

Reduced magnetic coercivity and switching field in conetic-alloy-based synthetic-ferrimagnetic nanodots

X. Li¹ , C. W. Leung² , K.-W Lin³ , M. S. Chan⁴ , P. W. T. Pong^{1, *}

¹ Department of Electrical and Electronic Engineering, The University of Hong Kong, Hong Kong

² Department of Applied Physics, The Hong Kong Polytechnic University, Hong Kong

³ Department of Materials Science and Engineering, National Chung Hsing University, Taichung 402,
Taiwan

⁴ Department of Electronic and Computer Engineering, Hong Kong University of Science and
Technology, Hong Kong * Tel: +852 28578491, Fax: +852 25598738, Email: ppong@eee.hku.hk

Abstract

The coercivity (H_c) and switching field (H_{sw}) of free layers increase remarkably with shrinking size, which reduces the sensitivity of spintronic devices. Conetic-alloy-based synthetic ferrimagnetic (SyF) trilayers are proposed to show reduced H_c and H_{sw} than single-layer in nanodots. The investigation on the thickness dependence reveals linear reliance of H_c and H_{sw} on amplification factor. H_c and H_{sw} are further reduced after field annealing at 200 °C. This work provides an approach to reduce the H_c and H_{sw} in nanomagnets.

Key words: synthetic-ferrimagnetic, nanodots, nanosphere lithography

Introduction

The development of spintronics has brought to various novel electronic devices such as magnetoresistance (MR) sensors, spin-torque oscillators (STO), and magnetic random access memories (MRAM). In these spin-valve based devices, ferromagnetic (FM) materials with low coercivity (H_c) and small switching field (H_{sw}) are required as free layers. However, the H_c and H_{sw} in nanostructures are typically much larger than that in the planar films due to the increased demagnetization field with shrinking size. By engaging synthetic-ferrimagnetic (SyF) free layer, the H_c of submicrometer magnetic tunnel junction can be reduced while high thermal stability is maintained [1]. Conetic alloy ($\text{Ni}_{77}\text{Fe}_{14}\text{Cu}_5\text{Mo}_4$) is an FM material with smaller H_c than traditional soft magnetic materials [2]. While the previous investigations on the magnetic properties of SyF are mainly based on NiFe, CoFe and CoFeB, whether smaller H_c and H_{sw} can be achieved in conetic-alloy-based SyF nanostructures remains unclear. Meanwhile, annealing is reported to be beneficial for reducing the H_c of MR sensor with conetic alloy free layer [3]. An experimental investigation on the magnetic properties and annealing effect of conetic-alloy-based SyF nanodots will be beneficial for evaluating the feasibility of engaging conetic-alloy-based SyF as free layers in nanometric spintronic devices. Besides, nanosphere lithography [4] is a cost-effective nanofabrication technique that is capable of parallel production of nanostructures on large area. Its high yield for sample fabrication made it an ideal method for this research. In this work, the Ru thickness in SyF planar films is firstly optimized to achieve small H_c and H_{sw} . Later, the magnetic properties of SyF nannodots with different FM thicknesses are compared with that of the single-layer nanodots. Finally, the SyF films and nanodots are field annealed to explore the annealing effect of H_c and H_{sw} .

Experimental

The SyF structures are Ta 5/Cu 5/ Ni₇₇Fe₁₄Cu₅Mo₄ tF/ Ru tRu/ Ni₇₇Fe₁₄Cu₅Mo₄ 3/Ru 5 (thickness in nanometers). The fabrication processes of nanodot arrays are schematically shown in Fig. 1. The films are sputtered on Si/SiO₂ substrate under in-plane magnetic field to define the easy axis (Fig 1(a)). Non-close-packed 120-nm polystyrene nanospheres are distributed on the surface by electrical static adsorption (Fig 1(b)). The nanodot arrays are formed after ion milling (Fig 1(c)) and cleaning to remove the residual spheres (Fig 1(d)). The vacuum annealing is conducted at 100 °C to 400 °C for 1 hour under 0.1 T magnetic field along the easy axis.

Results and discussions

The SEM images of the nanosphere masks and nanodot arrays are shown in Fig. 1(e) and (f), respectively. Non-close-packed spheres are uniformly distributed with average diameter of 120 nm and average center-to-center distance of 220 nm. The diameter of the dots are reduced to around 60 nm in the tilted ion milling process. The dot-to-dot interactions are relatively weak due to the large distances amongst the units. The thickness of Ru (tRu) in SyF planar films are varied from 0.5 to 1.1 nm while tF is maintained at 7 nm to investigate the influence of spacer thickness. Minimum H_c of 0.72 Oe and H_{sw} of 1.1 Oe can be achieved at Ru thickness of 0.5 or 1.3 nm (Fig. 2). In the following investigation, tRu = 1.3 nm is used. The magnetic properties are comparatively studied in SyF nanodots (tF = 5 – 11 nm) and single layer nanodots (6 nm). Curved hysteresis loops with low squareness are observed (Fig. 3 (a)), due to the increased demagnetization field in nanostructures [5]. The H_c (2.3 – 17 Oe) and H_{sw} (10 – 58 Oe) of the SyF nanodots are all smaller than the single-layer nanodot (H_c = 20 Oe and H_{sw} = 70 Oe). This is because the demagnetization field at the edges of the nanodots are reduced by the interlayer

coupling between the two FM layers [6, 7]. The H_c of SyF nanodot can be interpreted as the combined contribution of anisotropy and in-plane demagnetization [8]: $H_c = 2K_u/M_s \times (t_1+t_2)/(t_1-t_2) + N_d M_s/d \times (t_1-t_2)$ (1) where K_u is the uniaxial anisotropy, M_s is the saturation magnetization, t_1 and t_2 are the thickness of two FM layer, N_d is the demagnetization factor in easy axis, d is the size of the dot. Since circular dot is discussed here, N_d equals to 0. The term $(t_1 + t_2)/(t_1 - t_2)$ is defined as the amplification factor (I). As shown in Fig. 3 (b), the measured H_c and H_{sw} exhibit linear relationships with I ($I = (t_F+3)/(t_F-3)$). The intercept is believed to be resulted from the demagnetization in the edges. These results indicates that smaller H_c and H_{sw} is expected at larger thickness difference between two FM layers. The minor hysteresis loops of as-deposited and field-annealed SyF nanodots and planar films are shown in Fig. 4 (a) and (b), respectively. The squariness ratio decreases at high annealing temperature ($T_a > 300$ °C), indicating the possible loss of ferromagnetism. The H_c of nanodots and planar films present similar annealing effect (Fig. 4(c)). The H_c reaches minimum as T_a increases to 200 °C. The reduced H_c is attributed to the enhanced crystalline fitness at interfaces [2]. However, further increasing T_a to 400 °C results in remarkable increase in H_c and H_{sw} , especially in nanodots. This is possibly resulted from the reduced crystallinity due to oxidization or inter-diffusion at high temperature. This inference is confirmed by the atomic force microscopy and X-ray diffraction characterization, which will be shown in the full paper.

Conclusions

Conetic-alloy-based SyF thin films and nanodot arrays are prepared by nanosphere lithography, and the dependence of H_c and H_{sw} on Ru thickness, FM layer thickness, and annealing temperature are investigated. Smallest H_c and H_{sw} are observed in SyF thin films when $t_{Ru} = 0.5$ or 1.3 nm. The H_c and H_{sw} of SyF nanodots are much smaller than that of single-layer

nanodots, and exhibit linear relationship with I . Minimum H_c of 4.1 Oe and H_{sw} of 16 Oe is achieved in SyF nanodots after 200 °C field annealing, while higher annealing temperature tends to increase H_c . This work has proved that Conetic-alloy-based SyF structure is capable of providing low H_c and H_{sw} in nanometer regime, thus holding some promise as free layers of nanometric spintronic devices.

References

- [1] Y. M. Lee et al., J. Appl. Phys., 101, p. 023905, 2007.
- [2] J.-G. Choi et al., J Magn. Magn. Mater., 322, pp. 2191-2194, 2010.
- [3] J. Son et al., Jpn. J. Appl. Phys., 51, p. 033002, 2012.
- [4] H. Fredriksson et al., Adv. Mater., 19, pp. 4297-4302, 2007.
- [5] J. Tang et al., J. Appl. Phys., 89, pp. 7690-7692, 2001.
- [6] J. Janesky et al., Appl. Phys. Lett., 85, pp. 2289-2291, 2004.
- [7] K. Inomata et al., Appl. Phys. Lett., 81, pp. 310-312, 2002.
- [8] K. Inomata et al., Appl. Phys. Lett., 82, pp. 2667-2669, 2003.

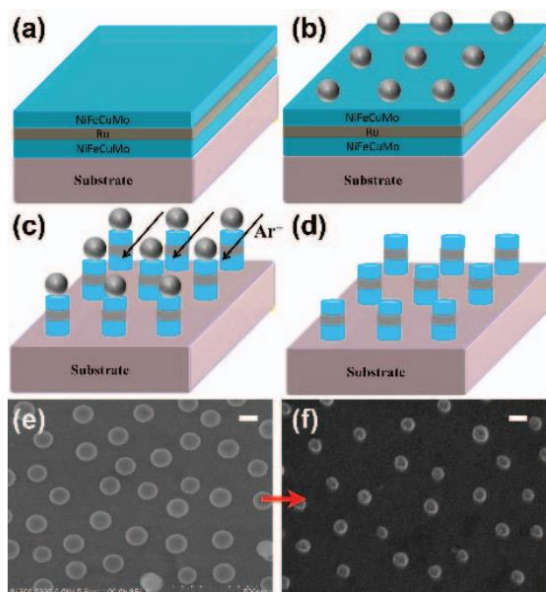


Fig. 1 (a)–(d) The fabrication processes of nanodots, and the SEM images of (e) nanosphere masks and (f) patterned nanodots

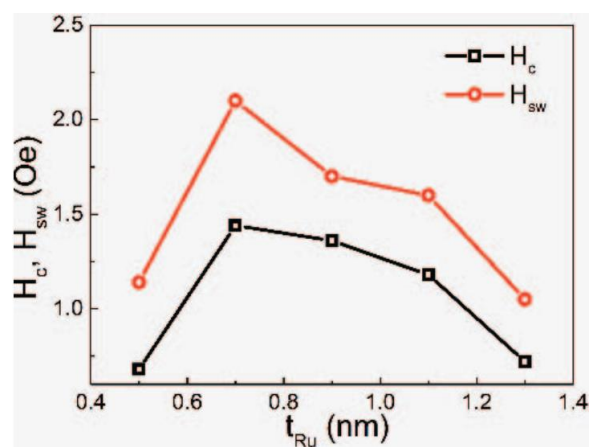


Fig. 2 H_c and H_{sw} of SyF planar films with different Ru thickness

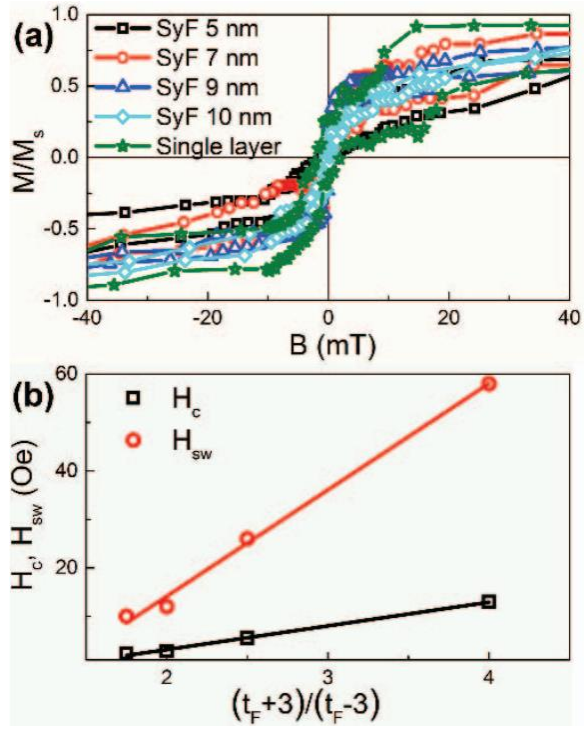


Fig. 3 (a) The minor hysteresis loop of nanodots of single layer and SyF at different t_F . (b) H_c and H_{sw} as a function of amplification ratio.

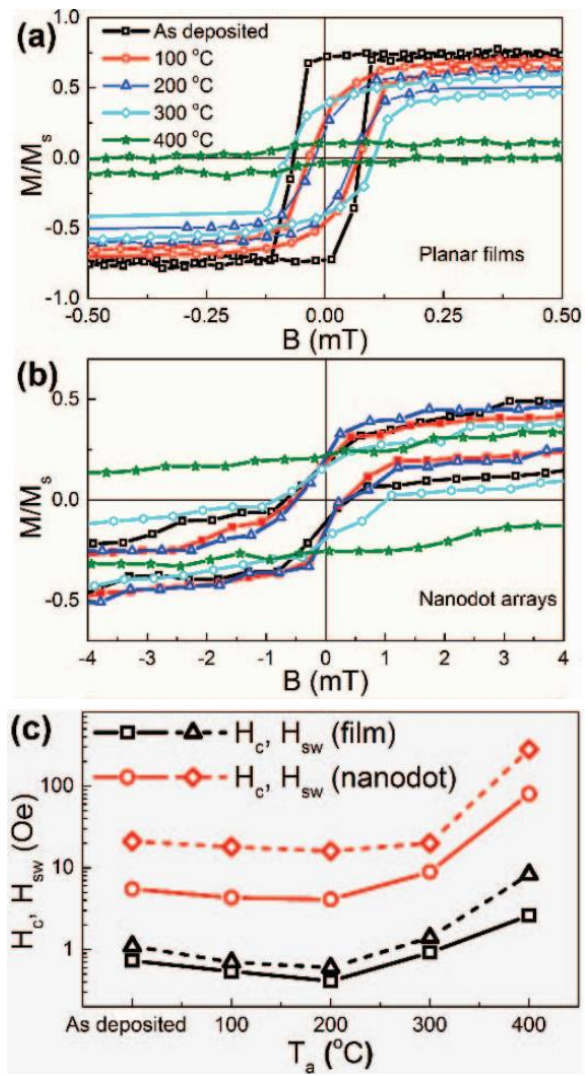


Fig. 4 The minor hysteresis loop of (a) nanodots and (b) planar films at different annealing temperature. (c) H_c and H_{sw} as a function of T_a .