the pentacene grain size on Nb oxide is the smallest, indicating that the oxygen vacancies at the dielectric surface can act as nucleation sites in the process of pentacene growth, and smaller grains are formed on the surface with more nucleation sites.[30–33] On the other hand, larger pentacene grains can be formed on the La-doped Nb oxide and La oxide films with fewer nucleation sites. Moreover, by comparing sample A with samples B and C, it can be concluded that the oxygen vacancies at the dielectric

TABLE I. Main parameters of OTFTs with different La concentrations extracted from the off-to-on sweep of transfer characteristics and properties of gate dielectrics. With 11 devices measured for each sample, the average values of the parameters are listed together with the highest carrier mobility and standard deviation (SD).

Sample No.	A	B	C	D
La/(Nb+La)	0	0.347	0.648	1
μ (cm^{2}V^{-1}s^{-1})	0.001	0.543	1.14	0.519
(SD)	(5.3×10^{-4})	(0.059)	(0.096)	(0.098)
Highest	0.002	0.612	1.30	0.690
V_T (V)	3.28	−0.85	−1.35	−2.25
(SD)	(0.495)	(0.214)	(0.189)	(0.289)
ΔV_T (V)	1.52	1.09	−0.13	−0.34
SS (V/dec)	3.41	0.279	0.208	0.346
I_{on}/I_{off}	753	4.78×10^4	2.90×10^5	2.93×10^4
I_{on} (μA) @$V_D = V_G = -5$ V	0.153	8.15	10.46	4.45
C_{ox} (μF·cm^2)	1.10	0.267	0.222	0.220
μC_{ox} (μF·V^{-1}s^{-1})	0.00122	0.145	0.253	0.114
t_{ox} (nm)	32.2	38.3	42.9	32.3
Dielectric constant	39.7	11.6	10.7	8.0
N_t (cm^{-2}eV^{-1})	3.87×10^{14}	6.15×10^{12}	3.46×10^{12}	6.61×10^{12}
γ^p (mJm^{-2})	14.7	25.4	26.8	29.5
γ^d (mJm^{-2})	39.9	38.5	37.2	35.3
γ_S (mJm^{-2})	54.6	63.9	64.0	64.8

surface play a more important role than the surface roughness of the gate dielectric in the nucleation process of pentacene growth.

Assuming the densities of deep bulk trap and deep interface trap are independent of trap energy,[34] the trap density at the interface can be extracted from *SS* as

$$N_t = \frac{C_{ox}}{e}\left(\frac{e \cdot SS}{kT\ln 10} - 1\right), \qquad (2)$$

where N_t is the density of deep bulk trap and interface trap per unit area and unit energy, k the Boltzmann's constant, e the electron charge, and T the temperature in Kelvin. Therefore, the trap density at the interface can be calculated from the measured *SS* value.

The transfer and hysteresis characteristics of the OTFTs are shown in Fig. 5. The off-on (forward) sweep was measured by decreasing the gate voltage from 3 V to −5 V, while the on-off (reverse) sweep was done by increasing the gate voltage from −5 V to 3 V. There was negligible pause time between the two sweeps. In Table I, the key parameters of the OTFTs are extracted from the off-to-on sweep of transfer characteristics and averaged over 11 devices. ΔV_T is defined as V_T of off-to-on (forward) sweep minus V_T of on-to-off (reverse) sweep as shown in Fig. 5(b). The OTFT with $La_{0.648}Nb_{0.352}O_y$ exhibits the highest carrier mobility of 1.14 cm^{-1}V^{-1}s^{-1}, which is about 1000 times and 2 times higher than those of the devices using Nb oxide and La oxide, respectively. The highest carrier mobility of sample C (La 64.8%) should be attributed to the largest pentacene grains (see Fig. 4) resulted from the best passivation of

oxygen vacancy in the dielectric (smallest *SS* or ΔV_T in Table I) by La incorporation and smooth dielectric surface (see Fig. 3(c)). With increasing Nb content, the carrier mobility firstly increases due to the smoother dielectric surface by Nb incorporation, but then decreases as more oxygen vacancies are generated by Nb incorporation, leading to smaller pentacene grains and more carrier scatterings at the grain boundaries.

Hysteresis effects are often observed in OTFTs during the forward and reverse sweeps of the gate voltage. Various mechanisms are employed to explain the phenomena.[35] The hysteresis of OTFTs is frequently attributed to the traps at the interface and in the bulk of dielectric. As shown in Fig. 5, the OTFT with Nb oxide shows a clear anticlockwise hysteresis with a very large ΔV_T. When the La content is increased to 64.8%, the hysteresis of the OTFT is reduced greatly to a negligible value of −0.13 V and has an opposite direction. Furthermore, the OTFT with La oxide shows a larger clockwise hysteresis.

Oxygen vacancies at the interface and in the bulk of the dielectric act as donor-like traps and are positively charged when they are empty, and neutral when filled with electron.[36–38] The process that the donor-like traps release electrons can be treated as that they trap holes. These donor-like traps (hole traps) capture the holes fast but release them slowly. During the on-to-off sweep, initially at $V_G = -5$ V, many donor-like traps capture holes, and as V_G decreases, the trapped holes are slowly released, hence compared with the off-to-on sweep, fewer mobile holes in the channel at a given V_G (see the inset graphs of Fig. 5(a)), resulting in smaller I_D. In addition, the trapped holes can generate a

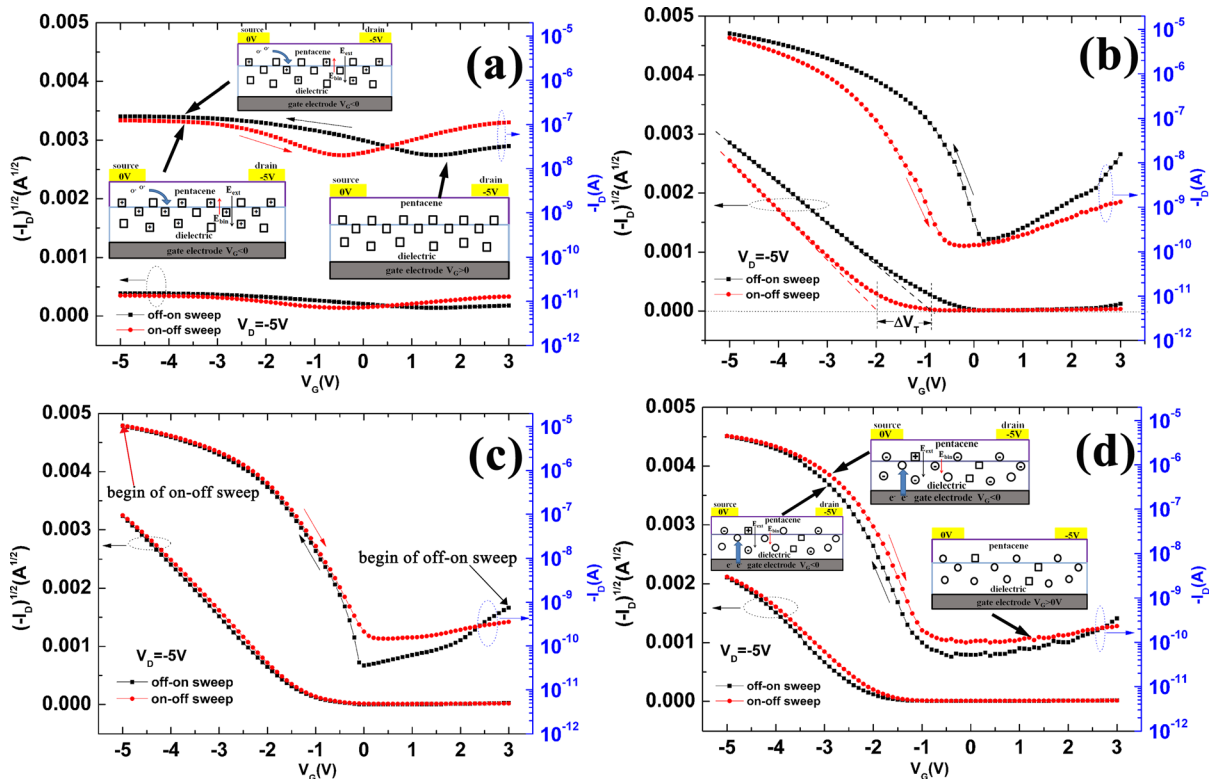

FIG. 5. Transfer and hysteresis characteristics of the OTFTs with different La contents: (a) sample A, (b) sample B, (c) sample C, and (d) sample D; □ and ⊞ represent the filled and empty donor-like traps, respectively; ○ and ⊖ are the empty and filled acceptor-like traps, respectively; O$^+$ and e$^-$ are hole and electron, respectively.

built-in electric field to oppose the external electric field, and thus, the threshold voltage is shifted to a negative direction, causing an anticlockwise hysteresis as shown in Figs. 5(a) and 5(b). On the other hand, La_2O_3 absorbs water and generates hydroxide (OH), which is frequently reported to capture electrons and thus can function as acceptor-like traps in dielectrics.[39,40] During the on-to-off sweep, electrons are injected from the gate electrode into the dielectric and trapped by the accepter-like traps at the interface and bulk of the dielectric. Upon sweeping V_G in the positive direction, the electrons remain in the channel and the bulk, generating a built-in electric field to enhance the external electric field, thus accumulating more holes in the channel, leading to larger I_D and a shift of the threshold voltage in the positive direction. The clockwise hysteresis is clearly observed in samples C and D (see Figs. 5(c) and 5(d)). Among the samples, sample C shows the smallest hysteresis. With La incorporation in Nb oxide, the hysteresis and thus the instability of threshold voltage can be suppressed. In short, the donor-like traps (originated from the oxygen vacancies induced by the Nb incorporation) cause the anticlockwise hysteresis of the OTFTs, but the acceptor-like traps (due to OH in La_2O_3 after absorbing moisture) produce the opposite hysteresis.

In conclusion, pentacene OTFTs using $La_xNb_{(1-x)}O_y$ as gate dielectric with different La contents ($x = 0$, 0.347, 0.648, 1) have been fabricated. The dielectric films were deposited using co-sputtering of Nb and La_2O_3 targets by varying the sputtering powers of the targets in an Ar/O_2 ambient. The binding-energy shift of the XPS spectrum demonstrates that the La incorporation can suppress the oxygen vacancies in Nb oxide, thus reducing the nucleation sites at the oxide surface to obtain larger pentacene grains grown. On the other hand, the Nb incorporation in La oxide can reduce the hygroscopicity of La oxide, resulting in a smoother dielectric surface, which can also give rise to larger pentacene grains. Therefore, with appropriate La content, the OTFT with $La_{0.764}Ta_{0.236}O_y$ as gate dielectric can achieve the largest pentacene grains as well as smoother dielectric surface, and thus the highest carrier mobility of $1.14\,cm^2V^{-1}s^{-1}$, which is about 1000 times and 2 times those of the devices using Nb oxide and La oxide, respectively. In addition, the anticlockwise hysteresis of the OTFT with Nb_2O_5 is induced by the donor-like traps originated from oxygen vacancies, while the clockwise hysteresis of the OTFT with La_2O_3 is caused by the hydroxyl ions in the dielectric after moisture absorption. In summary, La incorporation in Nb oxide can passivate its oxygen vacancies, thus obtaining high carrier mobility, small threshold voltage, and negligible hysteresis for the OTFT.

[1]P. F. Baude, D. A. Ender, M. A. Haase, T. W. Kelley, D. V. Muyres, and S. D. Theiss, Appl. Phys. Lett. **82**, 3964 (2003).
[2]M. H. Zhang, S. P. Tiwari, and B. Kippelen, Org. Electron. **10**, 1133 (2009).
[3]M. Devynck, P. Tardy, G. Wantz, Y. Nicolas, L. Vellutini, C. Labrugere, and L. Hirsch, Appl. Phys. Lett. **100**, 053308 (2012).
[4]J. Roh, C. M. Kang, J. Kwak, C. Lee, and B. J. Jung, Appl. Phys. Lett. **104**, 173301 (2014).
[5]W. Wang, J. H. Han, J. Ying, and W. F. Xie, IEEE Trans. Electron Devices **61**, 3507 (2014).
[6]H. S. Byun, Y. M. Xu, and C. K. Song, Thin Solid Films **493**, 278 (2005).
[7]C. Wang, W.-Y. Lee, R. Nakajima, J. Mei, D. H. Kim, and Z. Bao, Chem. Mater. **25**, 4806 (2013).
[8]M. E. Roberts, N. Queralto, S. C. B. Mannsfeld, B. N. Reinecke, W. Knoll, and Z. N. Bao, Chem. Mater. **21**, 2292 (2009).
[9]P. Cosseddu, S. Lai, M. Barbaro, and A. Bonfiglio, Appl. Phys. Lett. **100**, 093305 (2012).
[10]J. M. Choi, D. K. Hwang, S. H. Jeong, J. H. Park, E. Kim, J. H. Kim, and S. Ima, J. Electrochem. Soc. **154**, H331 (2007).
[11]M. Shtein, J. Mapel, J. B. Benziger, and S. R. Forrest, Appl. Phys. Lett. **81**, 268 (2002).
[12]R. E. I. Schropp, B. Stannowski, and J. K. Rath, J. Non-Cryst. Solids **299**, 1304 (2002).
[13]R. P. Ortiz, A. Facchetti, and T. J. Marks, Chem. Rev. **110**, 205 (2010).
[14]N. Umezawa, K. Shiraishi, S. Sugino, A. Tachibana, K. Ohmori, K. Kakushima, H. Iwai, T. Chikyow, T. Ohno, Y. Nara, and K. Yamada, Appl. Phys. Lett. **91**, 132904 (2007).
[15]D. Liu and J. Robertson, Appl. Phys. Lett. **94**, 042904 (2009).
[16]M. F. Chang, P. T. Lee, S. P. McAlister, and A. S. Chin, IEEE Electron Device Lett. **30**, 133 (2009).
[17]C. Y. Han, W. M. Tang, C. H. Leung, C. M. Che, and P. T. Lai, IEEE Electron Device Lett. **34**, 1397 (2013).
[18]W. K. Chen, J. Chem. Phys. **47**, 1144 (1967).
[19]Y. Zhao, M. Toyama, K. Kita, K. Kyuno, and A. Toriumi, Appl. Phys. Lett. **88**, 072904 (2006).
[20]T. L. Duan, H. Y. Yu, L. Wu, Z. R. Wang, Y. L. Foo, and J. S. Pan, Appl. Phys. Lett. **99**, 012902 (2011).
[21]S. Guha and V. Narayanan, Phys. Rev. Lett. **98**, 196101 (2007).
[22]K. Xiong, J. Robertson, M. C. Gibson, and S. J. Clark, Appl. Phys. Lett. **87**, 183505 (2005).
[23]P. S. Bagus, F. Illas, G. Pacchioni, and F. Parmigiani, J. Electron. Spectrosc. Relat. Phenom. **100**, 215 (1999).
[24]S. Steudel, S. De Vusser, S. De Jonge, D. Janssen, S. Verlaak, J. Genoe, and P. Heremans, Appl. Phys. Lett. **85**, 4400 (2004).
[25]S. E. Fritz, T. W. Kelley, and C. D. Frisbie, J. Phys. Chem. B **109**, 10574 (2005).
[26]D. K. Owens and R. C. Wendt, J. Appl. Polym. Sci. **13**, 1741 (1969).
[27]S. Y. Yang, K. Shin, and C. E. Park, Adv. Funct. Mater. **15**, 1806 (2005).
[28]P. R. Ribič, V. Kalihari, C. D. Frisbie, and G. Bratina, Phys. Rev. B **80**, 115307 (2009).
[29]B. J. Song, K. Hong, W. K. Kim, K. Kim, S. Kim, and J. L. Lee, J. Phys. Chem. B **114**, 14854 (2010).
[30]M. Voigt, J. Pflaum, and M. Sokolowski, Phys. Status Solidi A **205**, 449 (2008).
[31]M. C. Kwan, K. H. Cheng, P. T. Lai, and C. M. Che, Solid State Electron. **51**, 77 (2007).
[32]S. Godlewski and M. Szymonski, Int. J. Mol. Sci. **14**, 2946 (2013).
[33]C. Sanchez-Sanchez, G. Bavdek, D. Cvetko, M. F. Lopez, J. A. Martin-Gago, and L. Floreano, J. Phys. Chem. C **115**, 4664 (2011).
[34]W. L. Kalb and B. Batlogg, Phys. Rev. B **81**, 035327 (2010).
[35]M. Egginger, S. Bauer, R. Schwödiauer, H. Neugebauer, and N. S. Sariciftci, Monatsh Chem. **140**, 735 (2009).
[36]R. F. Janninck and D. H. Whitmore, J. Chem. Phys. **37**, 2750 (1962).
[37]H. Sawada and K. Kawakami, J. Appl. Phys. **86**, 956 (1999).
[38]H. B. Xiao, C. P. Yang, C. Huang, L. F. Xu, D. W. Shi, V. V. Marchenkov, I. V. Medvedeva, and K. Baürner, J. Appl. Phys. **111**, 063713 (2012).
[39]L. L. Chua, J. Zaumseil, J. F. Chang, E. C. W. Ou, P. K. H. Ho, H. Sirringhaus, and R. H. Friend, Nature **434**, 194 (2005).
[40]S. H. Kim, H. Yang, S. Y. Yang, K. Hong, D. Choi, C. Yang, D. S. Chung, and C. E. Park, Org. Electron. **9**, 673 (2008).